



**Inverted and thin NEG, where we are**

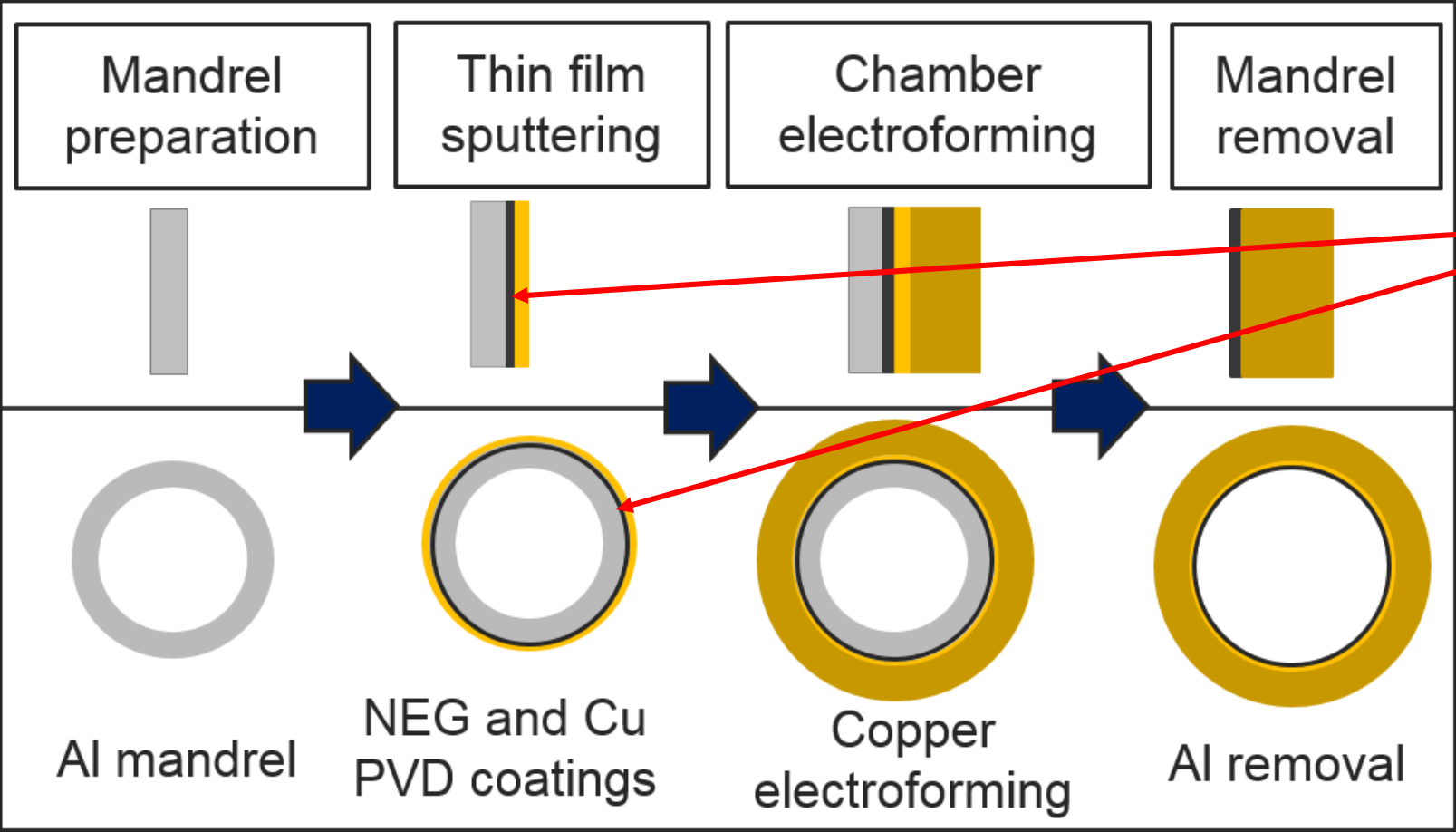
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P.Chiggiato**

**TE-VSC**

# Summary

- Inverted NEG scheme
- Performance of inverted NEG
- Performance of thin NEG < 1 $\mu$ m

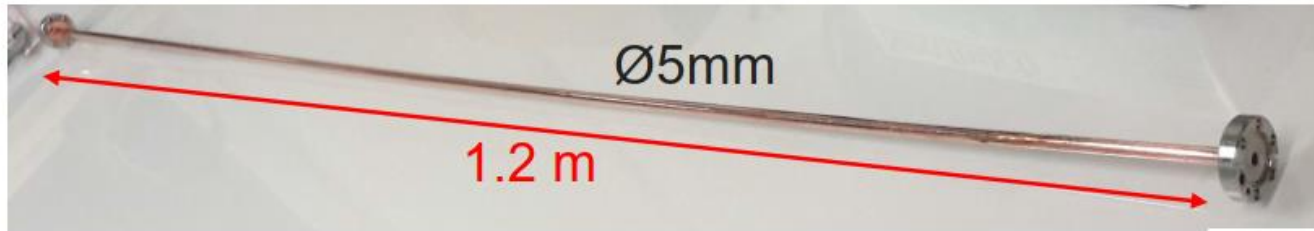
# Inverted NEG (Non Evaporable Getter) scheme



Additional NEG layer on the Al mandrel

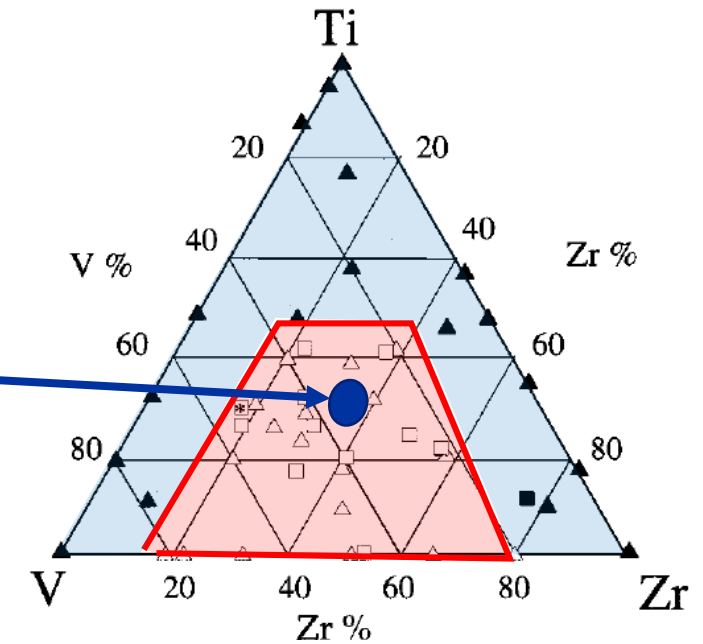
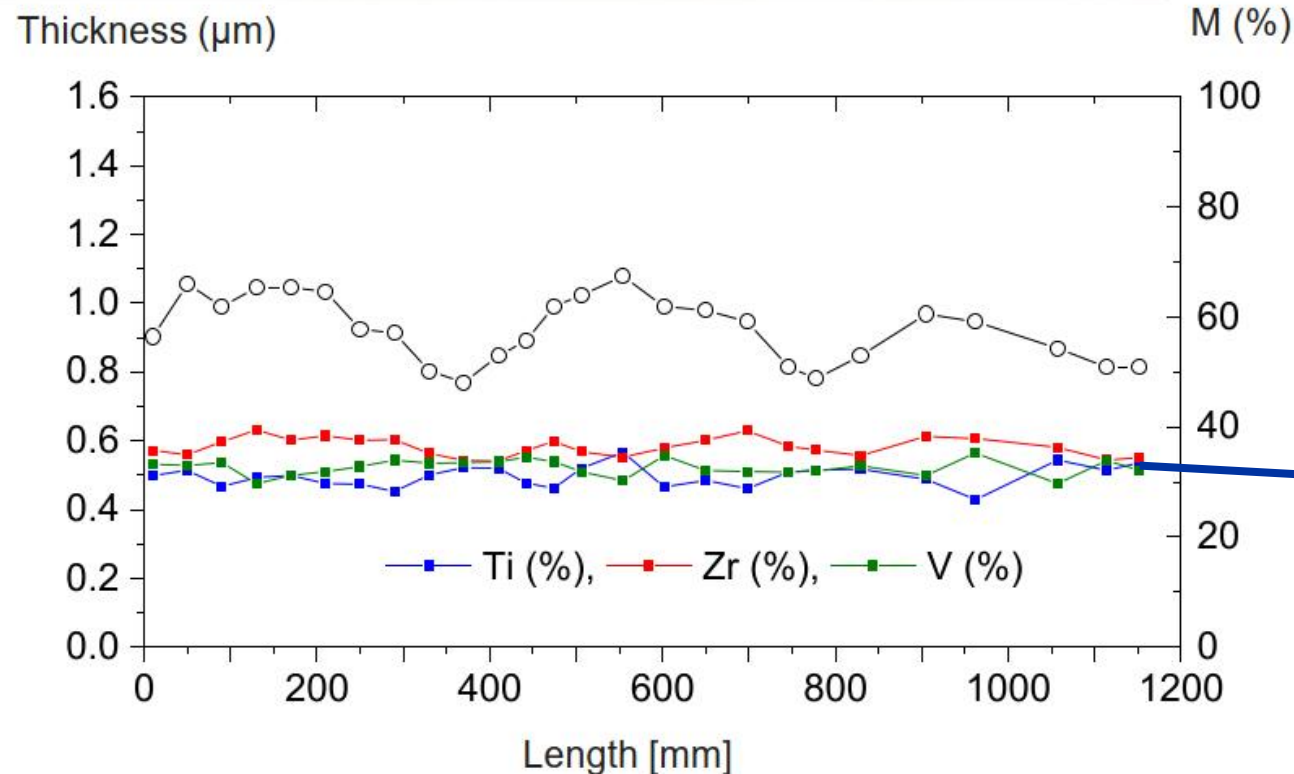
# Control of the NEG thickness

5mm internal diameter, 1.2 m length, TiZrV coated chamber



Analysis in XRF after cutting the chamber:

