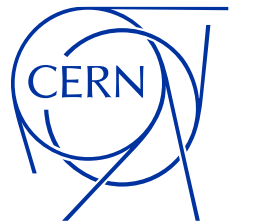


Advances with Nb₃Sn HiPIMS coatings for SRF cavities

Stewart Leith

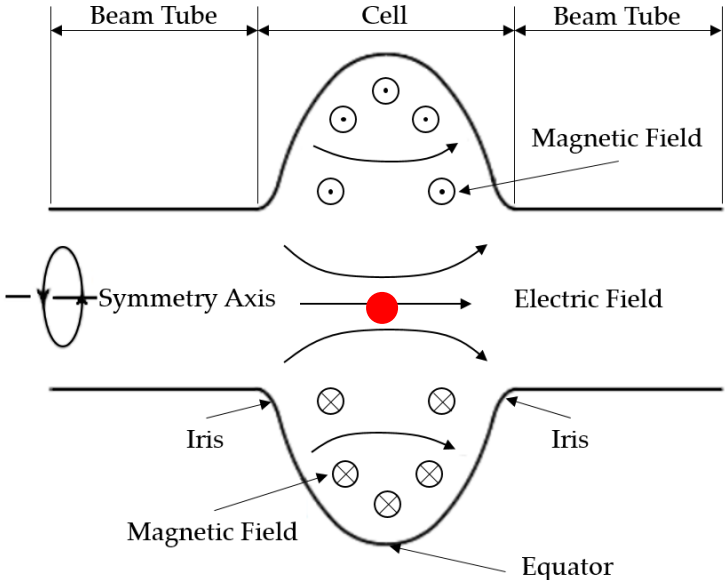
TE-VSC-SCC



Background

Superconducting Radio Frequency (SRF) Cavities

- Accelerate charged particles
- Bulk Nb vs. **Nb/Cu**



Figures of Merit

- Quality factor

$$Q_0 = G/R_s = \frac{\text{stored power}}{\text{dissipated power}}$$
- Surface resistance

$$R_s = R_{\text{BCS}} + R_{\text{res}}$$

Motivation

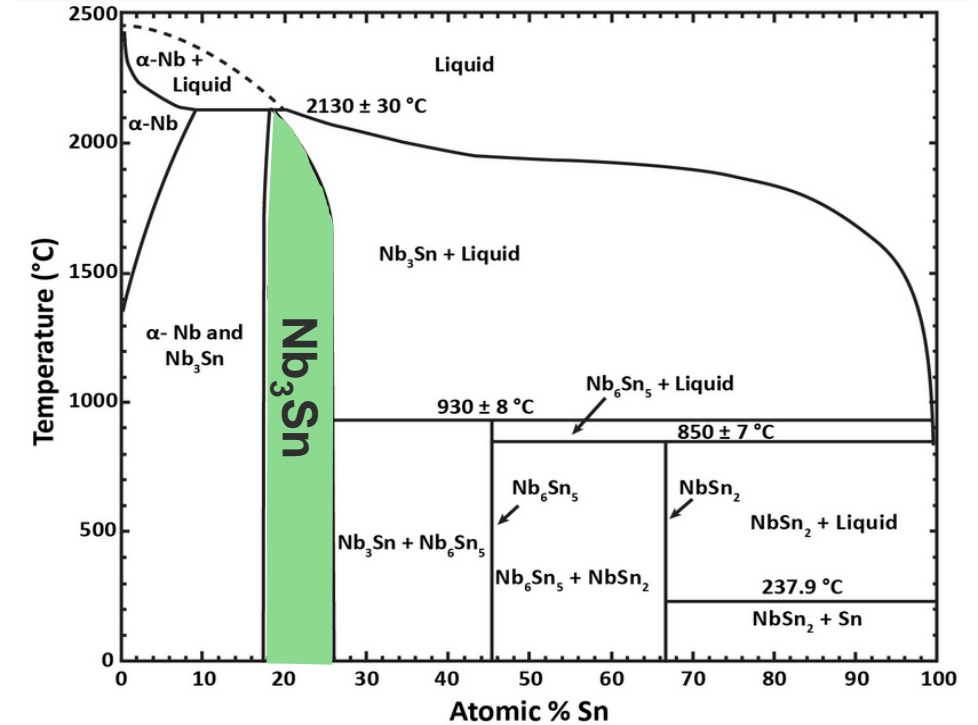
Nb₃Sn

- Q_0 at 4.2 K ~ bulk Nb at 2 K

T_c	Nb ~ 9.2 K
	Nb ₃ Sn ~ 18.3 K
R_{BCS} @4.2K and 500MHz	Nb ~ 45nΩ
	Nb ₃ Sn ~ 0.4nΩ

Challenges

- Superconducting Nb₃Sn phase formation
 - Stoichiometry control (Sn at. % 18 - 26 at. %)
 - High temperature reaction
- Copper substrate influence



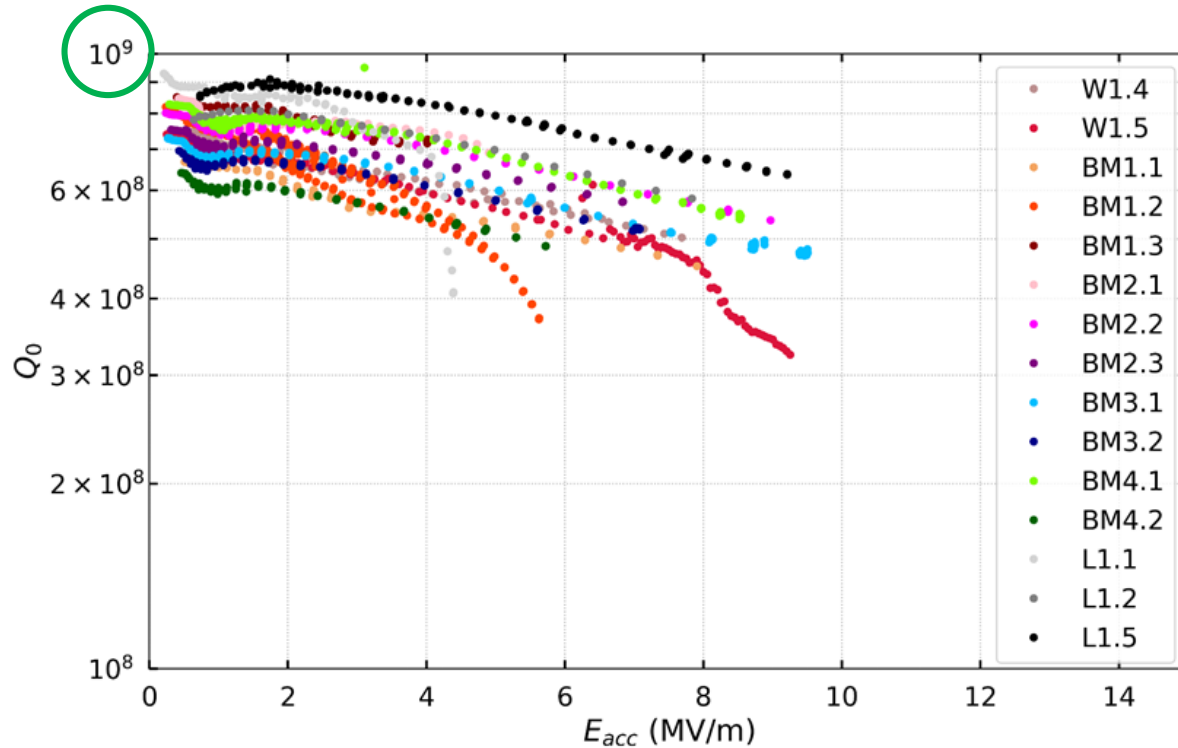
Binary phase diagram of the Nb-Sn system [1]

[1] J. Charlesworth, I. MacPhail, and P. Madsen, J. Mater. Sci. **5**, 580 (1970).

State of the art

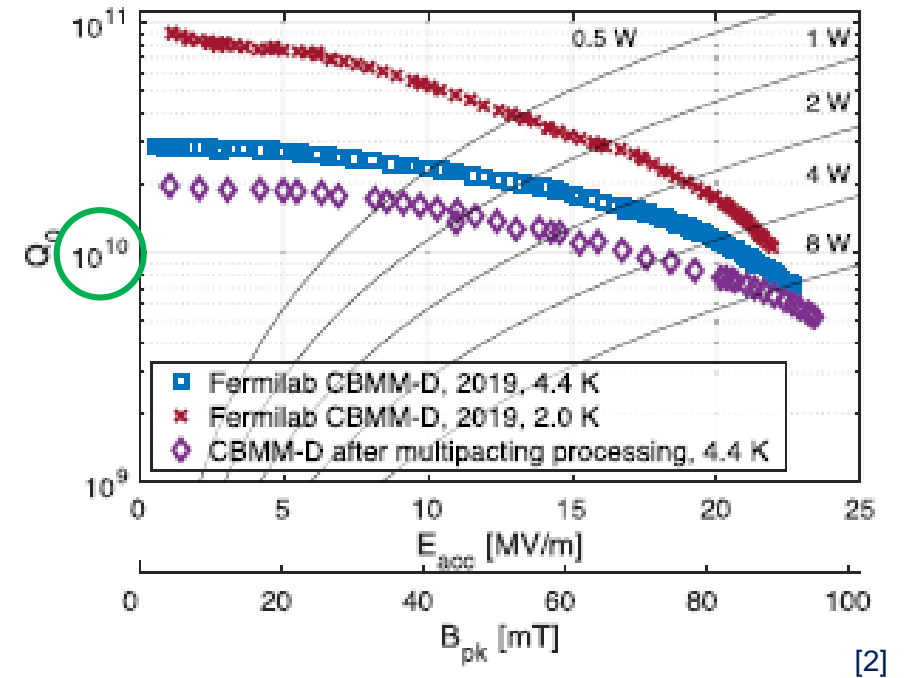
1.3GHz cavities

Nb/Cu @ 4.2K



Courtesy: L. Vega-Cid

Nb₃Sn/Nb @ 4.4K



Order of magnitude improvement in Q_0

[2] S. Posen et al, Supercond. Sci. Technol. **34** (2021)

