



ENGINEERING  
DEPARTMENT

# Materials for magnets & measurements - II

Lecture I, Mon Nov 27

- Introduction and bibliography
- Magnetic properties of materials: types of magnetic behaviour
- Families and behaviour of magnetic materials

Lecture II, Tue Nov 28

- Materials for magnets: structural, cryogenics, vacuum
- Non-magnetic materials, phase transformations and measurements
- Case study: a low permeability stainless steel for the CMS HG-CAL

***CAS course on "Normal- and Superconducting Magnets", 19/11 – 02/12/2023, St. Pölten, Austria***

***Stefano Sgobba***



But it is by far the magnet the most spectacular stone which I wrote about above. I am also pleased to add here a picture of a magnet, as by looking at the stone itself it could not be properly expressed the iron slag which she draws, sticking out of it, like if the needle would understand the iron to be there...

84  
Magnesij populi dicuntur, masc. gene-  
re. nam & regio ipsa Macedoniae Ma-  
gnetia vocatur. *μαγνήτις* verò & *μά-  
γνησσα*, gentilia foeminini generis sunt.  
Longè verò alia est Magnetis illa lapi-  
di speculari similis, de qua scripsi su-  
prà. Libuit &  
picturam Ma-  
gnetis hîc ad-  
dere, quæ cû  
per se rectè ex-  
primi nō pos-  
set, ferri sco-  
bem qua tra-  
hit, ab eo hærentem expressi, sicuti &  
acum, vt ferrum esse intelligeretur.



Gesner, De omni rerum fossilium  
genere... (1565)



Magnes the shepherd, Pliny's Naturalis Historia, Book XXXVI, "The Natural History of Stones", chapter 25 – "The Magnet":  
Nicander is our authority that it was called Magnes from the man who first discovered it on Mount Ida and he is said to have  
found it when the nails of his shoes and the ferrule of his staff adhered to it, as he was pasturing his herds.



# Types of magnetic behaviour, summary

Type	Material	At $T \cong 295 \text{ K}^a$	References and Comments	
Type	Material	At $T \cong 295 \text{ K}^a$	At $T = 4.2 \text{ K}^a$	References and Comments
Diamagnetic	Cu	$\kappa = -9.3 \times 10^{-6}$	$\kappa = -9.4 \times 10^{-6}$	Fickett (1976); very pure Cu
	Nb	Paramagnetic	$\kappa = -1$	Superconductor below 9 K
Paramagnetic	Al	$\kappa = 2.0 \times 10^{-5}$	$\kappa = 2.5 \times 10^{-5}$	Fickett (1976)
	Ce	$\kappa = 1.6 \times 10^{-3}$	$\kappa = 23 \times 10^{-3}$	Edelstein (1968)
	Potassium-chrome-alum <sup>b</sup>	$\kappa = 1.6 \times 10^{-3}$	$\kappa = 32 \times 10^{-4} \text{ c}$	DeHaas and Gorter (1930)
	Fe-22Cr-13Ni-5Mn stainless steel	$\kappa = 2.1 \times 10^{-3}$	Antiferromagnetic	Ledbetter and Collings (1979)
Ferromagnetic	Low-carbon steel	$\mu_r = 5250$ (max.)	$\mu_r = 4900$ (max.)	McInturff and Claus (1970a)
	Si steel (1.8% Si)	$\mu_r = 6250$ (max.)	$\mu_r = 9700$ (max.)	McInturff and Claus (1970a)
Ferrimagnetic	Fe <sub>3</sub> O <sub>4</sub>	$\mu_r \geq 5.3$	$\mu_r \geq 5.5$	Jacobs (1959) – 4 K; Nagata (1961) – RT Assumes saturation at 0.11 MA·m <sup>-1</sup> (1.4 kOe)
Antiferromagnetic	Fe-22Cr-13Ni-5Mn stainless steel	Paramagnetic	$\kappa = 6.0 \times 10^{-3}$	Ledbetter and Collings (1979)

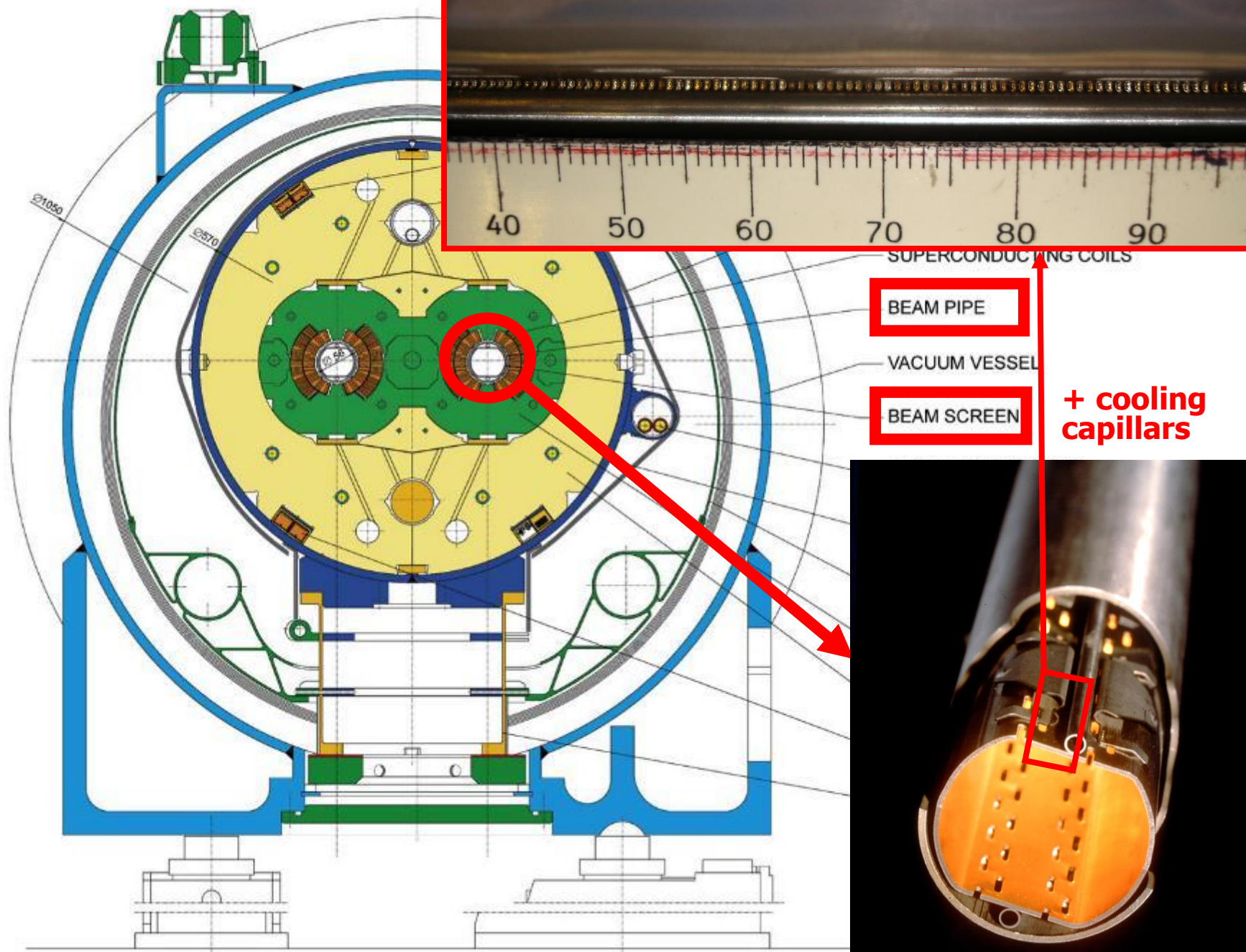
<sup>a</sup>Si units  
<sup>b</sup>Cr<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub>·K<sub>2</sub>SO<sub>4</sub>·24H<sub>2</sub>O  
<sup>c</sup>T = 14 K

<sup>a</sup>Si units  
<sup>b</sup> $\text{Cr}_2(\text{SO}_4)_3 \cdot \text{K}_2\text{SO}_4 \cdot 24\text{H}_2\text{O}$   
<sup>c</sup> $T = 14 \text{ K}$

R.P. Reed, A.F. Clark (1983)

# LHC DIPOLE : STANDARD

CERN AC/DI/MM - HE107 - 30 04 1999



## Materials for magnets: structural, cryogenics, vacuum: the LHC dipoles example

S. Sgobba, G. Hochoertler, A new non-magnetic stainless steel for very low temperature applications, Proc. Int. Congress Stainless Steel '99 : Science and Market, pp. 391-401

- Magnetic permeability 1.005 max in the whole T range (4.2 K to RT)
- From electroslag remelted ingots
- Rolled strip, forged bars ⇒ extruded + drawn pipes
- Laser welding

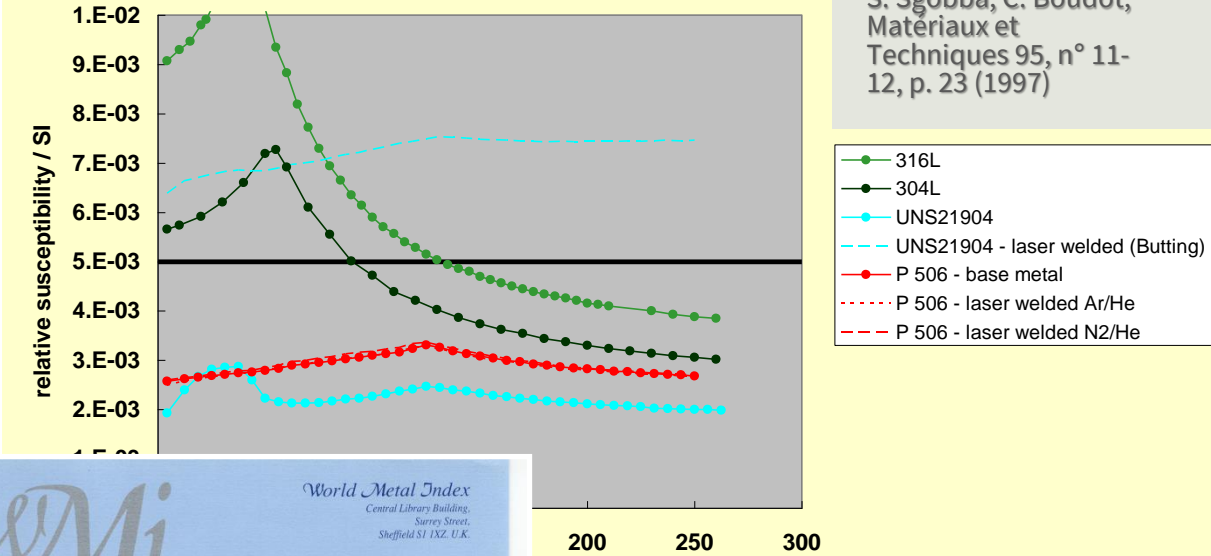


# Development of a new stainless steel (“P506”) for the LHC beam screen and the cooling capillaries

## Challenging scope:

- Magnetic susceptibility  $\leq 5 \cdot 10^{-3}$  at operating T in weld and parent material
- Fully stable
- Absence of hot cracking
- High strength and toughness (> 200 J)
- Thermal contraction not far from 316LN
- Corrosion behaviour  $\approx$  304L, 316LN
- Affordable price (several tens of km...)

Compared magnetic susceptibility of different austenitic SS and their laser weldments



S. Sgobba, C. Boudot, *Matériaux et Techniques* 95, n° 11-12, p. 23 (1997)

ADVANCED MATERIALS & PROCESSES/MARCH 2000

NICKEL, VOL. 15, NO. 1, SEPTEMBER 1999

resistance to corrosion in reducing acid environments. Timetal 5111 Pd is fully corrosion resistant in concentrated geothermal brine at 230°C (445°F) and pH 2 to 3, both with and without the addition of 1000 ppm of ferric ions. Applications include marine environments where toughness and corrosion resistance are essential.

For more information: Timet Corp., Henderson Technical Laboratory, 8000 Lake Mead Drive, Henderson, NV 89015; tel: 702/566-4403; Web site: [www.timet.com](http://www.timet.com). Circle 129

**A stainless steel that retains its non-magnetic properties even at very low temperatures has been developed in a joint project by Bohler Edelstahl GmbH, Austria, and CERN, Geneva, Switzerland. The alloy, designated P506, was developed specifically for a component of the Large Hadron Collider (LHC) under construction at CERN, the European Laboratory for Particle Physics. The alloy was de-**

signed for the “beam screen,” a tube that is to be 15 meters (50 ft) long and 44 mm (1.7 in.) in diameter, with a wall thickness of one millimeter. The screen is to be continuously formed and laser welded.

“The stainless steel is readily weldable, not susceptible to hot cracking, and has high tensile properties, ductility, and toughness down to 4.2 kelvins (-452°F, -269°C),” says CERN materials engineer Stefano Sgobba.

The screen must be totally non-magnetic, to prevent magnetic field distortion, and will be cooled to 10 to 20 kelvins (-442 to -423°F, -263 to -253°C).

For more information: Stefano Sgobba, CERN, 1211 Geneva 23, Switzerland; tel: 41 2276-79401; e-mail: [stefano.sgobba@cern.ch](mailto:stefano.sgobba@cern.ch); Web site: [www.cern.ch](http://www.cern.ch).

**Aluminum/H<sub>2</sub>O<sub>2</sub> fuel cells generate 20 times more power**

A fuel cell that produces electricity through chemical reactions between hydrogen peroxide and aluminum is

### Exploring the Unknown

A new stainless steel is developed for Europe's Hadron collider

Bohler Edelstahl GmbH of Austria and CERN in Geneva, Switzerland, have jointly developed a new stainless steel that stays non-magnetic even at very low temperatures and maintains its magnetic properties after welding. Stainless steel P506 contains 1% nickel and 12% manganese and was developed specifically for the “beam screen,” a component of the Large Hadron Collider (LHC) being constructed at CERN's European Laboratory for Particle Physics.

When it is switched on in 2005, the LHC will slam high-intensity proton beams together at energies previously impossible, thereby enabling scientists to observe, for the first time, phenomena that occurred in the early universe, just after the “Big Bang.”

The beam screen will consist of a one-millimeter-thick, 15-meter-long perforated tube, measuring 44 millimeters in diameter. The tube will shield the vacuum chamber (in which superconducting magnets are housed) from the synchrotron radiation emitted by the circulating proton beams. The screen must be totally non-magnetic, thereby preventing magnetic field distortion. In addition, to keep a thermal equilibrium on the screen, the screen will be cooled to 10-20° Kelvin by gaseous helium. The screen will be continuously formed and laser-welded.

Tests have shown that the new stainless steel has a low value of magnetic permeability of less than 1,000, which is maintained over a temperature range of 4.2° Kelvin to room temperature.

“The stainless steel is readily weldable, is not susceptible to hot cracking, and has high tensile properties, ductility and toughness down to 4.2° Kelvin,” says Stefano Sgobba, a materials engineer for the European Laboratory for Particle Physics.

These properties together with high resistance to corrosion, should prove useful in other low-temperature applications.



representing magnets from synchrotron radiation

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The work is expected to keep solid-state physicists busy for years to come.

with EBC [www.nature.com](http://www.nature.com)

### Quantum-Mechanical Spinning

Three solid-state physicists (two at Uppsala University in Sweden and one at the Sandia National Laboratories in Livermore, California) are offering a quantum-mechanical explanation for why brass “twins” a class of magnetic nickel alloys (P50600, for example) containing 20% nickel and 80% iron, exhibits almost no thermal expansion over a broad range of temperatures.

Due to this special behavior, laser alloys are used primarily in shadow masks for television and computer screens to prevent screen distortion (see page 30, Nickel, Vol. 13, No. 2, March 1998) and in other applications such as high-pressure instruments such as selenographs.

In the July 1, 1999, issue of *Nature*, Mark van Spittel, Ignor Alkhusen and Boris Johansson provide a microscopic explanation for the laser effect. For the first time, they showed noncollinear spin ordering in their quantum mechanical calculations of the electronic and magnetic structure of these metals. At a low temperature, the ground state is given by parallel ferromagnetic spin alignment. However, when the temperature increased (as indicated by a decrease in volume), they showed that spins gradually depart from parallel alignment, so that the spin directions become increasingly disordered, with lower local moment amplitudes.

For laser alloys, this decrease in volume compensates for the vibrational thermal expansion. The work is expected to keep solid-state physicists busy for years to come.

with EBC [www.nature.com](http://www.nature.com)

Australian Meat Processors

Dr. David Johnston, Director of NCI Australia, will present a paper on the technical aspects of stainless steel in the meat processing industry at The Country Meatworkers Association Annual Conference which is being held in Australia, October 21-23, 1999. The paper will cover alloy selection, fabrication/welding practice and the application of stainless in the handling of hot high chloride waters. The meat industry is a large user of nickel stainless steel, particularly in salubrious or slaughter houses.

with EBC [www.nature.com](http://www.nature.com)

**World Metal Index**  
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Survey Street,  
Sheffield S1 1XZ, U.K.

Telephone: +44 (0)114 273 4714 / 273 4744  
Fax: +44 (0)114 275 7405

WMIAEB  
S.S. 60

For more information: Stefano Sgobba, CERN, 1211 Geneva 23, Switzerland.

Dear Sirs,

**WORLD METAL INDEX**

The World Metal Index (WMI) is an information service which provides users with the identification of metal grade, trade name or alpha-numeric reference number. It has developed over a period of more than thirty years in response to enquiries from companies; its coverage is world-wide. It contains references to both ferrous and non-ferrous materials.

The index itself contains information related to over 200,000 grades of metal. Information is backed by files of company trade catalogues and data compilations in addition to the extensive standards collection Business and Technology Library. The service is staffed by expert personnel and is strictly confidential.

Appearance of your products in our index will advertise them to users.

In order to keep our collection of trade literature up to date, I would supply me with:

technical information of the following grades of metal

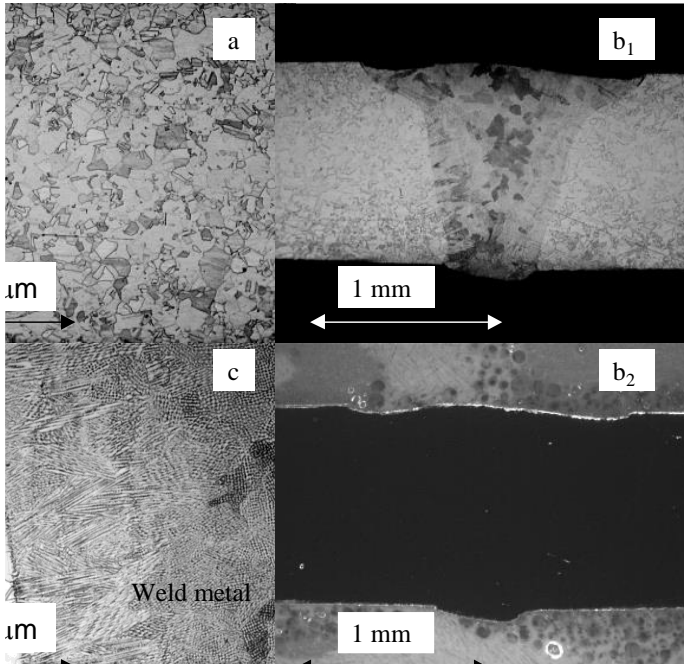
as advertised in

Yours faithfully

*Stefano Sgobba*  
Stefano Sgobba  
Development Officer  
World Metal Index

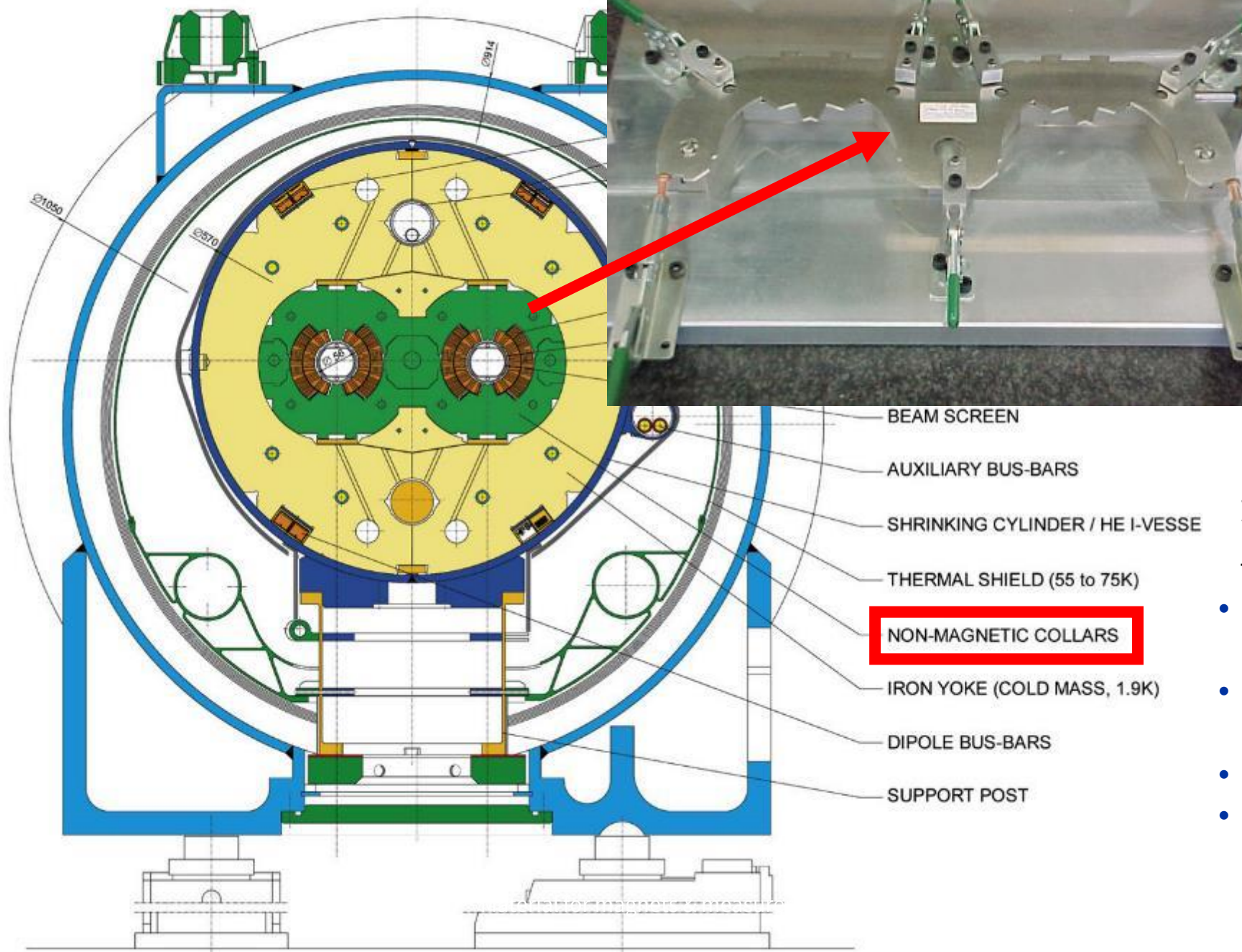
ADVANCED MATERIALS & PROCESSES/MARCH 2000  
p.12

