



# Current Deep Level Transient Spectroscopy (I-DLTS) technique applied to p-type silicon diodes for Acceptor Removal studies

## Authors:

HIMMERLICH, Anja (CERN); MOLL, Michael (CERN); GURIMSKAYA, Yana (Universite de Geneve (CH)); LIAO, Chuan (Hamburg University (DE)); FRETWURST, Eckhart (Hamburg University (DE)); PINTILIE, Ioana (NIMP Bucharest-Magurele, RO); MATEU, Isidre (Universitaet Bern (CH)); SCHWANDT, Joern (Hamburg University (DE)); MAKARENKO, Leonid (Byelorussian State University (BY)); CASTELLO MOR, Nuria (IFCA (ES)), MAULEROVA, Vendula (Hamburg University (DE))

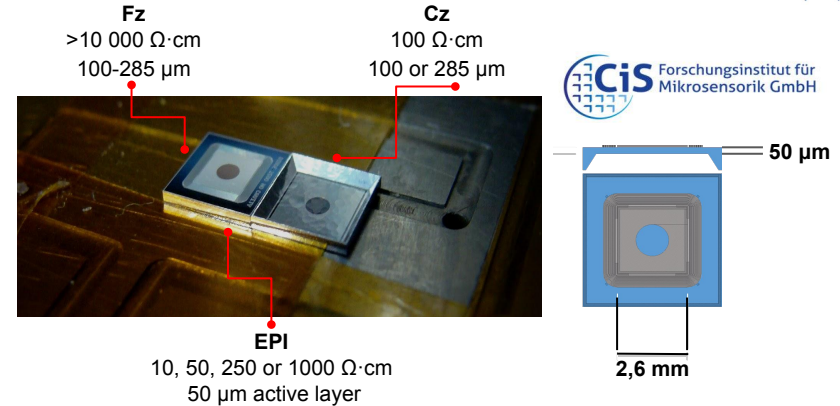
**Presenter:** GURIMSKAYA, Yana (Universite de Geneve (CH))

**Session Classification:** *Defect and Material Characterization – Acceptor removal studies*

# RD50 - Acceptor removal project

## Dedicated defect and material characterization experiment

- pursue the “acceptor removal project” to understand defect kinetics mechanisms;
- measure the ratio of point to cluster defects for various particle irradiations;
- compare microscopic defect formation to macroscopic effects on silicon sensors



### Electron irradiation

5.5 MeV



200 MeV



### Neutron irradiation



Reactor neutrons

### γ- irradiation



$^{60}\text{Co}$

50 kGy, 200 kGy and 1 MGy

### Proton irradiation



24 GeV /c

**Boston General Hospital** 230 MeV

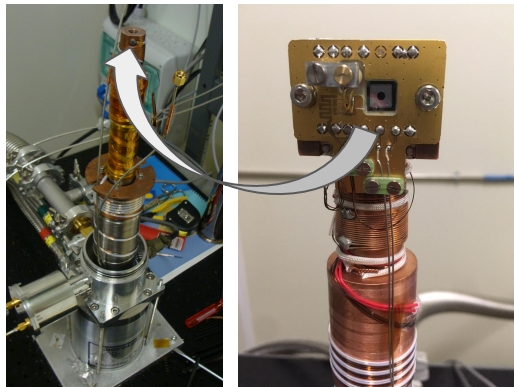


This project has received funding from the European Union's Horizon 2020 research and innovation programme under grant agreement No 654166

# Microscopic Damage: $N_{BiOi} / \Phi_{neq}$

TSC

I-DLTS



C-DLTS

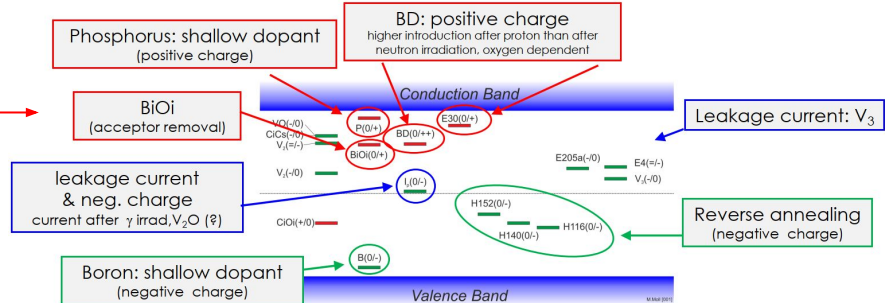
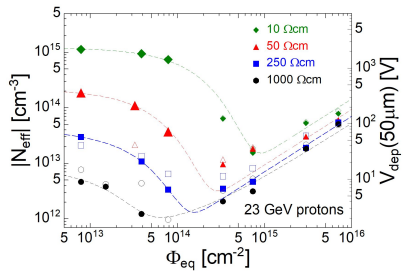
See previous talk by Anja Himmerich

TS-Cap

See next talk by Chuan Liao

I-DLTS looks into the current transient by carrier emission in a time scale of milliseconds (TSC - seconds, different filling procedure). TSC and I-DLTS can be complementary to each other by means of defect identification. Both - current-based microscopic defect analysis methods.

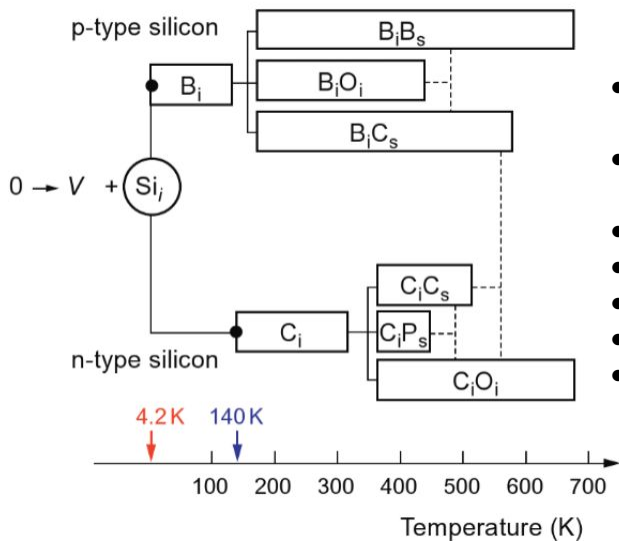
Defects mapping



Find the microscopic origin of the macro effects of radiation damage such as  $I_{leak}$  trapping and doping

# Defect kinetics and stability

## Simplification



### Burdens:

- Reaction is not complete
- Not electrically active
- Several levels
- bistability
- negative-U
- PF
- T-dependent CCS

## 'Reality'

I reactions	V reactions	$C_i$ reactions
$I + C_s \rightarrow C_i$	$V + V \rightarrow V_2$	$C_i + C_s \rightarrow CC$
$I + CC \rightarrow CCI$	$V + V_2 \rightarrow V_3$	$C_i + O_i \rightarrow CO$
$I + CCI \rightarrow CCII$	$V + O \rightarrow VO$	
$I + CO \rightarrow COI$	$V + VO \rightarrow V_2O$	
$I + COI \rightarrow COII$		
$I + V_2 \rightarrow V$		
$I + VO \rightarrow O$		
$VP + I \rightarrow P$	$V + P \rightarrow VP$	
$V_3O + I \rightarrow V_2O$	$V_2O + V \rightarrow V_3O$	
$VO_2 + I \rightarrow O_2$	$V + O_2 \rightarrow VO_2$	
$V_2O_2 + I \rightarrow VO_2$	$V + VO_2 \rightarrow V_2O_2$	
$I + O_2 \rightarrow IO_2$	$V + Y \rightarrow VY$	
$VY + I \rightarrow Y$	$V + VY \rightarrow V_2Y$	
	$V + I \rightarrow Si_s$	

- B and C competing for interstitials
- High  $\rho$  Si:  $O \gg C \gg B$  leading to the production of mainly  $C_iO_i$

Defect kinetics according to the Davies Model [Dav87] (first part). MacEvoy extension [Mac95] (second part). Oxygen dimer extension [Kra03] (third part).