DC MS Nb₃Sn Coatings @ CERN

DC Magnetron Sputtering (DC MS)

- Formation of A15 phase

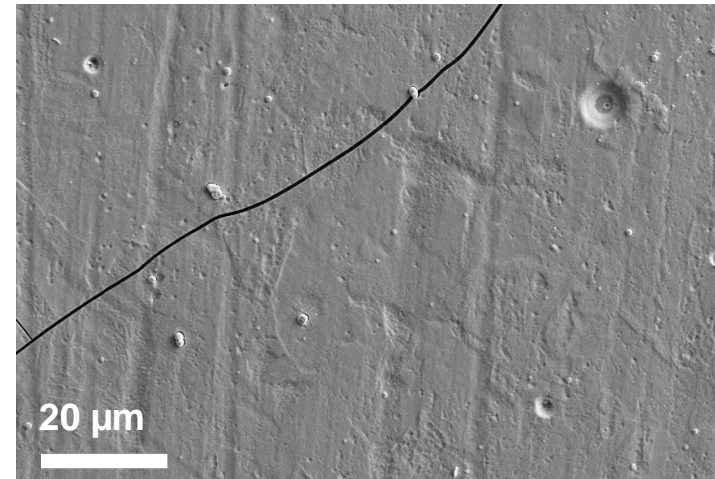
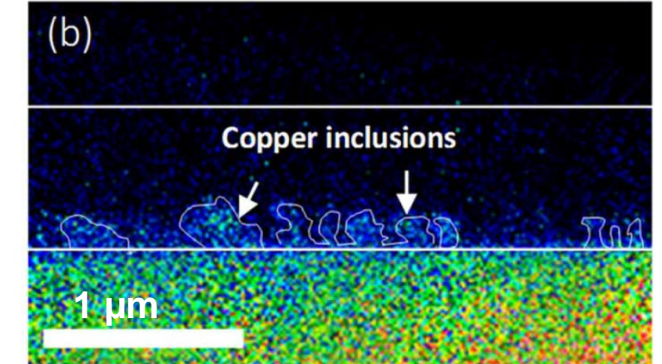
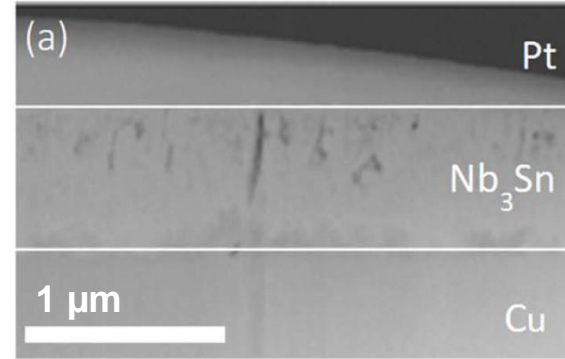
- Reacted **During** and **After** Coating
- High temperatures required (> 600°C)

- Cu Diffusion

- Barrier interlayer required (Ta or Nb)

- Surface cracking

- Mitigated with Kr

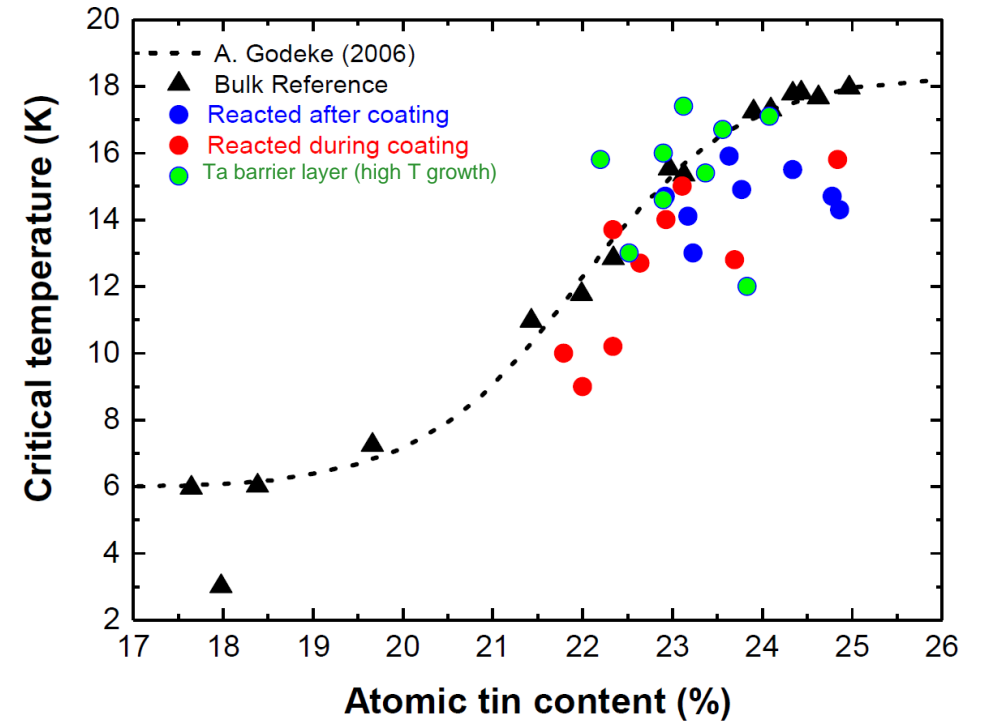


[3] K. Ilyina et al. *Supercond. Sci. Technol.* **32** (2019)

DC MS Nb₃Sn Coatings @ CERN

DC Magnetron Sputtering (DC MS)

- **Impressive T_c**
 - Consistently higher when reacted after coating
 - Sn dependent (increased with Kr)
 - Increased with Ta interlayer

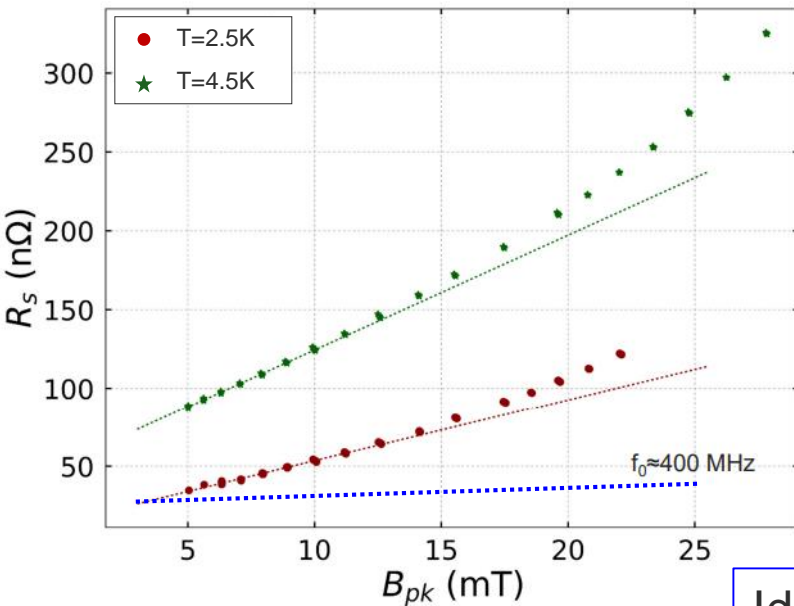


- **Quadrupole Resonator (QPR) Coatings**

- Reduction of low field R_s
- Q-slope



Switch to HiPIMS



Ideal behaviour

[3] K. Ilyina et al. *Supercond. Sci. Technol.* **32** (2019)

[4] M. Arzeo et al. FCC Week 2018

Bipolar HiPIMS Nb₃Sn/Ta/Cu

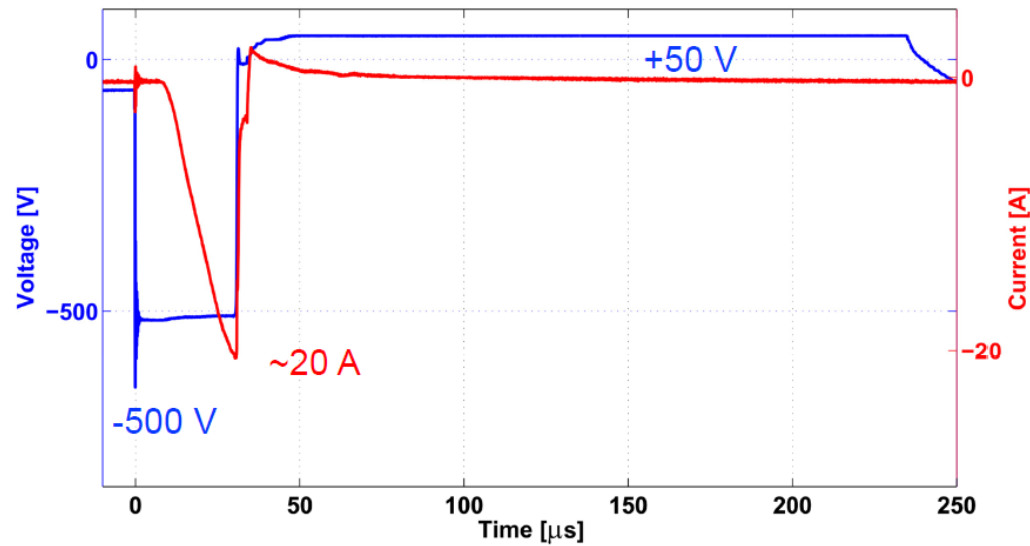
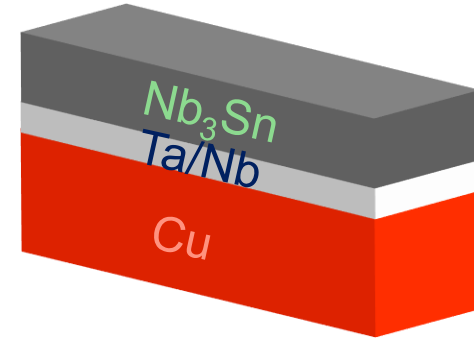
Reacted **During** Coating