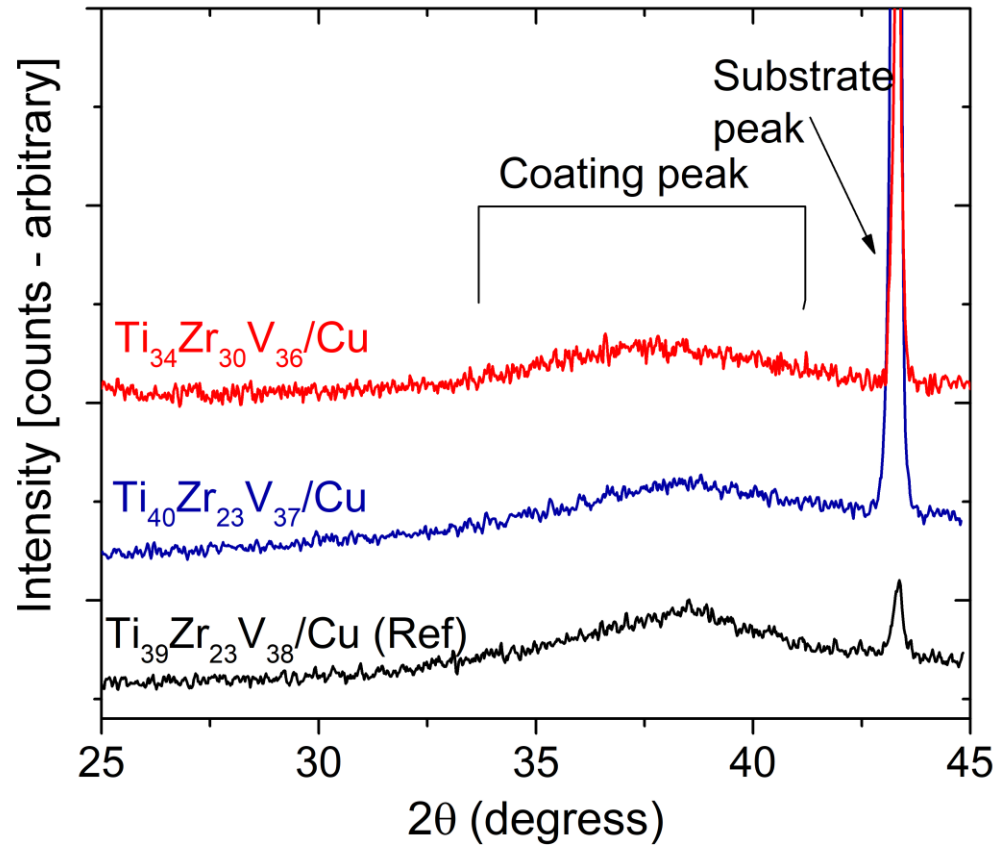
- The thickness is uniform
- Composition is uniform and correct



A. E. Prodromides et al. Vacuum 60 (2001) 35, 41

# Crystalline Structure

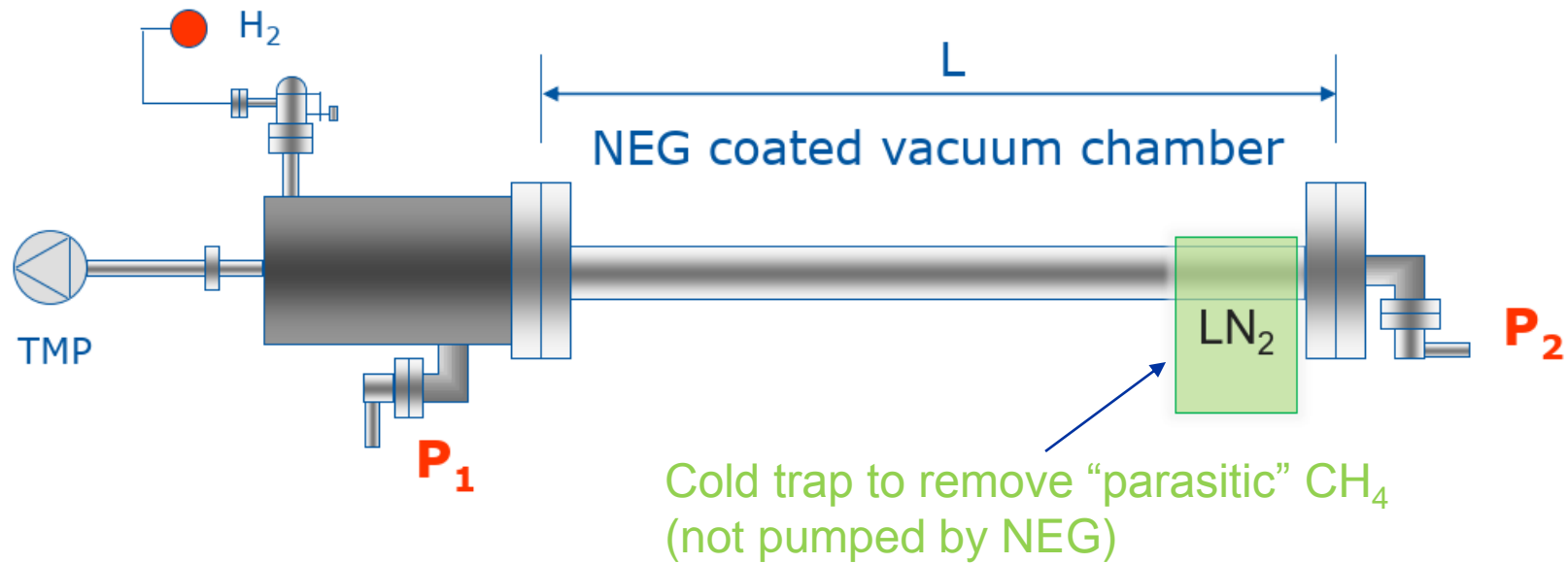
## XRD analysis



All samples show the small crystal size (estimated around 3 nm)

← Ref. prepared by the conventional coating process

# Pumping performance: transmission method

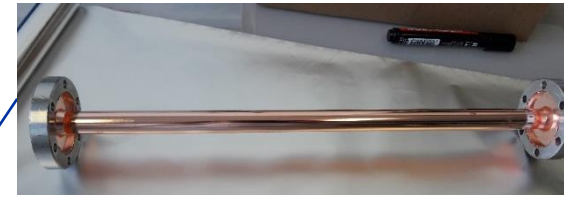
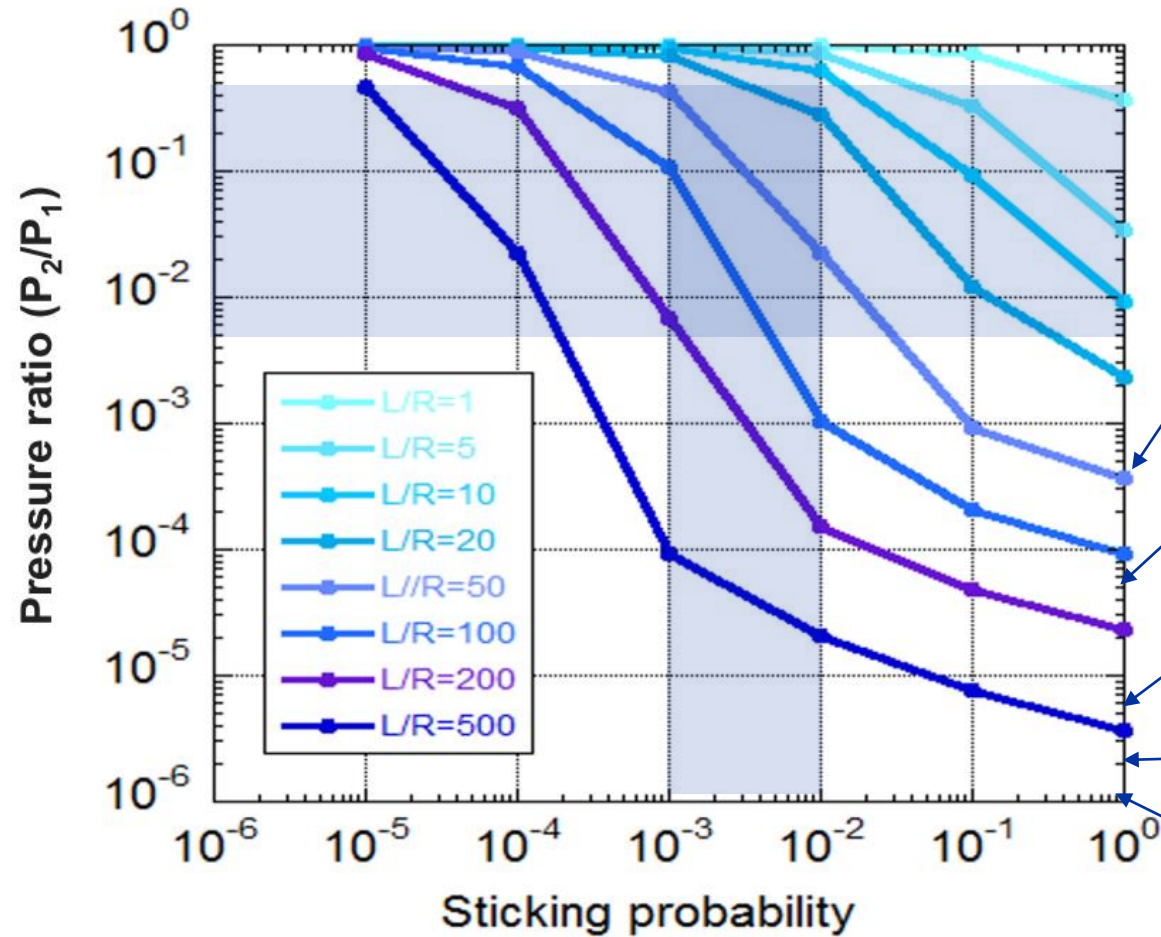


Steady state upon H<sub>2</sub> injection:

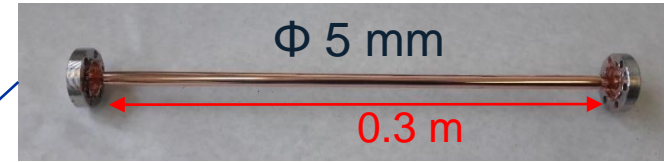
$P_2/P_1$  + Monte Carlo calculation (Molflow) : enable to get the sticking coefficient for hydrogen

# Pumping performance: what can be measured

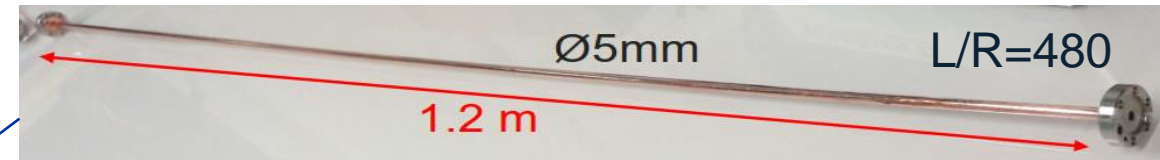
We expect a sticking of  $10^{-3}$  -  $10^{-2}$  and can easily measure a pressure ratio of 0.5-0.005: **not for too slim chambers!**