# Initial impurity content

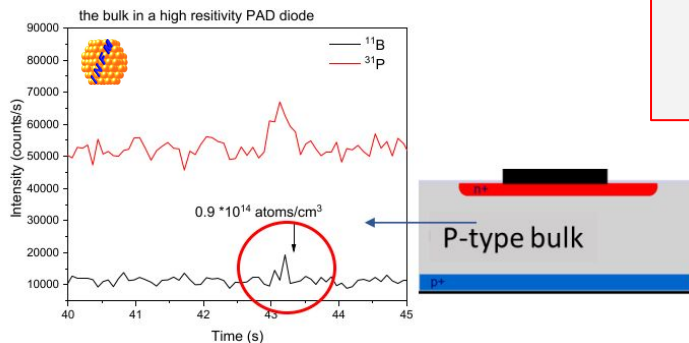
- **SR (a) and SIMS (b): ITME and ITE, Warsaw, Poland**

[B<sub>s</sub>] content could be detected by SIMS only in EPI diodes with  $\rho < 50 \Omega \cdot \text{cm}$

Similar C content in EPI and CZ material  $\sim 1.3 \cdot 10^{16} \text{ cm}^{-3}$ , for FZ  $\sim 10^{15} \text{ cm}^{-3}$

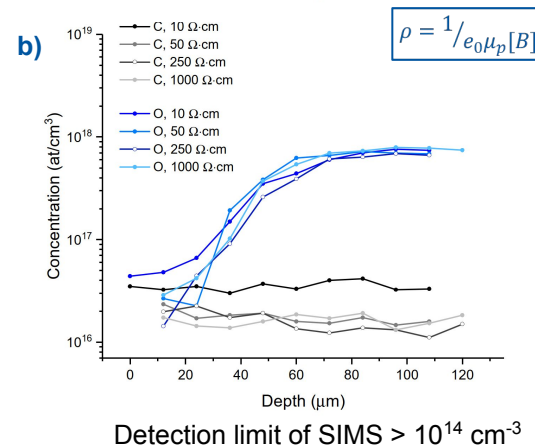
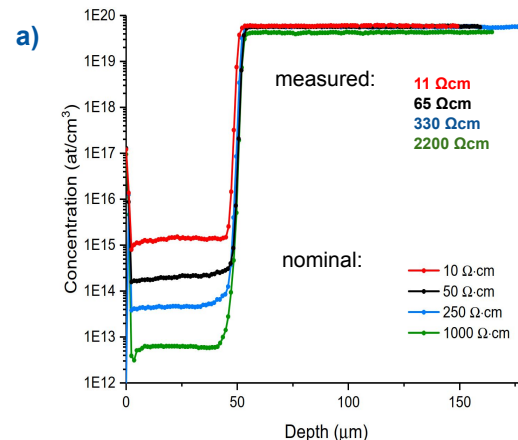
- **LA-ICP-MS: NIMP, Bucharest-Magurele, Romania**

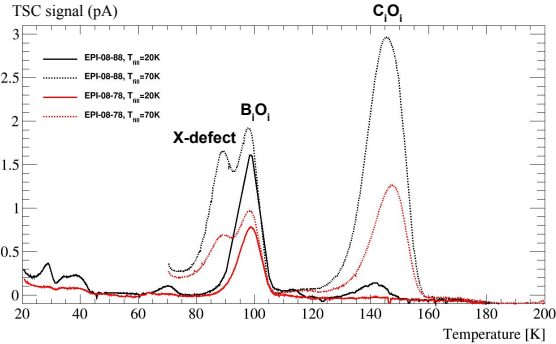
Stefan Neatu et al., 37th RD50 Workshop, 2020



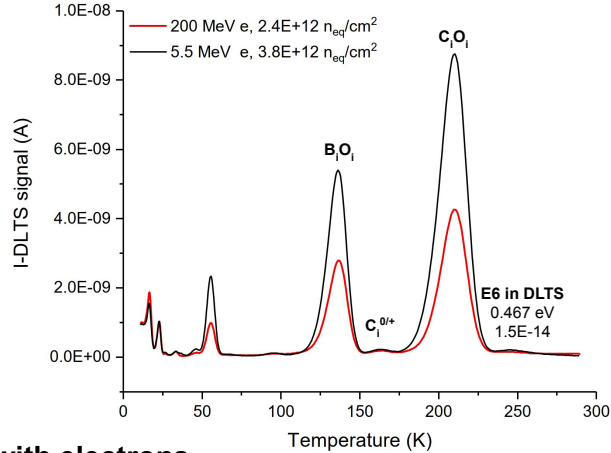
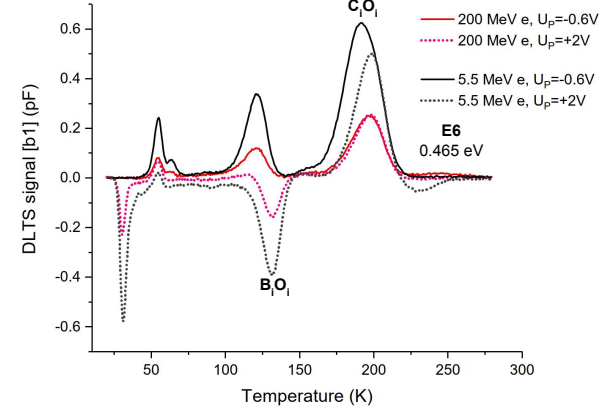
Looking for new techniques and facilities...

**Both P and B are detected in Si bulk**



**TSC**Max  $V_{\text{bias}}$  250VTSC:  $U_R = -100V$ ,  $I_{\text{inj}} = 1mA$ ,  $t_{\text{inj}} = 60\text{sec}$ ,  $T_{\text{fill}} = 20K, 70K$ 

## 2 EPI pad diodes [ $250\Omega \cdot \text{cm}$ ] irradiated with electrons

**I-DLTS**Max  $V_{\text{bias}}$  100VI-DLTS:  $U_R = -100V$ ,  $U_p = +1.5V$ ,  $T_w = 10\text{ms}$ ,  $t_p = 10\text{ms}$ **C-DLTS**1 MHz AC signal  
Max  $V_{\text{bias}}$  100VDLTS:  $U_R = -10V$ ,  $t_p = 1\text{ms}$ ,  $T_w = 20\text{ms}$ 

- Bias voltage up to 300V;
- Presence of the shoulder (X-defect) by  $T_{\text{fill}}$  variation
- 'Full' concentrations



- Can detect shallow defect levels - at least 11 in total;
- Arrhenius in one T-scan
- Separate type of carriers



- Terrific sensitivity, at least 10 defect levels in total;
- No need to fully deplete device
- Separate type of carriers