High **P**ower **I**mpulse **M**agnetron **S**puttering (HiPIMS)

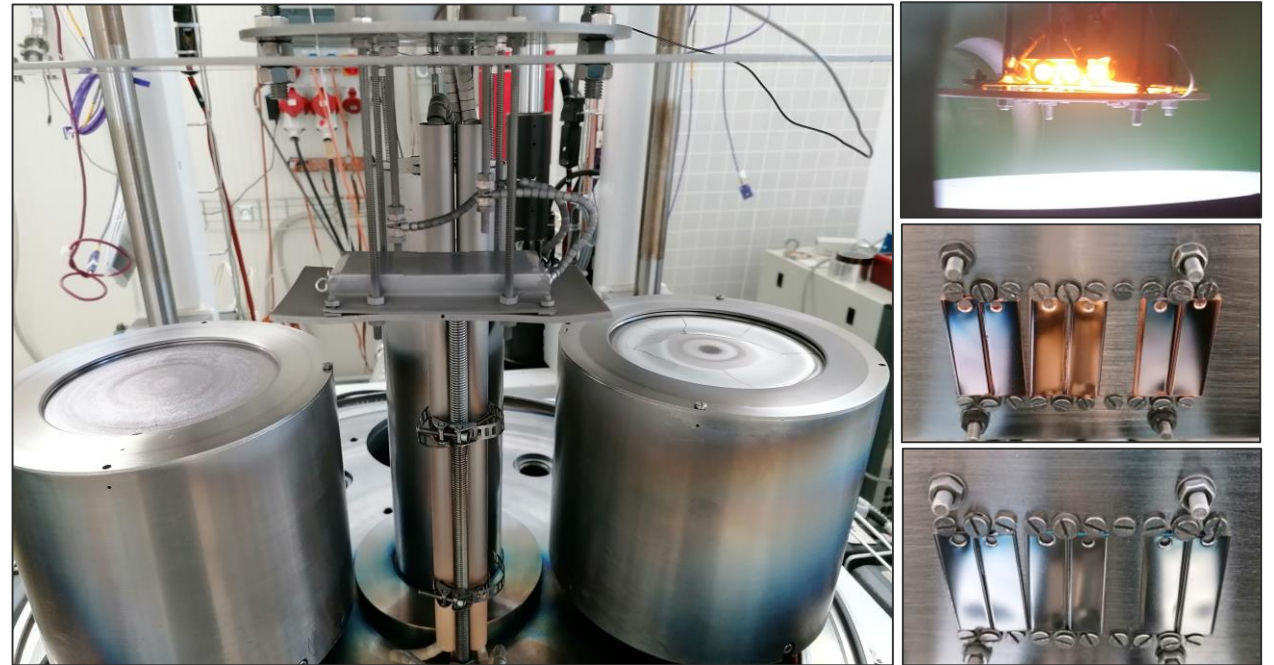
- Improved density required for RF performance
- Proven with Nb/Cu

Coating parameters:

- Gas: Kr
- T_s : 500 ... 750°C
- P : $7 \cdot 10^{-4}$... $5 \cdot 10^{-2}$ mbar
- PP : 35 ... 100 V
- Post anneal: 0 ... 72 hrs



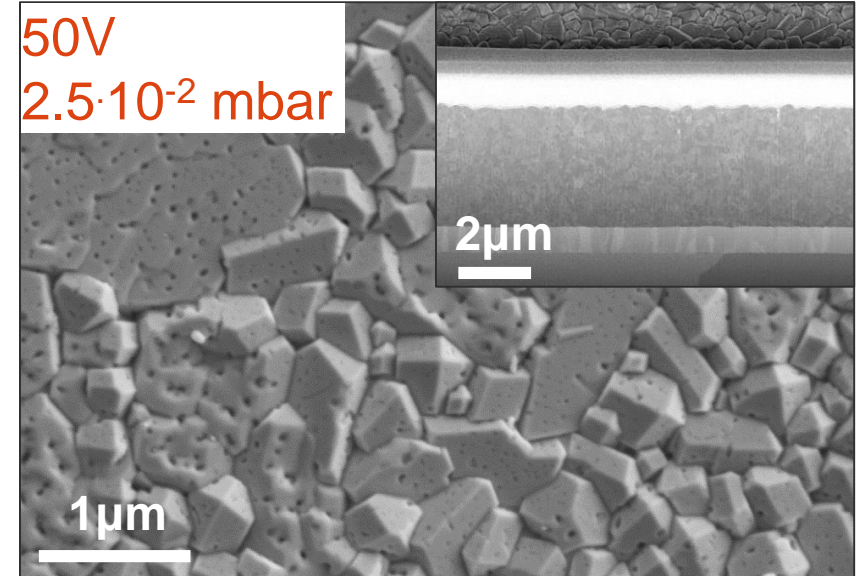
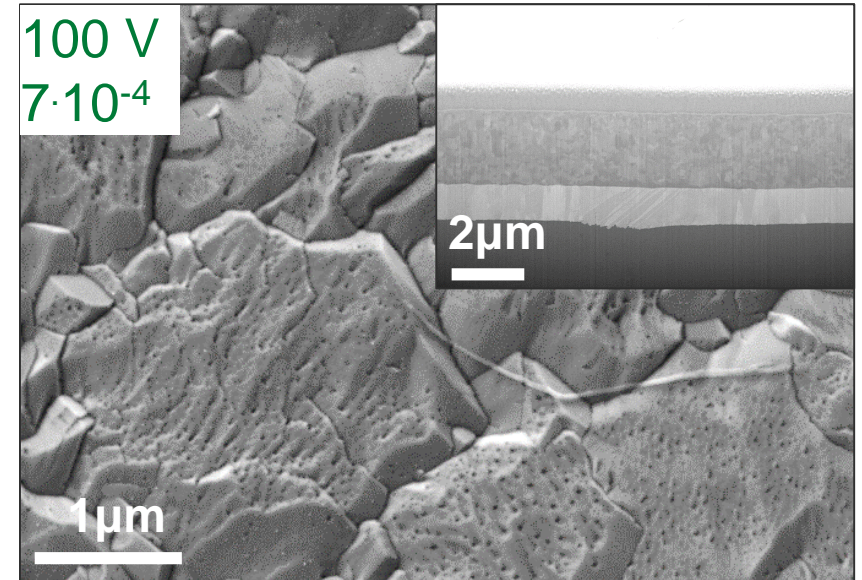
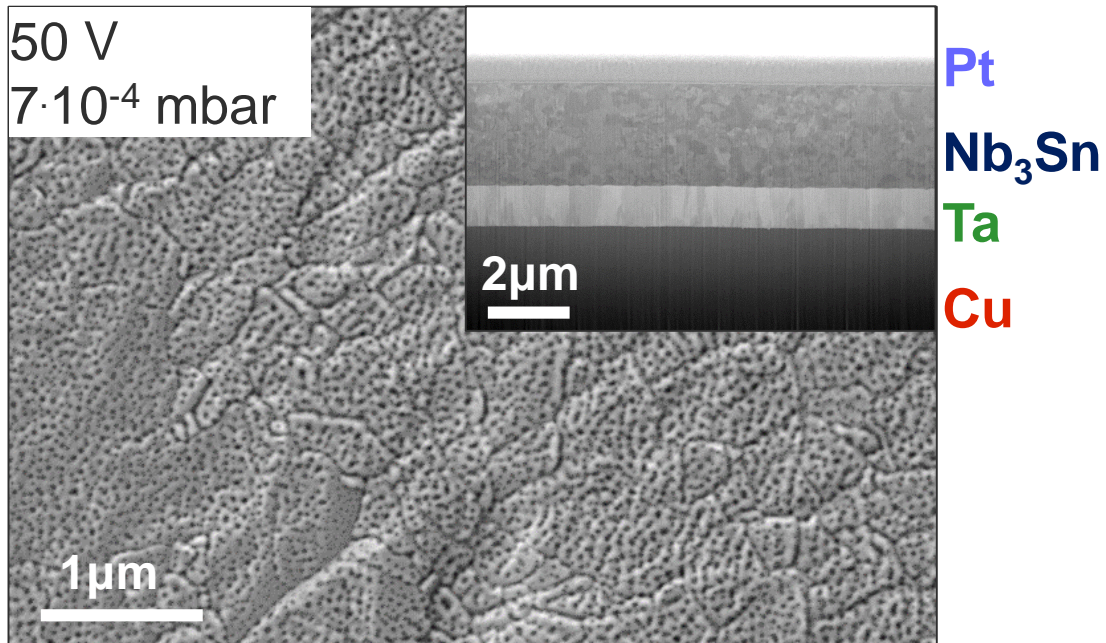
[5]