$\Phi$  12 mm x 0.3 m  
L/R=50



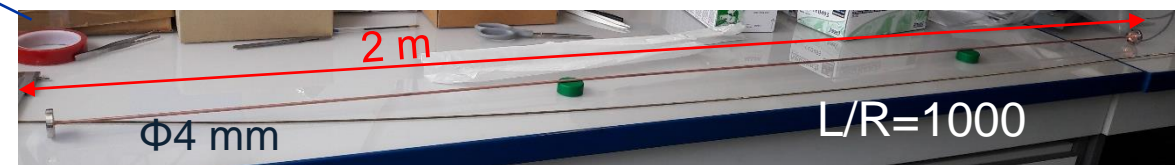
L/R=120



L/R=480



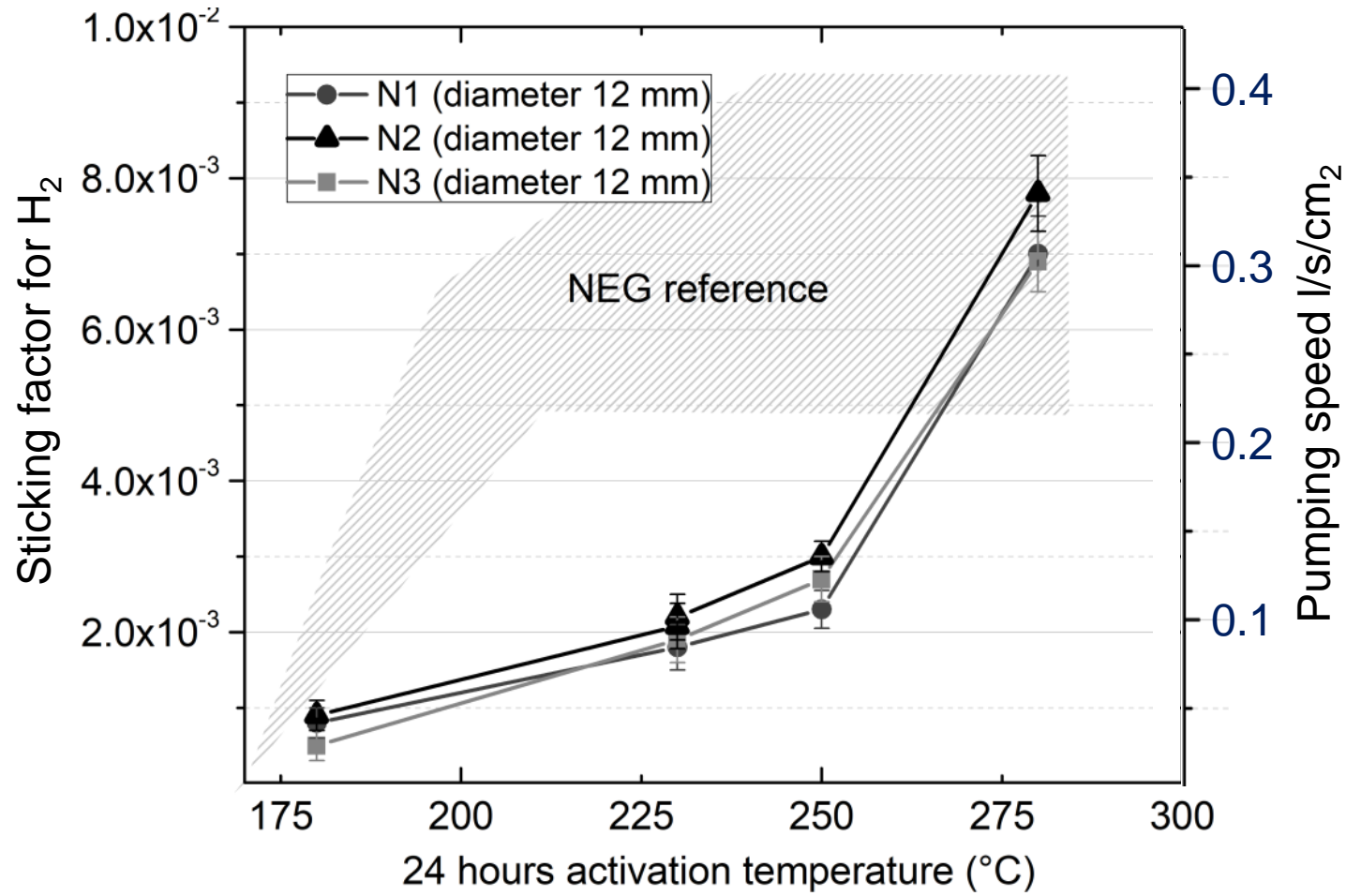
L/R=666



L/R=1000



# Pumping performance: sticking factor and speed



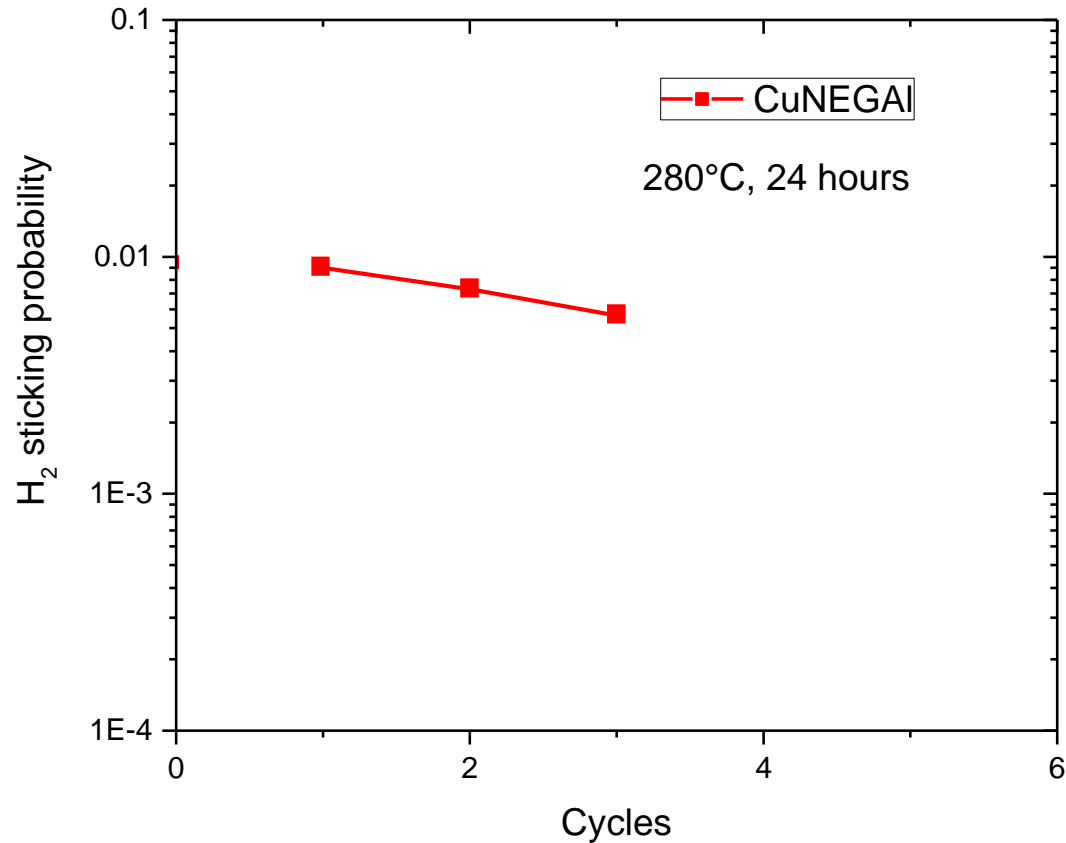
- Reproducible
- Reaches reference sticking and pumping speed (Ex: For a chamber of 6 mm diam. the pumping speed for H<sub>2</sub> will be 5.6 l/s/m)
- Activation is delayed in temperature compared to standard reference