- Noise
- High  $I_{\text{leak}}$  from 220K



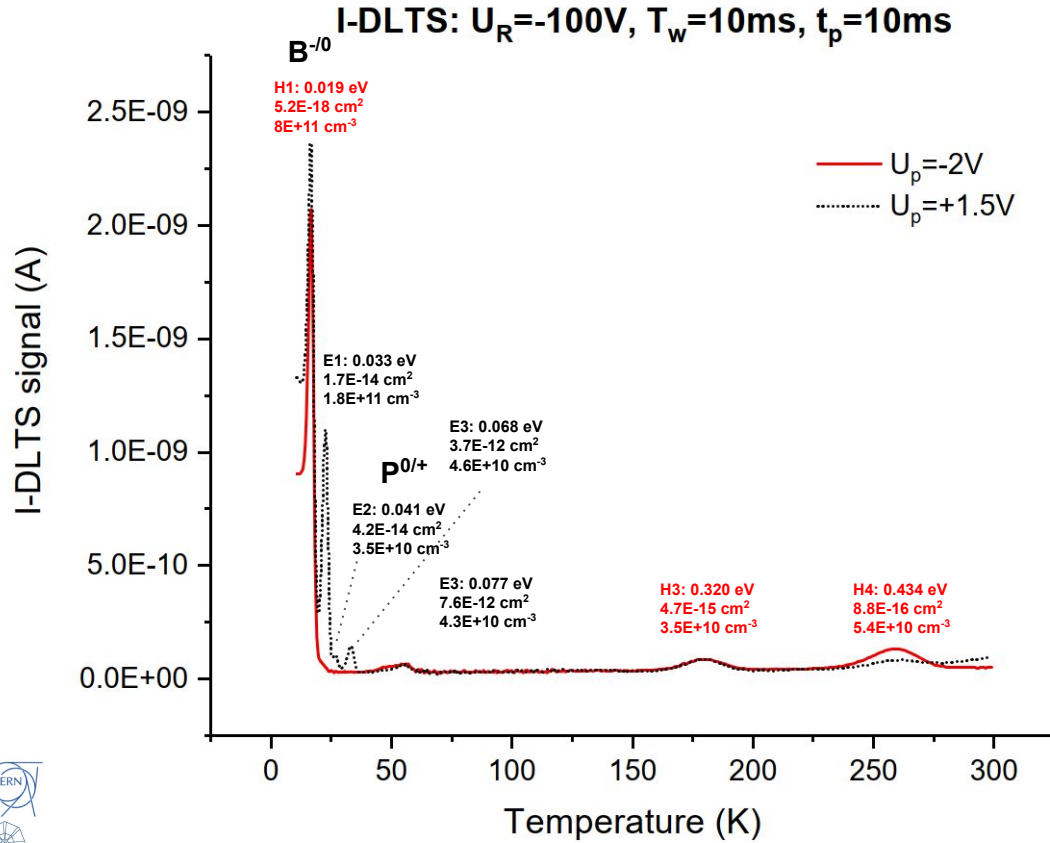
- Amplitude of transient is T-dependent



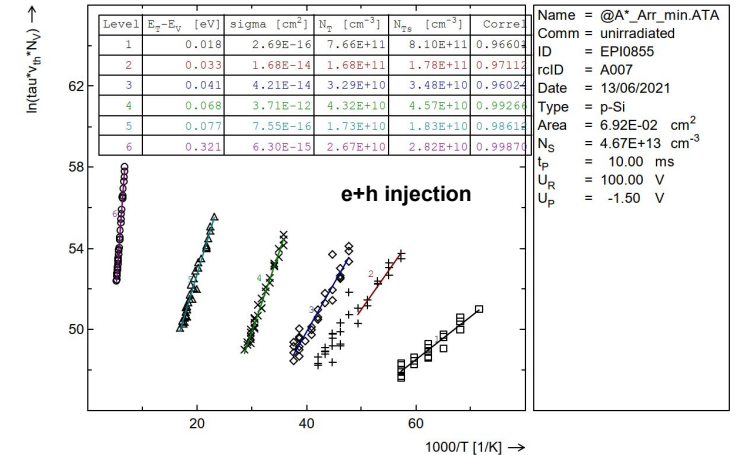
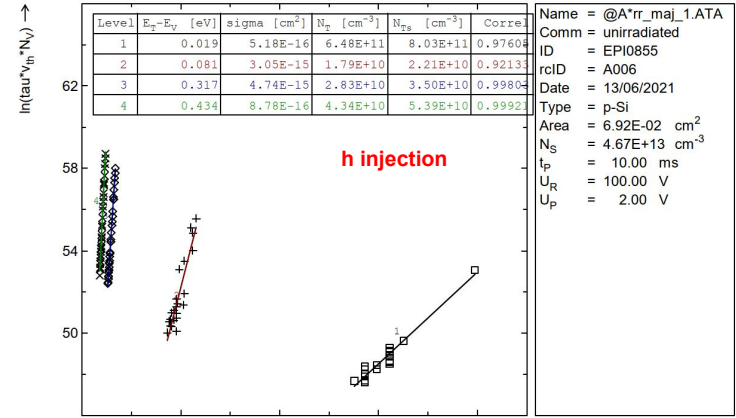
- Limitation:  $N_T \ll N_S$
- Carrier freeze-out

# I-DLTS on unirradiated 250 Ω · cm EPI pad diode

$$\ln(\tau_e v_{th,n,p} N_{C,V}) = -\ln(\sigma_{n,p}) + \frac{E_a}{k_B T}$$

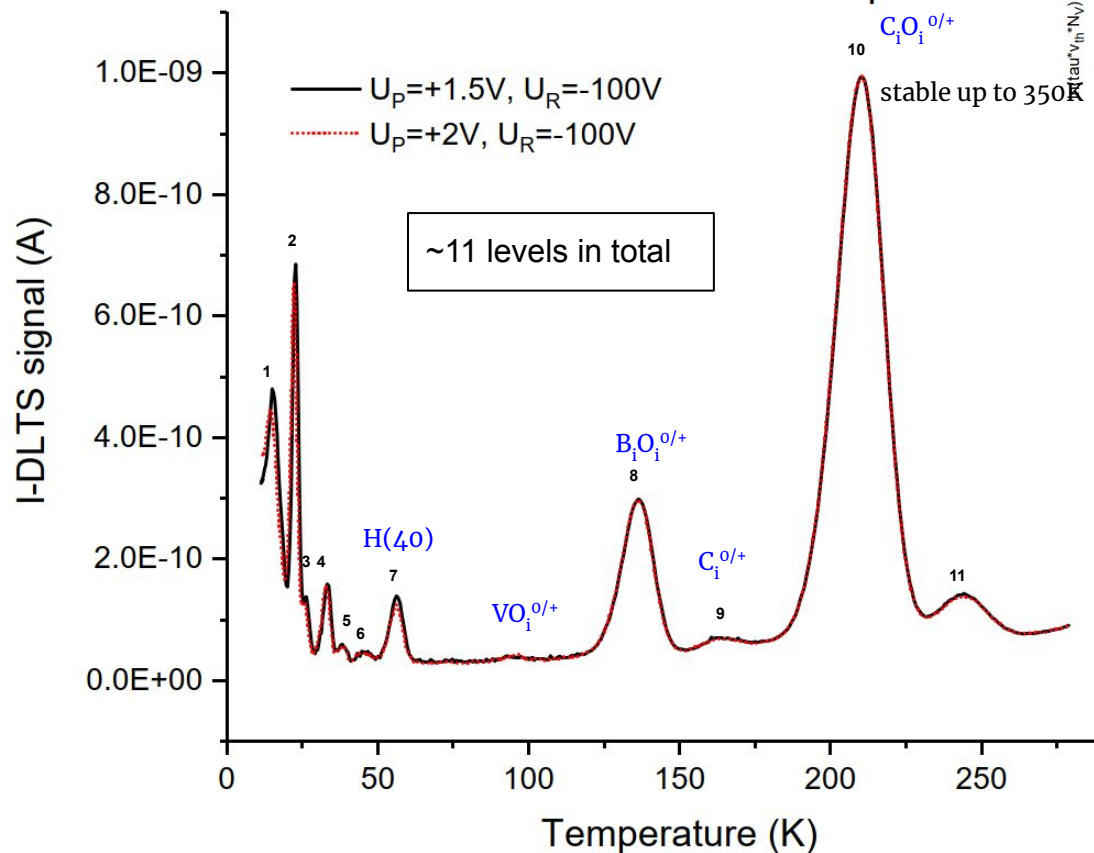


If  $\rho \sim 330 \Omega \cdot \text{cm}$  we expect to have  $4E+13$  of  $B_s$

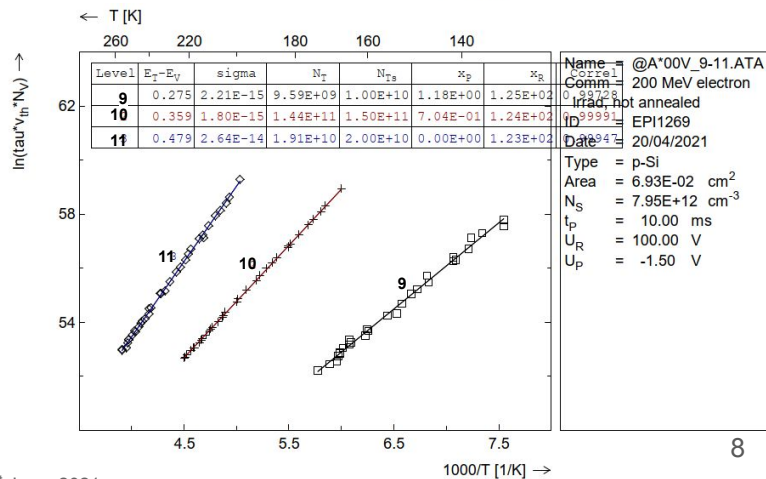
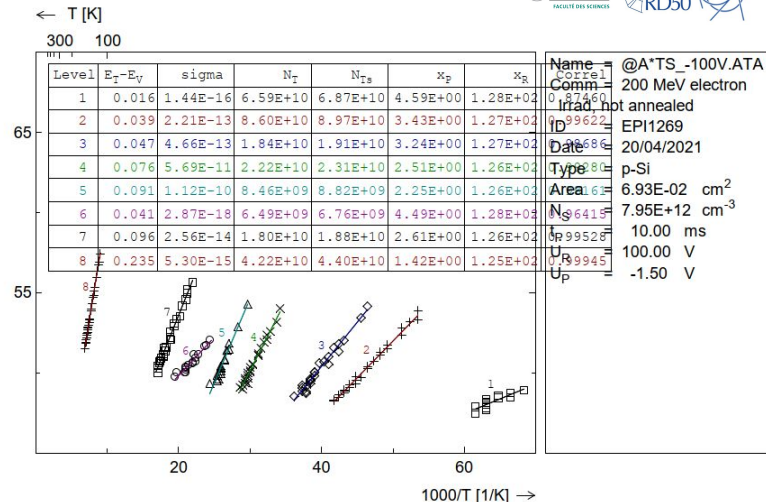