H/WMI/Grade



NICKEL, VOL. 15, NO. 1, SEPTEMBER 1999





BEAM SCREEN

AUXILIARY BUS-BARS

SHRINKING CYLINDER / HE I-VESSE

THERMAL SHIELD (55 to 75K)

NON-MAGNETIC COLLARS

IRON YOKE (COLD MASS, 1.9K)

DIPOLE BUS-BARS

SUPPORT POST

## Materials for magnets

Table I. Chemical composition of melts (% weight)

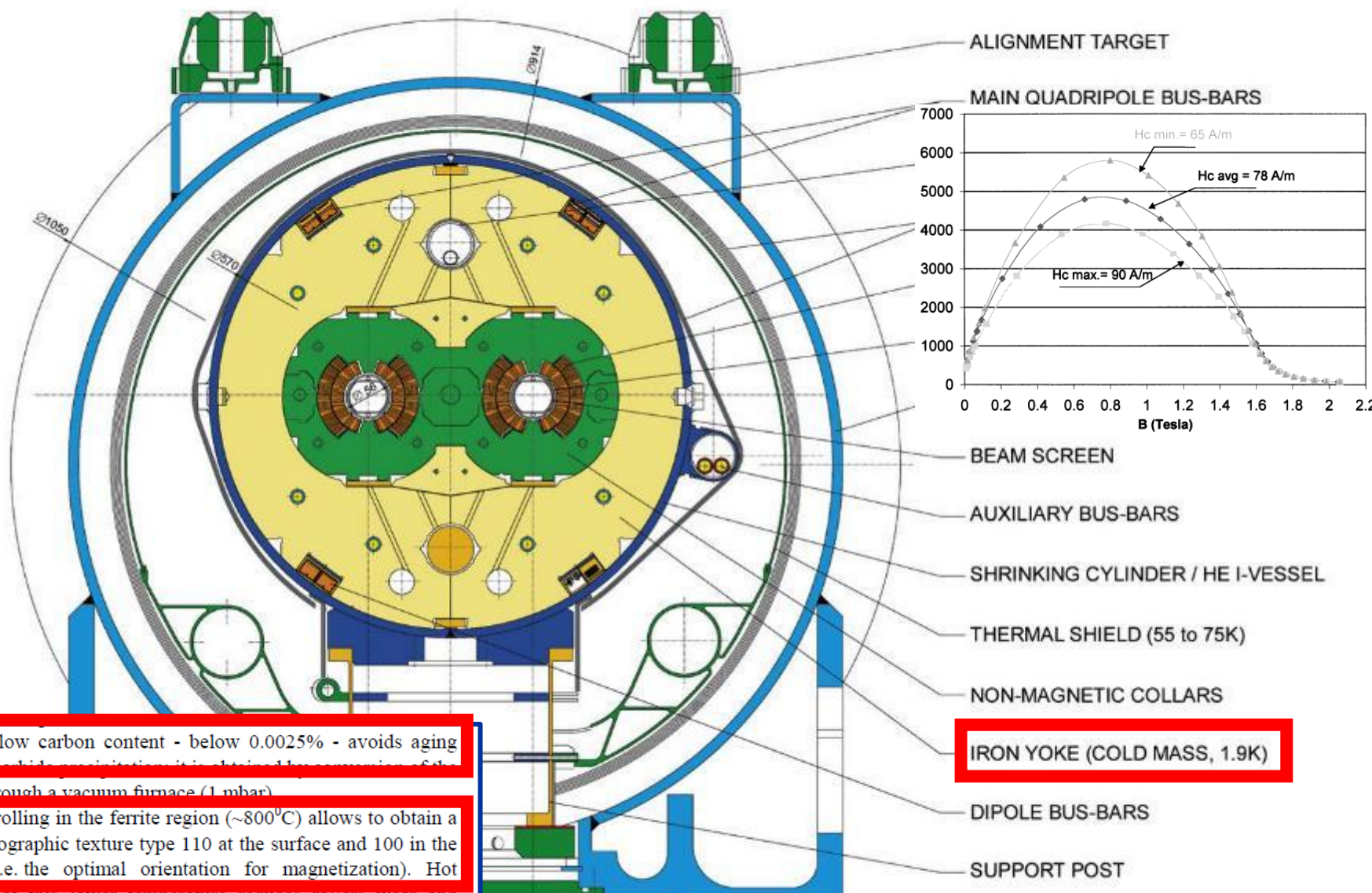
C	Mn	Ni	Cr	Mo	Si	N
0.086	11.5	6.61	17.77	0.09	0.42	0.30
( $\sigma=0.005$ )	( $\sigma=0.2$ )	( $\sigma=0.06$ )	( $\sigma=0.07$ )	( $\sigma=0.02$ )	( $\sigma=0.04$ )	( $\sigma=0.01$ )

- Stainless steel “Nitronic 40” type (YUS 130S ~ UNS 21904)
- In kind contribution from Japan (Nippon Steel)
- Rolled strip, 11000 t
- Twelve million collars



# LHC DIPOLE : STANDARD CROSS-SECTION

CERN AC/DI/MM - HE107 - 30 04 1999



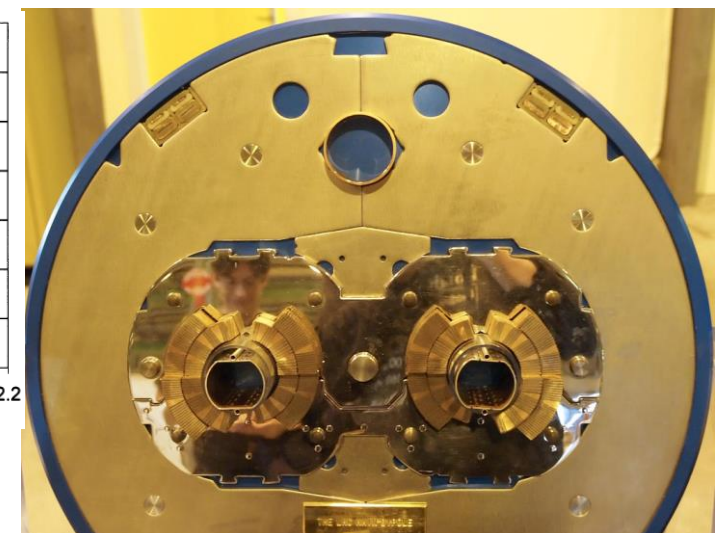
The low carbon content - below 0.0025% - avoids aging due to precipitation, it is obtained by remelting of the melt through a vacuum furnace (1 mbar).

Hot rolling in the ferrite region ( $\sim 800^\circ\text{C}$ ) allows to obtain a crystallographic texture type 110 at the surface and 100 in the core (i.e. the optimal orientation for magnetization). Hot rolling at this lower temperature reduces rollers wear and fabrication costs, improves steel surface quality and flatness, and allows the formation of a stable, protective layer of mill scale.

During the annealing heat treatment, the energy stored in the steel with hot rolling allows grain size growth.

F. Bertinelli et al., "Production of Low-Carbon Magnetic Steel for the LHC Superconducting Dipole and Quadrupole Magnets," in IEEE Transactions on Applied Superconductivity, vol. 16, no. 2, pp. 1777-1781, June 2006, doi: 10.1109/TASC.2006.873236

## Materials for magnets



S. Babic et al., Toward the Production of 50 000 Tonnes of Low-Carbon Steel Sheet for the LHC Superconducting Dipole and Quadrupole Magnets, <https://doi.org/10.1109/TASC.2002.1018621>

- Ultra Low Carbon steel "Magnetil" type
- Rolled strip, 50000 t

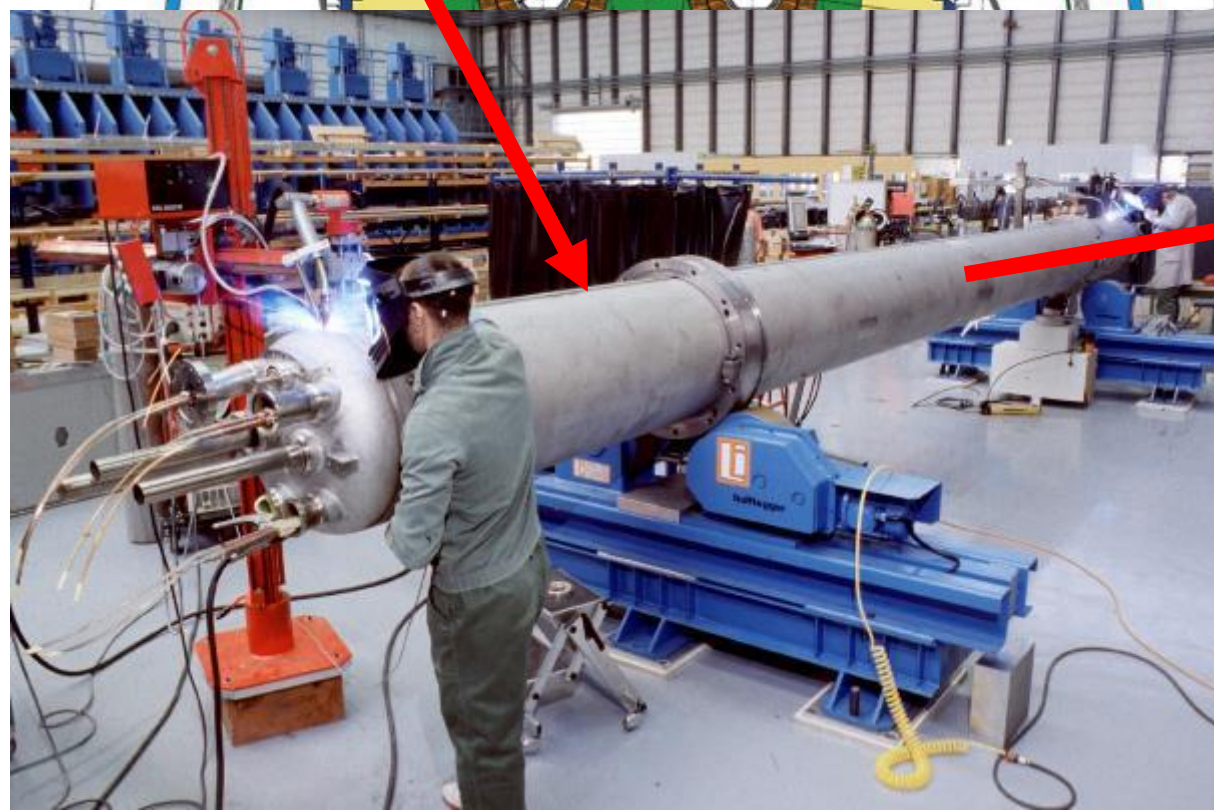


# LHC DIPOLE : STANDARD CROSS-

CERN AC/DI/MM - HE107 - 30 04 1999

Courtesy Dr. N. Pauze, ArcelorMittal Industeel

## Materials for magnets: structural, cryogenics, vacuum: the LHC dipoles example



BEAM SCREEN  
AUXILIARY BUS-BARS  
**SHRINKING CYLINDER / HE I-VESSEL**  
THERMAL SHIELD (55 to 75K)  
NON-MAGNETIC COLLARS  
IRON YOKE (COLD MASS, 1.9K)  
DIPOLE BUS-BARS  
SUPPORT POST

- 3000 tonnes of 1.4429 (316LN) stainless steel plates delivered by ArcelorMittal Industeel /FR
- 41 km of half-shells, Ø 550 mm, 15 m length, 10.1 mm thick produced by Butting /DE
- **Ingot casting ⇒ Forged slabs ⇒ rolled plates**
- **MIG/MAG welding (STT) – special cryogenic filler**





After capsule removal by pickling and heat treatment, before machining



### PM Design Competition Award Winner

Grand Prize, reception at the PM Part Competition at PowderMet 2007, The International Conference on Powder Metallurgy and Particulate Materials, Denver-CO, May 13-16, 2007

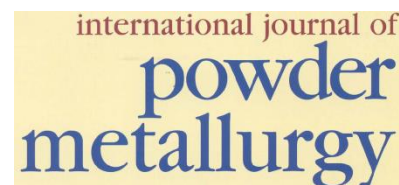
Award: Grand Prize  
Year: 2007



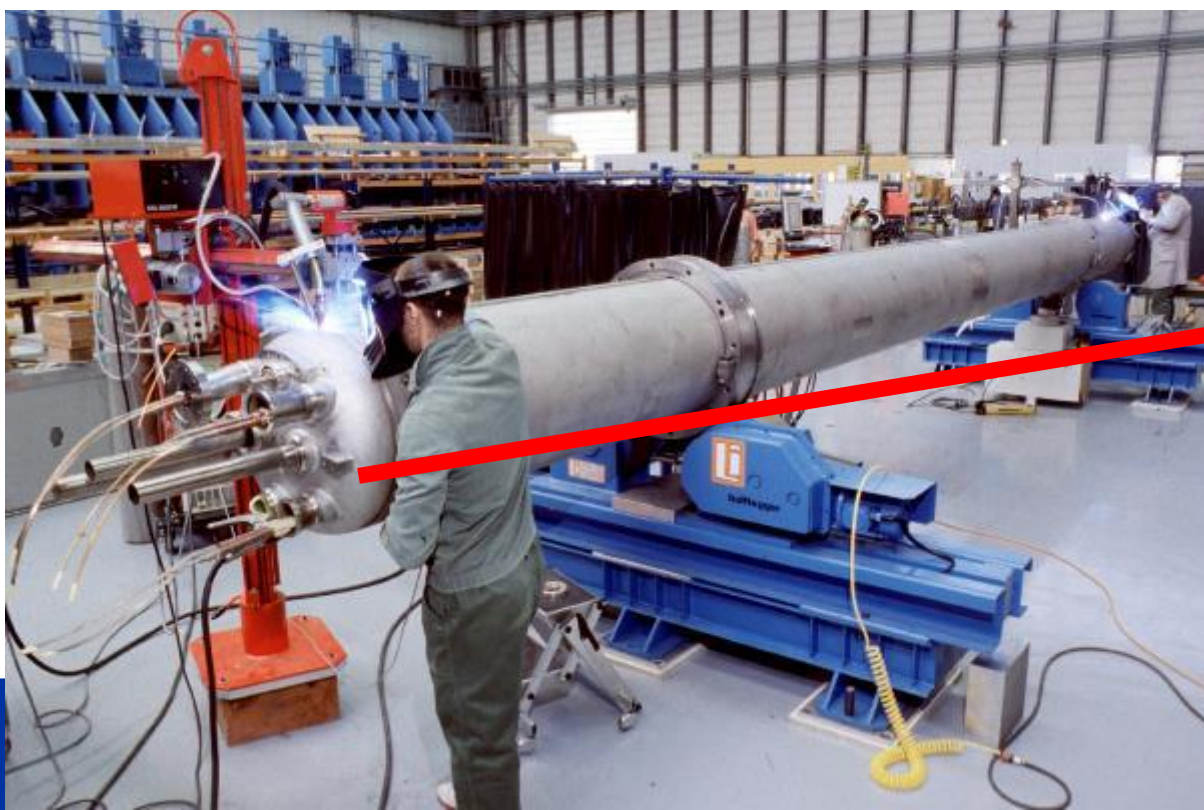
Award of the American Society for Metals (ASM),  
30/01/2004



Award of the American Society for Metals (ASM International, Finland) for Outstanding Achievements in Powder Metallurgy, to the paper "Powder HIP End Covers for CERN LHC Project, High Technology for Low Temperatures", 6th Annual Powder Metallurgy Network Seminar, Engineering Solutions Based on Powder Metallurgy, Tampere 29th - 30th January, 2004



Metso Corporation 2003



SHRINKING CYLINDER

HE I-VESSEL

THERMAL SHOCK

end covers

NON-MAGNETIC COLLARS

IRON YOKE (COLD MASS, 1.9K)

DIPOLE BUS-BARS

SUPPORT POST

S. Sgobba et al.  
Stainless Steel  
Temperatures  
Metallurgy W  
Kyoto, Japan, vol. 2, p. 1002-1005

- Leak tight to gaseous He at 300 K
- Leak tight to superfluid He at 1.9 K
- 25 thermal cycles down to 1.9 K

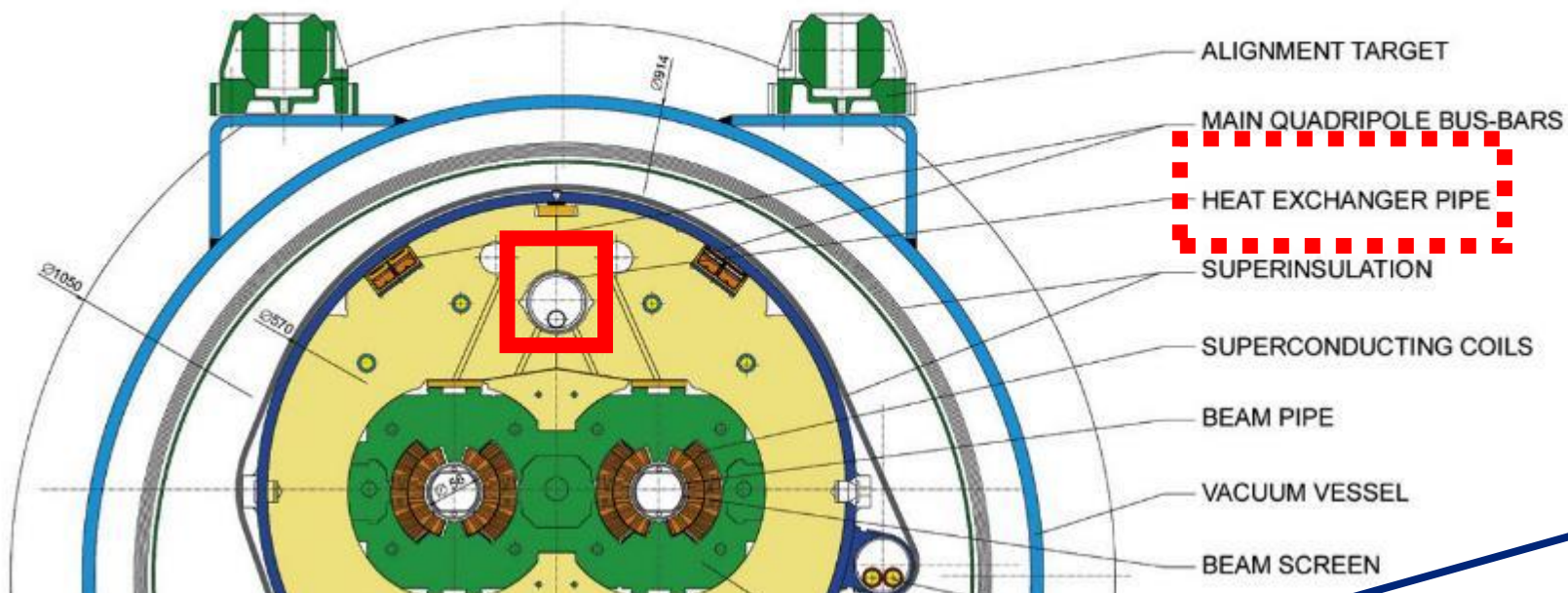
- High ductility and toughness at low T
- Compatible with its environment (316LN)
- Cost effective

- Powder metallurgy + HIPing
- MIG/MAG & TIG welding

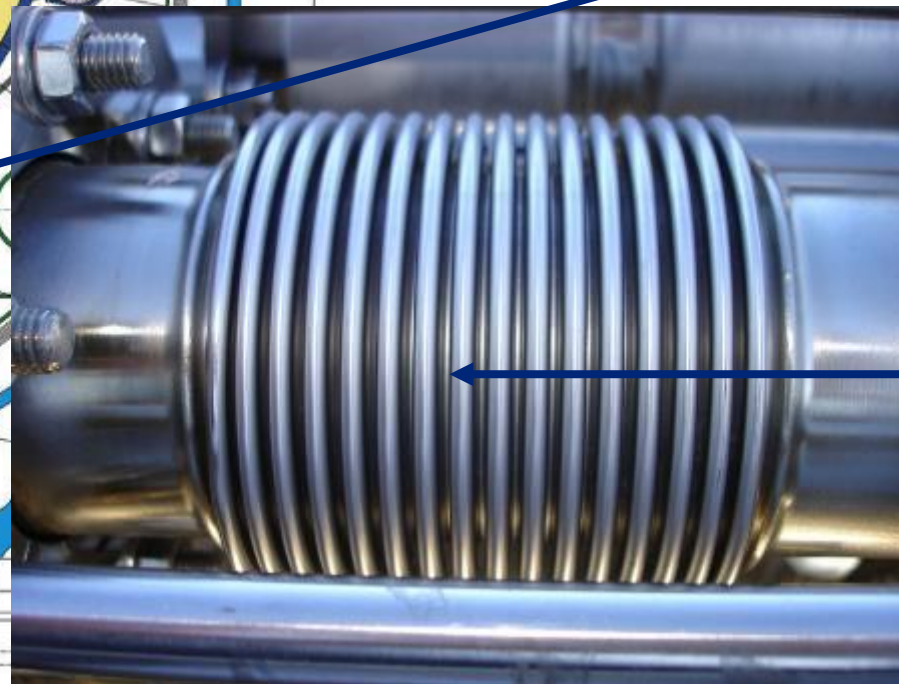


# LHC DIPOLE : STANDARD CROSS-SECTION

## Materials for magnets...



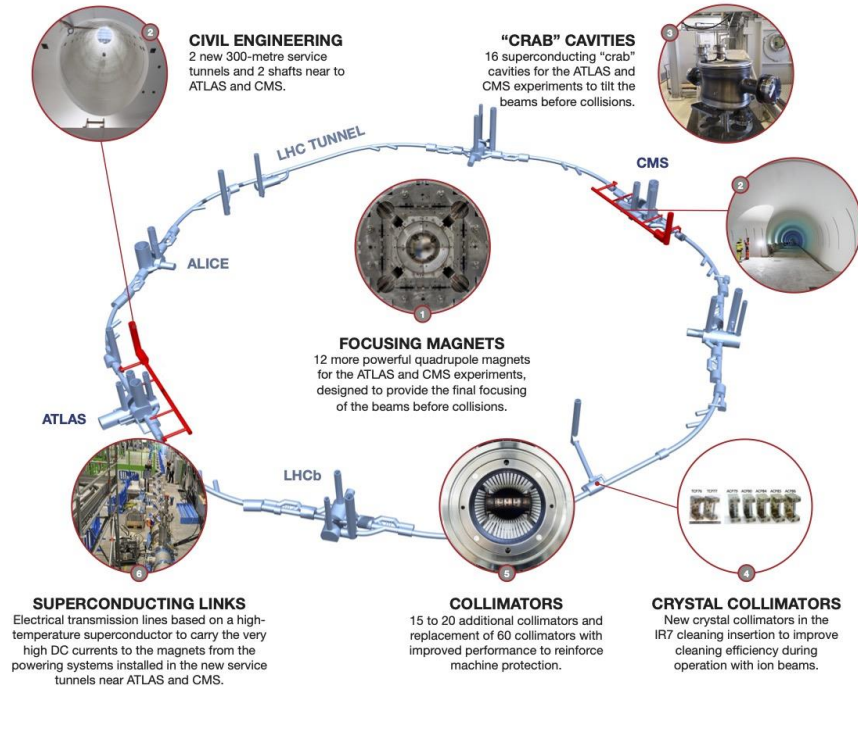
- Multidirectionally forged 316LN (1.4429) flanges/sleeves
- From electroslag remelted ingots



- Special 316L (1.4435, high Ni+Cr), from electroslag remelted ingots
- Strip
- Hydroformed bellows



## NEW TECHNOLOGIES FOR THE HIGH-LUMINOSITY LHC



# Materials for magnets: HL-LHC

## Fine-blanked collars

F712 contract

- More than 450 tonnes of austenitic stainless steel strips
- Same stainless steel specification as for the LHC

## Beam screen

HL-LHC implies:

- 3.1 km of finished strip;
- 4600 m of seamless cold-drawn cooling tubes in lengths of up to 14 m
- Same stainless steel as for the LHC

IT-4203/TE/HL-LHC

### 3.1.1 Special austenitic grade stainless steel strip (CERN supply)

The chemical composition of the CERN supplied stainless steel strip is given in table 2.