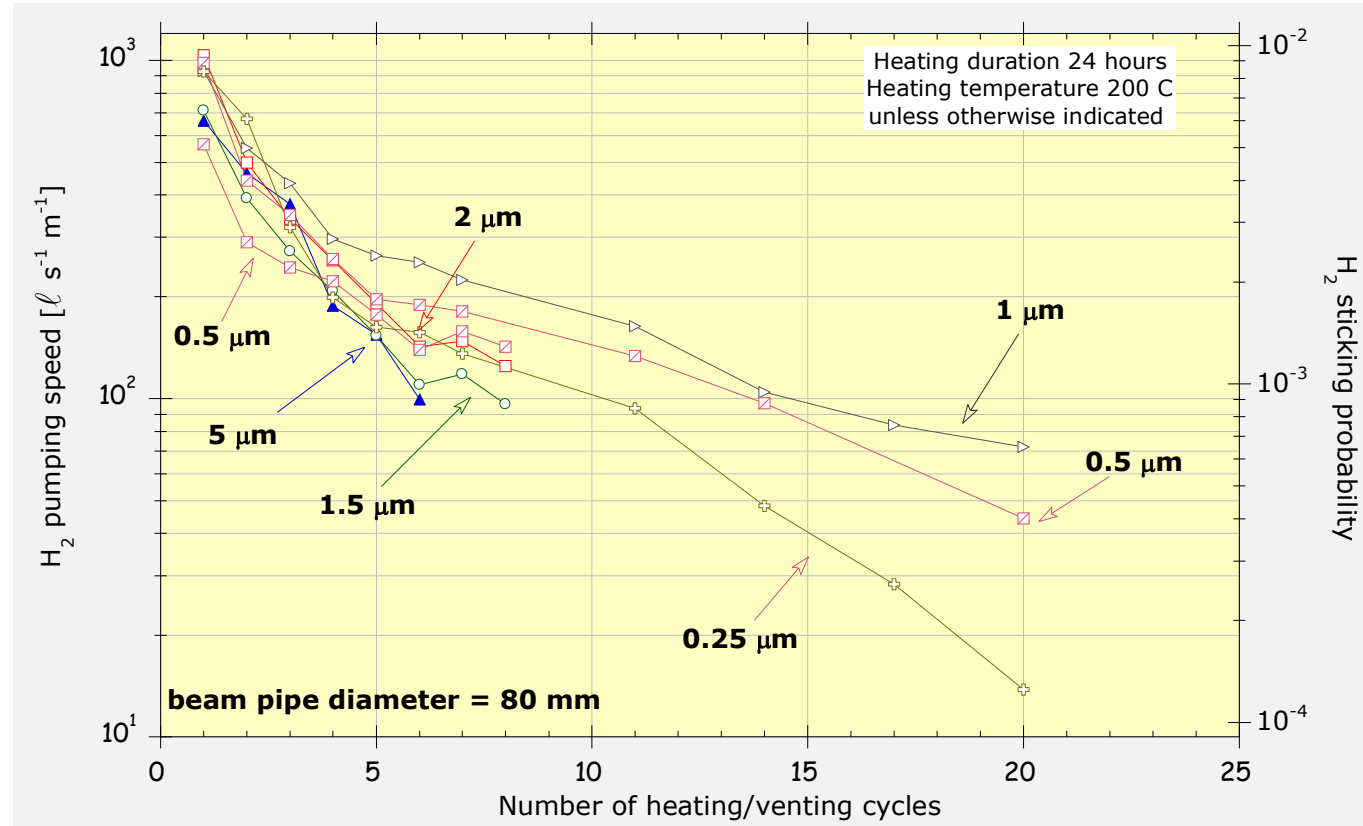
# Pumping performance: aging



Aluminium mandrel 0.5 mm

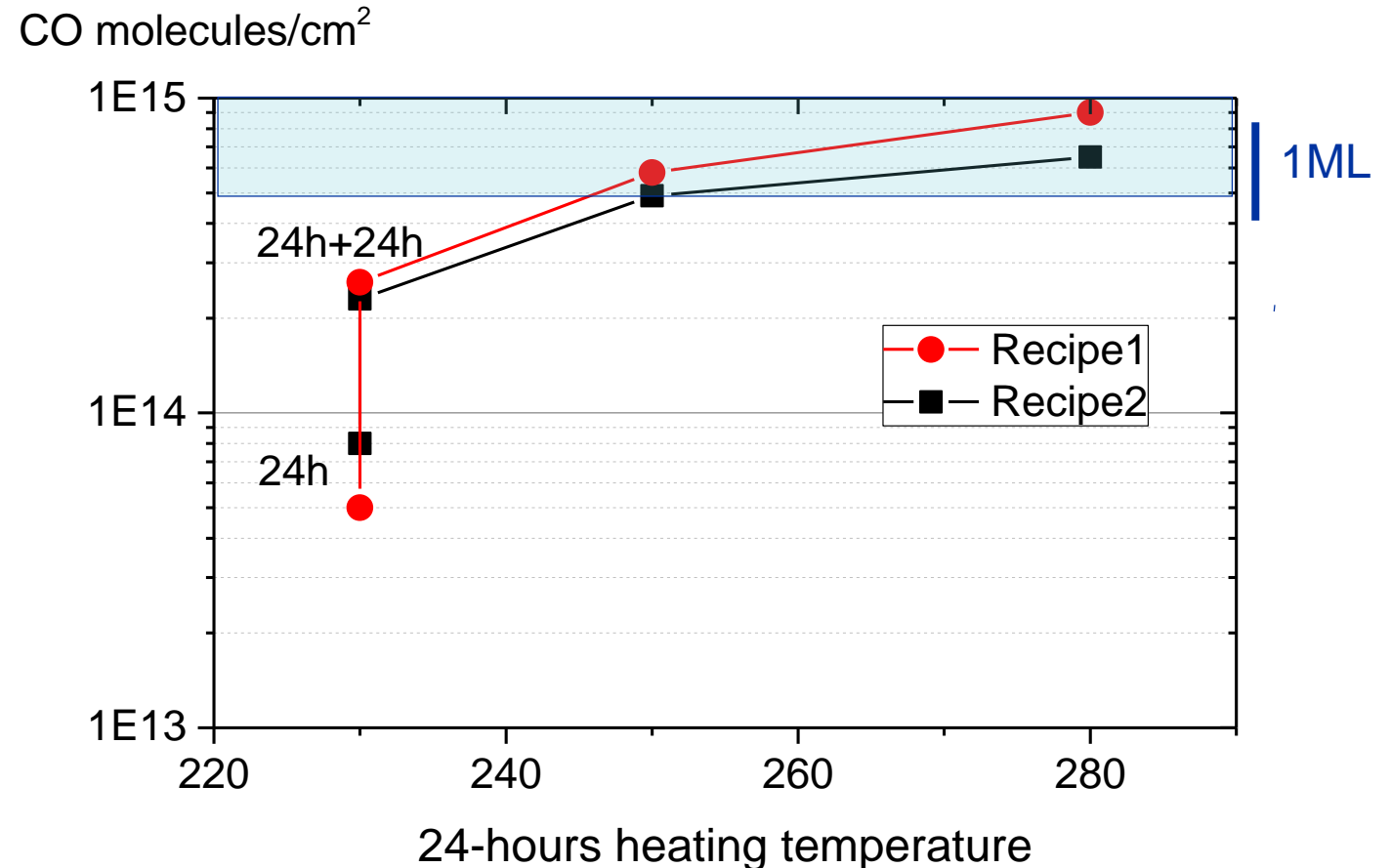
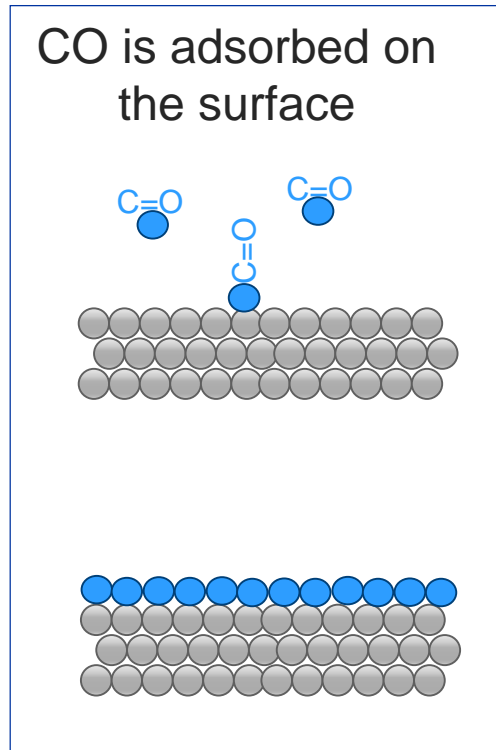


Expected for standard NEG



# Pumping performance: CO capacity

Saturated surface: 1 Monolayer ML  
CO (defined as  $5 \times 10^{14} - 1 \times 10^{15}$ )



- The expected capacity( as reference) is reached
- Also in this case there is a delay in the temperature

# Hypotheses for the origin of delay

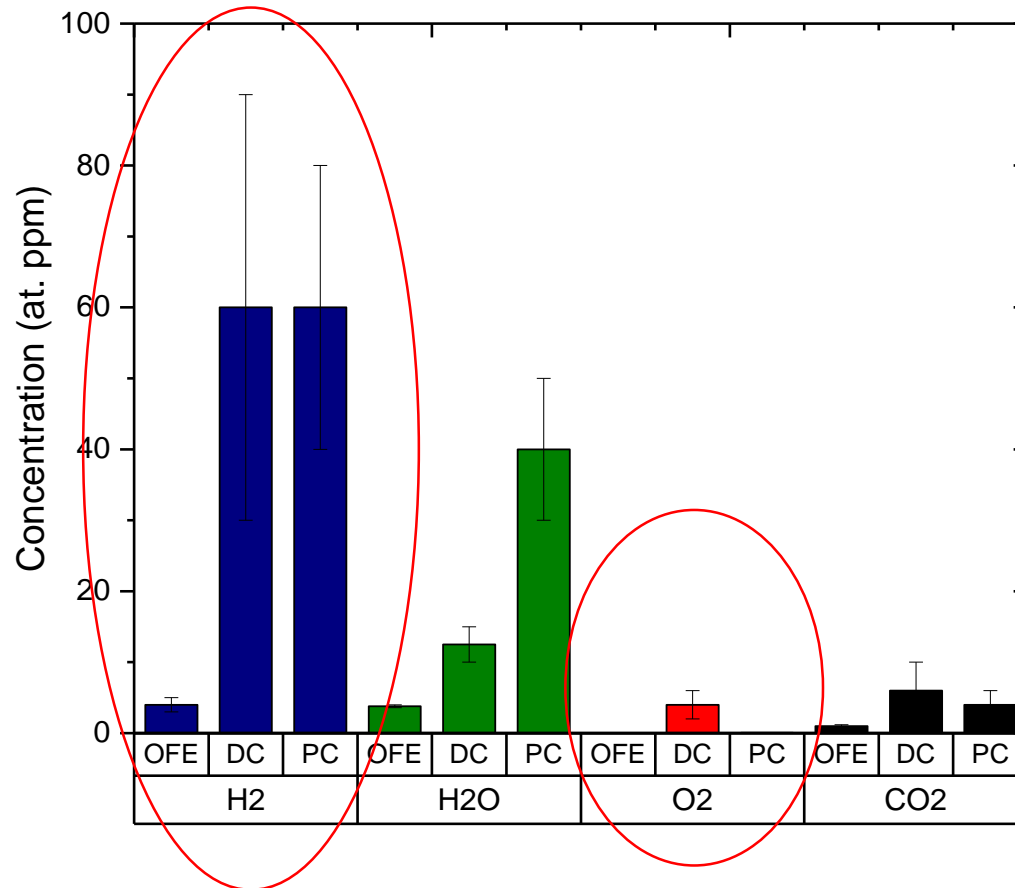
Impurities incorporated during the process?

- transferred to the NEG from the electroplated copper substrate
- coming from the chemical baths

# Origin of delay: comparison of substrates



Electroformed copper has in general few more impurities than OFE:

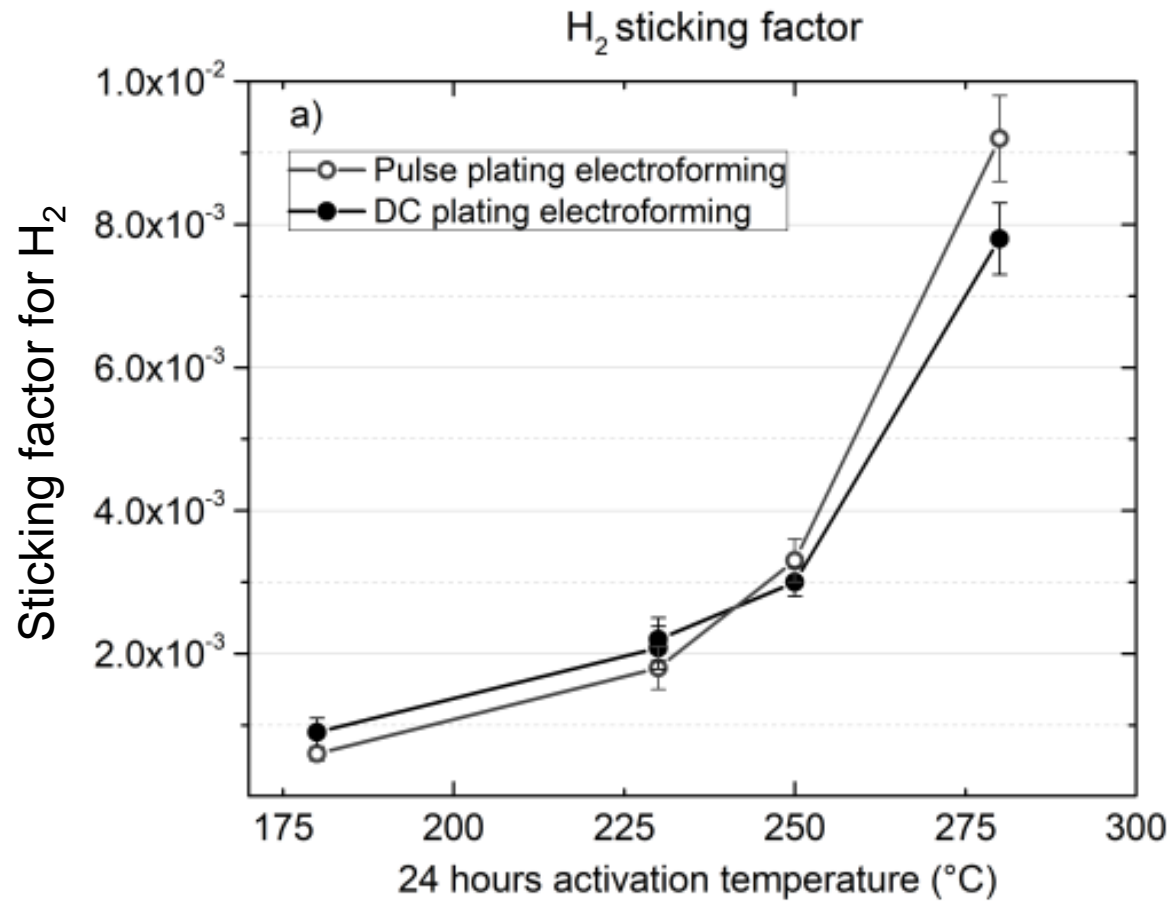


Measured in Thermal Desorption Spectroscopy on 10x10x1 mm samples

Still there is no evidence that O related impurities those play a role for the activation



# Origin of delay: different plating recipes?



Not much statistics, but ...no marked difference with electroforming in pulse mode or DC (with brightener)

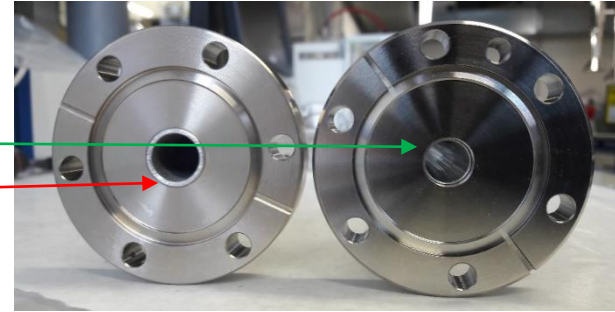
In spite of the different purity of the copper the performance of NEG is similar and delayed with respect to the reference NEG

# Origin of delay: etching and mandrel thickness

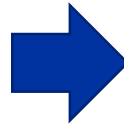
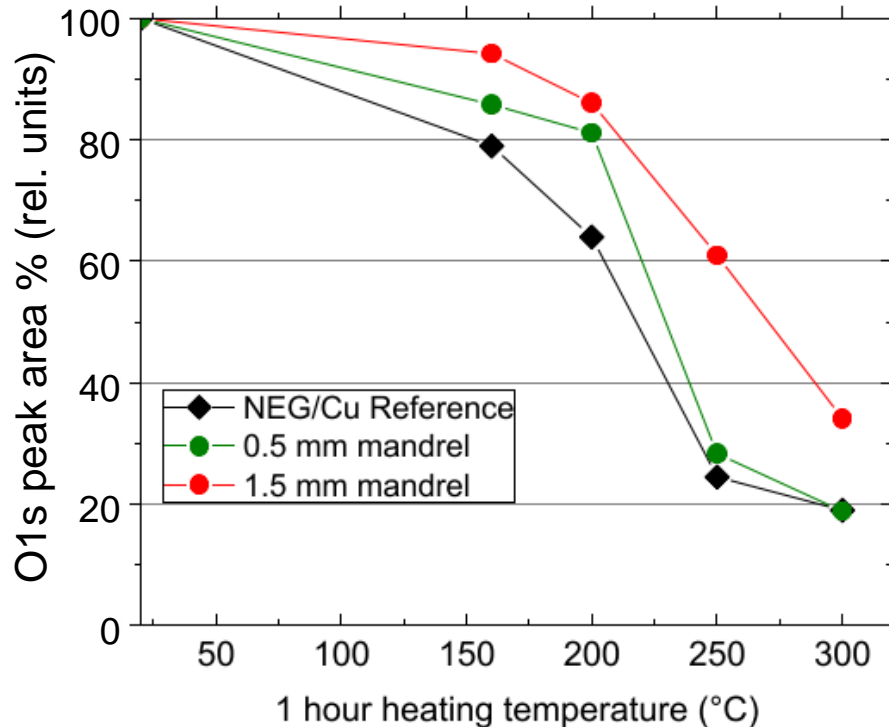
Thickness of aluminium mandrel

0.5 mm, 7h etching

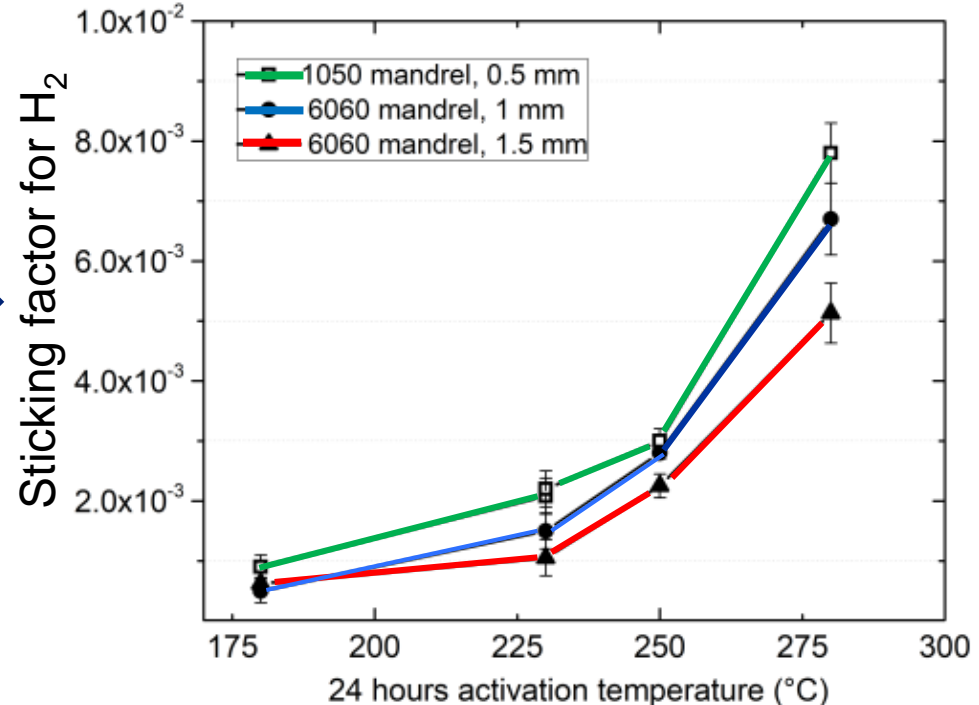
1.5 mm, 24 h etching



Decrease of O in XPS



H<sub>2</sub> sticking factor



Better activation with **thin** mandrel

# Making the NEG thin (if necessary for impedance)

For a film with much lower conductivity ( $\sigma_f$  NEG) than the substrate ( $\sigma_s$  Cu) the impedance is only governed by film thickness  $d$

$$\frac{Z_{\parallel}(\omega)}{C} \simeq \frac{Z_0 \omega}{4\pi c b} \left\{ [\text{sgn}(\omega) - i] \delta_s - 2i d \left( 1 - \frac{\sigma_f}{\sigma_s} \right) \right\} \quad (3)$$

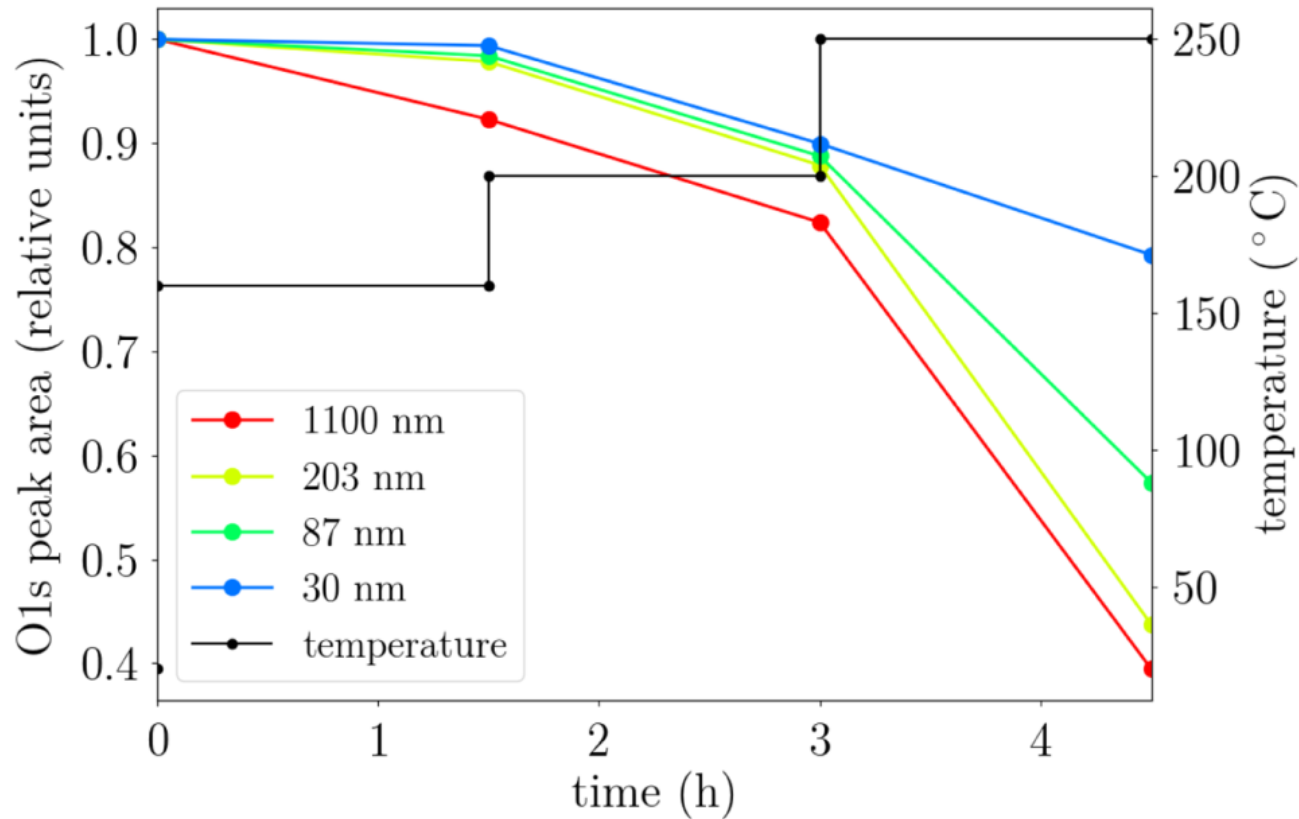
E.Belli et al  
PR Accel Beams  
21 111002 (2018)

$$\frac{Z_{\perp}(\omega)}{C} \simeq \frac{Z_0}{2\pi b^3} \left\{ [1 - i \text{sgn}(\omega)] \delta_s - 2i d \text{sgn}(\omega) \left( 1 - \frac{\sigma_f}{\sigma_s} \right) \right\} \quad (4)$$