# Optimal I-DLTS conditions for irradiated high-resistivity material:

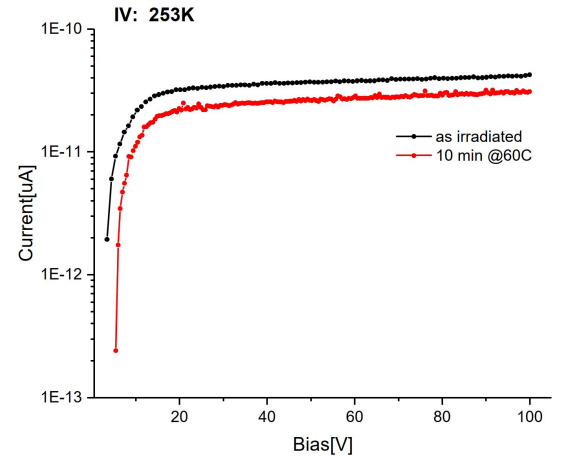
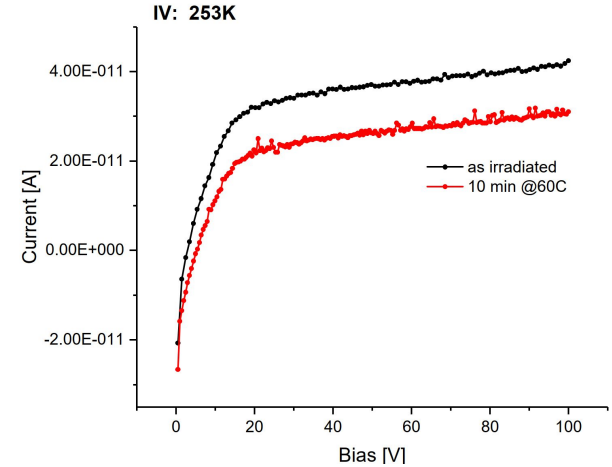
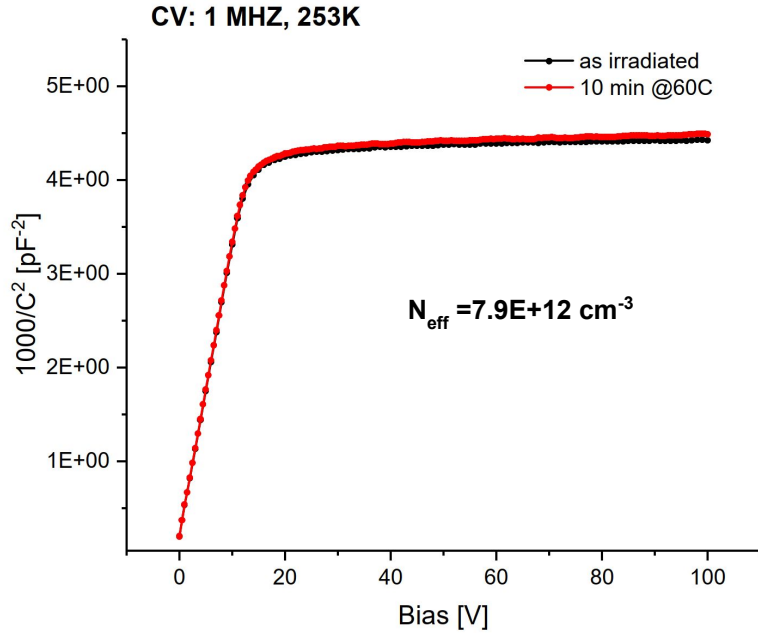
I-DLTS:  $U_R = -100V$ ,  $T_w = 10ms$ ,  $t_p = 10ms$



CIS16-EPI-12-50-DS-69 (1 kΩcm)  
 -- 200 MeV electrons,  $1.2E+12 n_{eq}/cm^2$



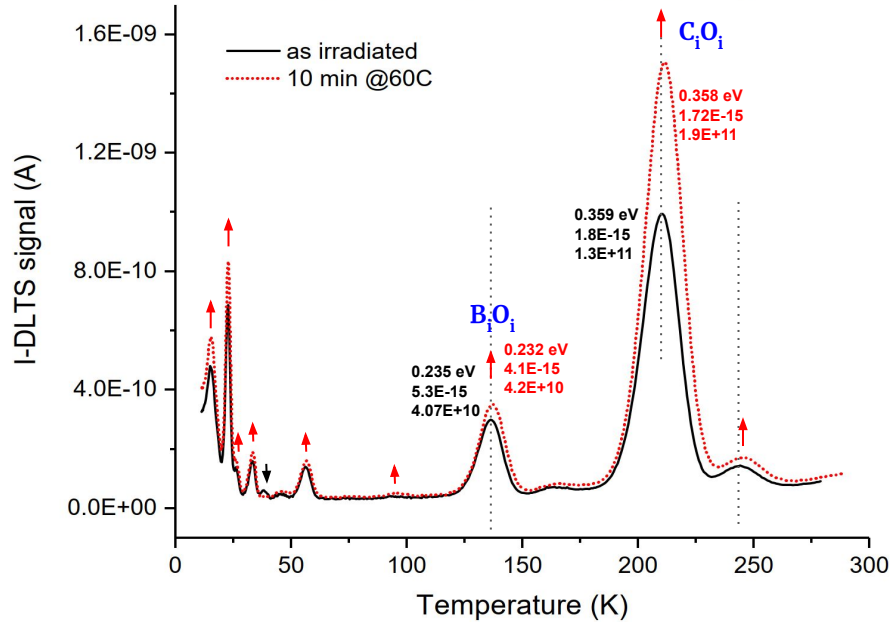




1<sup>st</sup> annealing step (10 min, 60°C) causes no change in  $V_{dep}$  (1MHz)  $\rightarrow N_{eff}$   
but pronounced change in  $I_{leak}$

EPI-12-69

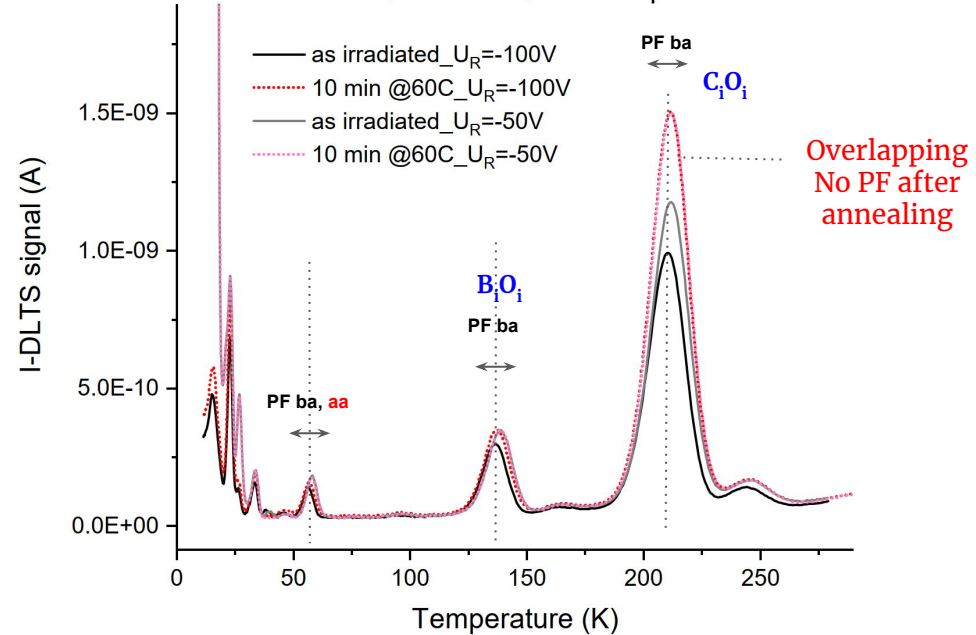
I-DLTS:  $U_R=-100V$ ,  $U_P=+1.5V$ ,  $T_w=10ms$ ,  $t_p=10ms$



Sink for electrically inactive C ?

EPI-12-69

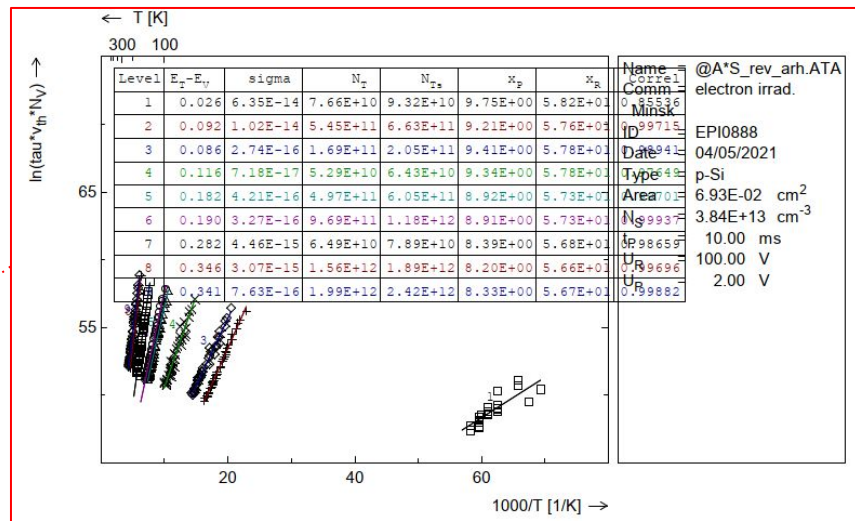
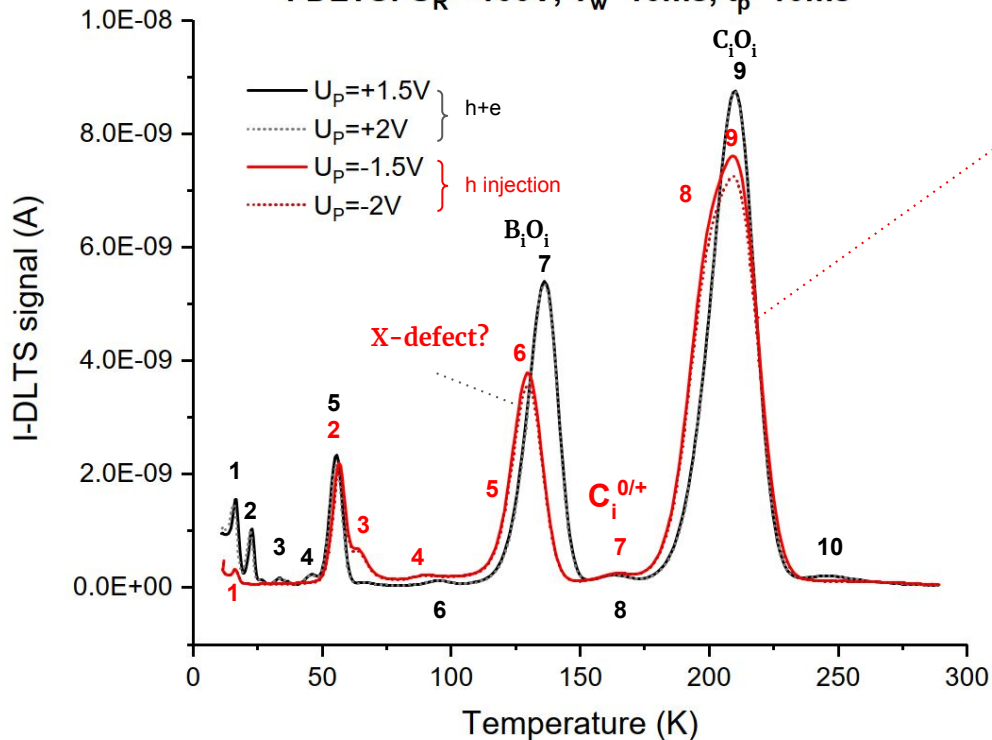
I-DLTS:  $U_P=+1.5V$ ,  $T_w=10ms$ ,  $t_p=10ms$