Table 2 - Typical chemical composition (weight-%) of the CERN supplied stainless steel strip.

%	C	Cr	Mo	Ni	Mn	Si	N	Cu	S	P	B	Co
Min		19.0	0.8	10.7	11.8		0.30					
Max	0.03	19.5	1.0	11.3	12.4	0.5	0.33	0.15	0.002	0.02	0.002	0.1



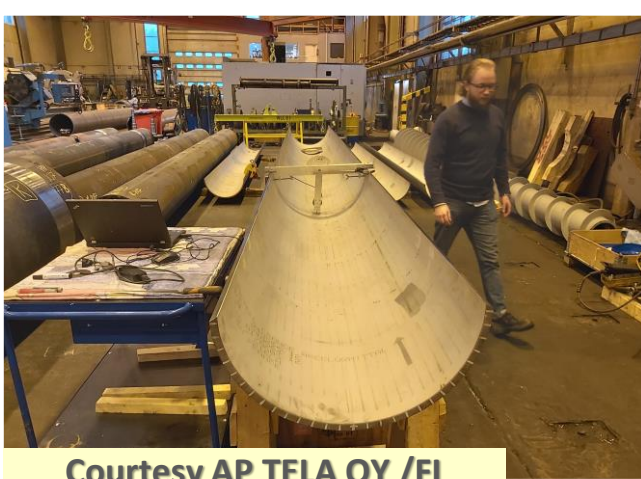
HL-LHC beam screens



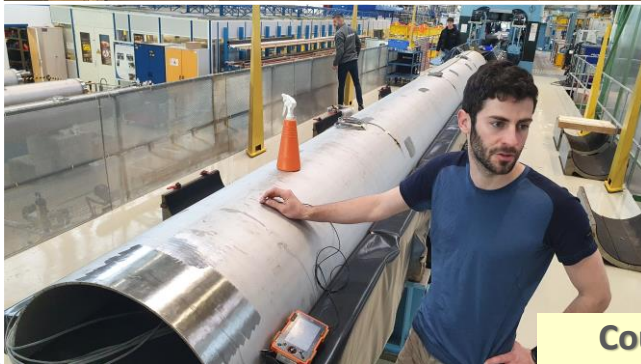
# Materials for magnets: HL-LHC

HL-LHC further 316LN plate production and forming (2015-16)

- Supplied by Arcelor Mittal Industeel
- 15 mm x 6.5 m (11T project)
- 8 mm x 6.5 m and 10 m for WP3 (Q2, CP et D2)



Courtesy AP TELA OY /FI



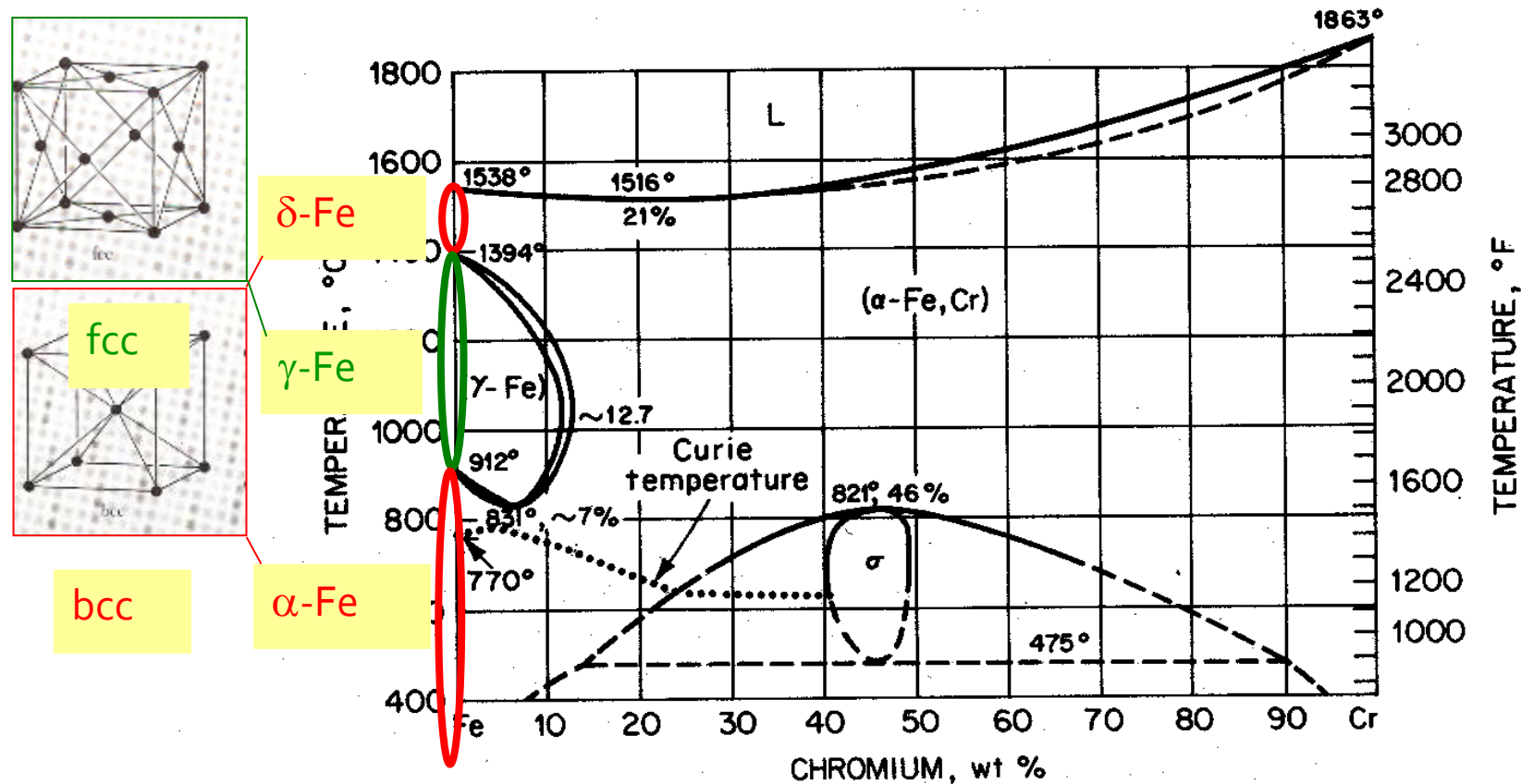
Courtesy H. Prin /CERN-TE



Shrinking  
and inertia  
cylinders



# Materials for magnets: stainless steels



**Fig. 2** The iron-chromium phase diagram. (From "Metals Handbook," vol. 8, p. 291, 8th ed., American Society for Metals, Metals Park, Ohio.)

**Stainless steel: iron alloys containing a minimum of approx. 11 % Cr**



# Materials for magnets: austenitic stainless steels

**AISI 304, the "18-8" or "18-10"  
stainless (18%Cr, 8-10%Ni)**

fork & knives

- form fcc FeC
- $\gamma$ -lo
- $\gamma$ -pl enl
- form sup
- form
- trans can sup allo

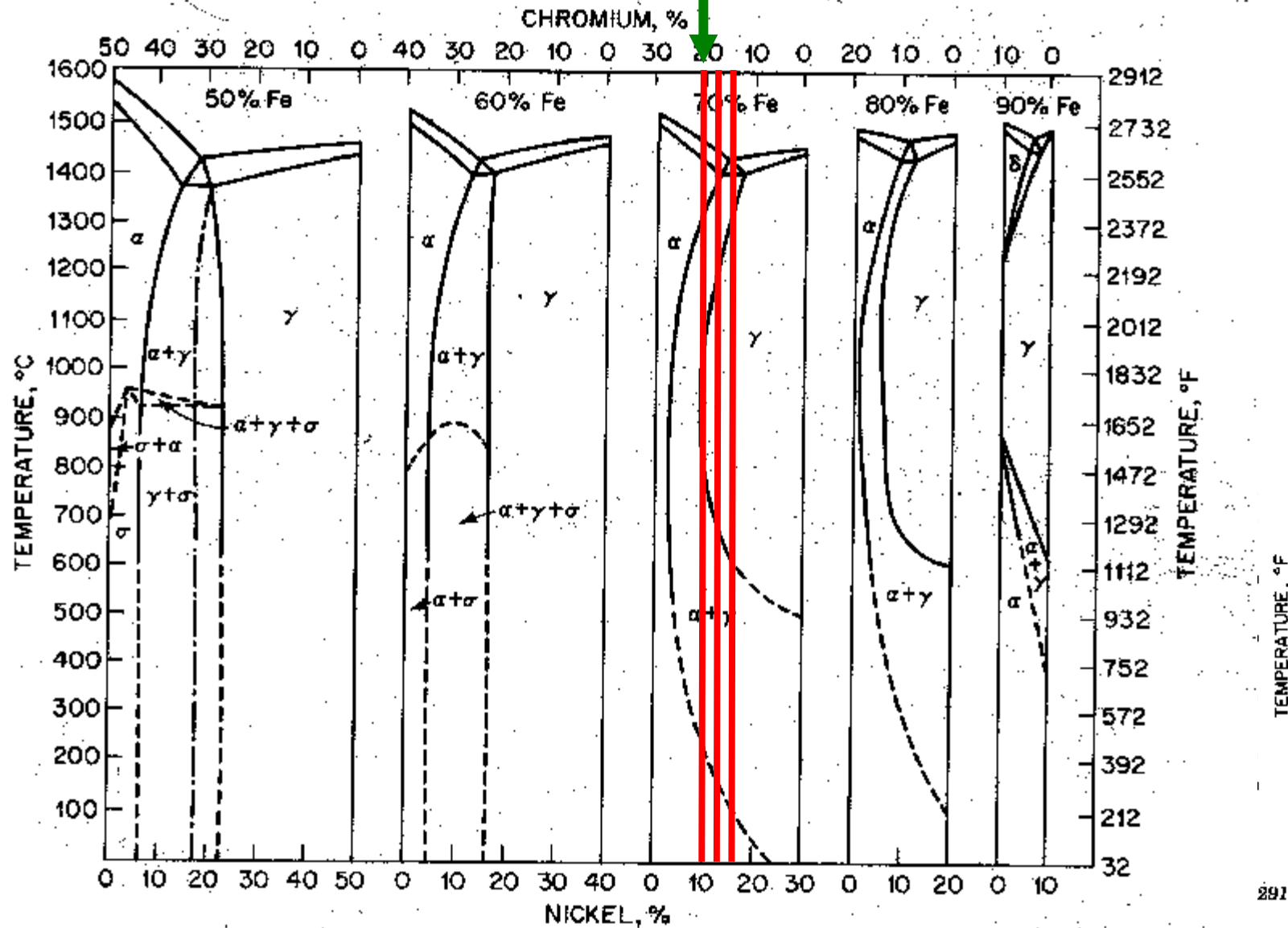


Fig. 14 Cross sections of Fe-Cr-Ni ternary.<sup>26</sup>



# Materials for magnets: austenitic stainless steels

**More:** <https://cds.cern.ch/record/2723993?ln=en>  
&:

Mechanical Materials Engineering

2 - 15 June 2024

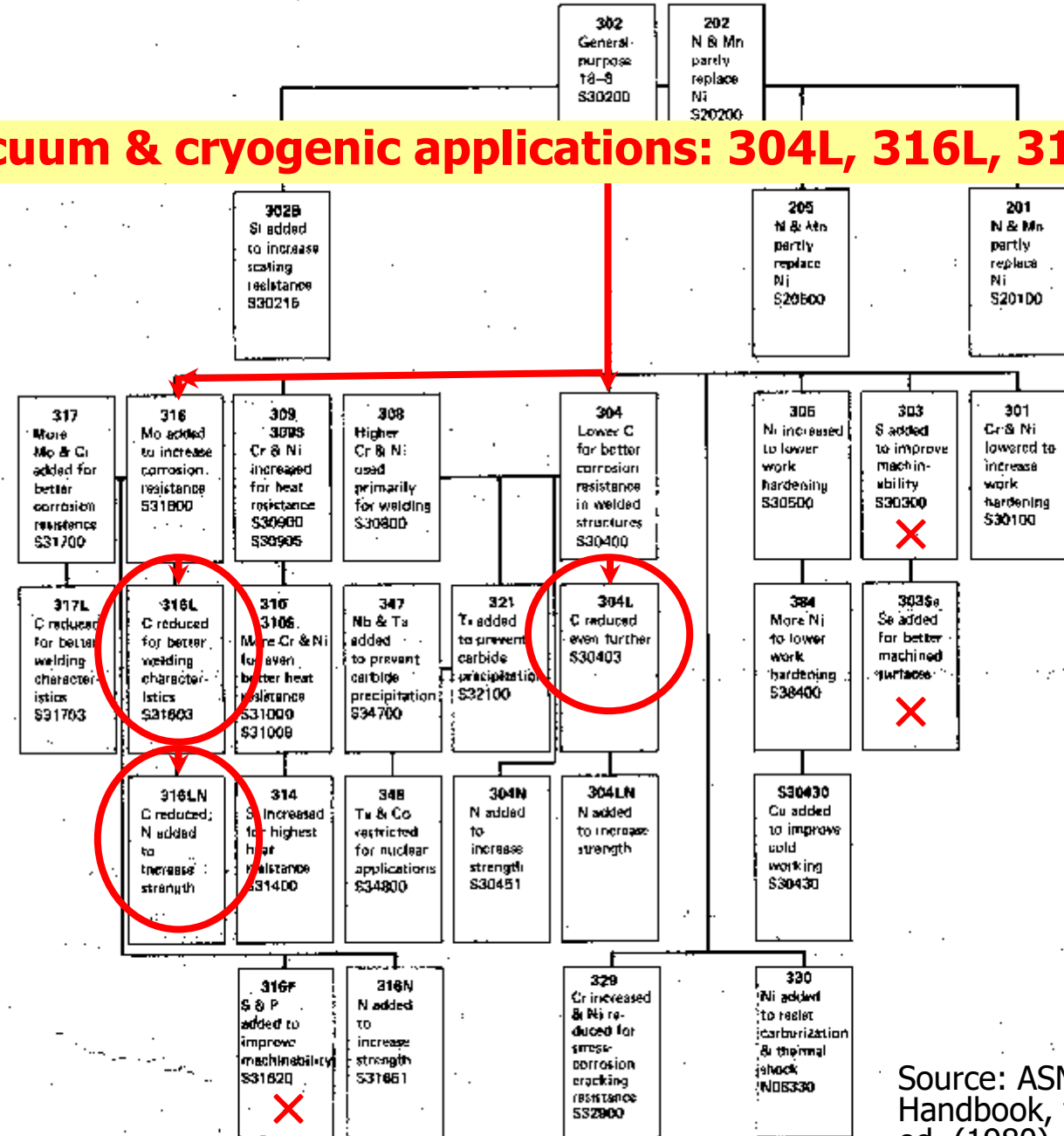
Sint-Michielsgestel, Netherlands

Registration will open soon



**Fig. 2 Family relationships for standard austenitic stainless steels**

**Vacuum & cryogenic applications: 304L, 316L, 316LN**



Source: ASM Metals  
Handbook, vol. 3, 9th  
ed. (1980)

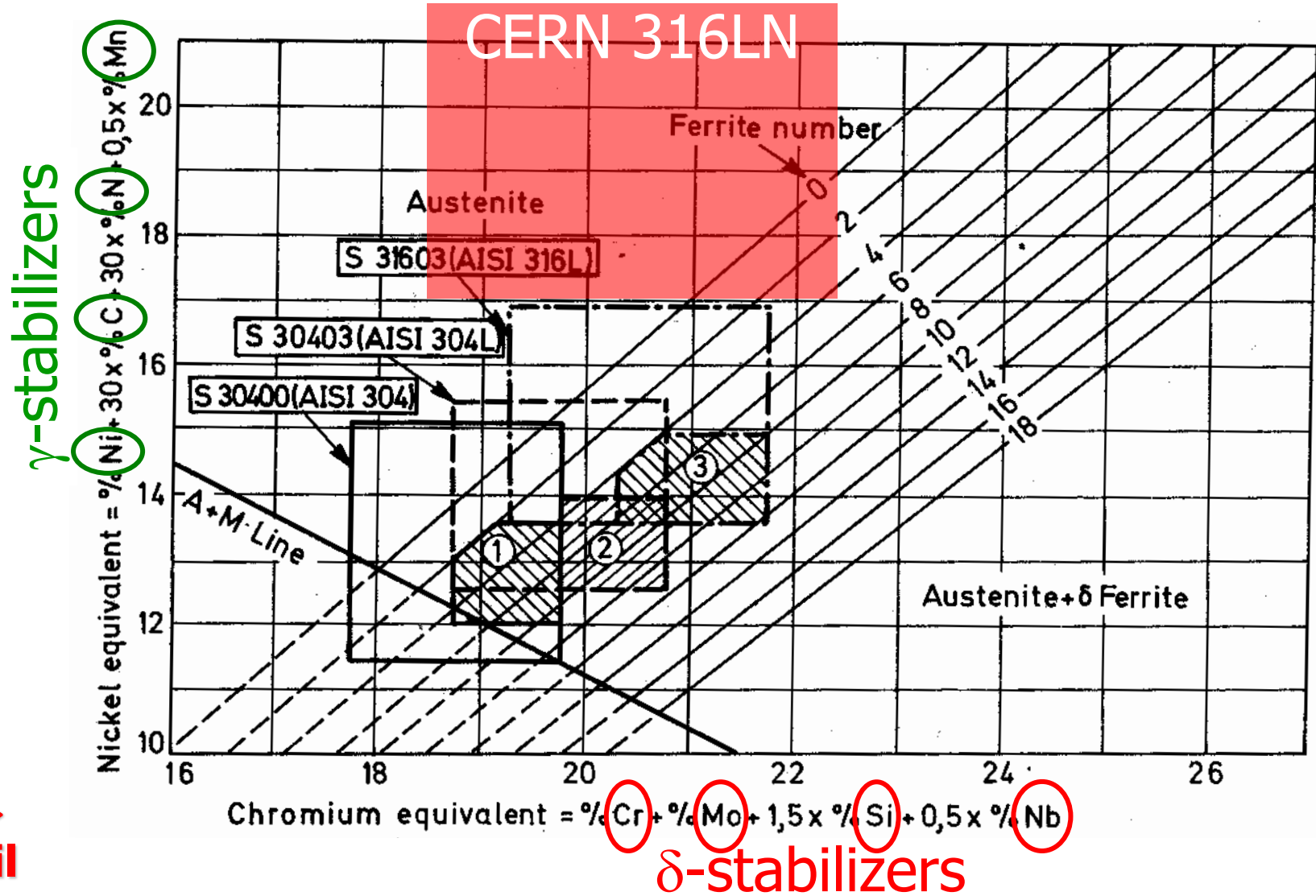


# Materials for magnets: austenitic stainless steels

CERN 316LN

Element	Chemical composition (product analysis) % by mass
Cr	16.00 – 18.50*
Ni	12.00 – 14.00*
C	0.030 max.
Si	1.00 max.
Mn	2.00 max.
Mo	2.00 – 3.00*
N	0.14 – 0.20*
P	0.030 max.*
S	0.010 max.*
Fe	Remainder