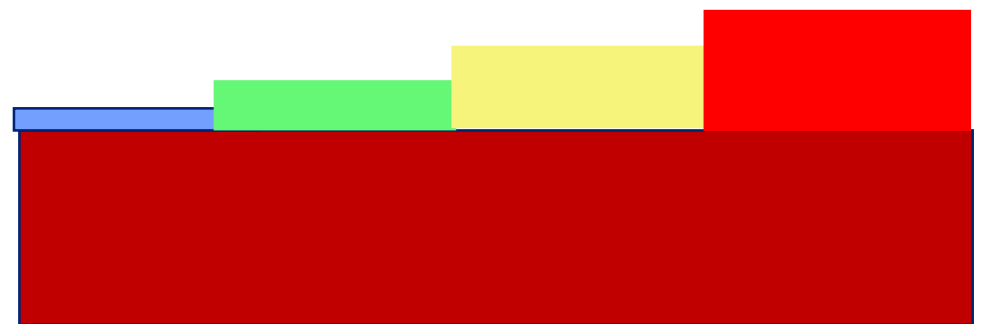
# How much cycles for thin NEG?

The expected effect is a **higher sensitivity to venting/activation cycles** for thin NEG: indeed.....



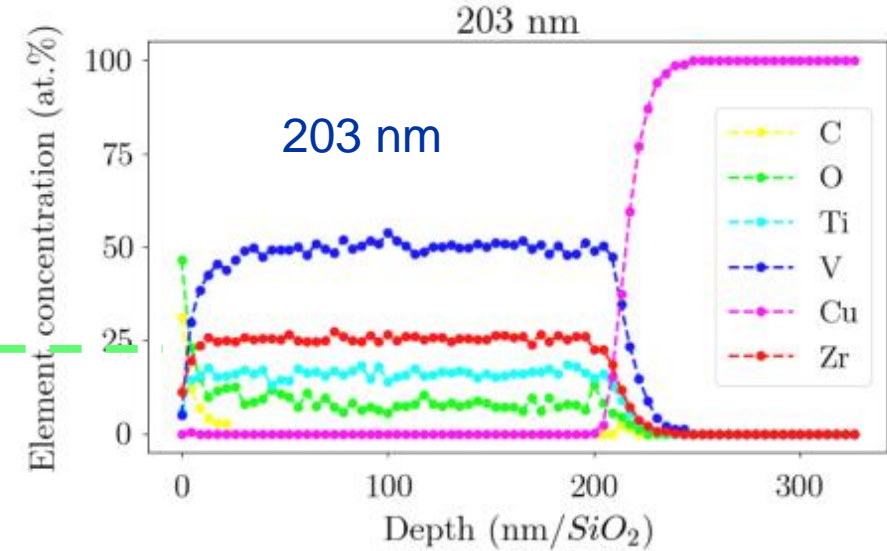
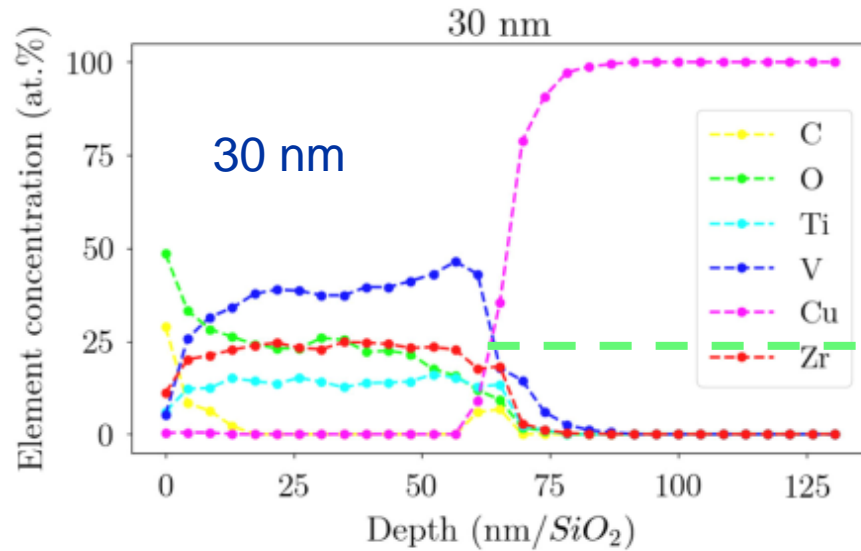
After **4 cycles** of venting /activation the activation of thinner films requires more time

→ It is more difficult for the surface oxygen to diffuse into the film



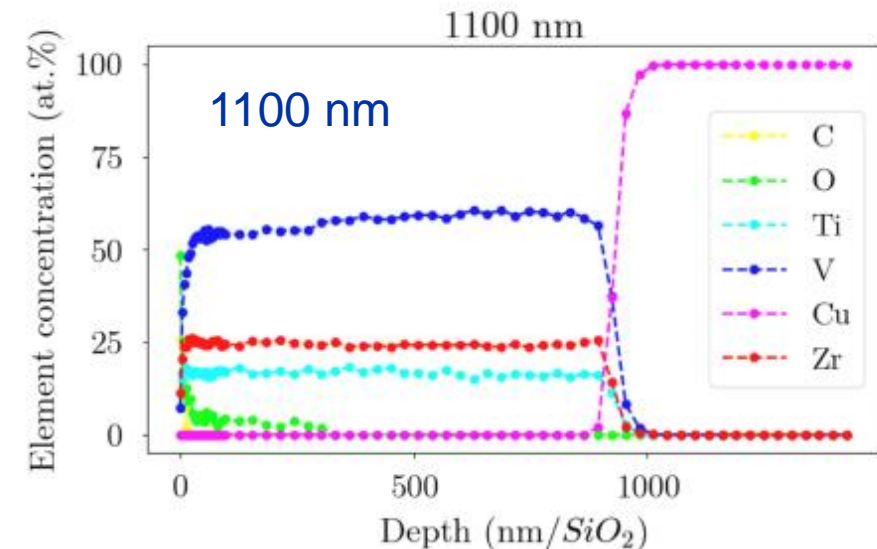
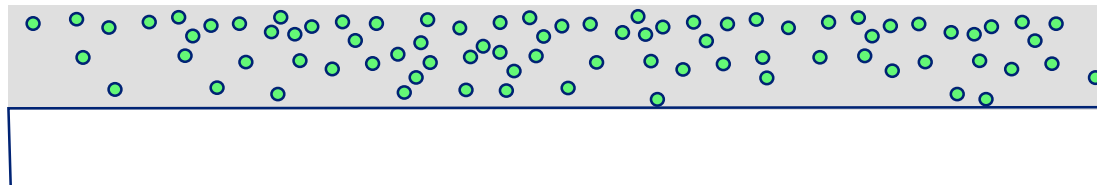
# How much cycles for thin NEG?

XPS  
depth profiles



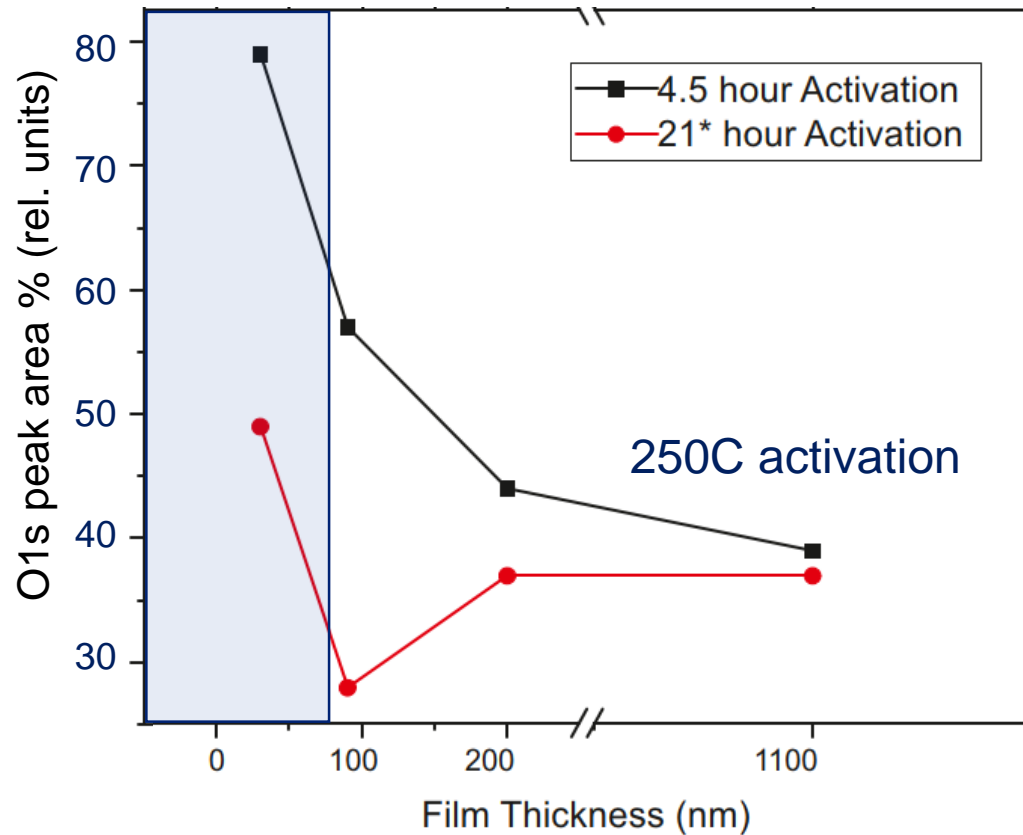
As expected after some cycles the concentration of dissolved **oxygen** is higher in thinner films

The lower concentration gradient slows down the further diffusion and activation



# How much cycles for thin NEG?

Reduction of the surface oxides  
(X-ray Photoelectron Spectroscopy)



- For film thickness  $\geq 100$  nm the activation after few (4) cycles can be recovered for 21h of activation at 250C