Seems that  $C_iO_i$  changes configuration to 'stable' with annealing

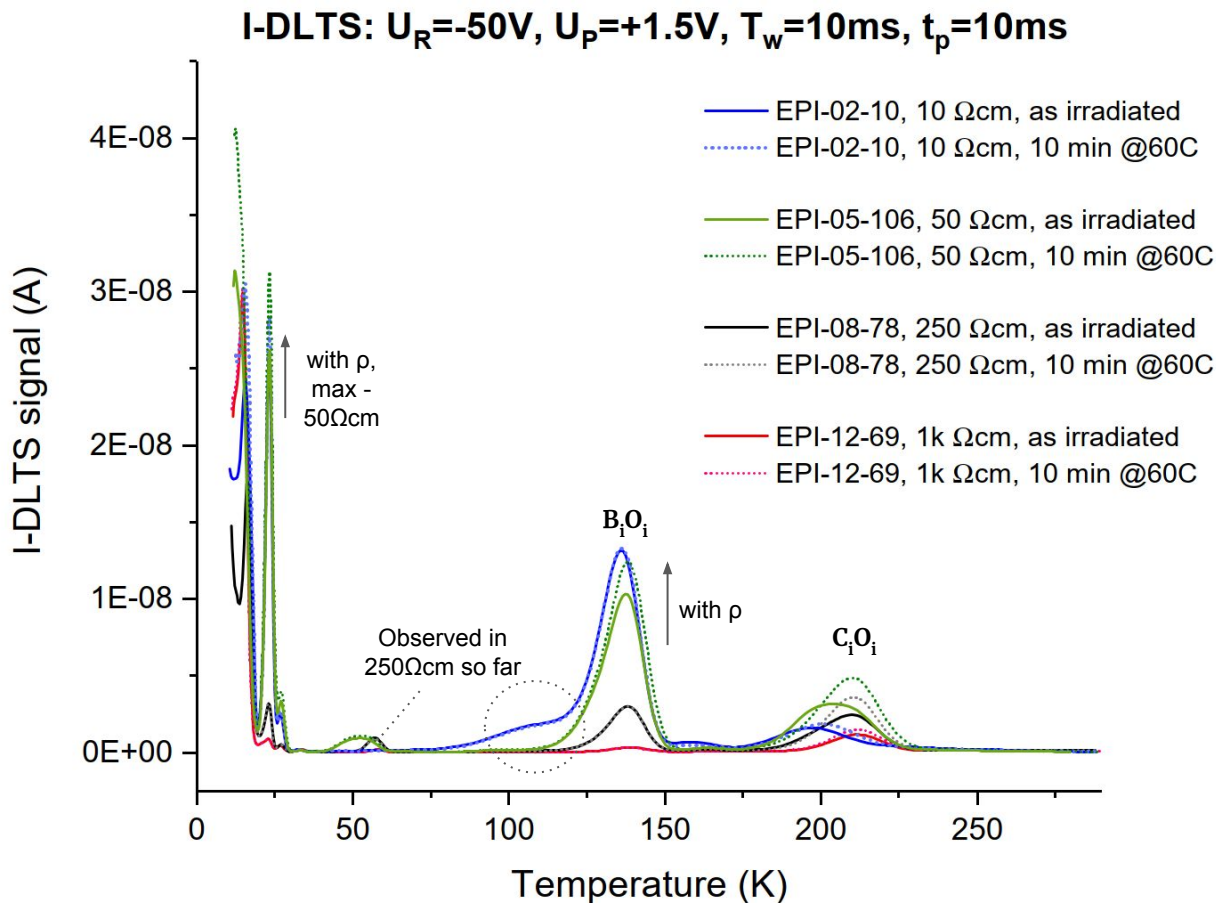
# Injection dependence in I-DLTS spectra

I-DLTS:  $U_R = -100V$ ,  $T_w = 10ms$ ,  $t_p = 10ms$



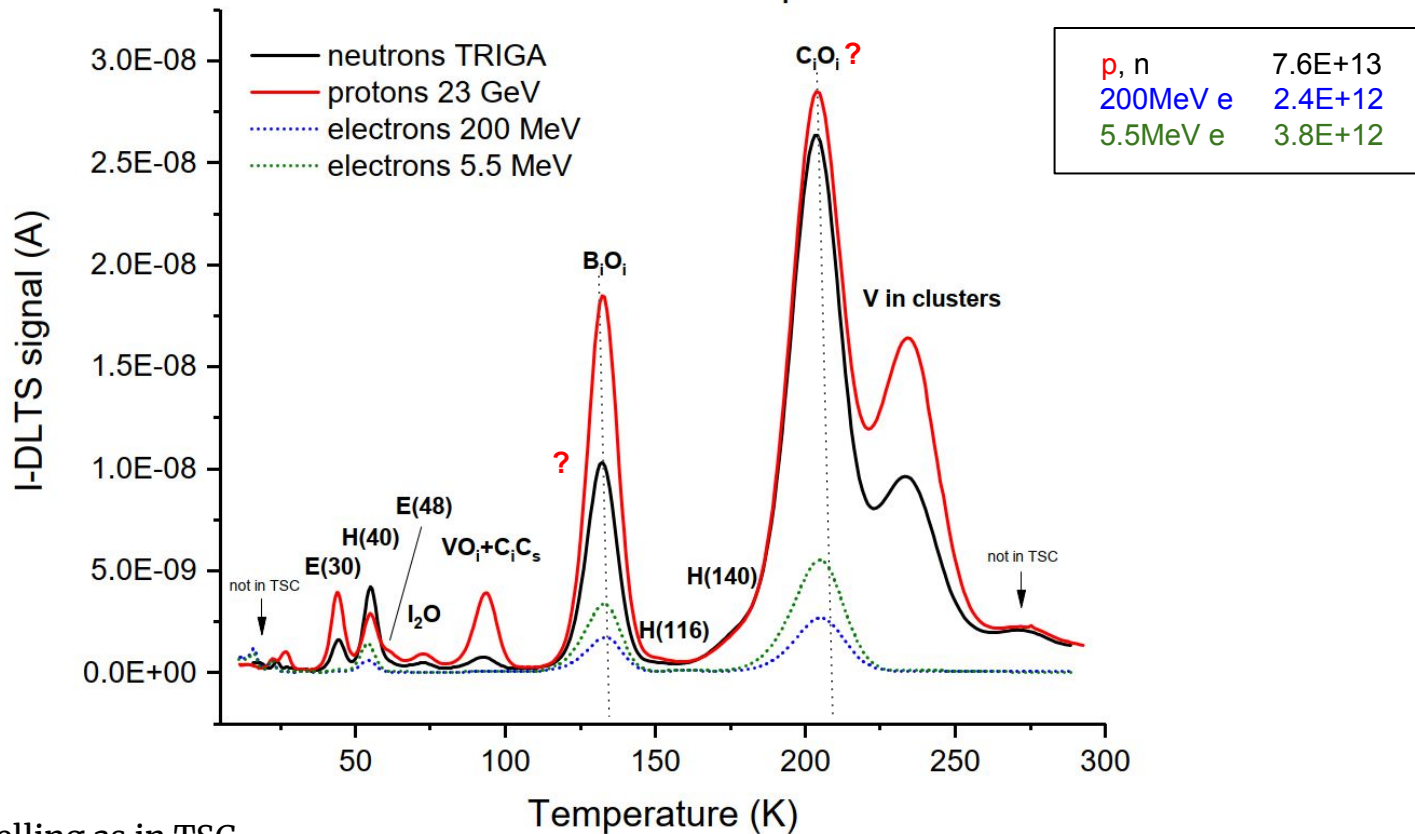
Was kept at RT, need to perform annealing to catch  $C_i$  kinetics