\* CERN requirement

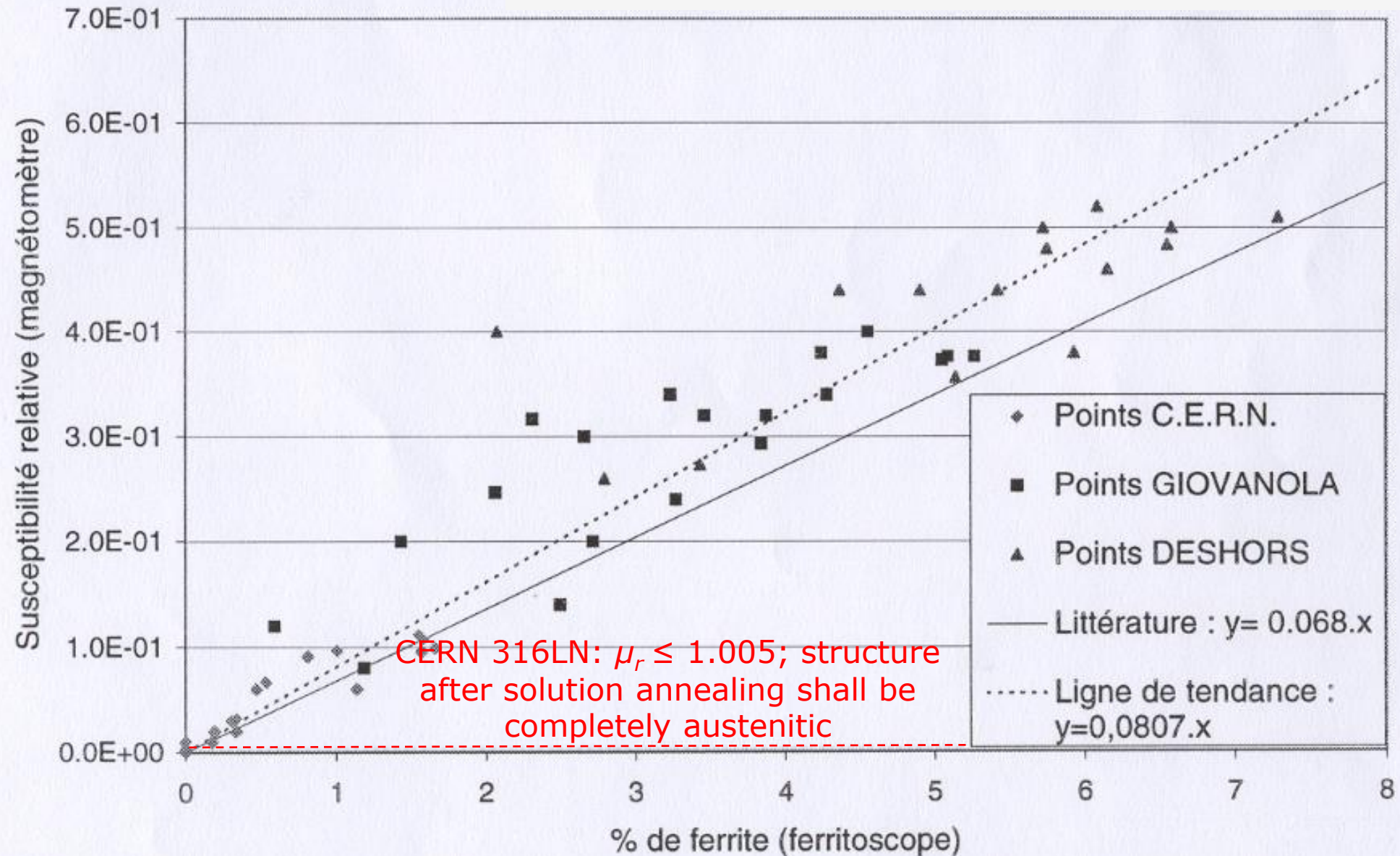


**Maximum allowed magnetic permeability  $\mu_r = 1.005$  at RT → allowed content of  $\delta$ -ferrite is nil**



# Materials for magnets: austenitic stainless steels

S. Sgobba and C. Boudot, Matériaux et Techniques 95, vol. 11-12, p. 23 (1997)





# Materials for magnets: austenitic stainless steels – ATLAS feet

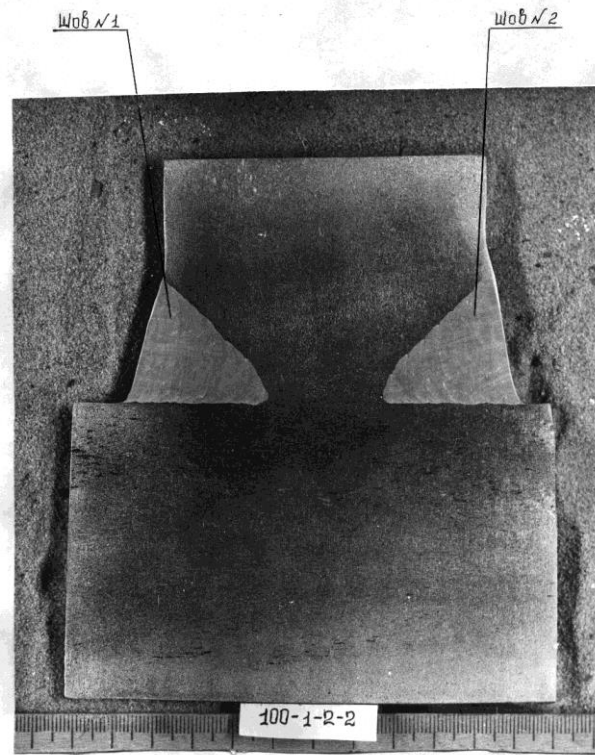
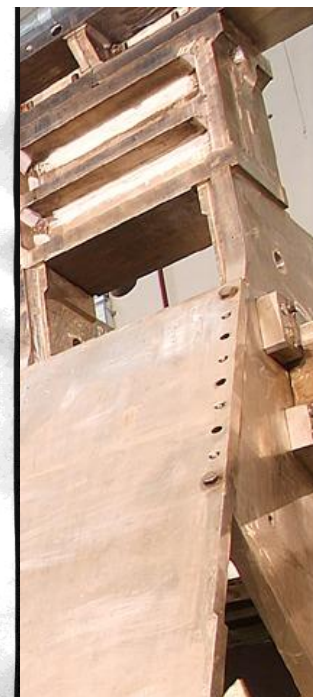
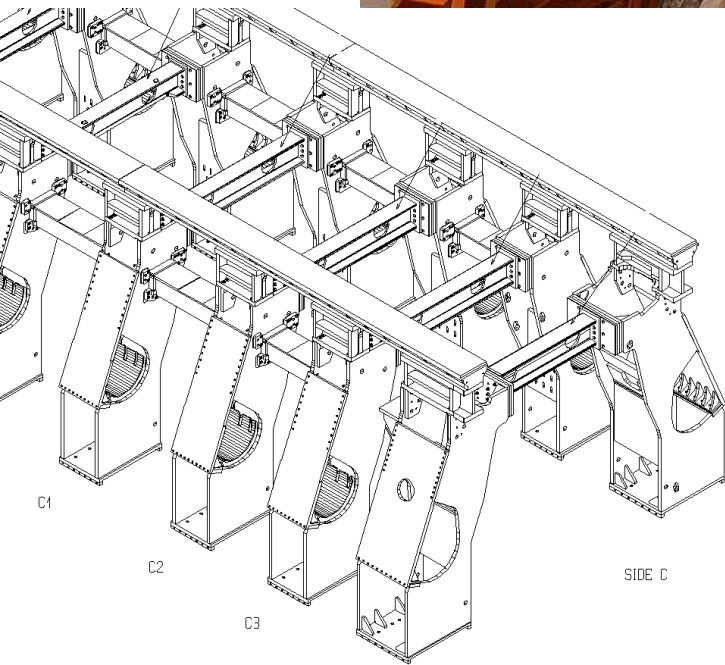


Рис.1. Макроструктура сварного соединения шерт.8002.41.25.100 СБ.

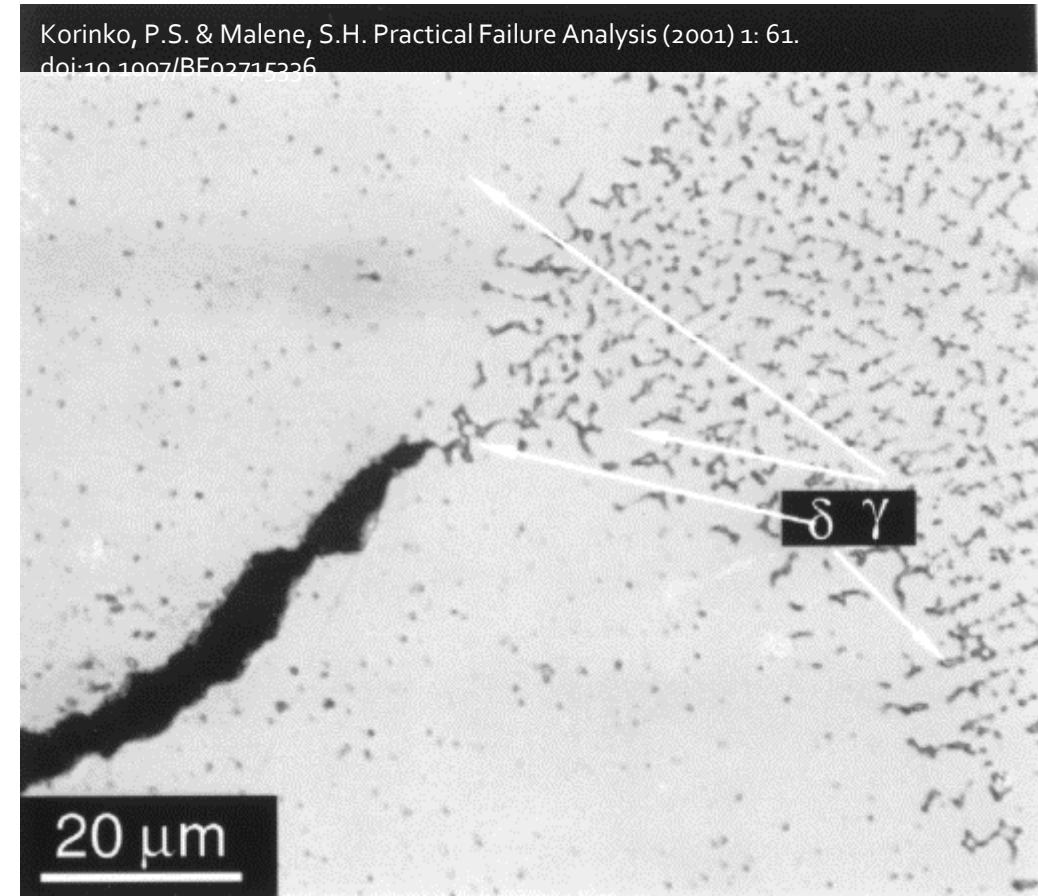
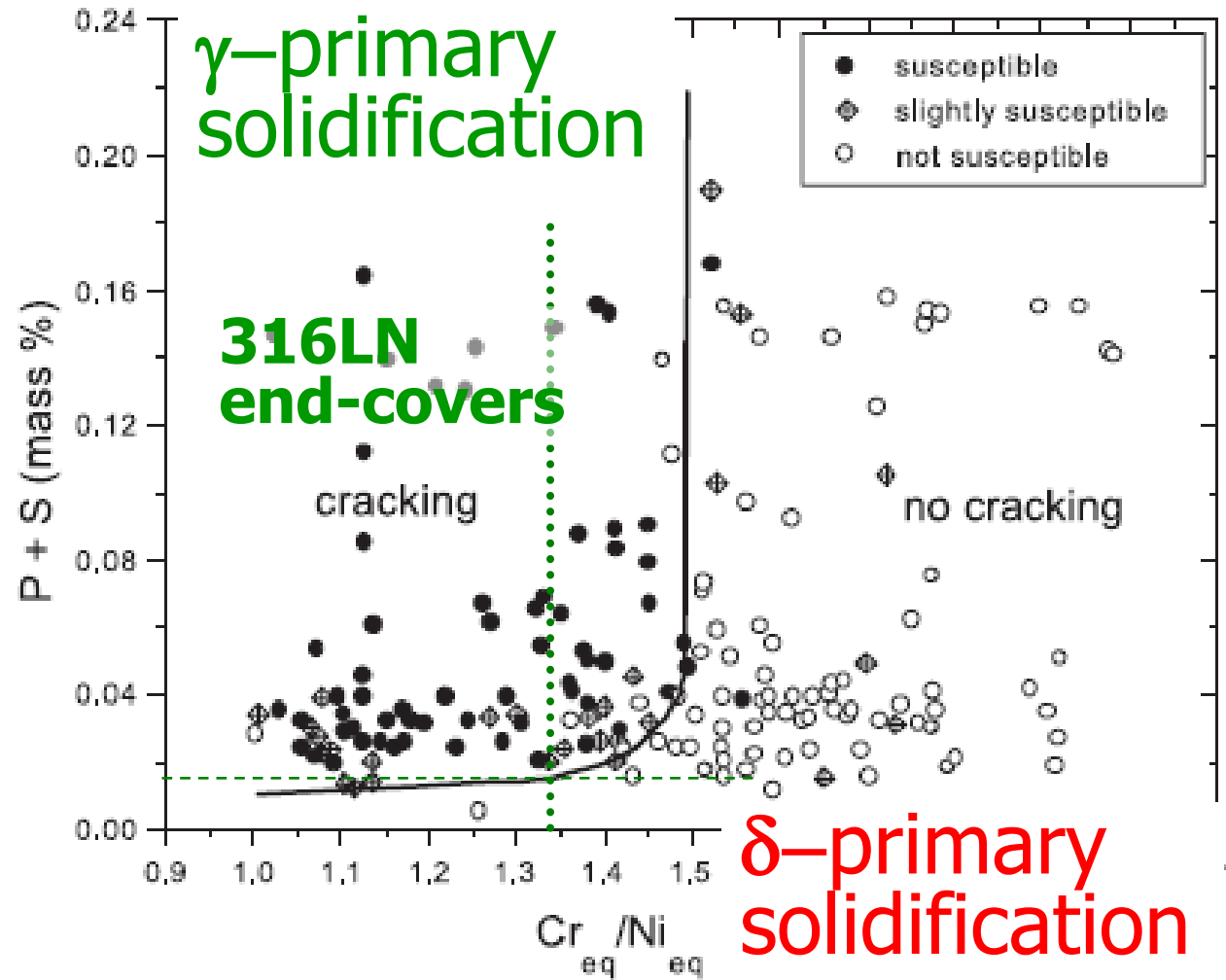
ОАО "ИЗ" Отчет №32/556 от 25.05.2001г. Лист 6 Листов 17



15 tons of filler material



# Materials for magnets: austenitic stainless steels



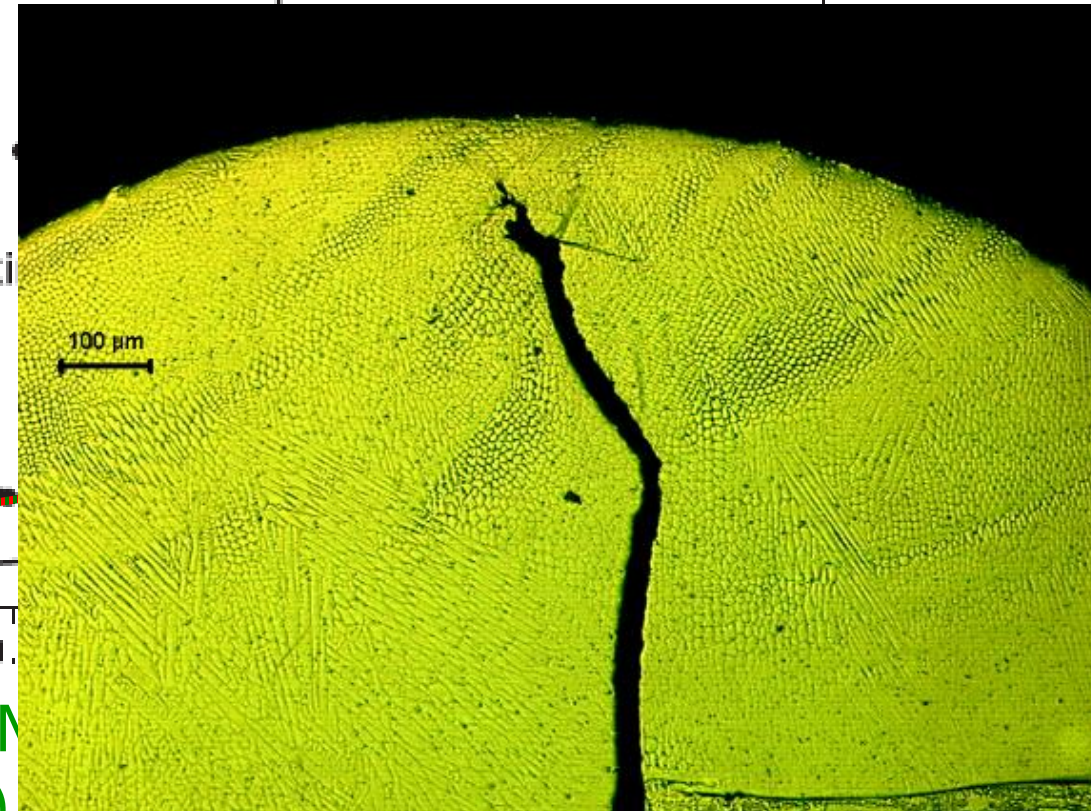
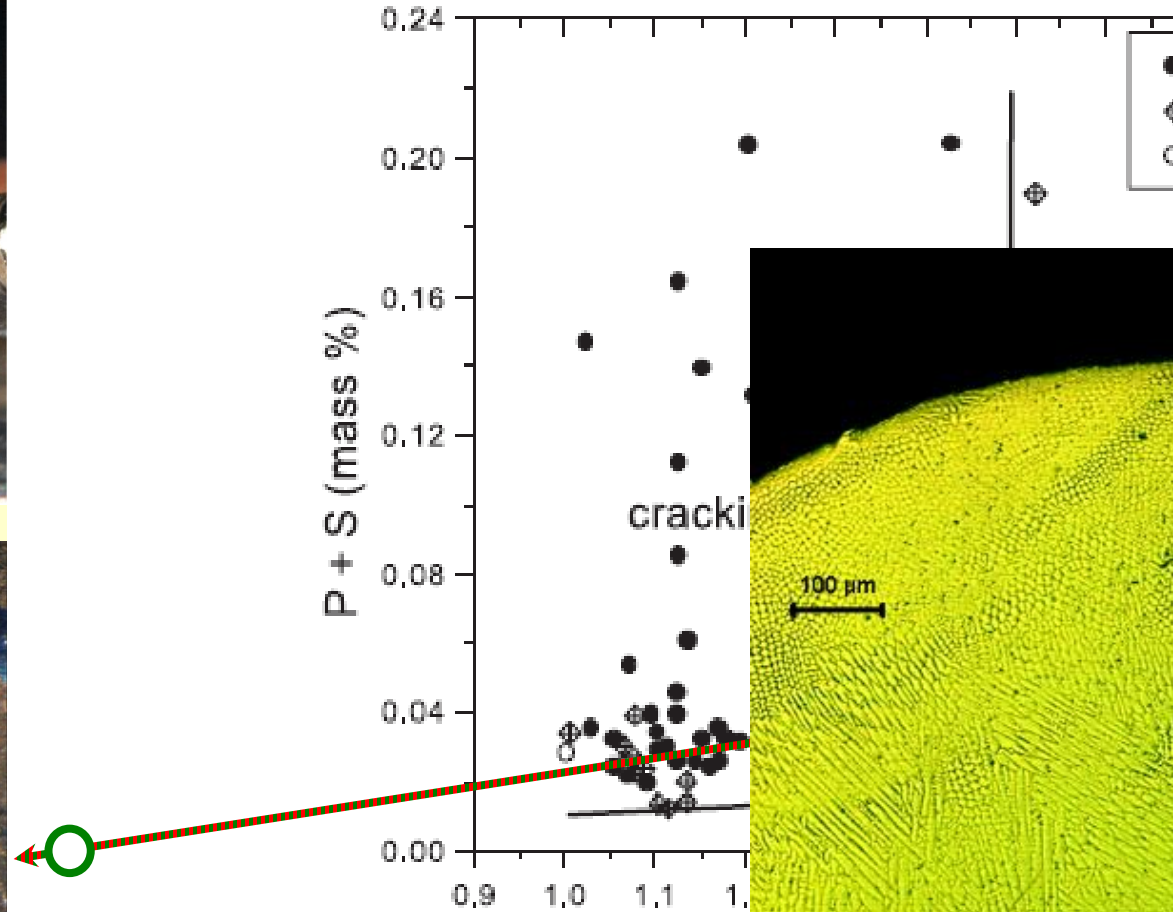
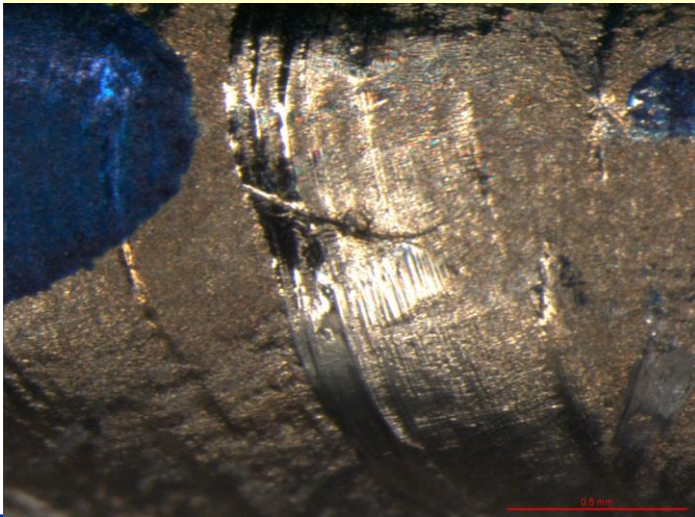
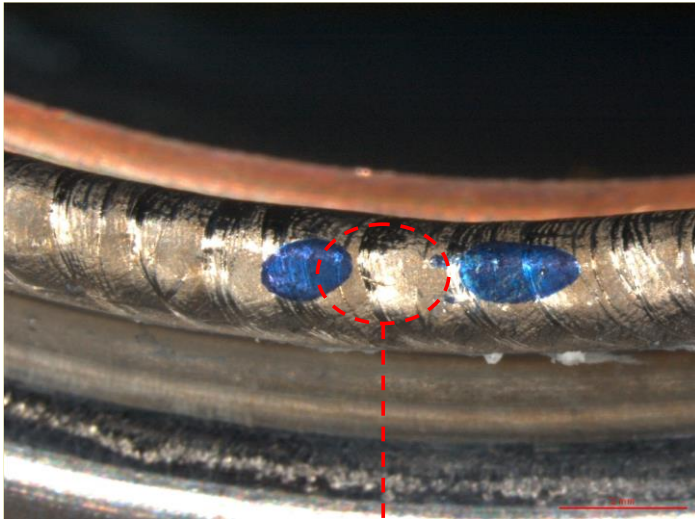
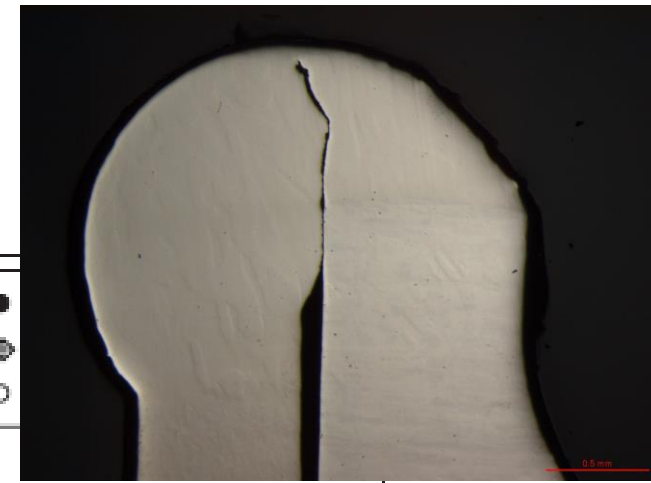
Schaeffler equivalent formulae for  $Cr_{eq}$  and  $Ni_{eq}$