[5] F. Avino *et al* 2019 *Plasma Sources Sci. Technol.* **28** 01LT03

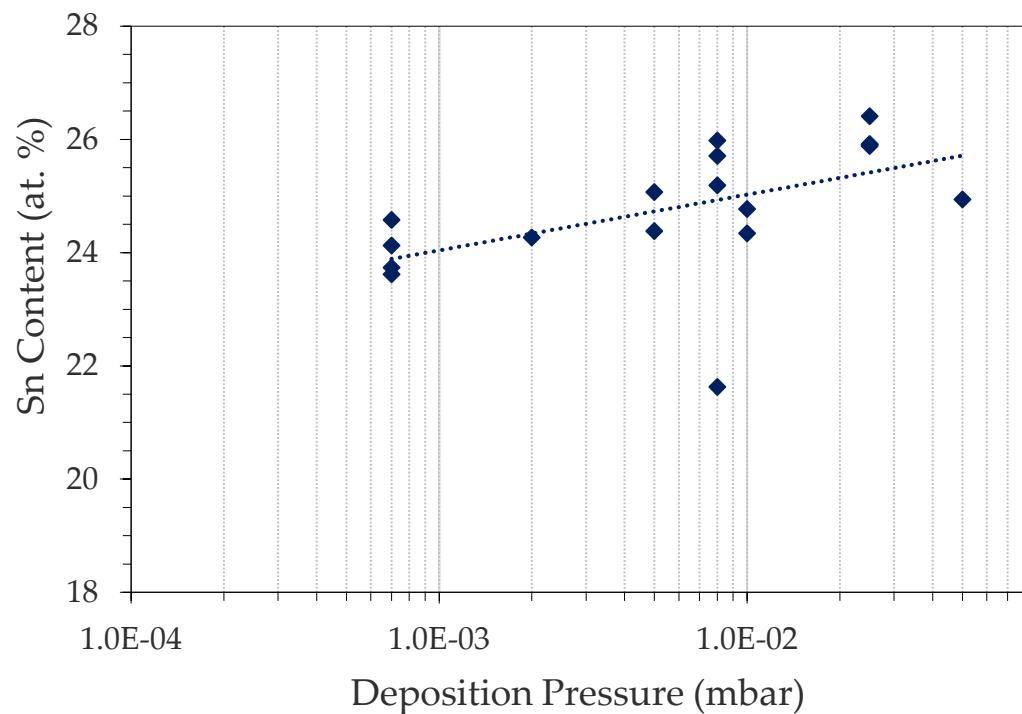
Morphology and Crystallinity

- Correct A15 phase for all samples
- Dense, crack free, porous surface
- No effect due to annealing

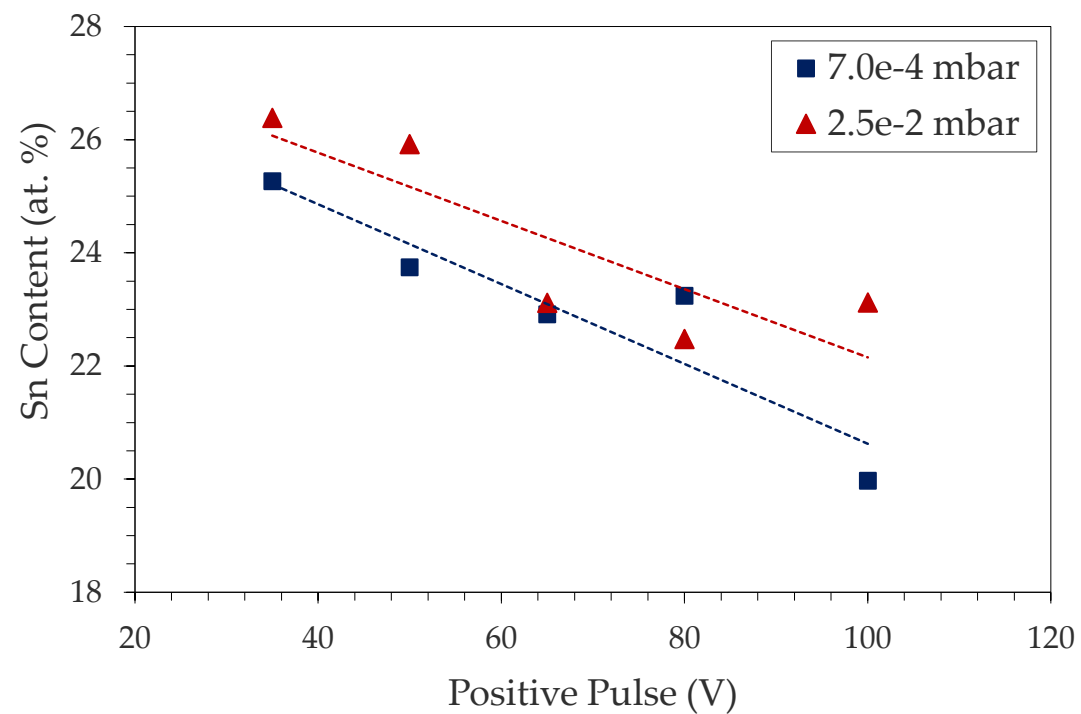


Compositional Analysis

- Sn % = 20 to 26.4 at. % (EDS)
- Coating temperature shows no obvious trend
- Post-coating annealing has no effect on Sn %



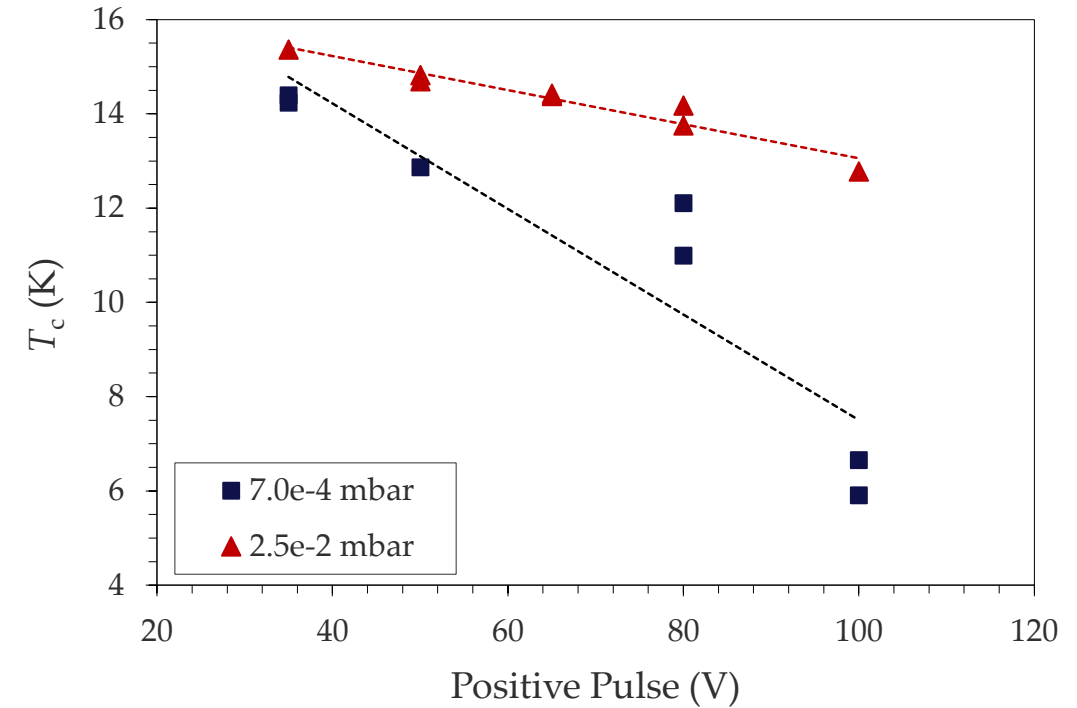
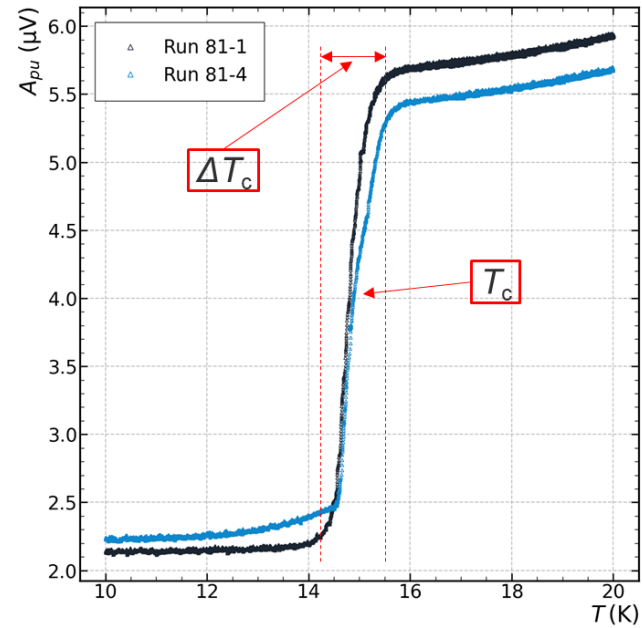
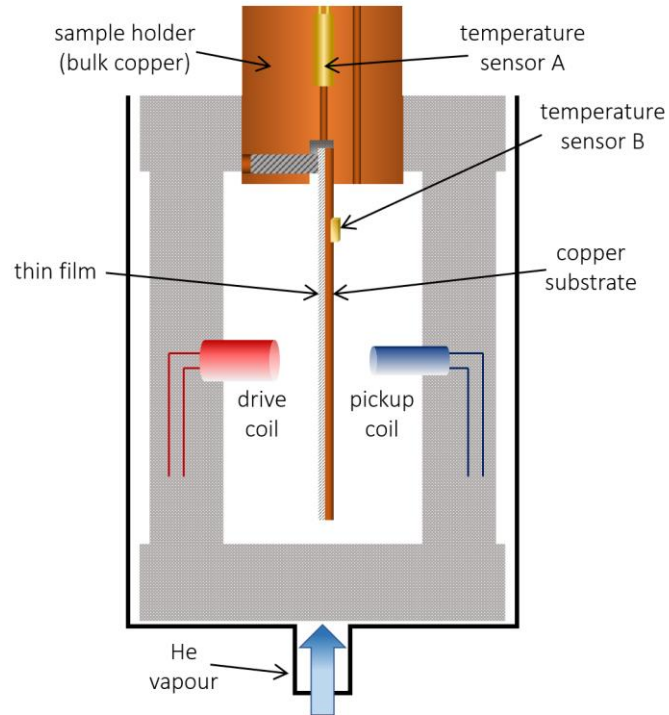
Resputtering or Nb/Sn ion ratio