# Photon stimulated Desorption on thin NEG

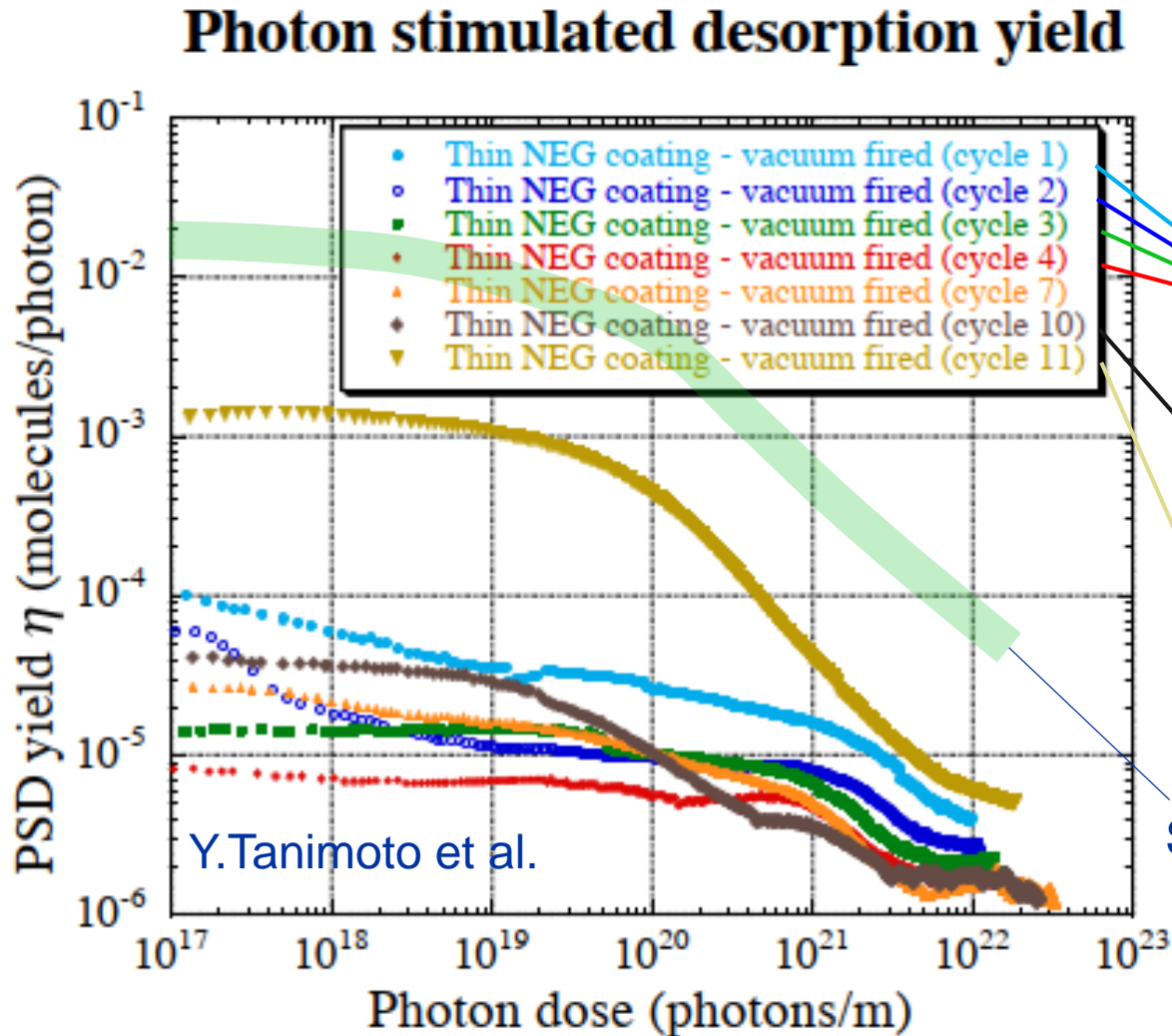
Critical energy: 4keV  
Incident angle: 10mrad  
Power density: 20W/m  
**150 nm thick NEG**

Cycles 1-4: PSD decrease with venting/activation (250C x 4h)/irradiation cycles

Cycle 10: slight increase when venting/activating without further irradiation, but conditions quickly (irradiation helps to reduce H content)

Cycle 11: venting without activation

**For PSD the thin NEG is robust upon cycling**

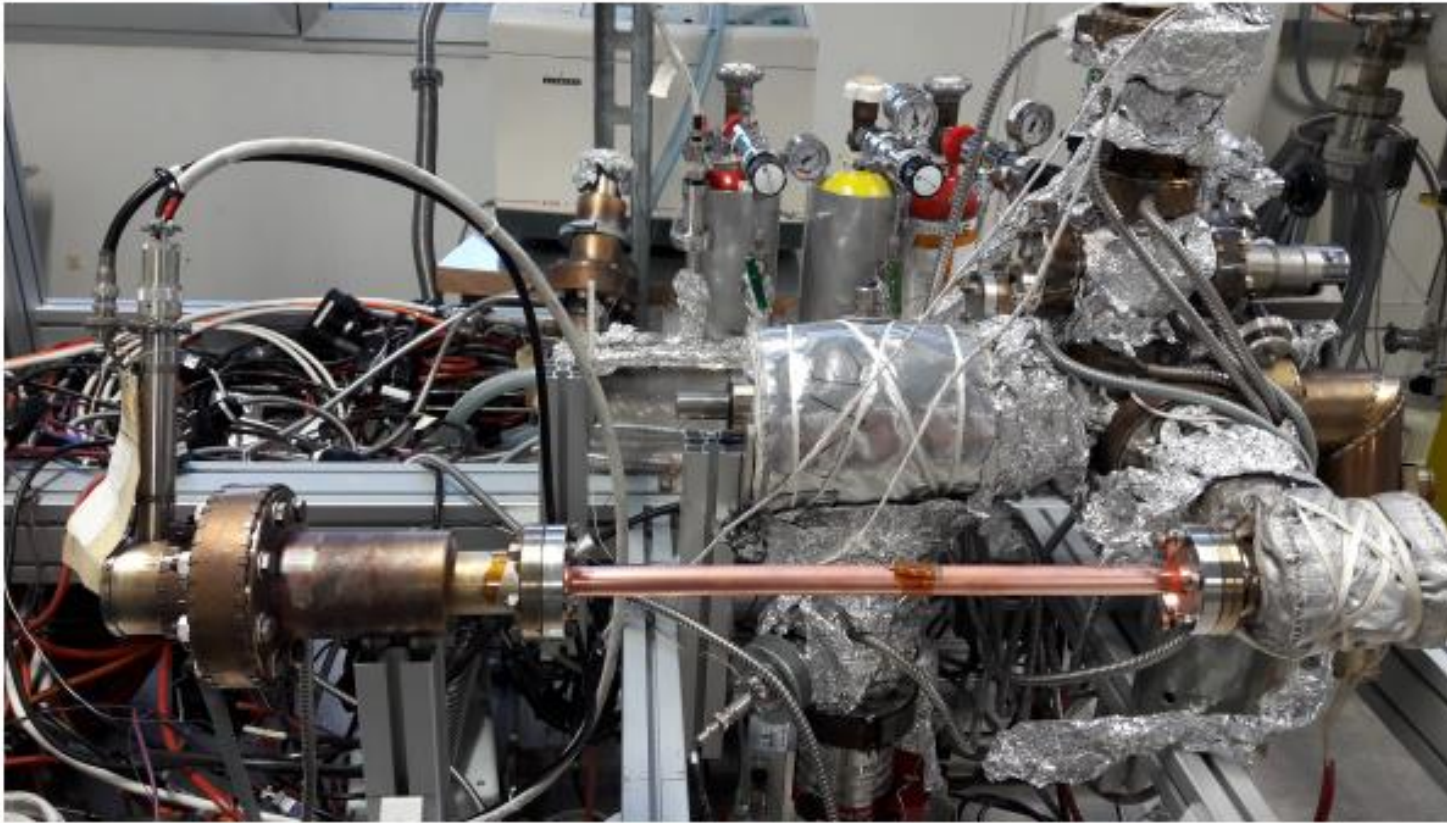


# What do we know on thin and inverted NEG

Property/NEG	Standard NEG	Thin NEG	Inverted NEG
1st activation O decrease	250C x 1h	Lowest thickness 30nm	250C x 1h
N cycles: O decrease	(see pumping)	>100 nm 250C x 24h N ≤ 4	?
1st activation pumping speed	180-200C x 24h	Reduced for 200 nm	275C x 24h
pumping speed: N cycles	Sticking 1/N , N ≤ 20 200C x 24h	?	Sticking as expected 1/N, N ≤ 3 , 275Cx24h
PSD	Ec = 20keV, 250C x 24h	150 nm, Ec=4KeV, 250C x 4h, N=10	?
SEY	1.1 for 250 C x 1h	>30 nm 1.2, 250C x 24h	1.2 for 250C x 1h
SEY, N cycles	1.2 , 250C x 1h for N=3 recovers at 300 x 1h for N=10	≥200 nm 1.2, 250C x 24h , N = 4	?

Thank you!