$$Cr_{eq} = Cr + 1.5Si + 1.37Mo$$

$$Ni_{eq} = Ni + 0.31Mn + 22C + 14.2N$$

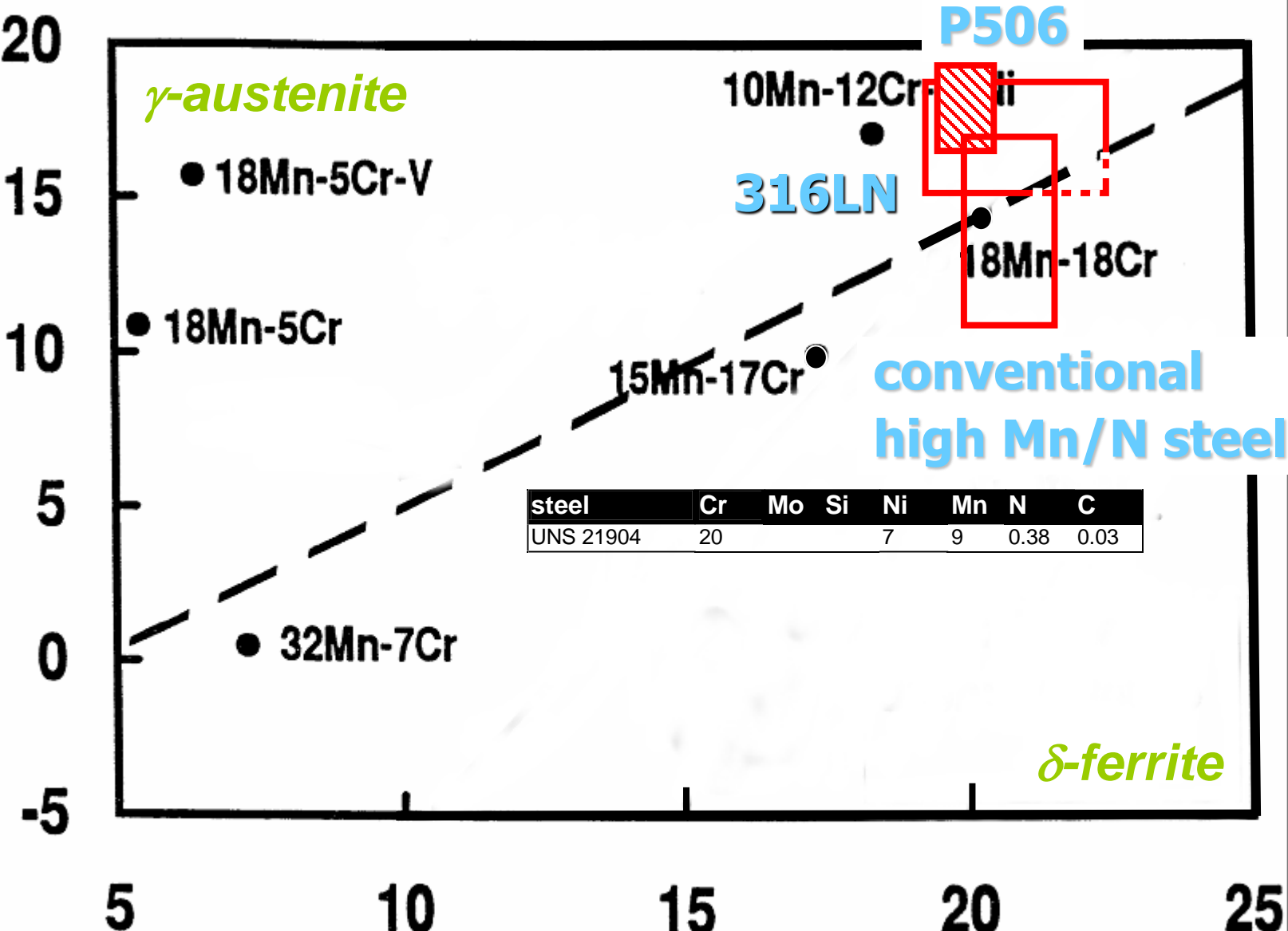


# Materials for magnets: austenitic stainless steels



"mumetal", Ni80Mo0.5Mn  
 $S < 0.0005$ ;  $P = 0.003$  (!)

$$[\text{Ni}]_{\text{eq}} = \text{Ni} + 0.11\text{Mn} - 0.0086\text{Mn}^2 + 0.41\text{Co} + 0.44\text{Cu} + 18.4\text{N} + 24.5\text{C}$$



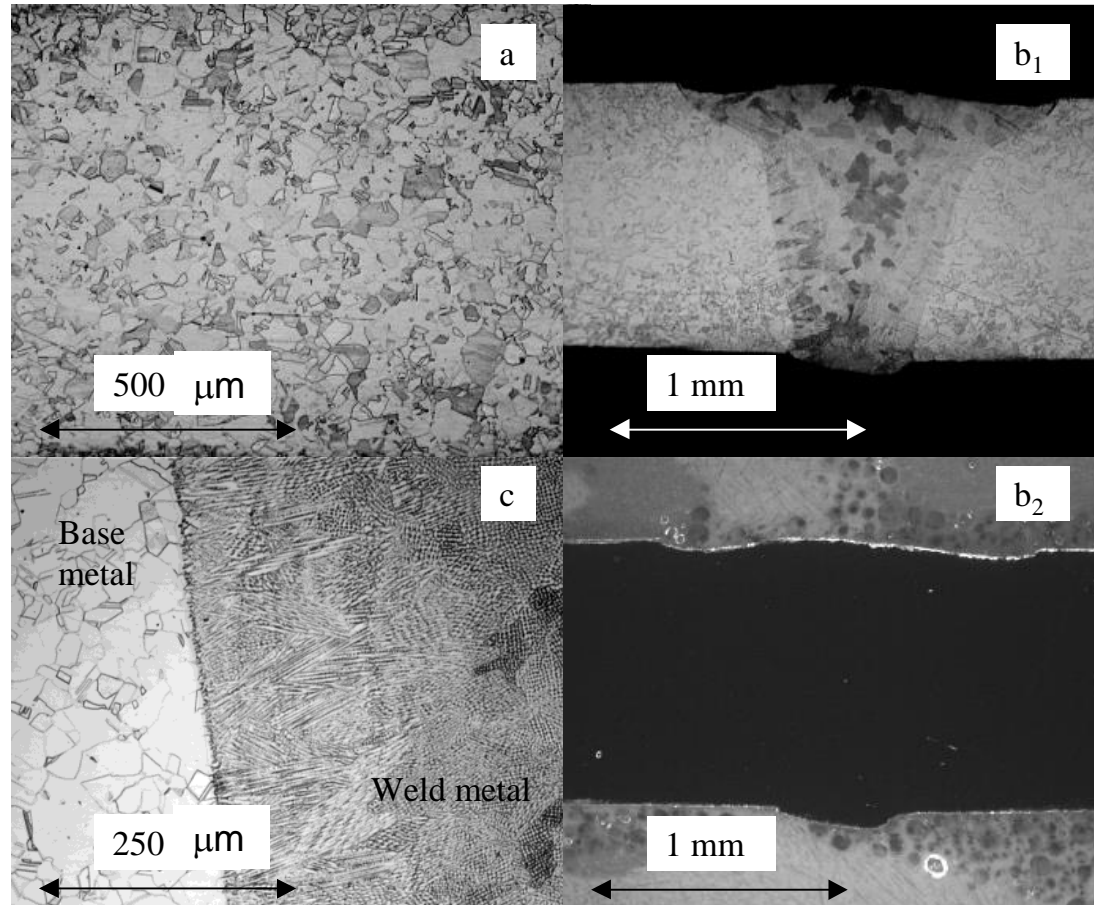
$$[\text{Cr}]_{\text{eq}} = \text{Cr} + 1.21\text{Mo} + 0.48\text{Si} + 2.27\text{V} + 0.72\text{W} + 2.20\text{Ti} + 0.14\text{Nb} + 0.21\text{Ta} + 2.48\text{Al}$$

Materials for magnets:  
austenitic  
stainless steels



# Materials for magnets: austenitic stainless steels

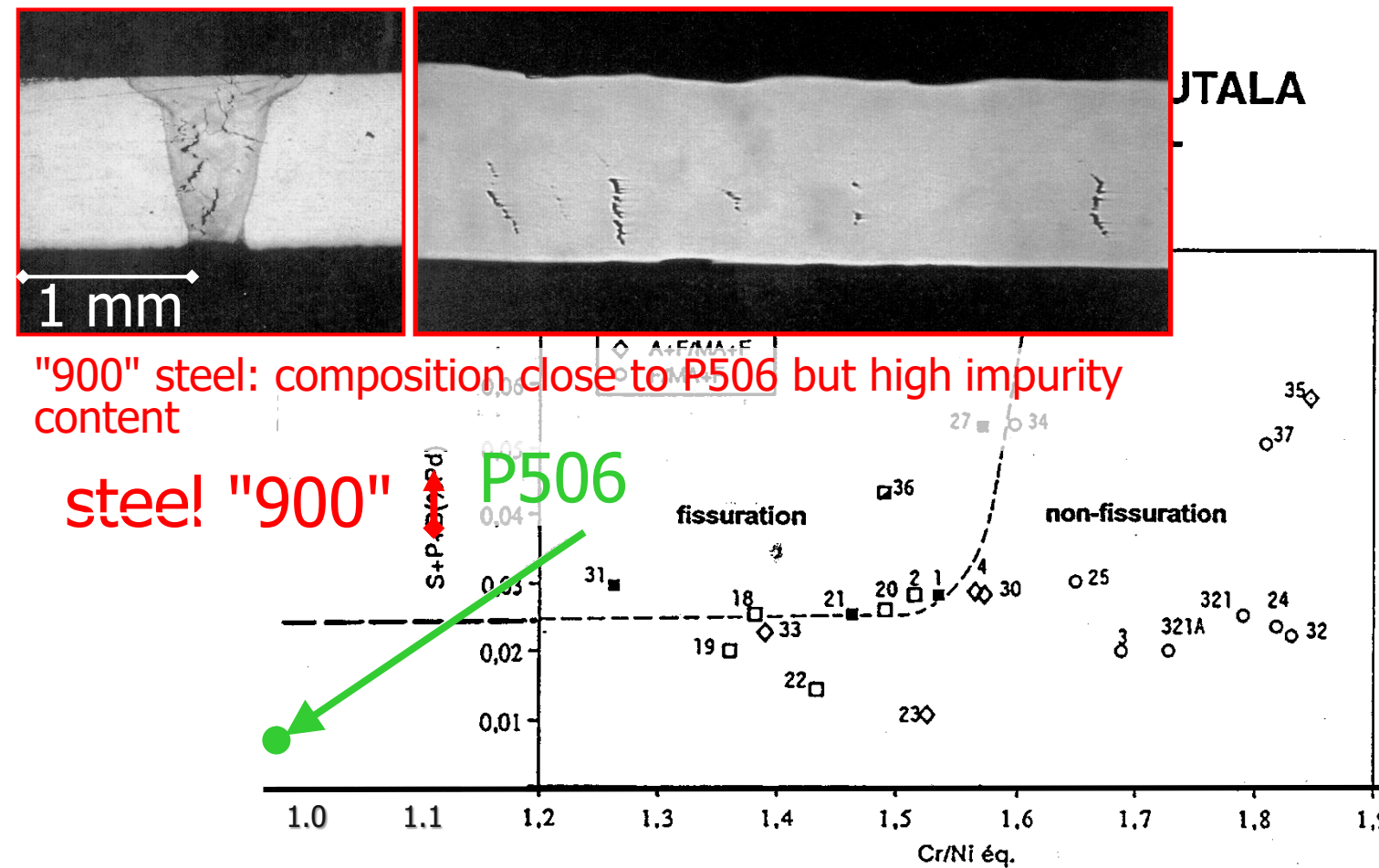
Steels	C	Mn	Ni	Cr	Mo	Si	N	P	S	B
P506	0.012	12.05	10.90	19.18	0.86	0.23	0.33	0.005	0.001	<0.001



S. Sgobba, C. Boudot,  
Soudabilité laser d'aciers  
inoxydables austénitiques,  
Matériaux et Techniques  
95, n°11-12, p. 23 (1997).

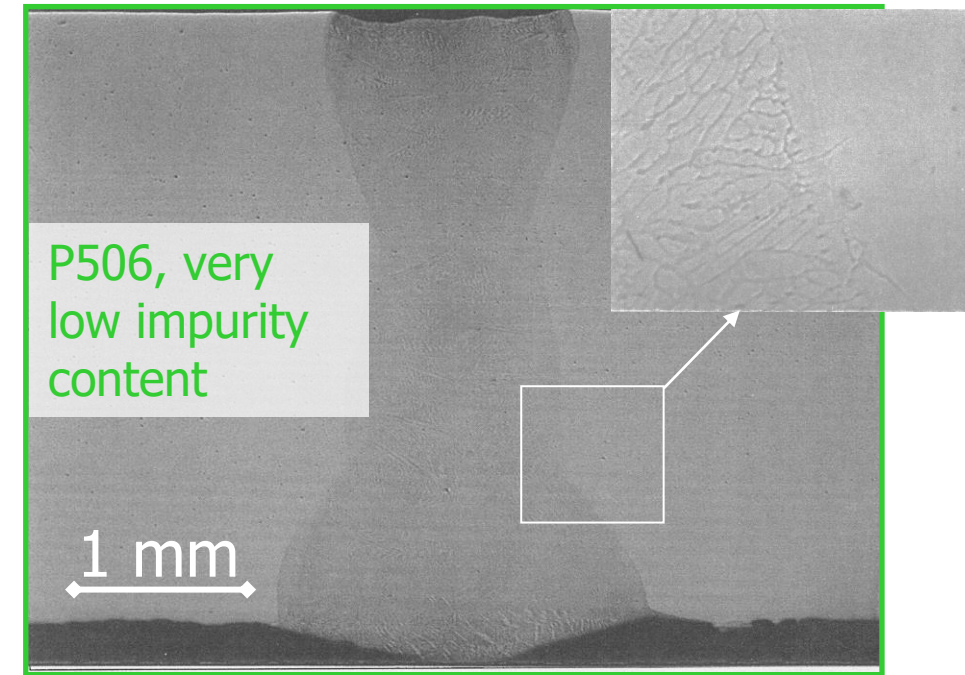
J.P. Bacher and S. Sgobba,  
TIG Weldability of Special  
Stainless Steels for the  
Beam Screen of the Large  
Hadron Collider, Bulletin du  
Cercle d'Etude des Métaux,  
XVI, p. 13.1 (1995)

# Materials for magnets: austenitic stainless steels



"900" steel: composition close to P506 but high impurity content

steel "900" P506

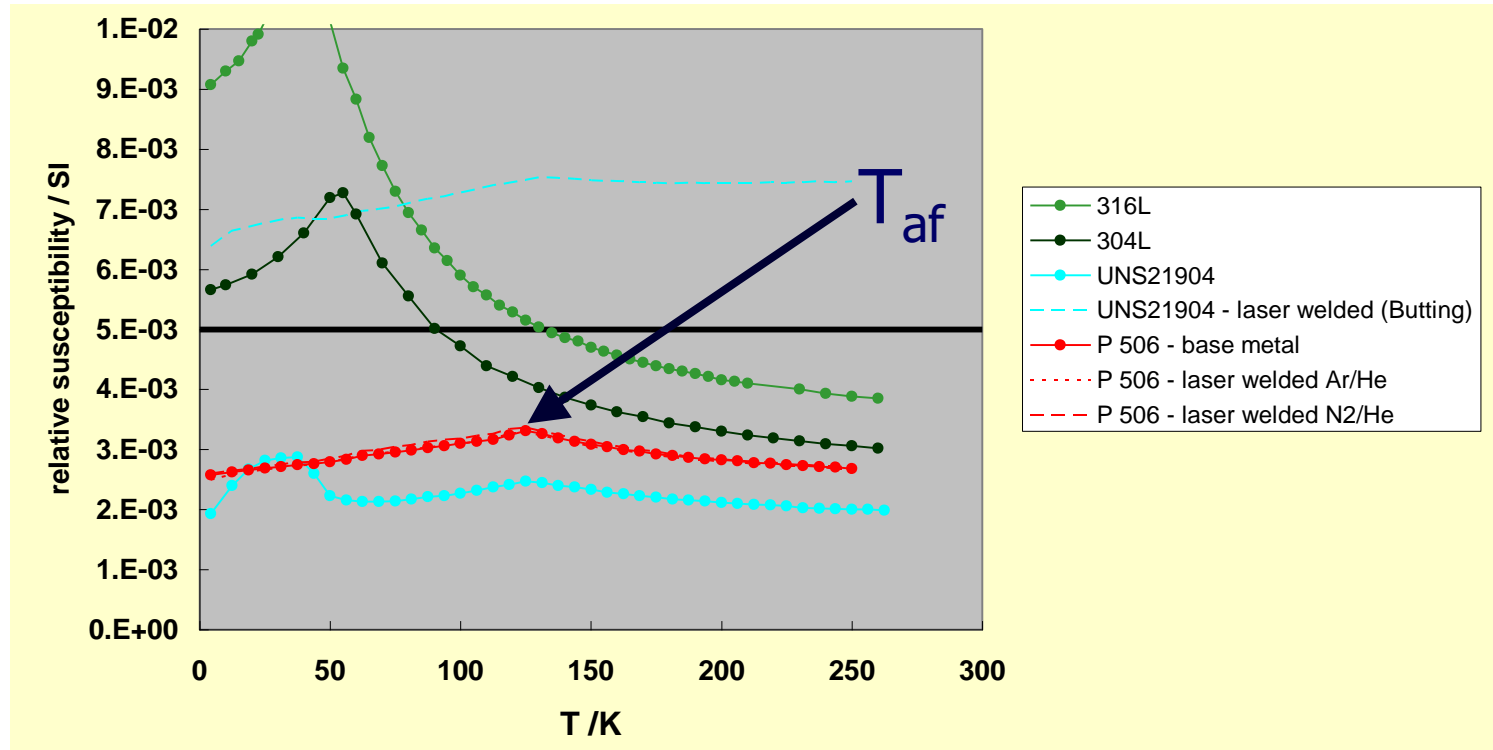


$$\frac{\text{Cr}_{\text{eq.}}}{\text{Ni}_{\text{eq.}}} = \frac{\text{Cr} + 1,37\text{Mo} + 1,5\text{Si} + 2\text{Nb} + 3\text{Ti}}{\text{Ni} + 0,31\text{Mn} + 22\text{C} + 14,2\text{N} + \text{Cu}}$$