Superconducting Performance

Inductive T_c measurement

- We want high T_c and low ΔT_c



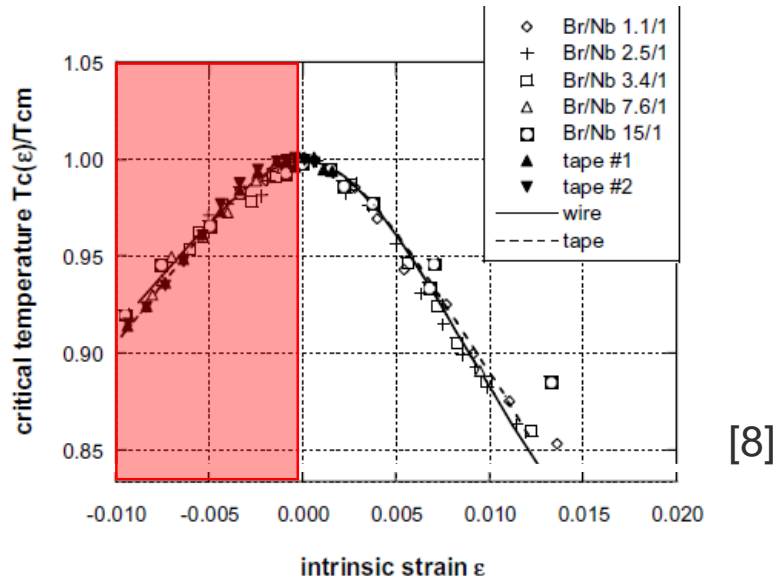
[6] D. Fonnesu - PhD Thesis (to be published)

Superconducting Performance

Nb₃Sn T_c suppression

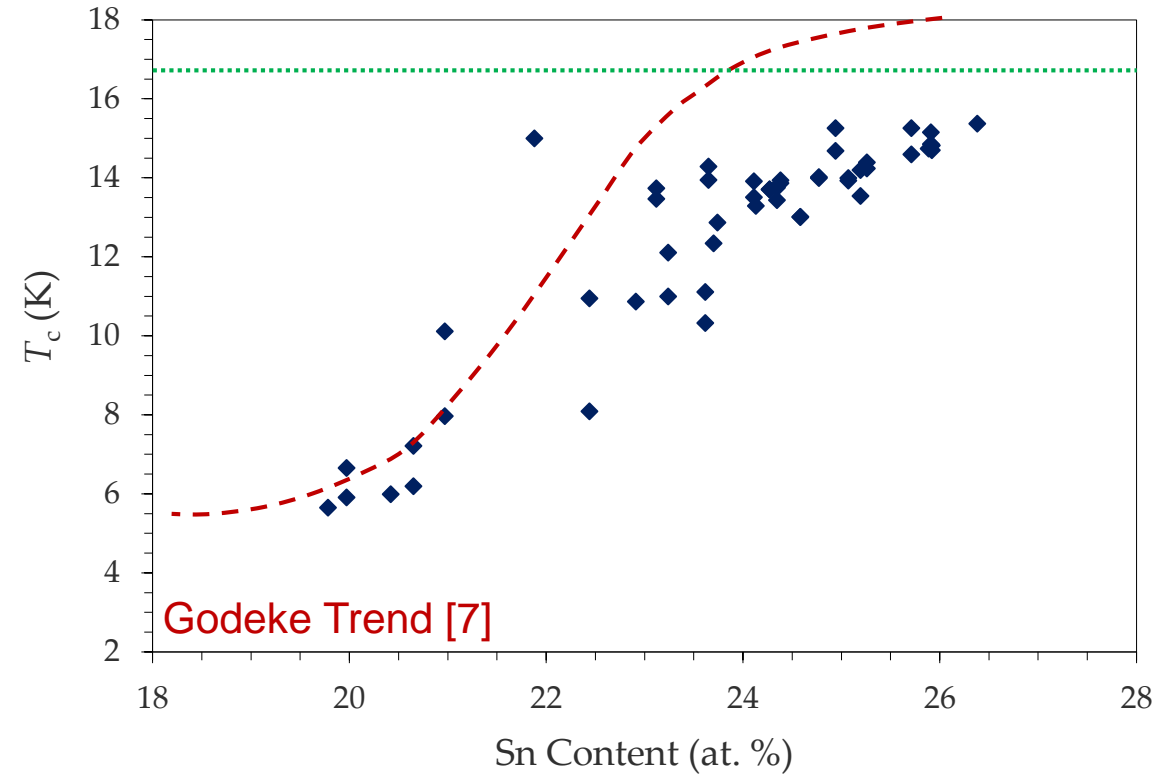
- Sn content in required range
- Residual stress
 - Any stress reduces T_c with Nb₃Sn
 - σ_{Ave} ($2.5 \cdot 10^{-2}$ mbar) = 1.5 GPa
- Further issues involved

Compressive residual stress



[8]

Theoretical Max T_c @ 1.5 GPa



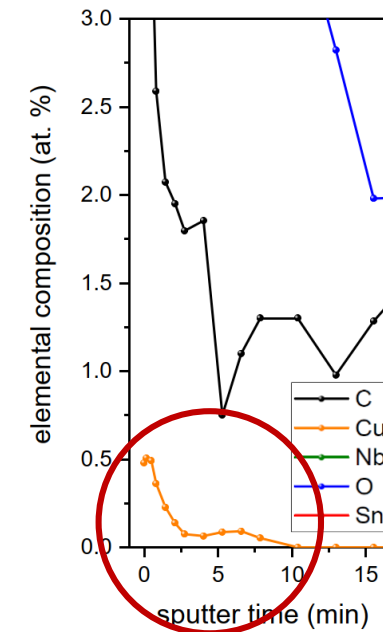
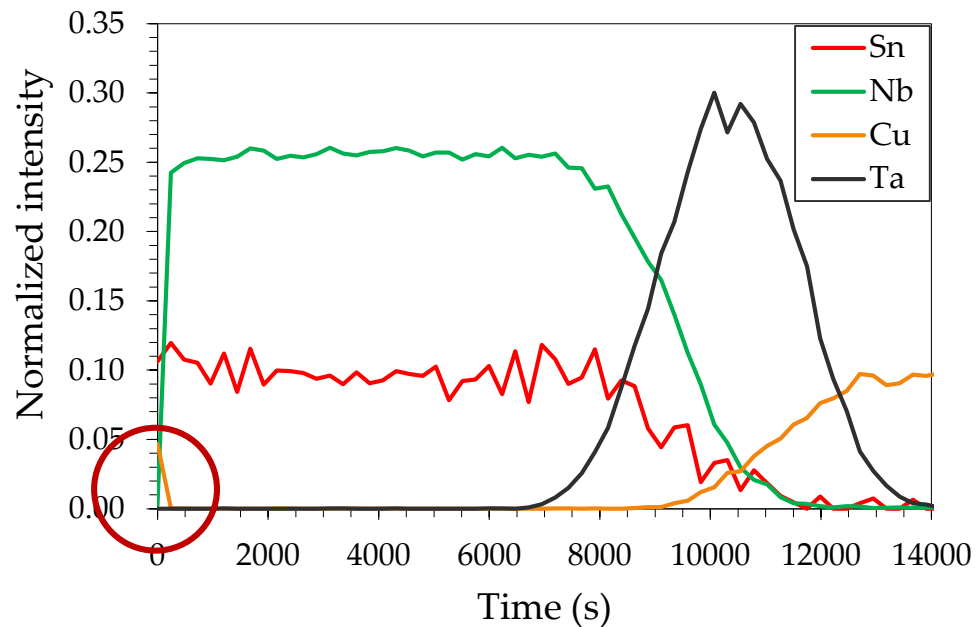
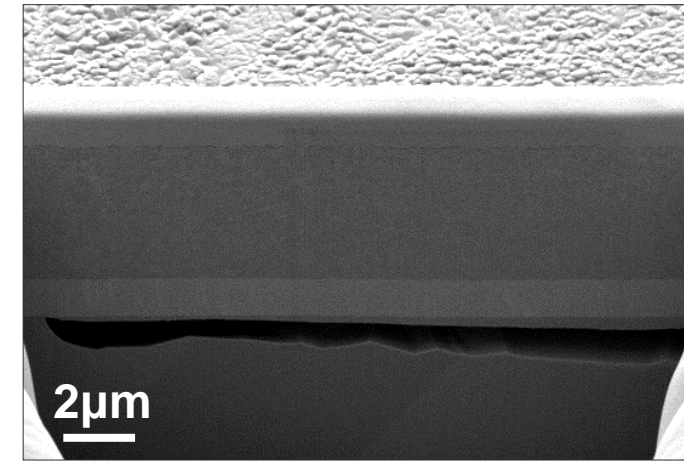
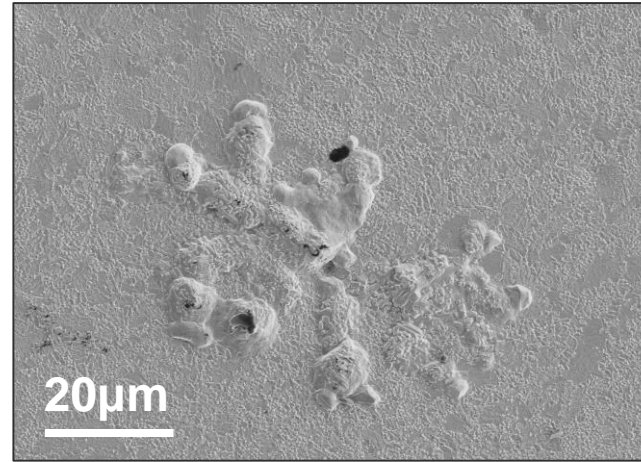
Godeke Trend [7]

[7] A. Godeke. *Supercond. Sci. Technol.* **19** (2006)

[8] G. De Marzi et al. *J. Phys.: Condens. Matter* **25** (2013)

Surface Cu Contamination

- Not always visible in SEM
- Often **undetected by EDS** – **SIMS / XPS**
- Localised to film surface



Removal of Cu contamination

- Diffusion length
 - o Increased Nb₃Sn film thickness

$t = 6 \mu\text{m vs. } 3 \mu\text{m}$



No

- $T_c \sim 16 \text{ K}$ (1 K increase)
- Cu surface contamination

- Diffusion dictated by temperature and time
 - o Decreased coating temperature

$T_{\text{coat}} = 500^\circ\text{C}$



No

- $T_c \sim 10 \text{ K}$
- Too low for A15 formation

- Surface Chemistry
 - o Ammonium persulfate rinse (15 min) - masked

Cu at. %	Pre Rinse	Post Rinse
Sample 1	3.9	0.6
Sample 2	5.3	1.6

- Still detectable
- How deep?