# p-Si pad diodes, 200 MeV electron irradiation



# 250 $\Omega$ cm pad diodes - different irradiation

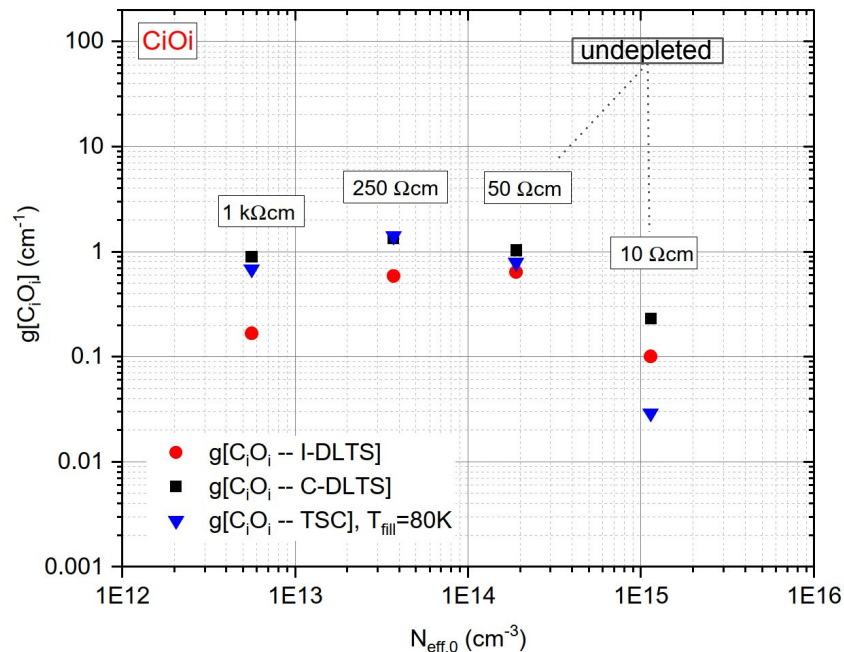
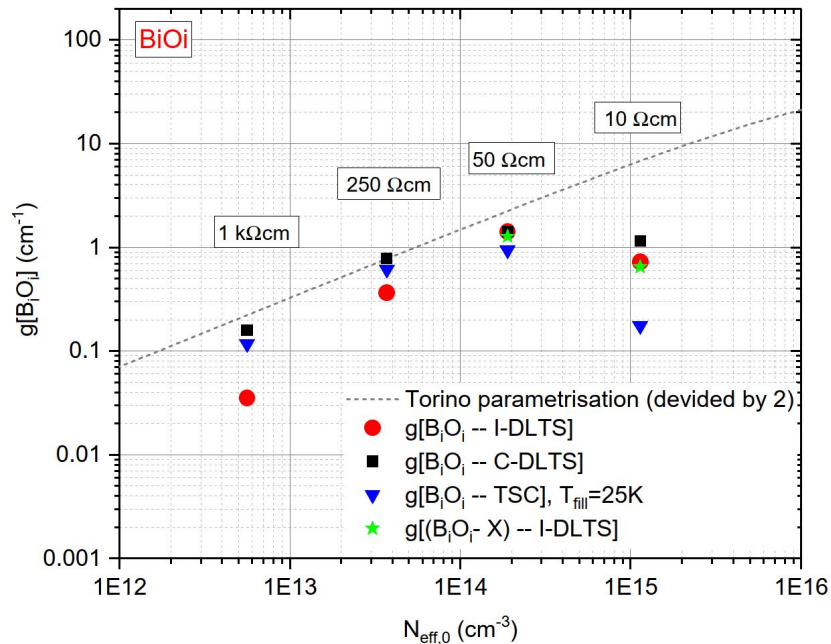
I-DLTS:  $U_R = -100V$ ,  $U_P = +1.5V$ ,  $t_p = 10ms$ ,  $T_w = 20ms$



If we use same labelling as in TSC

# Introduction rates of 2 defects of main interest - $B_iO_i$ and $C_iO_i$

200 MeV e



EPI-02-103 [11Ω · cm] :  $3.9\text{E}+13 \text{ n}_{\text{eq}}/\text{cm}^2$   
 EPI-06-105 [65Ω · cm] :  $7.1\text{E}+12 \text{ n}_{\text{eq}}/\text{cm}^2$   
 EPI-08-78 [330Ω · cm] :  $2.4\text{E}+12 \text{ n}_{\text{eq}}/\text{cm}^2$   
 EPI-12-69 [2200Ω · cm] :  $1.2\text{E}+12 \text{ n}_{\text{eq}}/\text{cm}^2$

# Summary

- It has been shown that I-DLTS is sensitive technique for surveying deep and shallow levels able to discover traps that have been overlooked in C-DLTS and TSC; but it has some limitations
- We shall certainly add I-DLTS to the measurements protocol especially for high resistivity devices and high fluence irradiations when C-DLTS is not possible
- Optical injection should be implemented to overcome the drawback of uncertainties in traditional voltage pulse filling with forward bias  $I_{fill}$  in both I-DLTS and TSC
- Optimal filling conditions at the saturation for defect levels with a special attention to  $B_iO_i$ ,  $C_iO_i$  and X-defect are the key for a success measurements campagne

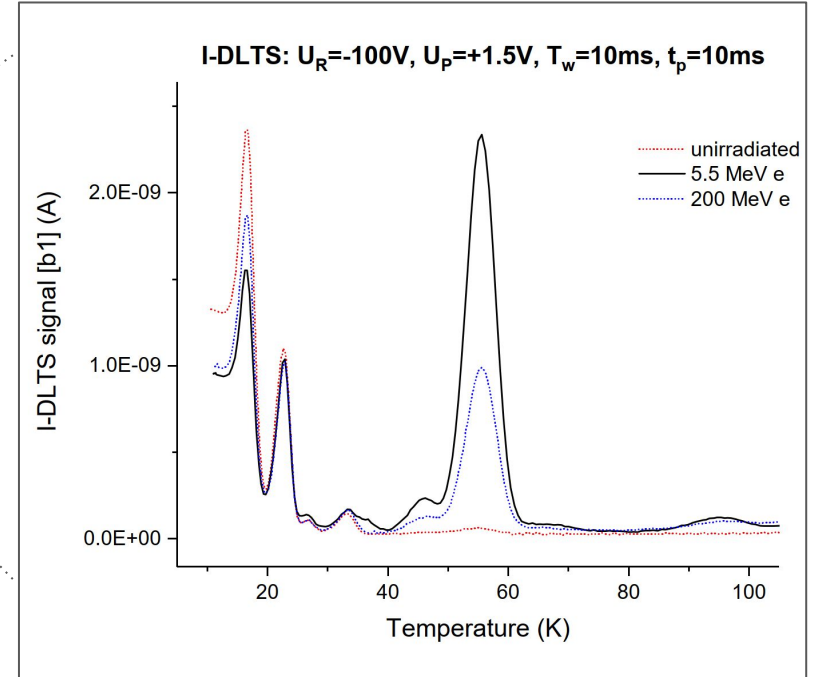
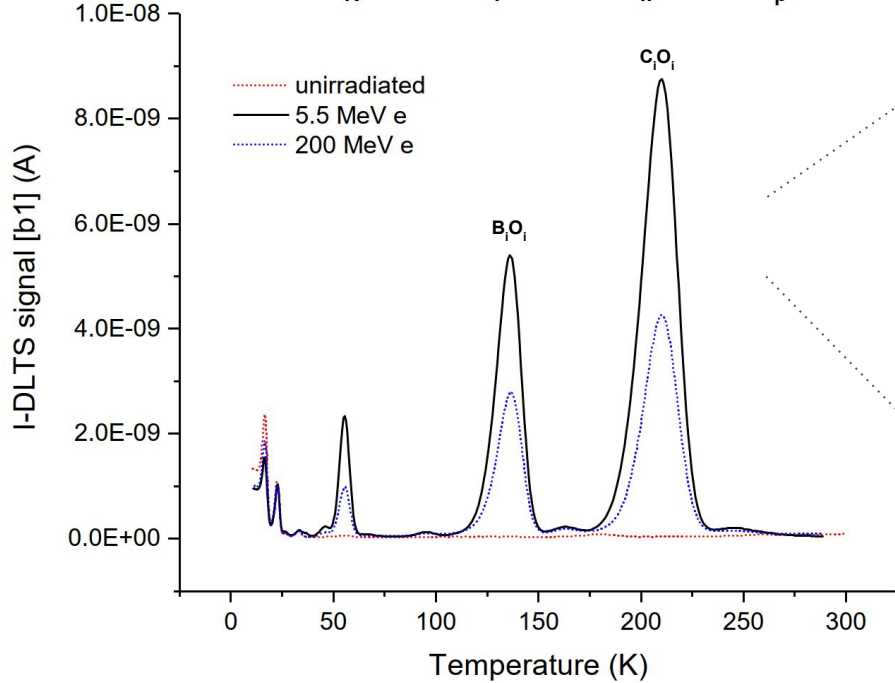
Thank you!