S. Sgobba: proc. Cycle Métaux et Procédés, CIP - Tramelan /CH, 1996, p. 8/1-10



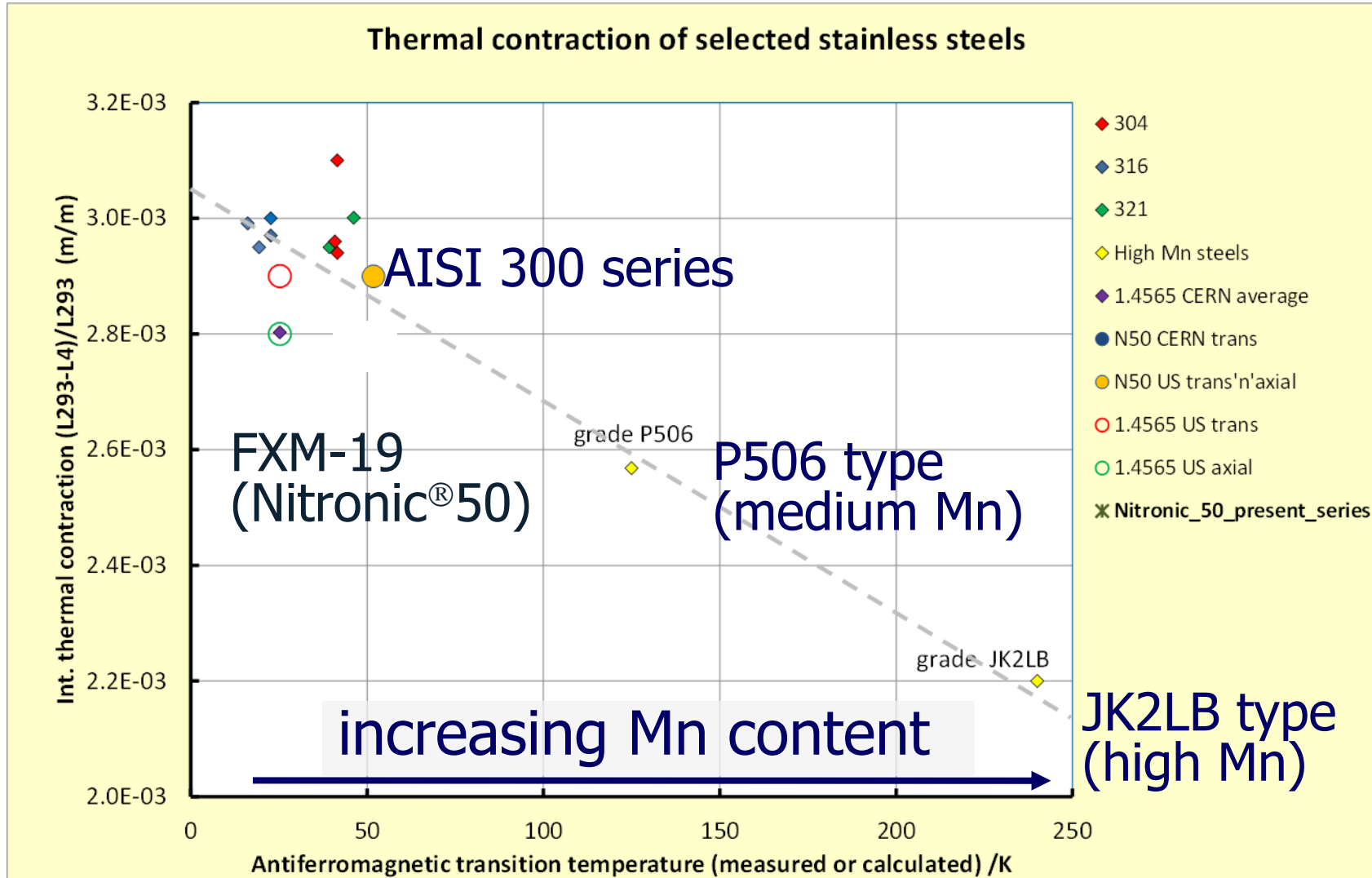
# Materials for magnets: austenitic stainless steels



Warnes law:  $T_{af}(K) = 90 - 1.25Cr - 2.75Ni - 5.5Mo - 14Si + 7.75Mn$

P506, predicted  $T_{af} = 121.5$  K, measured 125.7 K

# Materials for magnets: austenitic stainless steels





# ITER TOKAMAK

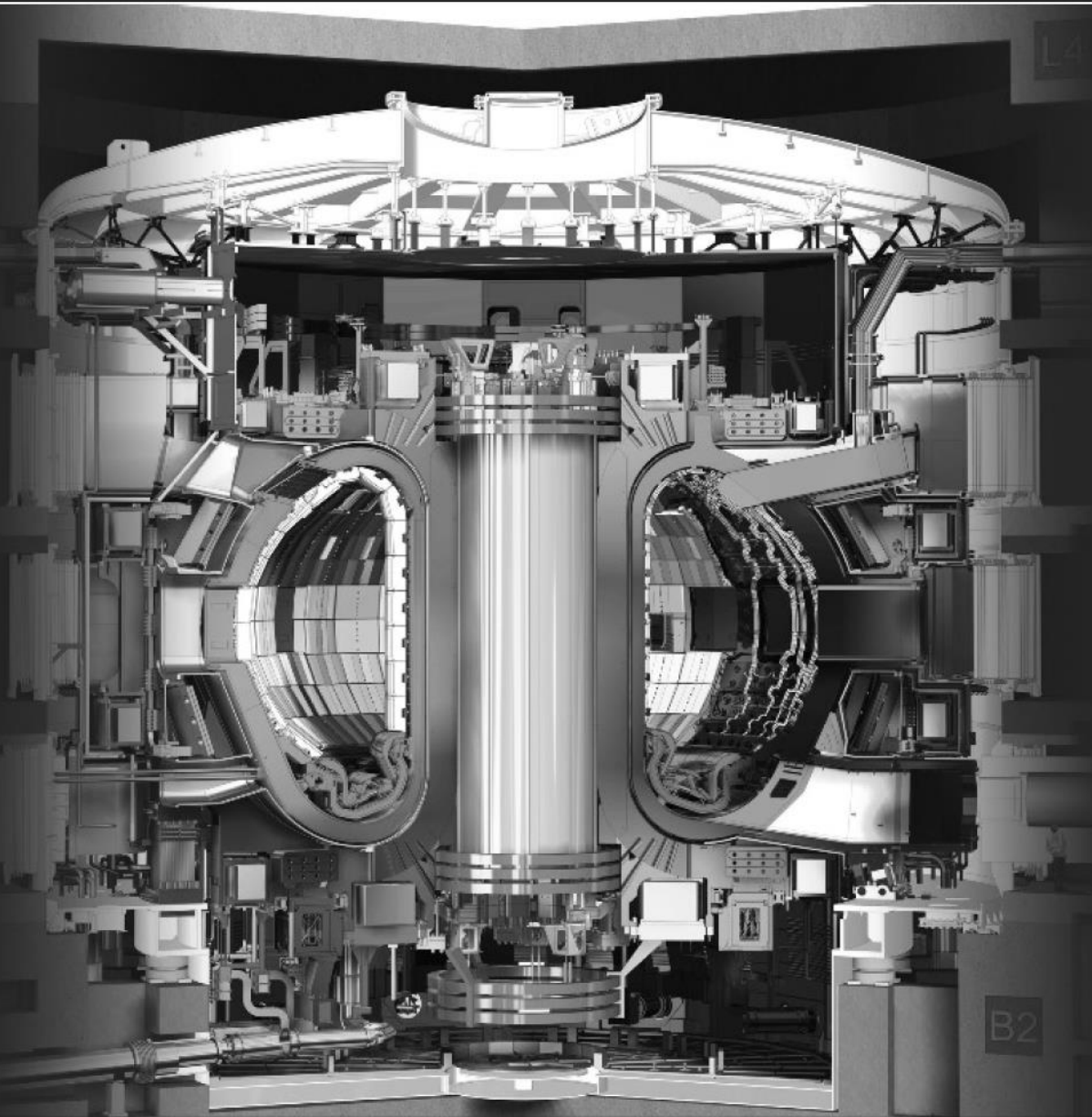
AN INTERNATIONAL PROJECT FOR A NEW & CLEAN ENERGY

Courtesy of ITER,  
@iter.org

ITER represents the future of nuclear power where the fission reaction is replaced by a fusion reaction, the nuclear reaction that powers the sun and the stars, a safe, non-carbon emitting and virtually limitless energy.

With its millions of components, ITER will be the largest and most powerful tokamak ever built.

35 countries will collaborate during 35 years



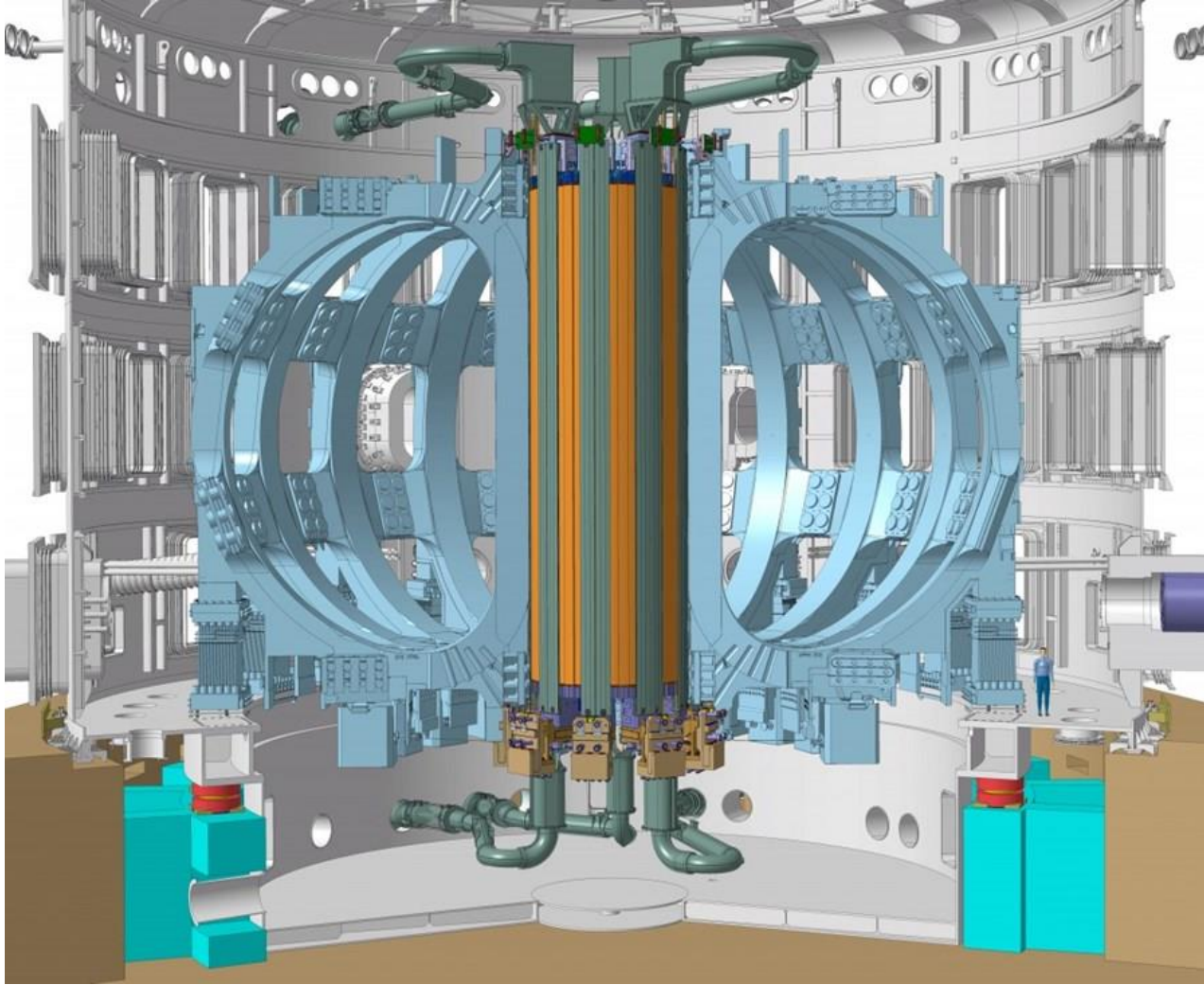
A GIANT  
23 000 T

10X the core  
of the sun  
150 millions ° C  
plasma temperature

FUSION ENERGY  
500 MW  
Output power

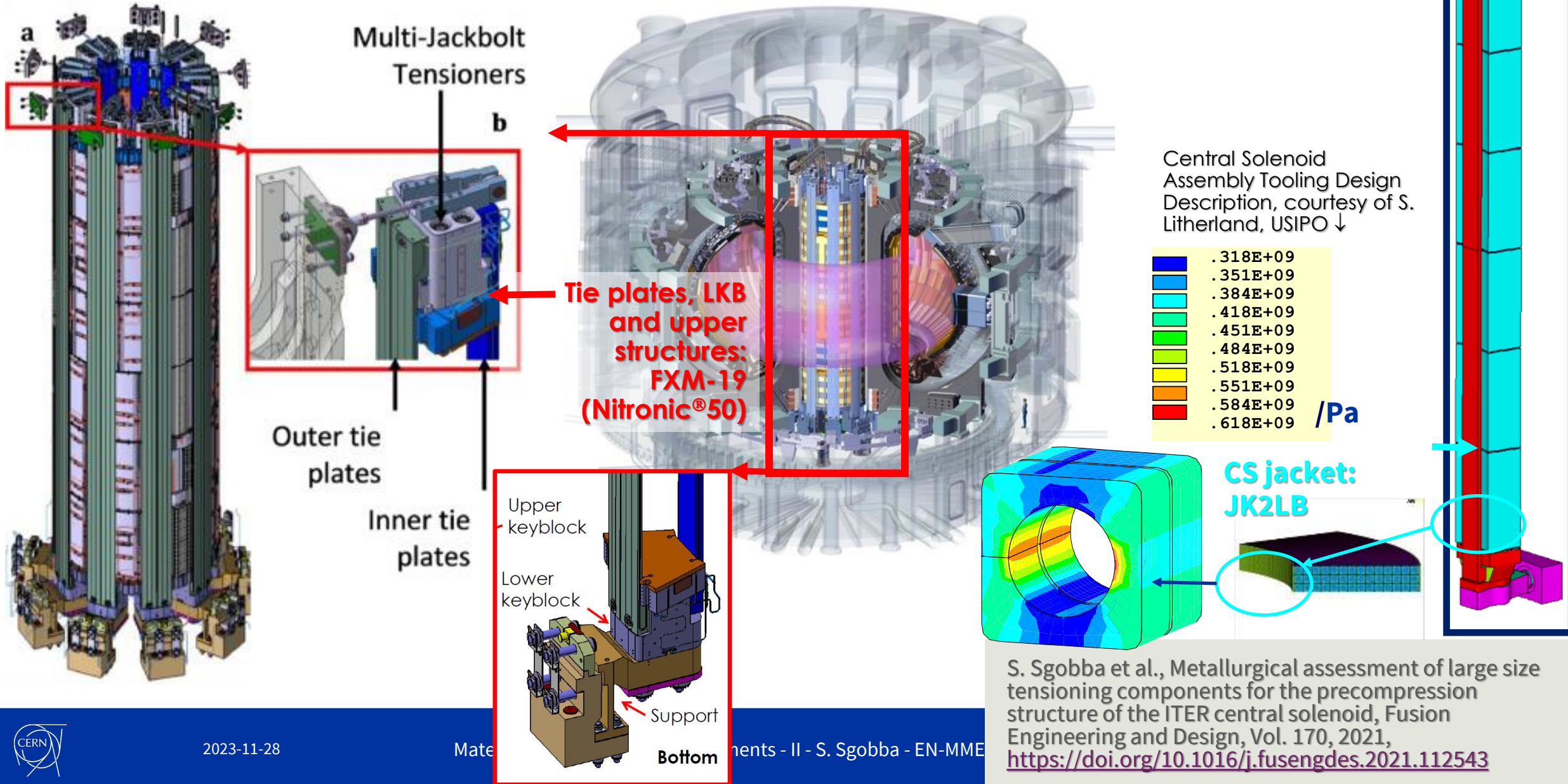


# Materials for magnets: austenitic stainless steels





# Materials for magnets: austenitic stainless steels





# Materials for magnets: austenitic stainless steels

Forged tie-plate dimensions:  
Length: 15.2 m  
width : 0.51 m  
thickness: 280 mm at heads,  
171 mm at central part  
material: Nitronic 50

- Very large multidirectionally forged components

*Lower Support brackets:  
from material  
specification to 100%  
volumetric inspection.  
Courtesy of Monchieri*

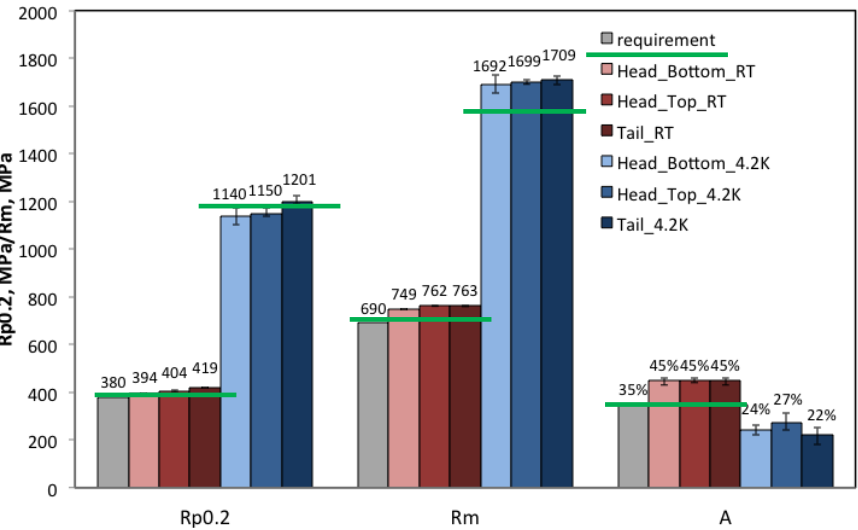
*Lower key block in  
test assembly.  
Courtesy of US  
ITER and Petersen  
Inc.*

*Tie plate single  
piece forging.  
Courtesy of Rolf  
Kind GmbH*



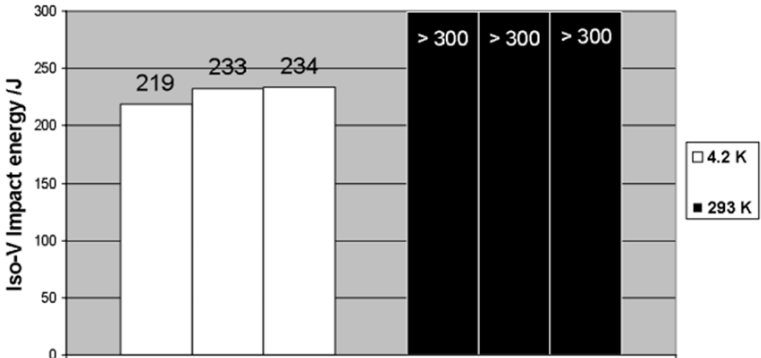
# Materials for magnets: austenitic stainless steels

## Mechanical properties FXM-19



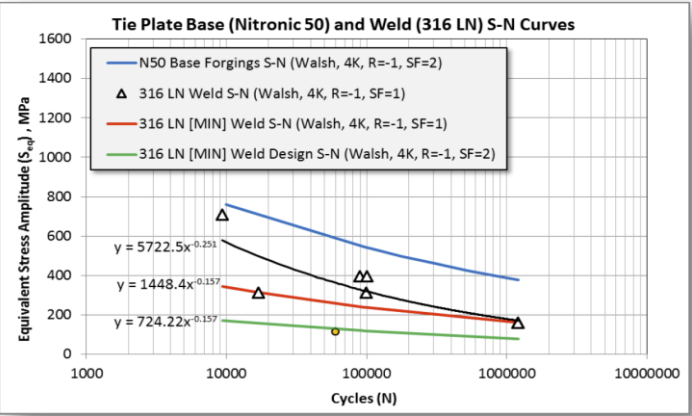
S.A.E. Langeslag et al., "Extensive Characterisation of Advanced Manufacturing Solutions for the ITER Central Solenoid Pre-compression System," Fusion Eng. Des. (2015), <https://doi.org/10.1016/j.fusengdes.2015.06.007>

Temperature [K]	direction	$\sigma_{y0.2}$ [MPa]	$\sigma_r$ [MPa]	$\epsilon_r$ [%]
293	Long.	450 ± 10	835 ± 20	61 ± 3
	Transv.	440 ± 10	825 ± 5	45 ± 4
77	Long.	1180 ± 10	1760 ± 20	57 ± 4
	Transv.	1120 ± 50	1715 ± 100	45 ± 3
4.2	Long.	1620 ± 50	2115 ± 60	18 ± 3
	Transv.	1700 ± 100	2105 ± 90	15 ± 4



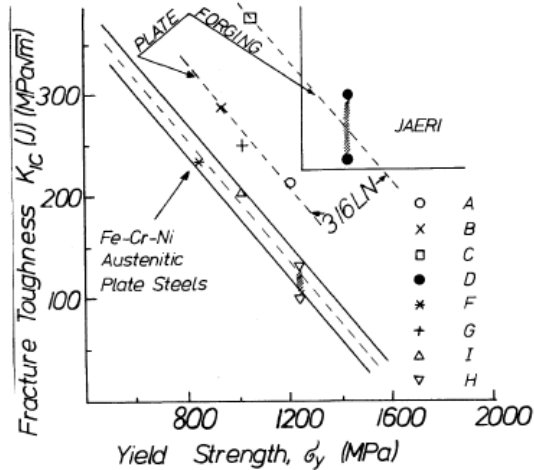
P506

	Specimen location	Specimen orientation	$K_{IC}$ [MPa√m]	$J_{IC}$ [N/mm]
Single piece forged	Head (top)	LT	170	130
	Head (bottom)	LT	190	161
		LS	197	172
	Tail (slab)	LT	188	157
		LT	226	239
Welded solution	weld	weld direction	112	62



Courtesy of R.P. Walsh, NHMFL

# Materials for magnets: austenitic stainless steels



4 K toughness-strength relation of nitrogen strengthened stainless steels.

A. Nyilas, P. Komarek - Cryogenic Tensile and Fracture Properties of Structural Materials for Superconducting Magnets in Fusion Technology (1989)