Maybe

- XPS depth profile

Post coating reaction to optimise A15 phase formation and minimise Cu diffusion

Bipolar HiPIMS Nb₃Sn/Ta/Cu

Reacted **After** Coating

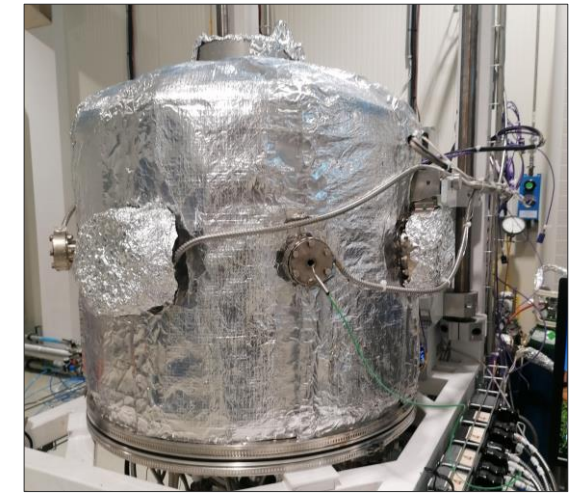
- Low temperature coatings (< 250°C)
 - Plasma self-heating
- High pressure ($2.5 \cdot 10^{-2}$ mbar)
- Low bias (50 V)

Annealing pressures	$5 \cdot 10^{-6}$ - $1 \cdot 10^{-7}$ mbar (ex-situ) $2.5 \cdot 10^{-2}$ mbar (Kr) (in-situ)
Annealing temperatures	450 - 750°C (1 - 24 hrs)

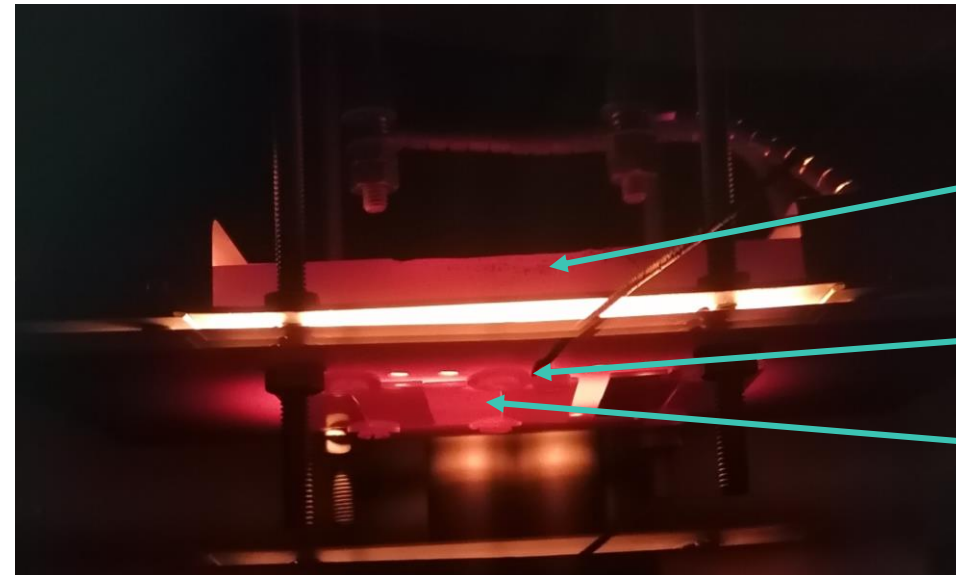
- Ex-situ coatings require air exposure
 - Oxygen leads to smaller crystallites
- In-situ completed in Kr atmosphere



Ex-situ (vacuum furnace)



In-situ (coating system)



Heater

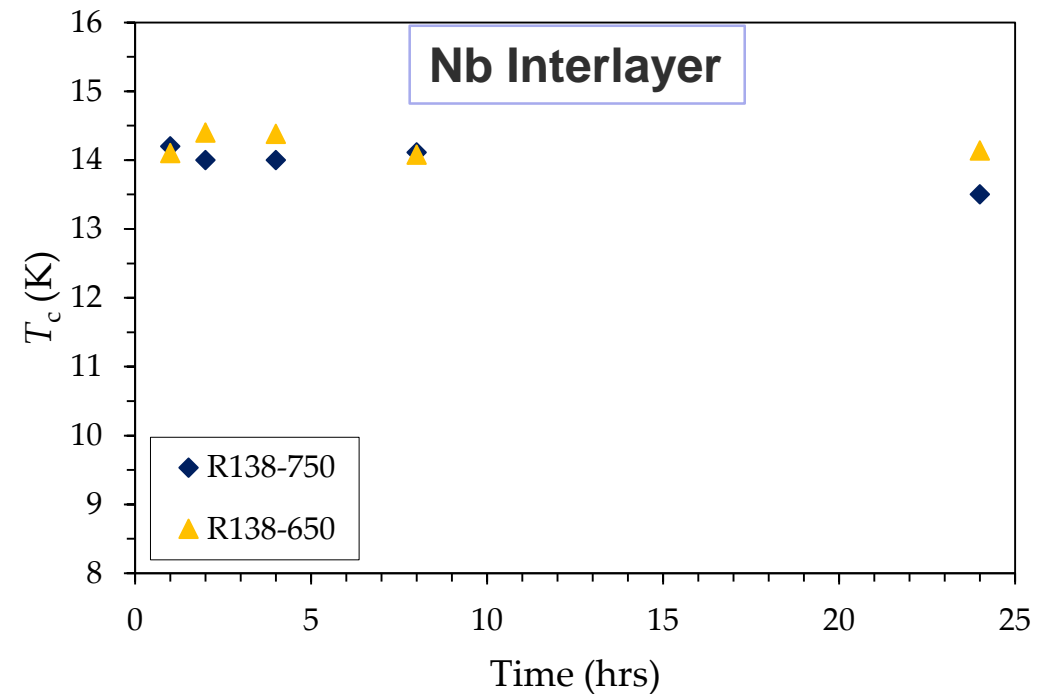
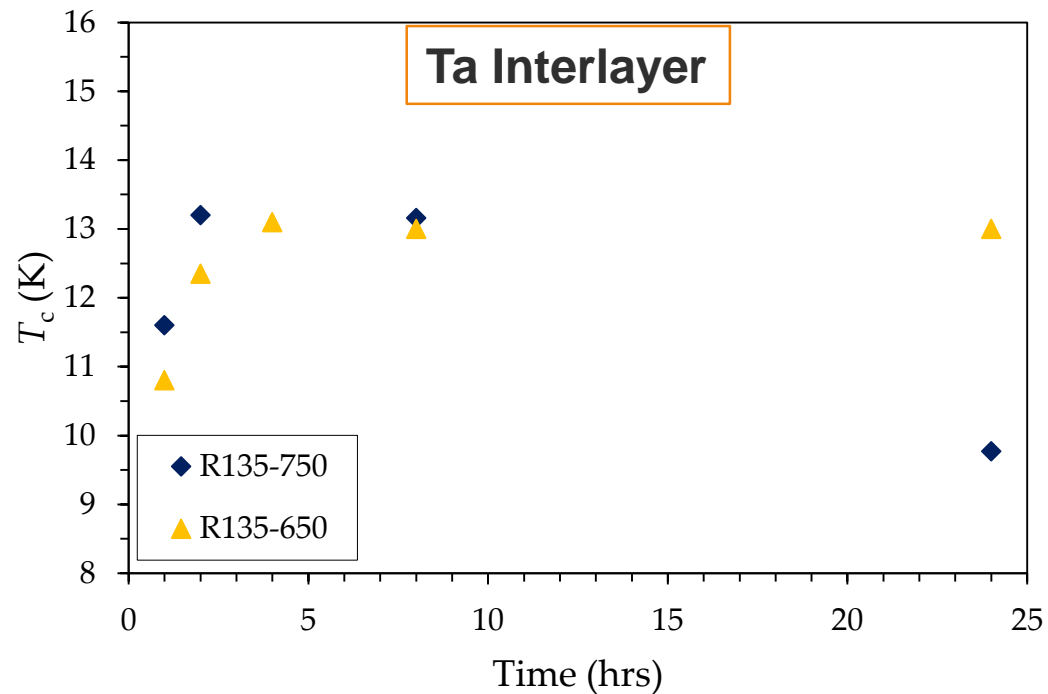
Thermocouple