**Spare slides**

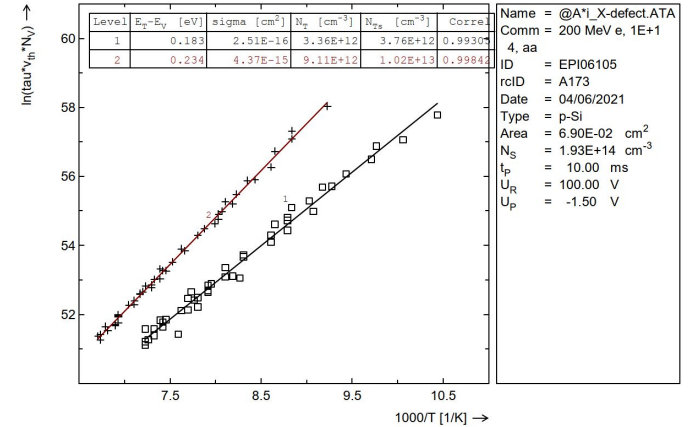
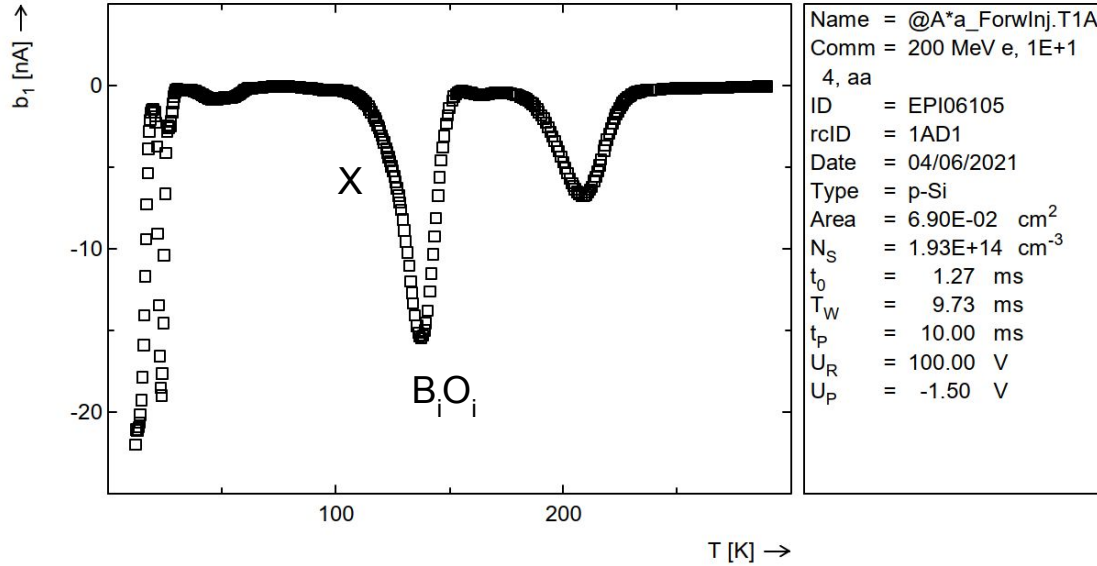


# Comparison of I-DLTS for unirradiated and e-irradiated 250 ohm.cm EPI diodes

I-DLTS:  $U_R = -100V$ ,  $U_P = +1.5V$ ,  $T_w = 10ms$ ,  $t_p = 10ms$

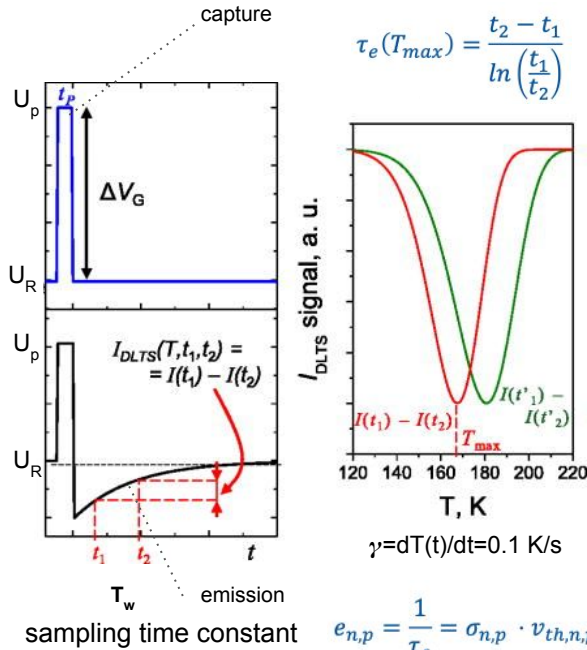


# Arrhenius for X-defect and $B_iO_i$ in 50 ohm.cm EPI diode

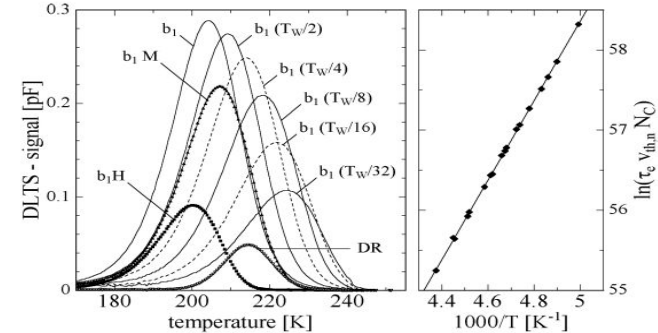
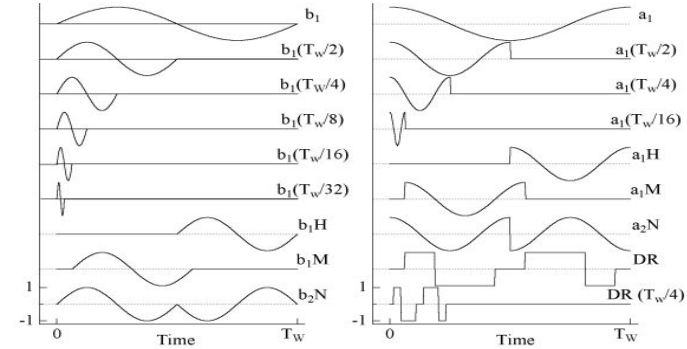


- I-DLTS spectrum is obtained from transients measured at different temperatures

# I-DLTS



## Correlator functions → Arrhenius



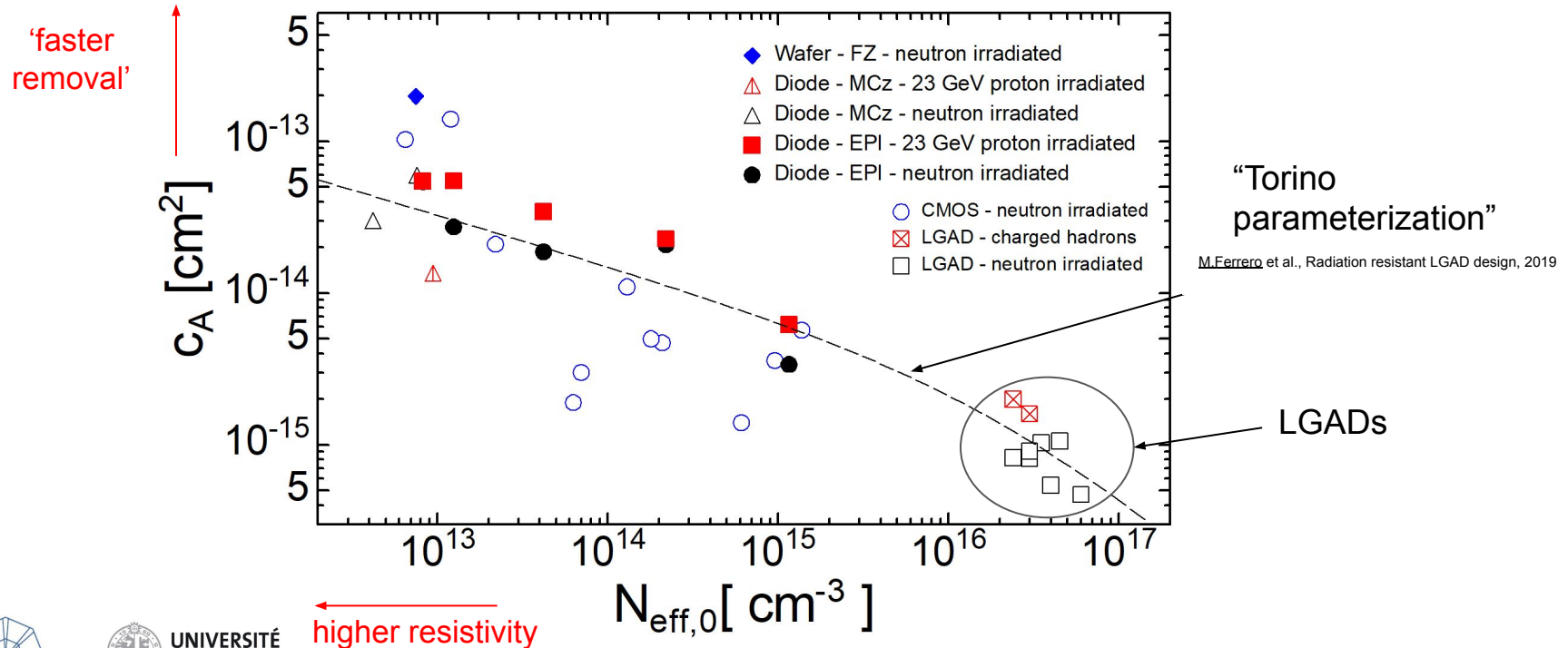
$$I(t) - I_R = \frac{1}{2} q S w_R N_T \frac{1}{\tau_e} e^{-t/\tau_e}$$