Courtesy Dr. N. Pauze,  
ArcelorMittal Industeel

## Nuclear physics & Fusion research

ITER PROJECT SUMMARY

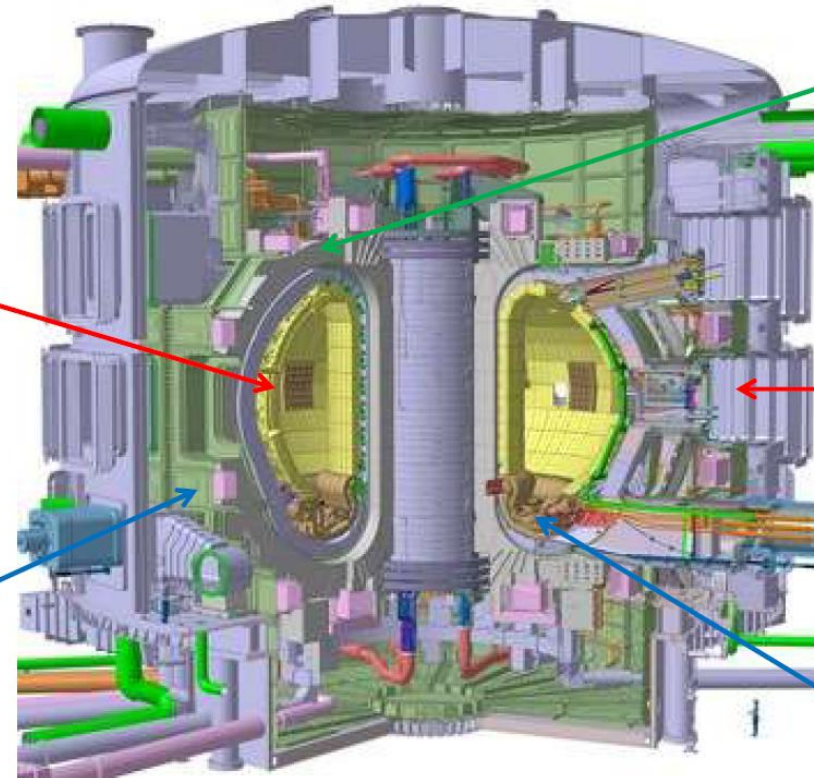
### VV, IWS, Ports & Components, Port plugs

HHI, Efremov, Iter India, AMW, ENSA, CNIM-LNT

316L(N)-IG  
304/304L,  
304B4,  
430  
7300T

### THERMAL SHIELD:

SFA  
304LN  
1700T



TFCoil  
(CP and TFCS)  
MHI, TOSHIBA, CNIM  
316LNH  
1200T

CRYOSTAT:  
LNT  
304L Iter  
2500T

DIVERTOR:  
Hollming, CNIM,  
Walter Tosto  
316L(N)-IG  
20T

> 12 500T booked



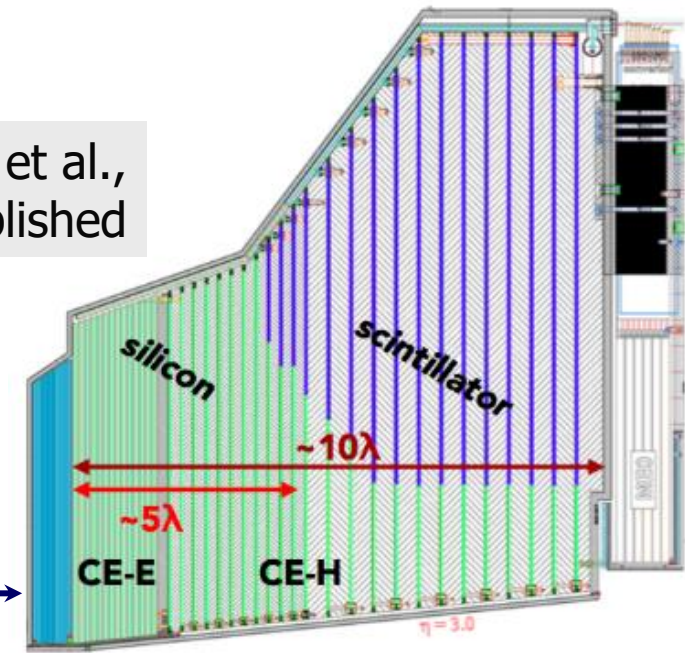
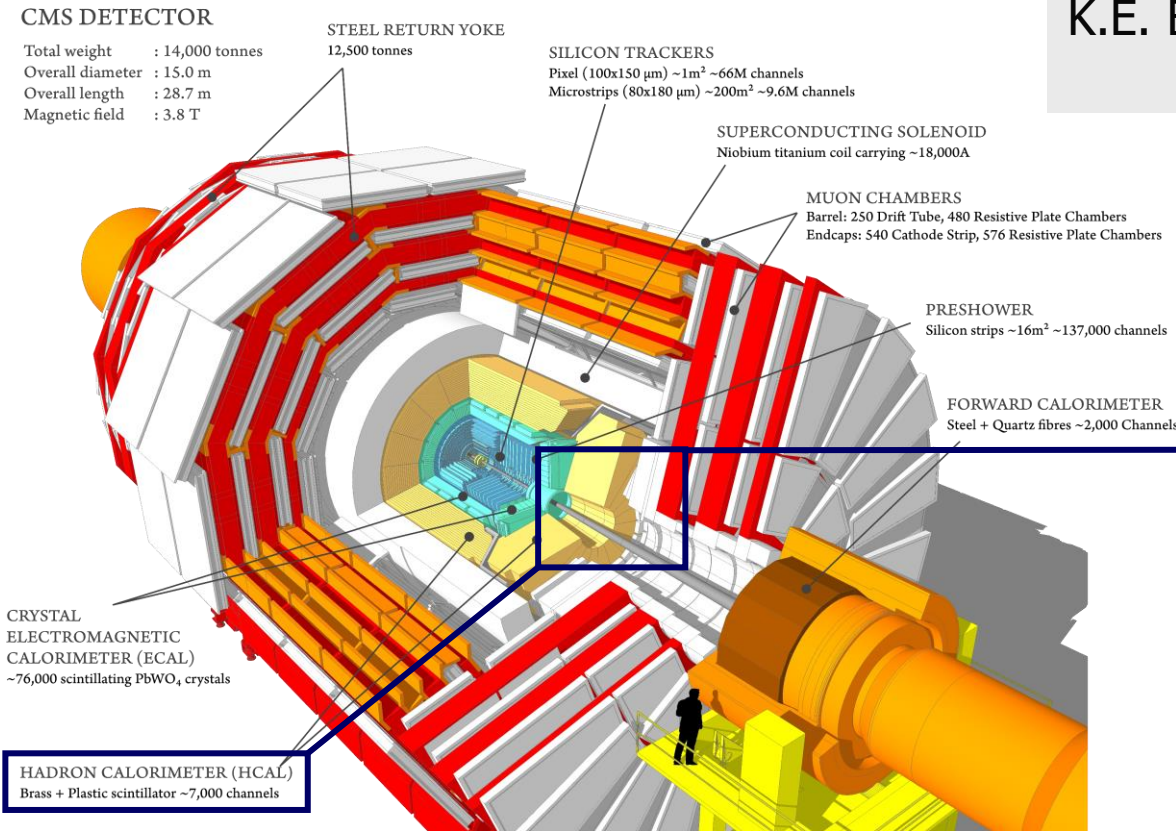
# Materials for magnets: case study. Austenitic stainless steels: a low permeability steel for the new CMS HG-CAL detector



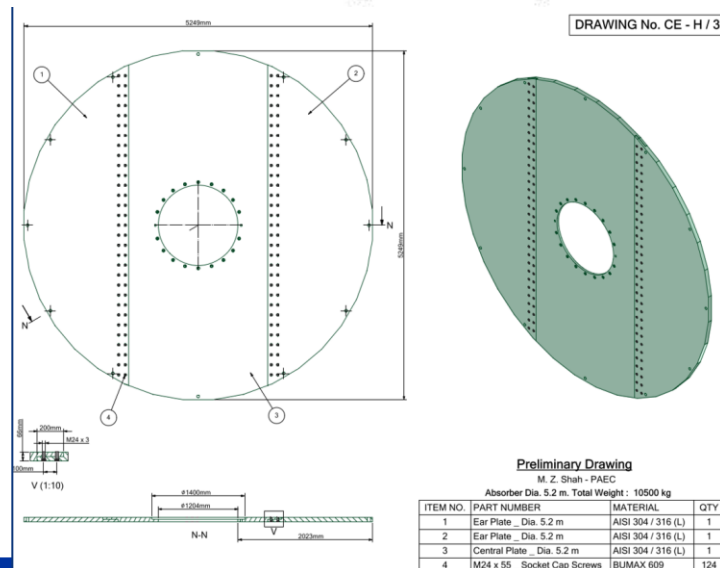


# Case study: a low permeability steel for CMS HG-CAL

K.E. Buchanan, S. Sgobba et al.,  
to be published



Schematic view of the High Granularity Calorimeter

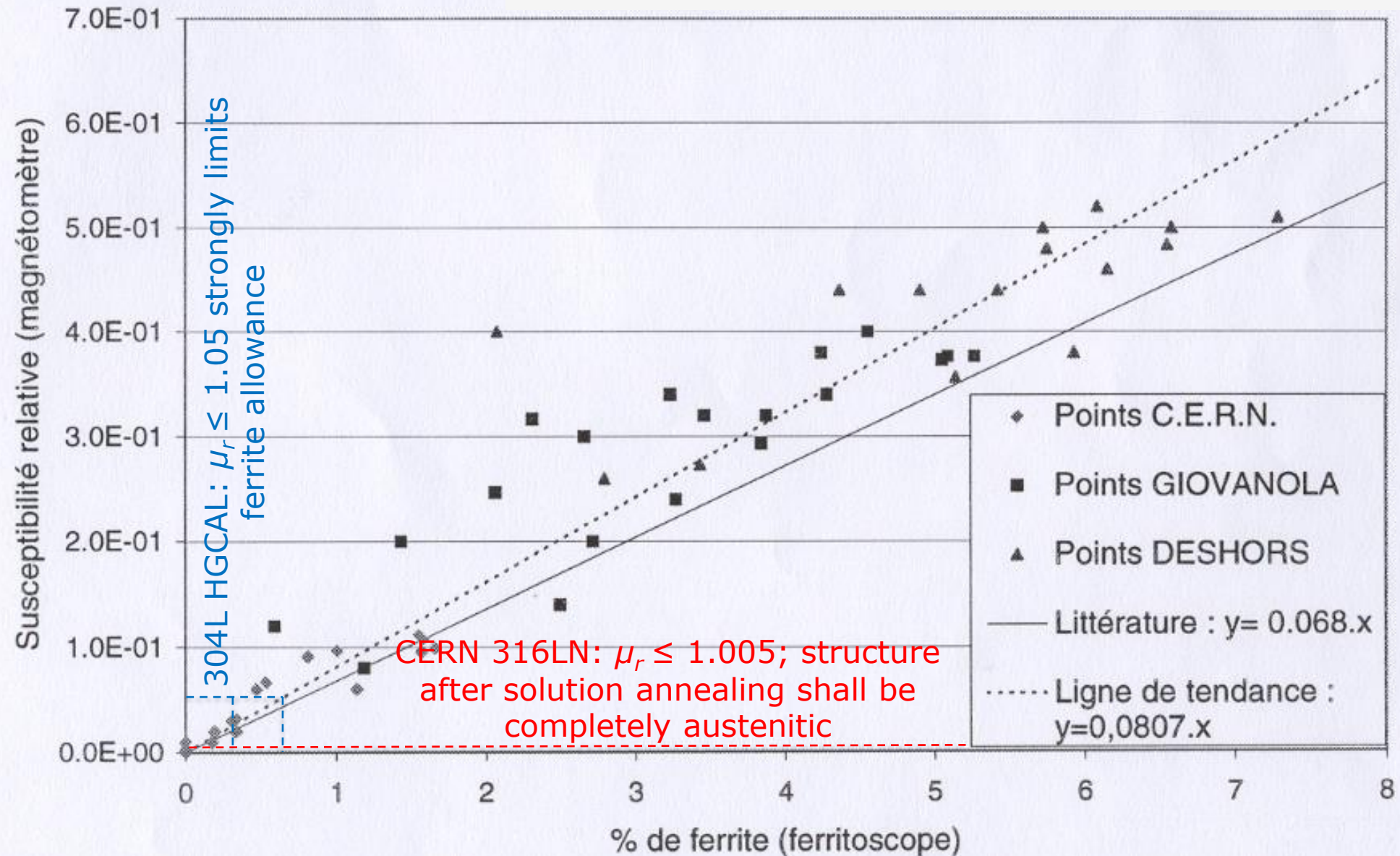


- 564 t of stainless steel required for the CE-H cassettes, thickness 45  $\sim$  110 mm
- The relative magnetic permeability in the bulk plate material shall not exceed **1.05**
  - Stringent control of ferrite content
  - Stability against martensitic transformations



# Case study: a low permeability steel for CMS HG-CAL

S. Sgobba and C. Boudot, Matériaux et Techniques 95, vol. 11-12, p. 23 (1997)



# Case study: a low permeability steel for CMS HG-CAL

NUMERICAL DESIGNATION  
1.4306

Material designation:  
X2CrNi19-11  
Country/Standard:  
European Union / EN  
Group of Materials:  
Metals  
Subgroup:  
EN 10028-7 Flat products made of steels for pressure purposes - Part 7:  
Stainless steels  
Comment:  
Austenitic stainless steel

1.4306:  
"High alloy" 304L

Criteria	Min.	Max.	Approx.
C		0.03	
Mn		2.00	
P		0.045	
S		0.015	
Si		1.00	
Ni	10.0	12.0	
Cr	18.0	20.0	
N		0.10	

Very clean heat,  
143 ppm S+P

"High alloy" 1.4306,  
Ni 10.417 %

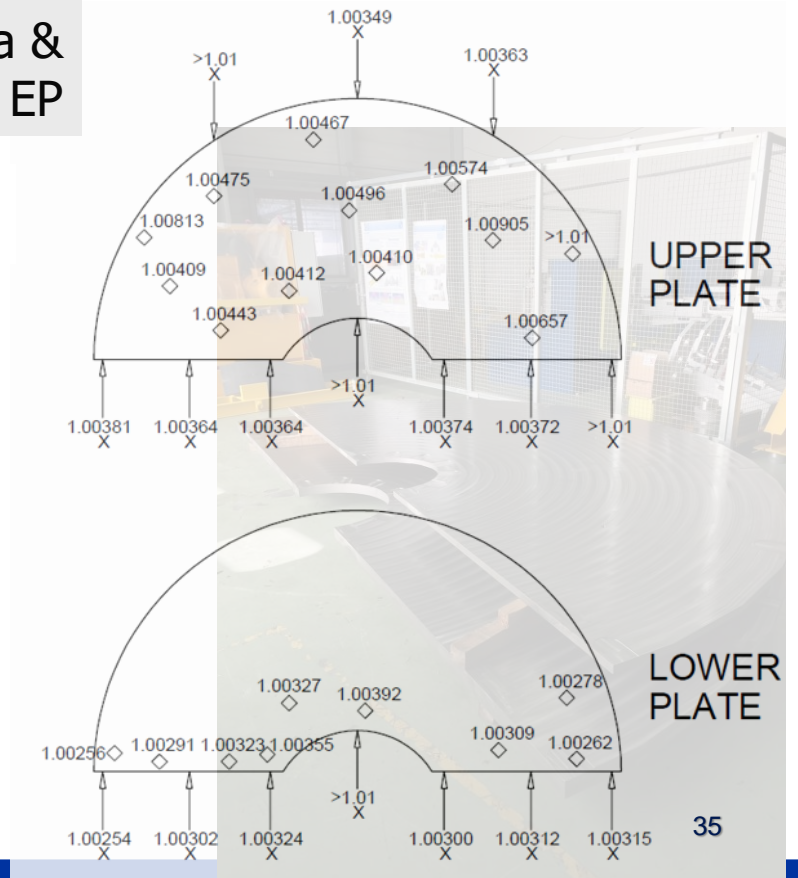
- Low magnetic permeability:
  - Stringent control of ferrite content (composition / steelmaking route)
  - Stability against martensitic transformations (grade selection)

Etat thermique de livraison - Heat treatment state of delivery - Wärmebehandlung Lieferzustand  
Hypertrempe eau ( 1050 °C - 0.5 min/mm ) - Solution annealed and water quenched  
Procédé d'élaboration - Melting process - Erschmelzungsart  
Electric-arc furnace - VOD - Finish n° 1 - 1D - HRAP

Nuance - Grade - Werkstoff X5CrNi18-10 (1.4301)  
X5CrNi18-10 (1.4301)  
UNS S30400 (304)  
UNS S30403 (304L)  
UNS S30400 (304)  
UNS S30403 (304L)  
X2CrNi19-11 (1.4306)  
X2CrNi19-11 (1.4306)

Courtesy of S. Moccia &  
M. Pentella /CERN EP

ArcelorMittal Industeel





← Ingot casting →

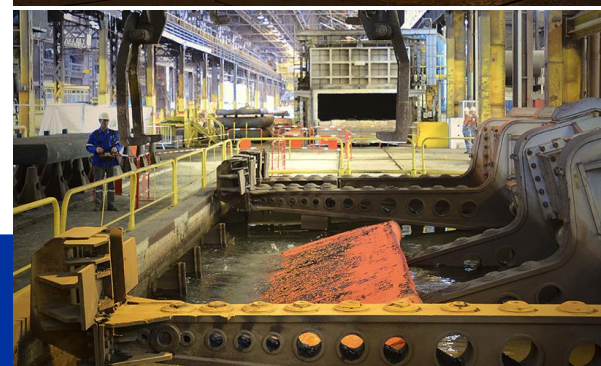


© 2023 Vesuvius



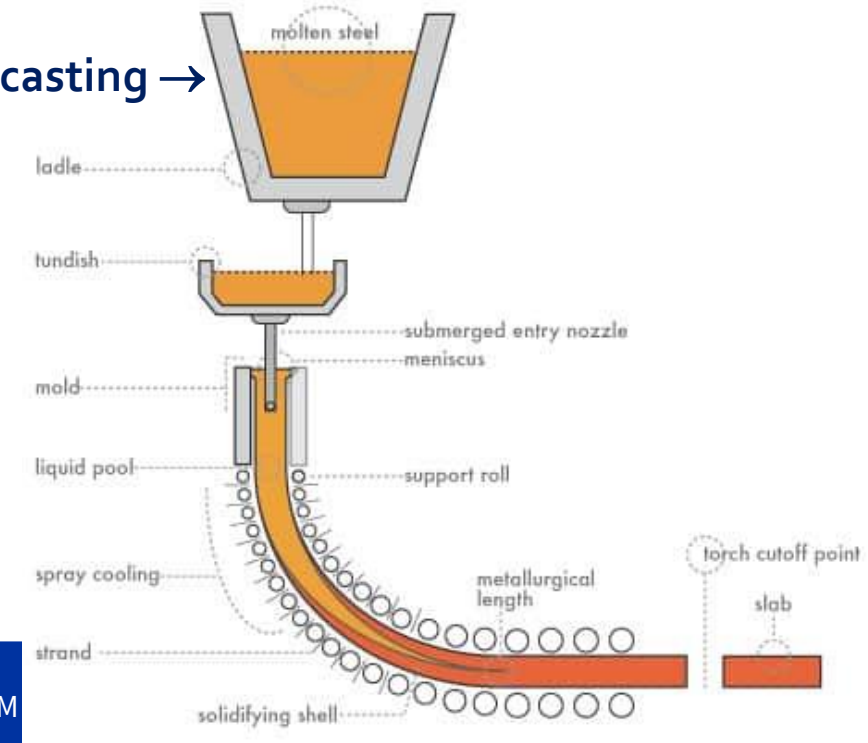
Rolled slab

Rolling of quarto plates



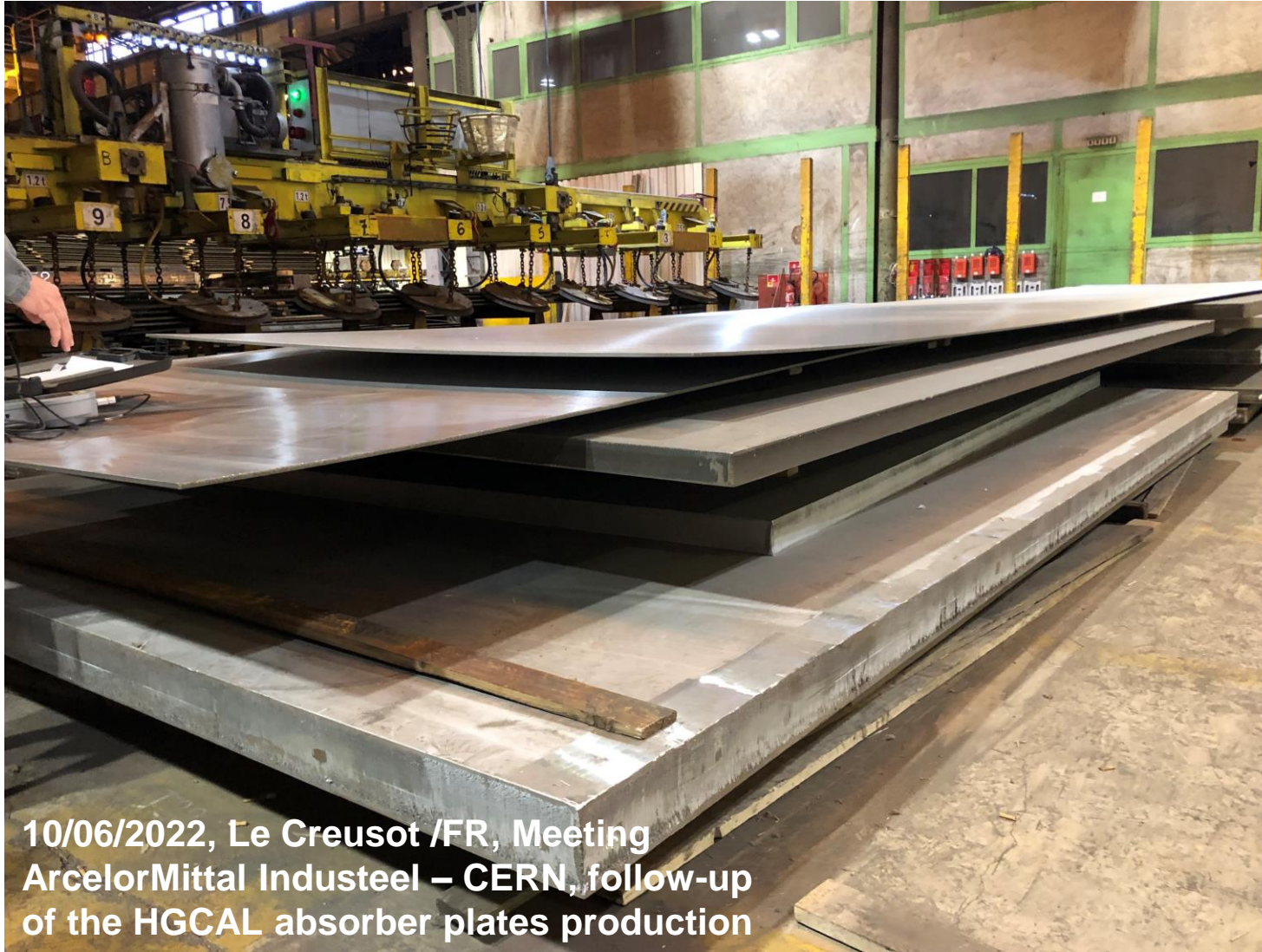
Solution annealing and finishing (descaling, grinding, skin pass rolling where applicable)