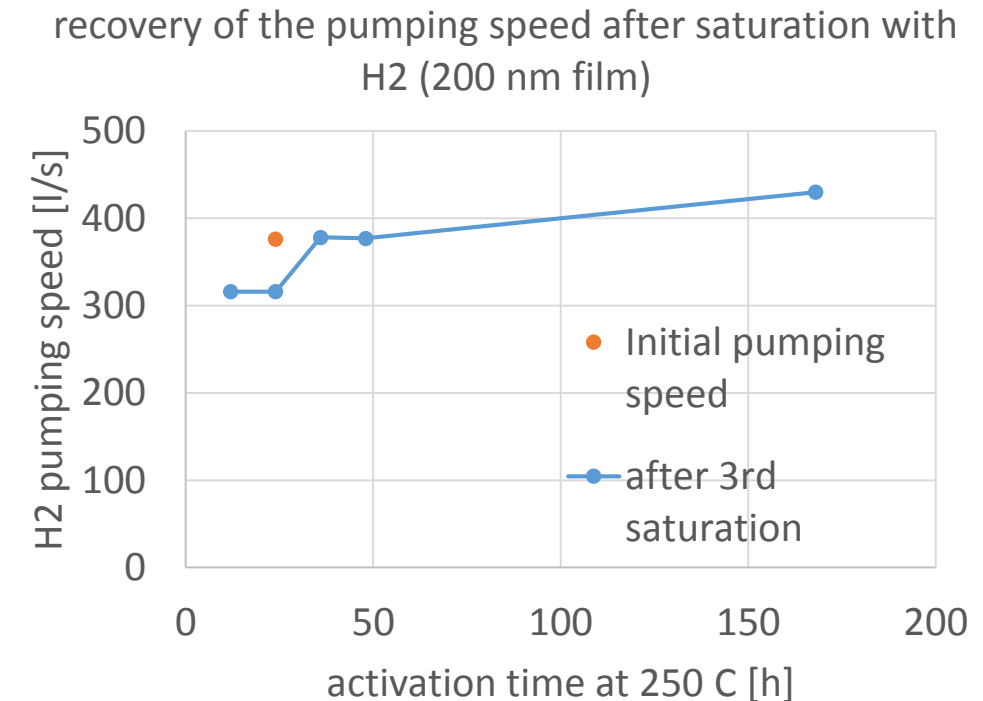
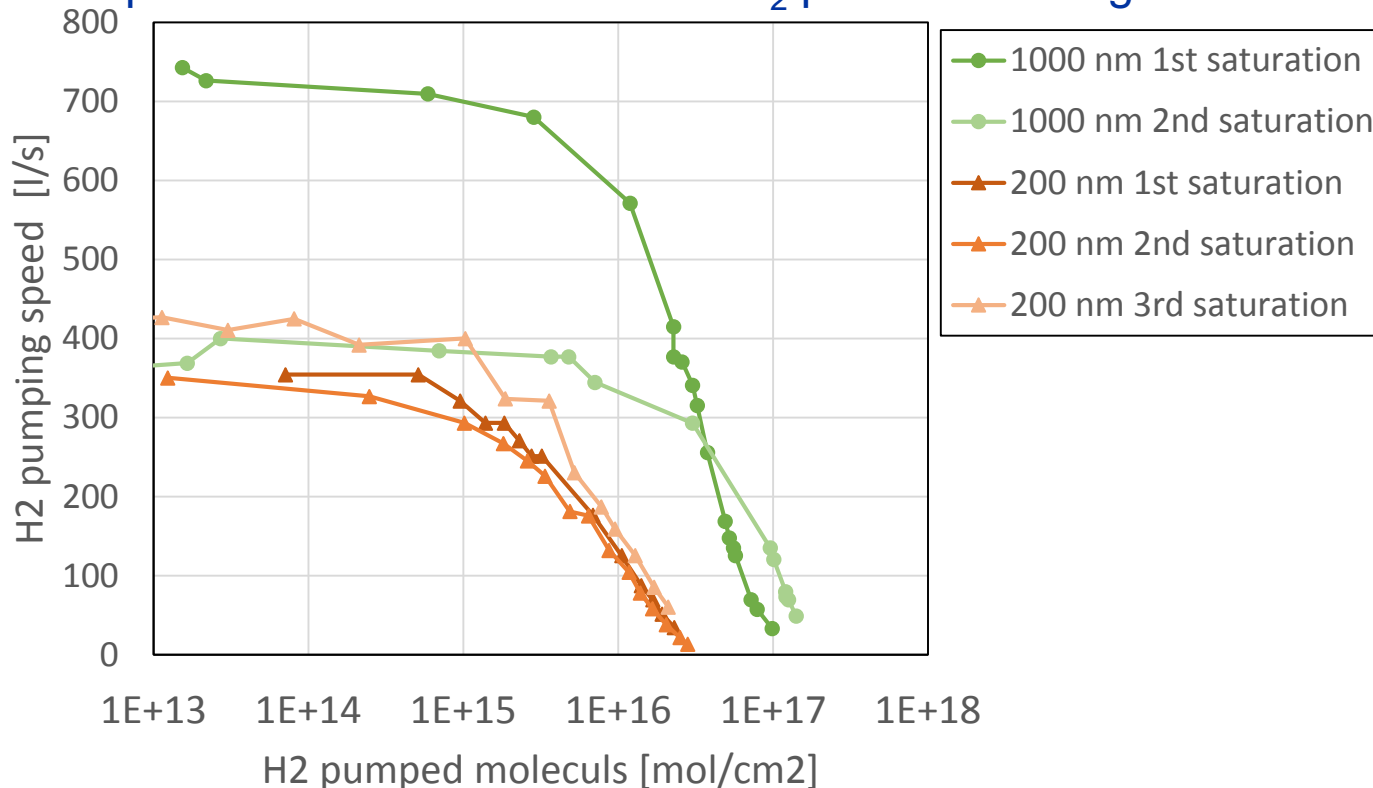
**Figure 4.24.** Chamber installed in transmission system.



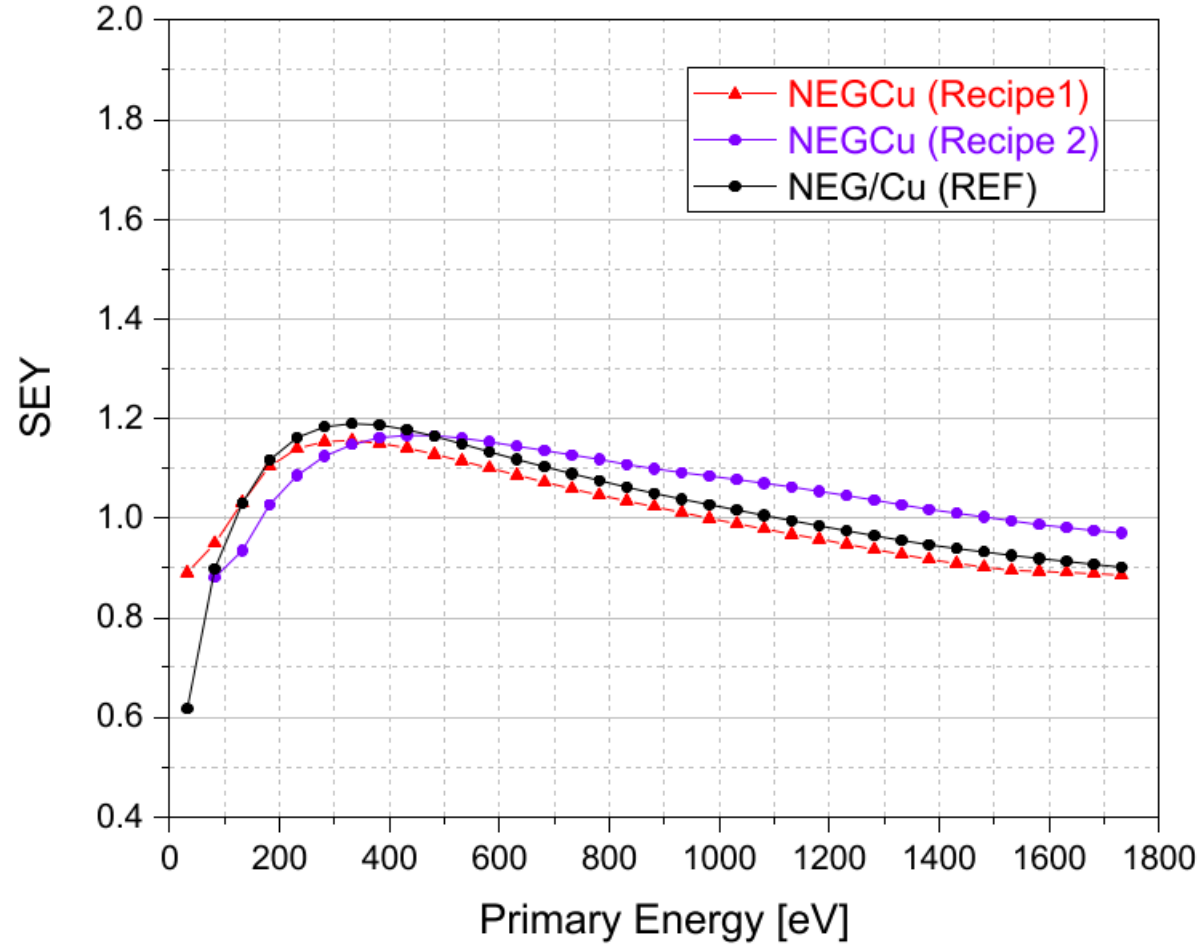
# Pumping speed and H<sub>2</sub> saturation of NEG



- For 1000 nm the pumping speed is as expected: after saturation a regeneration (no venting) of 24h x 2500 recovers only **half** of the initial pumping speed.
- For 200 nm the pumping speed is about half of reference: after saturation and regeneration (no venting) it recovers the same value for several saturation (no venting) cycles
- Atomic concentration of H in the film is similar for both thicknesses (~4% at. H in the film): did we reach equilibrium of H dissolved and H<sub>2</sub> pressure during activation?



# SEY after activation 1h at 250°C

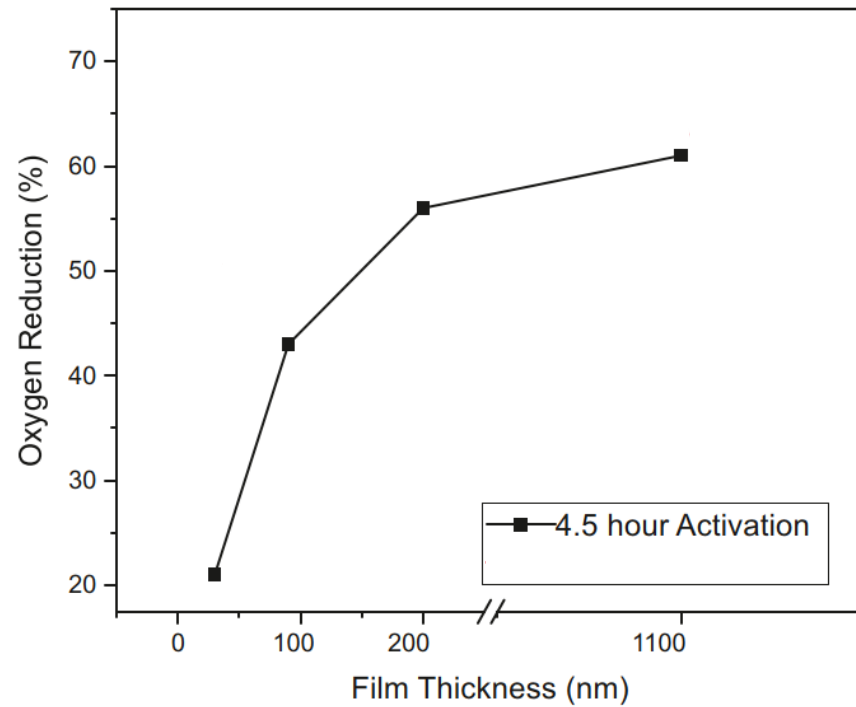


Inverted NEG:  
Shows a decrease of SEY which  
is not delayed

# 3 – Studies on “very thin” NEG films

4<sup>th</sup> activation cycle at 250 °C

Reduction of the surface oxides  
(X-ray Photoelectron Spectroscopy)



Maximal secondary electron yield