Amplitude of I-transient (trap concentration) is T-dependent, important to find good conditions ( $T_w, V_R, t_p$ )

# Acceptor removal coefficient $c$

$$N_B = N_{B0} \exp(-c_A \Phi)$$

As reported in literature + our data (different measurements techniques used, different devices, different Si material ([C], [O], [B]) → strong scattering of the data

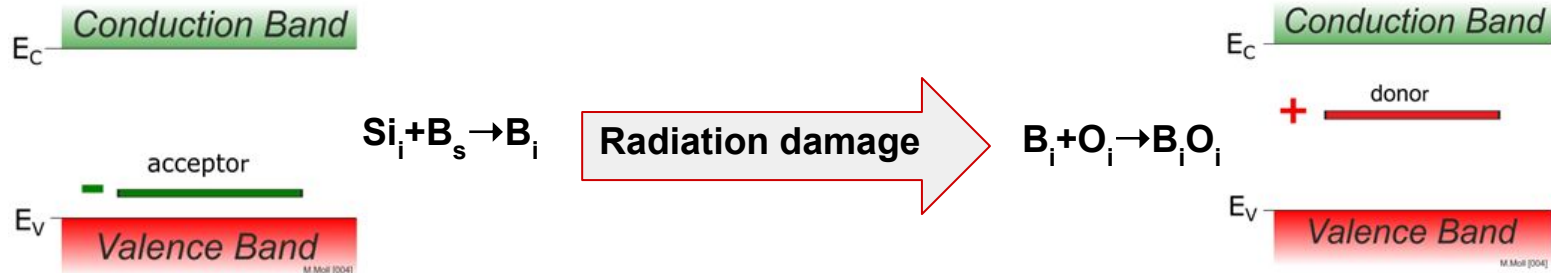


# Acceptor removal: reminder



- Radiation induced removal of  $B_s$  from the substitutional lattice site, its deactivation as a shallow dopant leading to the change of  $V_{fd}$  and  $N_{eff}$  on the macroscopic level
- Originated from  $B_iO_i$  complex formation on the microscopic level

- Most typical radiation induced reaction:



$B_iO_i$  - donor in the upper part of  $E_g$  (contributes with '+' space charge)  
For every removed Boron an acceptor is erased and a donor is created (factor of 2! in space charge)

# Possible defect kinetics in Si: reminder

Assumption:  $[O] \gg [B], [C]$

- Boron can be removed by the reactions:

$V + B_s \rightarrow VB$  (anneals out @  $T=0^\circ\text{C}$ ) - no role to play

$I + B_s \rightarrow B_i \rightarrow B_i + O_i \rightarrow B_i O_i$

$B_i$  - highly reactive!

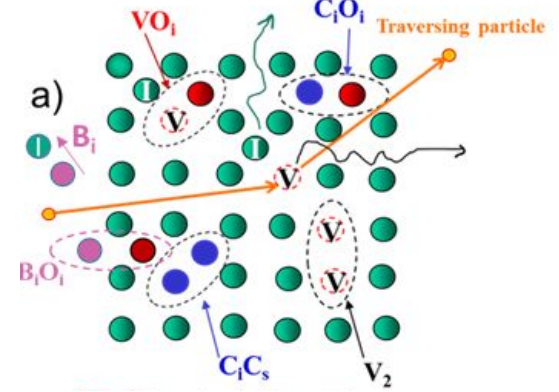
- $Si_i$  are shared between  $B_s$  &  $C_s$

Concurrent reaction channel:  $I + C_s \rightarrow C_i \rightarrow C_i + O_i \rightarrow C_i O_i$   
 (Increasing  $C_s$  will protect  $B_s$  from removal)

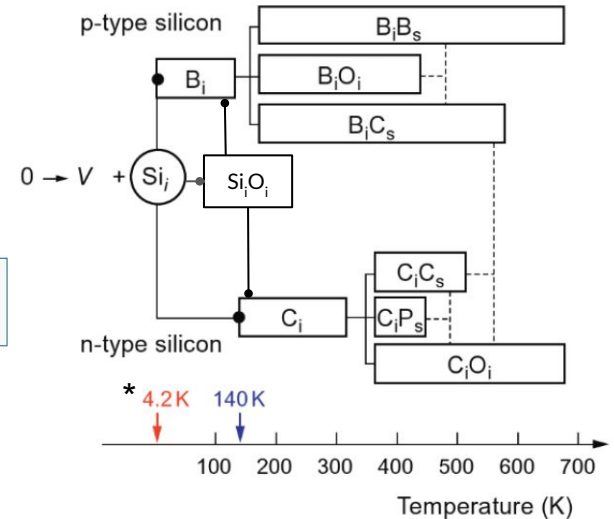
- $Vs$ :  $V + O_i \rightarrow VO_i$  (remains more  $Si_i$  available)
- Initial Boron removal rate (i.e. rate of  $BiO_i$  formation at low fluence):

$$g_B = g_{BiO_i} \approx g_I \left( 1 + \frac{k_{IC}[C_s]}{k_{IB}[B_s]} \right)^{-1}$$

- Generation of interstitials (outside clusters):  $g_I \approx 1-3 \text{ cm}^{-1}$  (high resistivity silicon)
- Sharing of interstitials between  $B_s$  and  $C_s$ :  $k_{IB}/k_{IC} \approx 1-7$
- $[C_s] \approx 1 - 5 \times 10^{15} \text{ cm}^{-3}$



Competing reactions with  $Si_i$  involving B, C and O



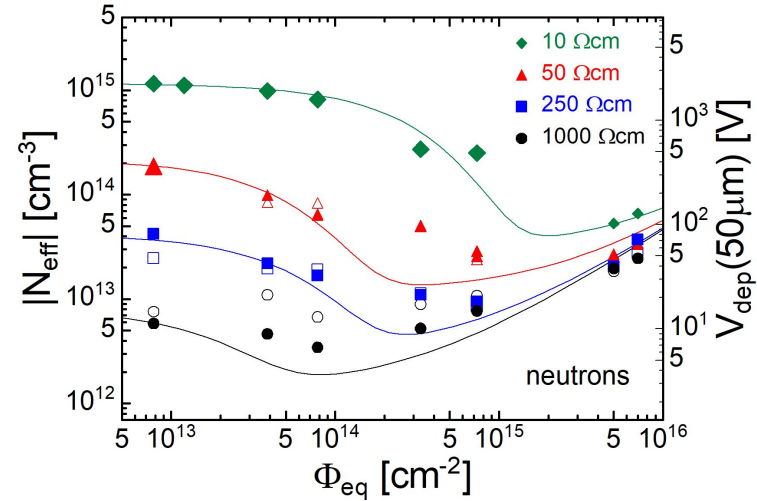
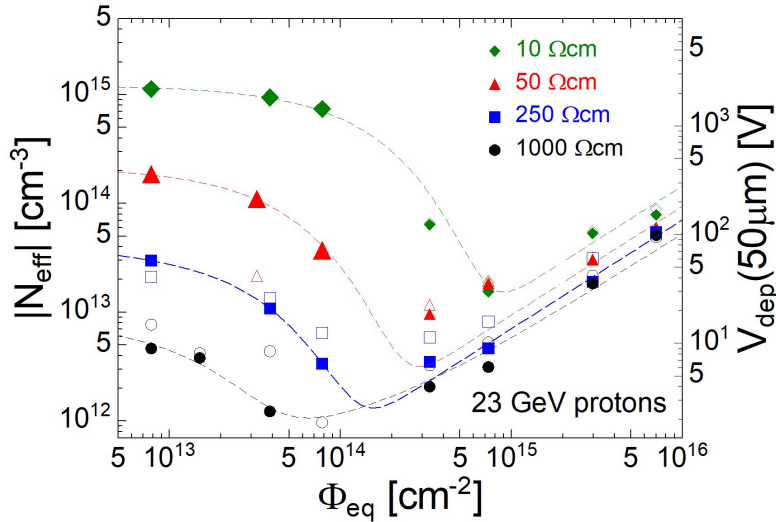
\*Watkins, G.D., Lattice defects in semiconductors, 1974

# Macroscopic Damage: $N_{eff}$

- $N_{eff}$  extracted from CV measurements (10 kHz, -20C)
- can be affected by errors for lower and higher irradiated sensors!

## Assumptions:

- $-V_{fd}$  is a valid parameter for evaluation of  $N_{eff}$
- $-N_{eff}=const$  throughout the bulk



- Samples initially differing by more than 2 orders of magnitude in resistivity behave very similar after very high radiation levels
- Very complex behavior after proton irradiation (“type inversion” ← TCT measurements)
- Parameterization of the data gives  $c$  - “removal coefficient” for each resistivity

$$N_{eff}(\Phi_{eq}) = N_{eff,0} \cdot \exp(-c\Phi_{eq}) + g_c\Phi_{eq}$$