Substrates

Reacted After Coating

Ex-situ

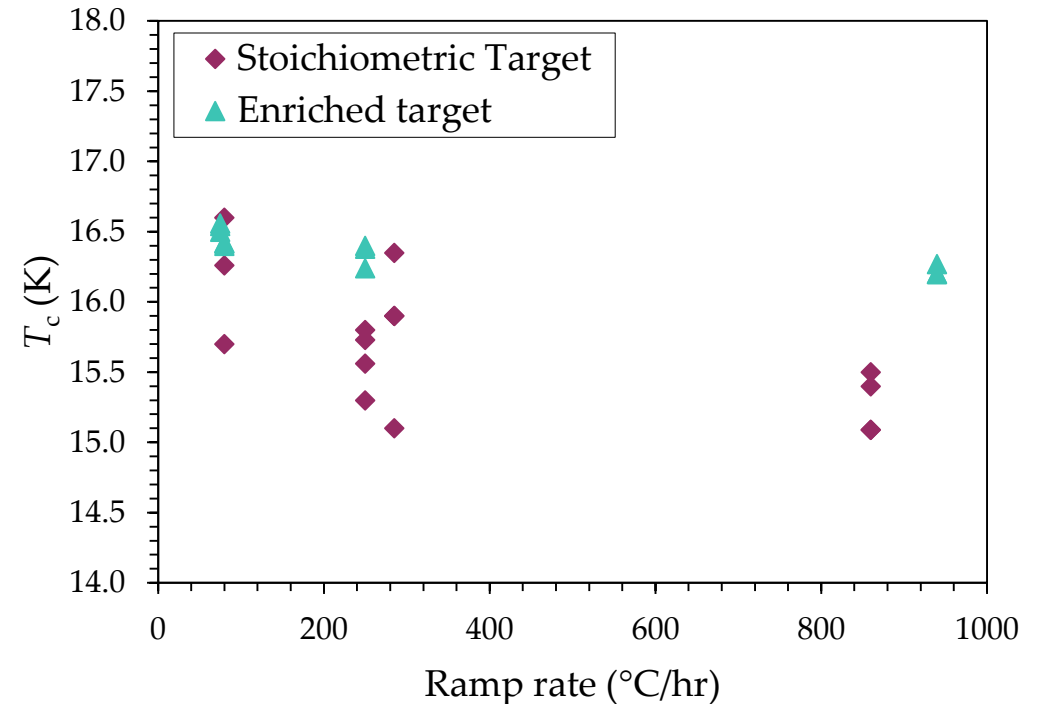
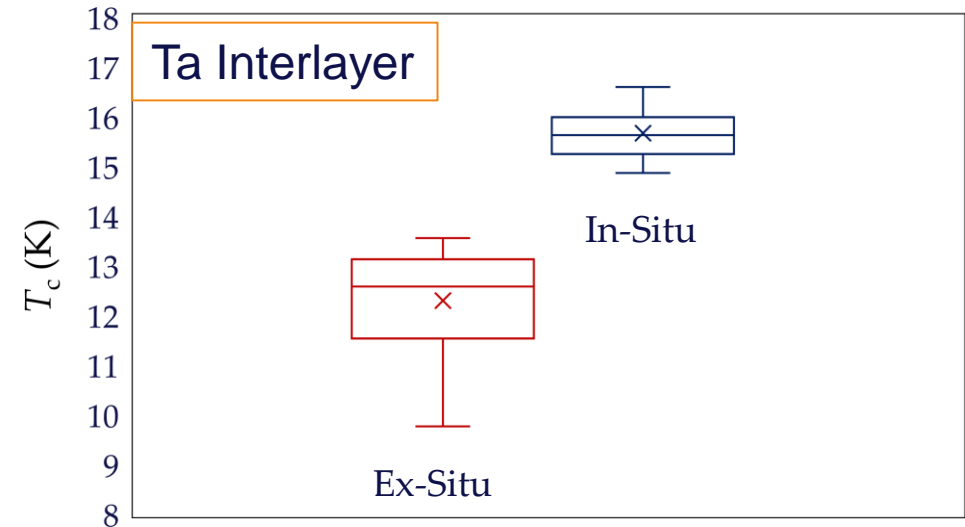
- $T_c < 10$ K for $T_{\text{anneal}} = 450/550^\circ\text{C}$ (24 hrs)
 - Focus on 650/750°C
- Discrepancy between Ta and Nb interlayers



Reacted After Coating

In-situ

- Higher T_c than ex-situ (Stoichiometric target)
 - Highest $T_c = 16.6 \text{ K}$
 - Effect of air exposure?
 - Sn $P_v = 1.91 \cdot 10^{-7} \text{ mbar}$
- Ramp rate dependency
 - Similar to Nb_3Sn magnets
- Two targets (Ta interlayer initially)
 - **Stoichiometric** (25 % Sn)
 - **Enriched** (27% Sn)
 - Enriched target reduces T_c spread

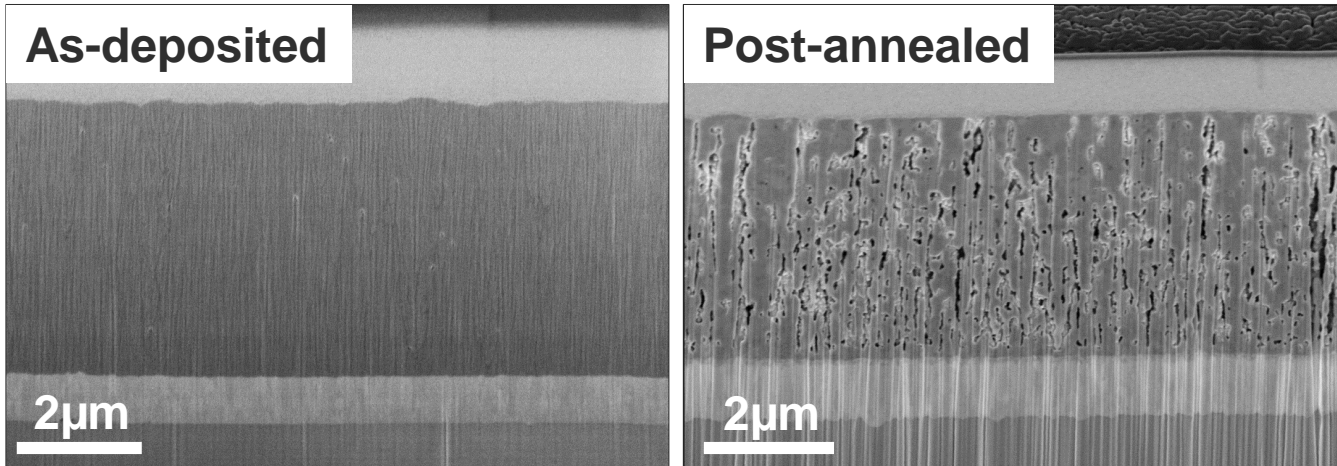


Reacted After Coating

In-situ

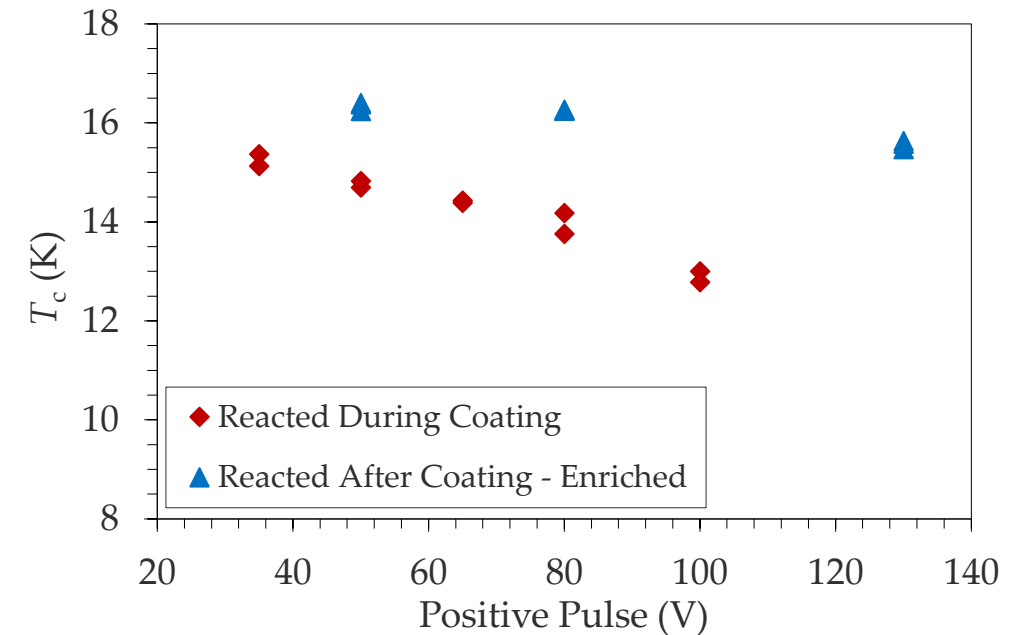
- Porous film post-anneal with both targets
- Increased **Cu** surface contamination with enriched target
- Instances of delamination

Stoichiometric Target



Solutions?

- Increase positive pulse
 - **Reduced** decrease in T_c
 - Cu contamination still **high**
- Ongoing
 - Decrease annealing temp/time
 - Deposition parameter investigation



Summary

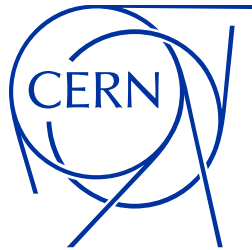
- Reacted **After** Coating shows consistently higher T_c
 - Similar to DC MS observations
 - T_c pushed to 16.6 K
 - Enriched target provides better film performance
- Surface **Cu** contamination still an issue
 - Effect on RF still outstanding

Future Work

- Process optimisation required
 - Deposition parameters to be explored to densify film
 - Further enriched target (30%) to be attempted
- Deposition of further QPR samples required
 - Before and after ammonium persulfate

Thank you for your attention

Questions?



Big thanks to:

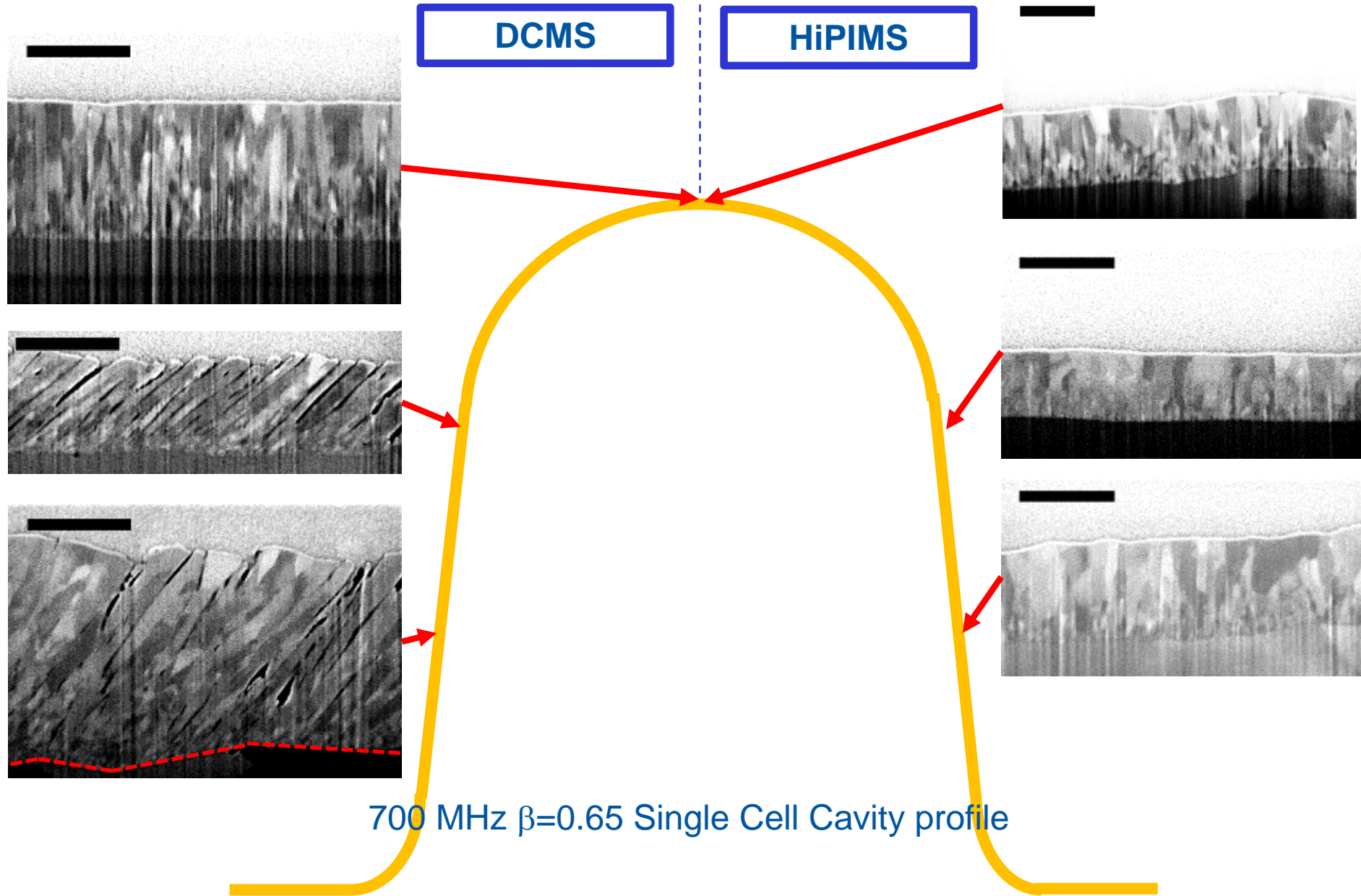
C.P.A. Carlos, G. Rosaz, P. Garritty, M. Watkins, M. Himmerlich, M. Taborelli, S. Forel (TE-VSC)

D. Fonnesu, E. Reches, T. Koettig (TE-CRG)

B. Ruiz Palenzuela, S. Pfeiffer, A. Moros, A. -T. P. Fontenla, F. Motschmann (EN-MME)

L. Vega-Cid, A. Bianchi, G. Pechaud, W. Venturini-Delsolaro (SY-RF)

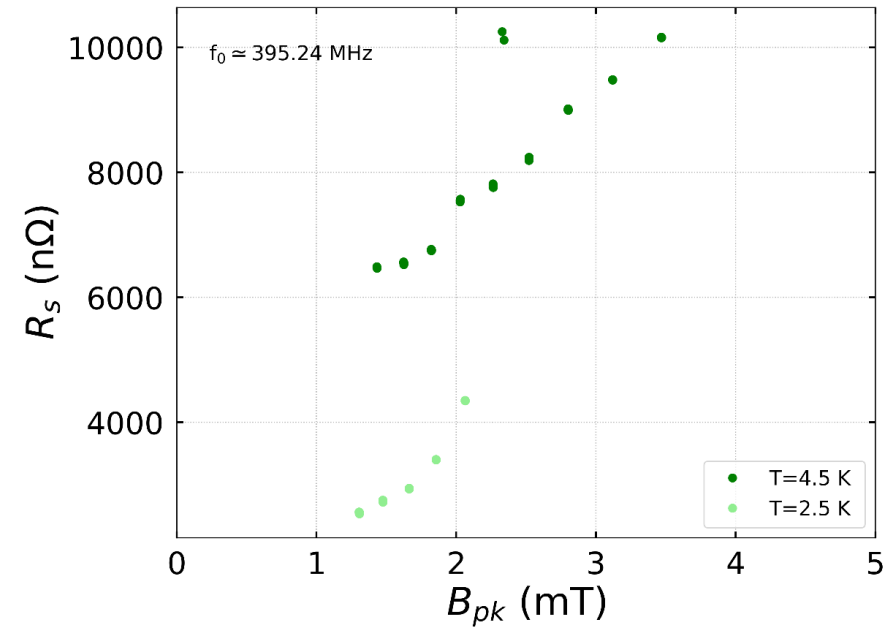
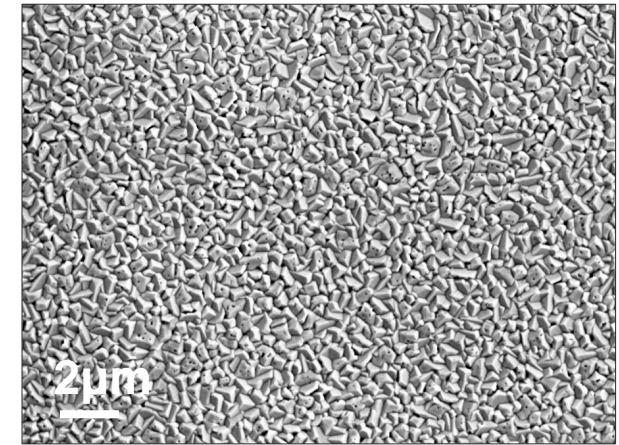
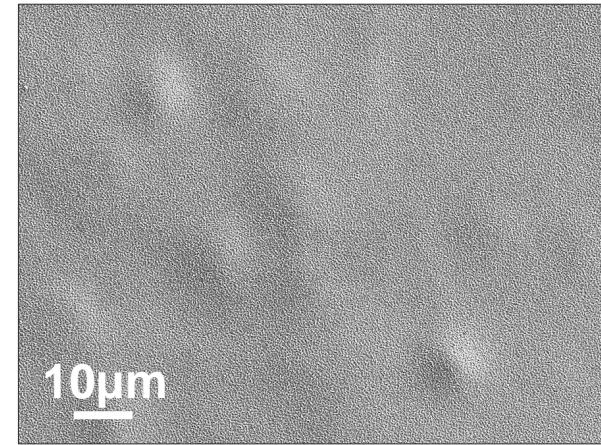
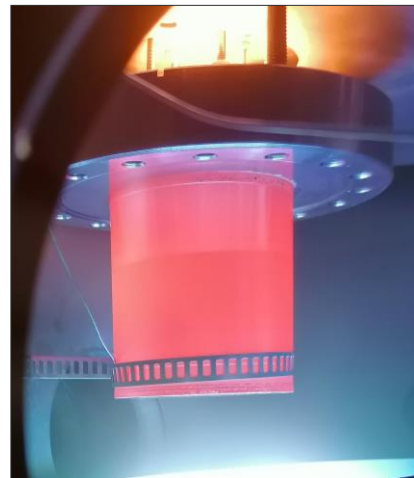
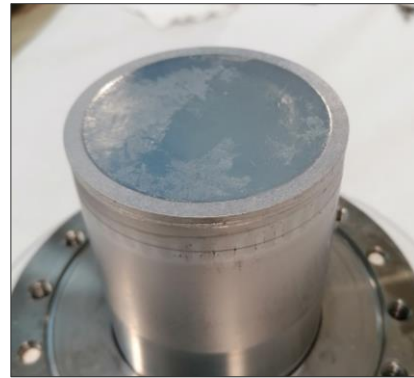
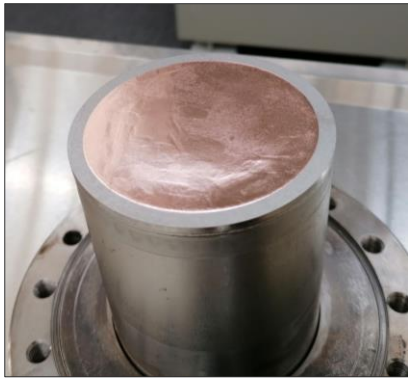
Back up slides



QPR Coatings

QPR Coating for RF measurements

- Sn % = 26.08 ± 0.21 %
- $T_c \sim 16$ K



Courtesy: L. Vega-Cid