Versus continuous casting →





# Case study: a low permeability steel for CMS HG-CAL





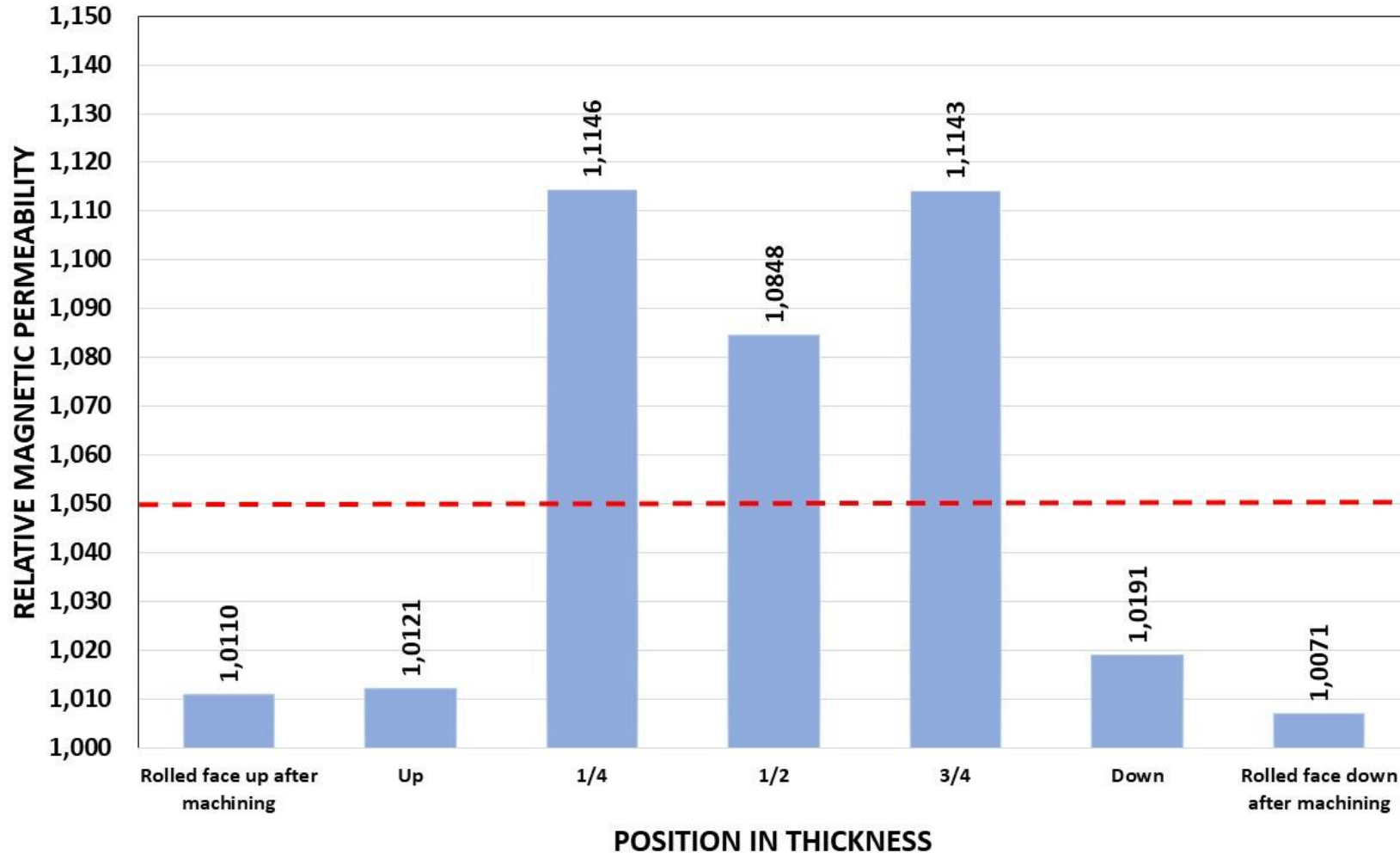
## Quartering of a sample issued from a 120 mm plate



- This sample was issued from a 120 mm thick plate and symmetrically machined down to 100 mm
- Other dimensions unchanged (50 mm x 100 mm)
- This machining operation is representative of the fabrication of the HG-CAL plates, whose thickness will be reduced with respect to the original as-rolled thickness

# Case study: a low permeability steel for CMS HG-CAL

New\_Relative\_magnetic\_permeability\_against\_position\_in\_thickness



- Double peak around mid-thickness (at  $\frac{1}{4} t$  and  $\frac{3}{4} t$ , respectively) as metallurgically expected
- The three central layers feature values above 1.05

- Average of layers in thickness, including rolled faces:

$$\mu_r = 1.0519$$

- Excluding rolled faces:

$$\mu_r = 1.0690$$



# Case study: a low permeability steel for CMS HG-CAL

## Double peak metallurgically expected

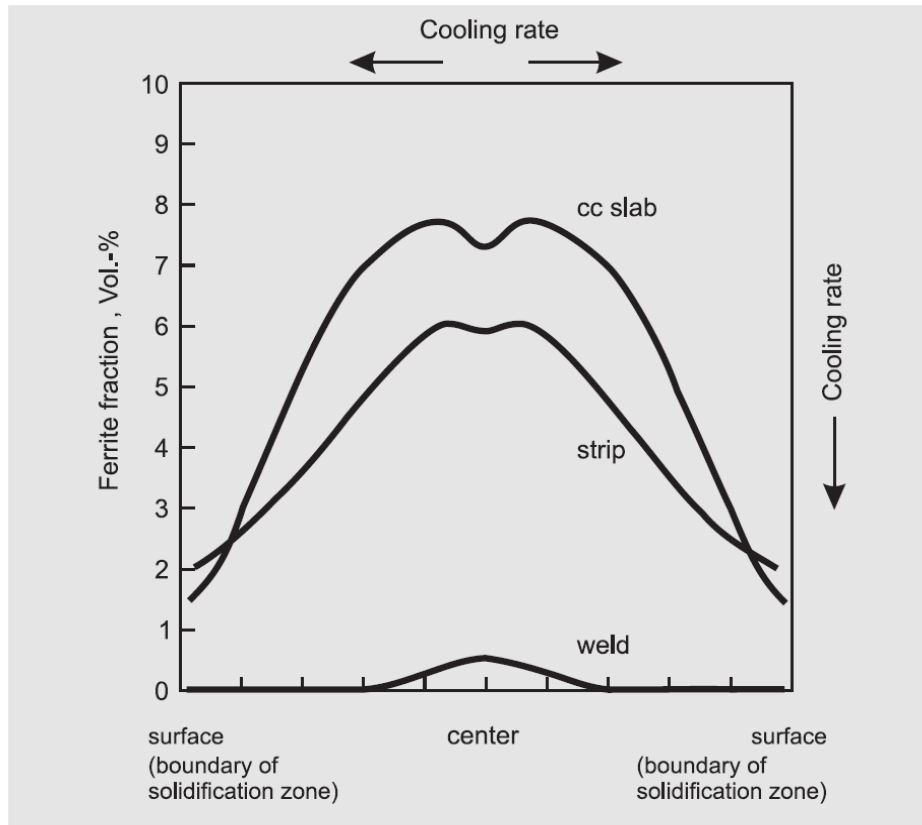
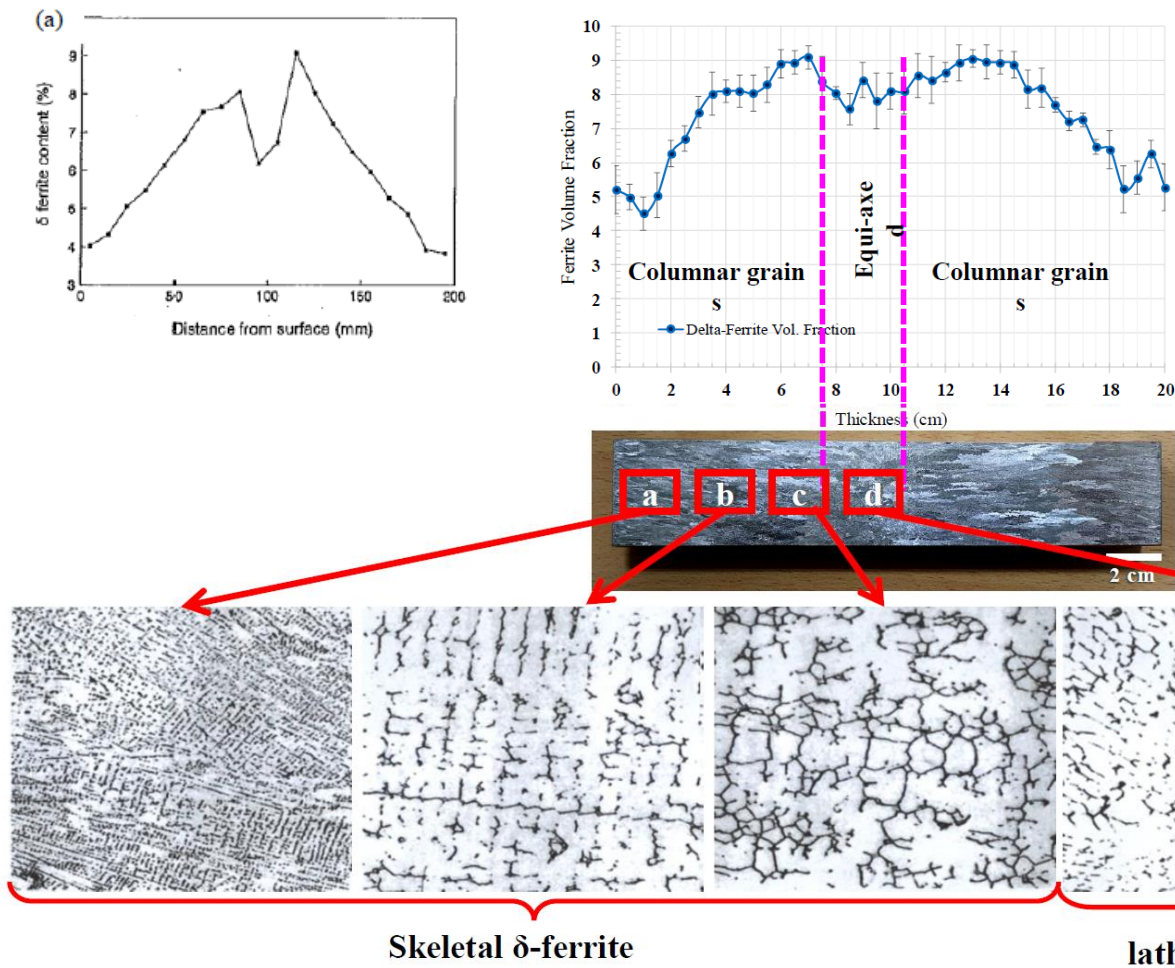


Fig. 1. Typical distribution of  $\delta$ -ferrite across the solidification cross section.

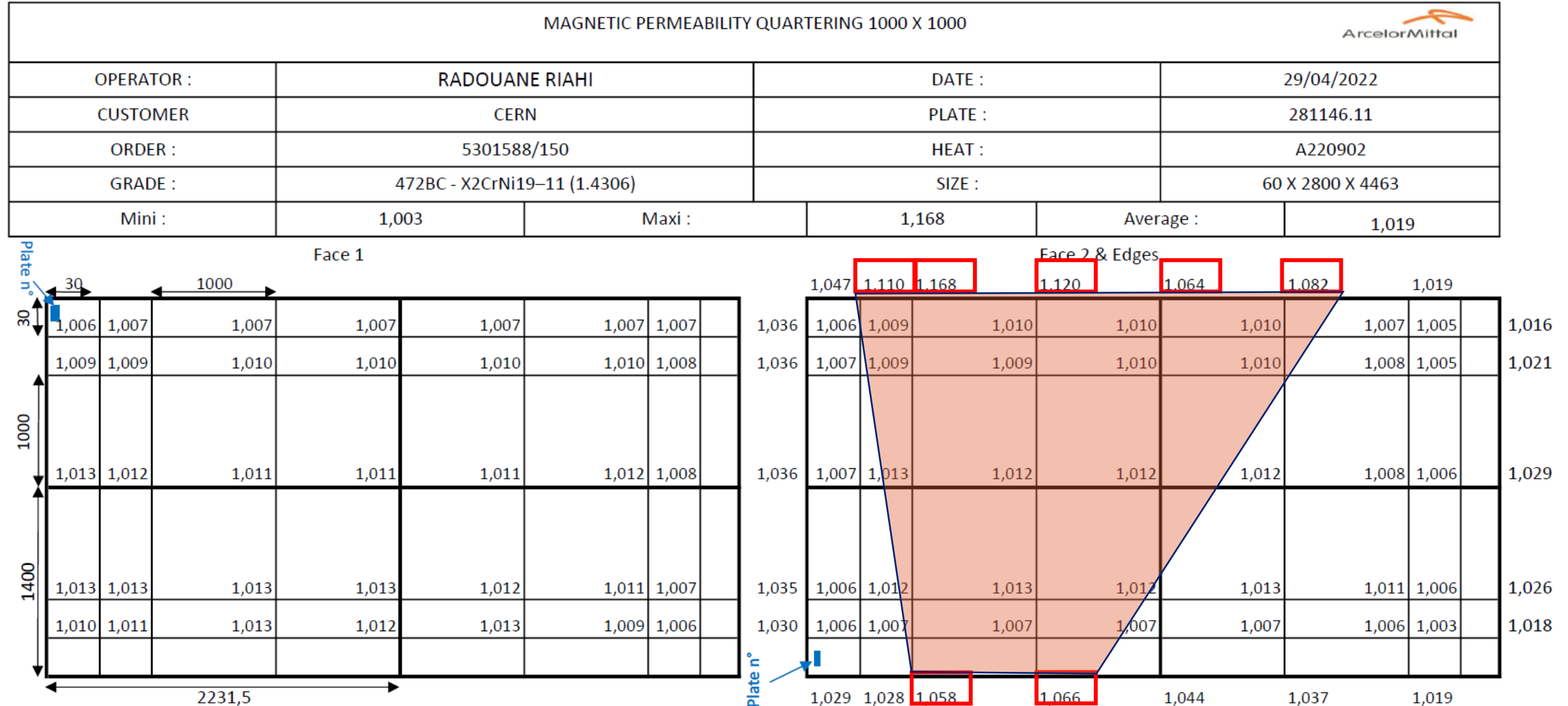
From Piotr R. Scheller, Roman Flesch and Wolfgang Bleck, *Solidification Morphology and Microstructure Properties at Increased Cooling Rates for 18-8 Cr±Ni Stainless Steel*, *ADVANCED ENGINEERING MATERIALS* 1999, 1, No. 3±4, p. 209ff.

From S. Kim, Y. Shin, N. Kim, Distribution of  $\delta$  ferrite content in continuously cast type 304 stainless steel slabs, *Ironmak. Steelmak.* 22 (1995) 316–325



# Case study: a low permeability steel for CMS HG-CAL

## ArcelorMittal Industeel results on this 60 mm plate





# Case study: a low permeability steel for CMS HG-CAL

- **Volumetric measurements** through static-sample magnetometry
- All the measurements were performed using the Foerster Magnetoscop 1.069 and a **Static-Sample Magnetometer (SSM)**

Specimen #1



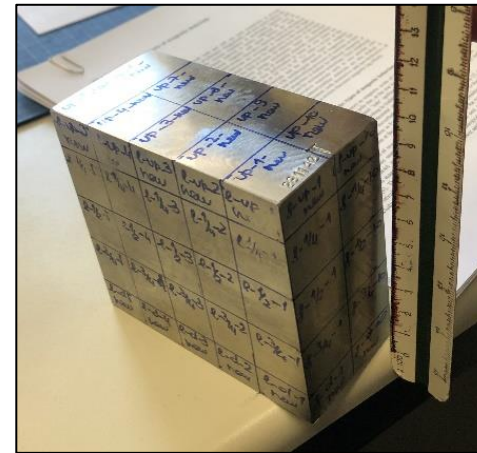
120 mm x 50 mm x 100 mm

Specimen #2



60 mm x 48 mm x 48 mm

Specimen #1 after machining

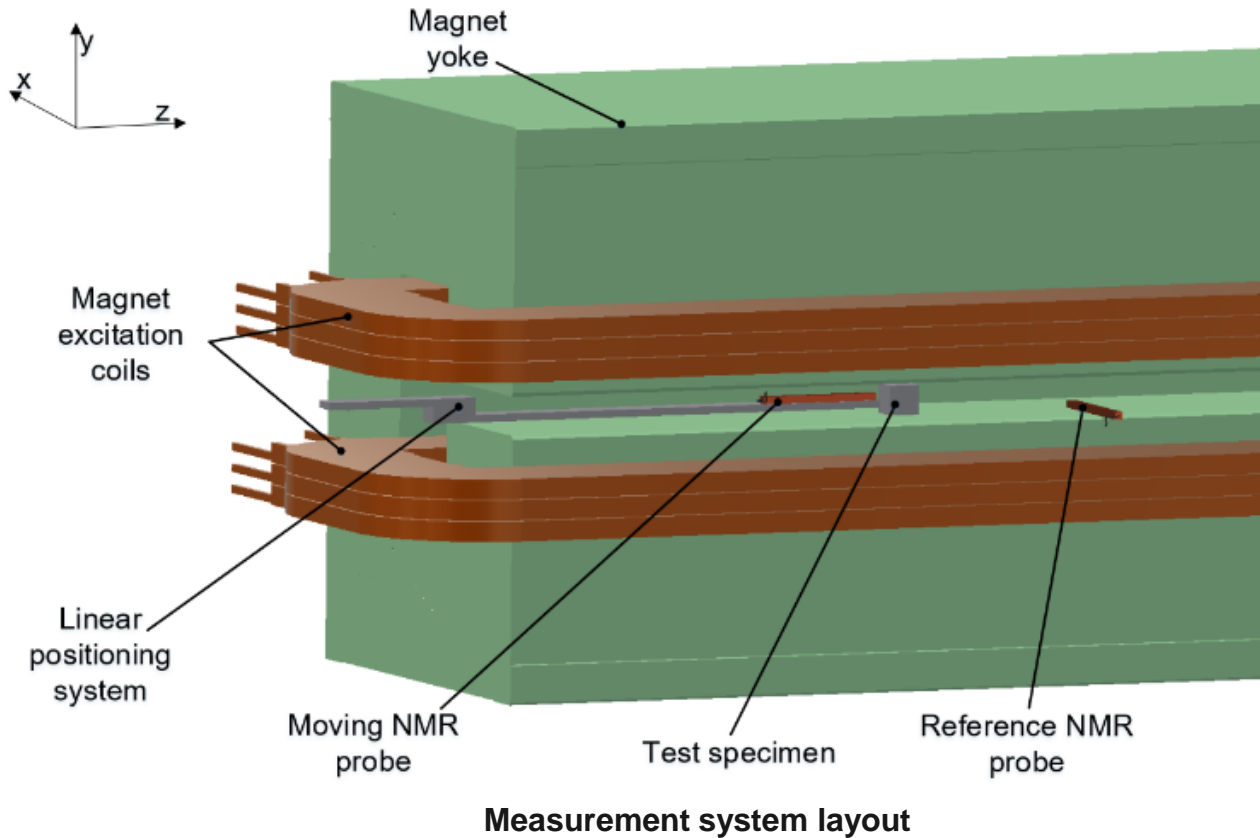


100 mm x 50 mm x 100 mm

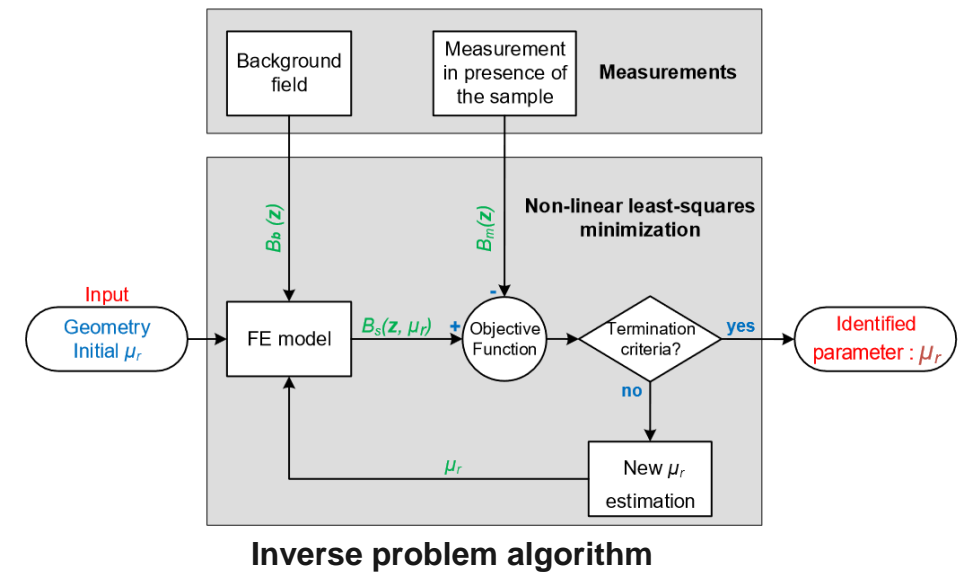
- A second goal is to **establish a valid measurement method** for **in-situ inspection** of the absorber plates using the Foerster localised measurements

# Case study: a low permeability steel for CMS HG-CAL

- The **SSM** works on the same principle of the Foerster Magnetoscop 1.069 (**flux-distortion detection**)

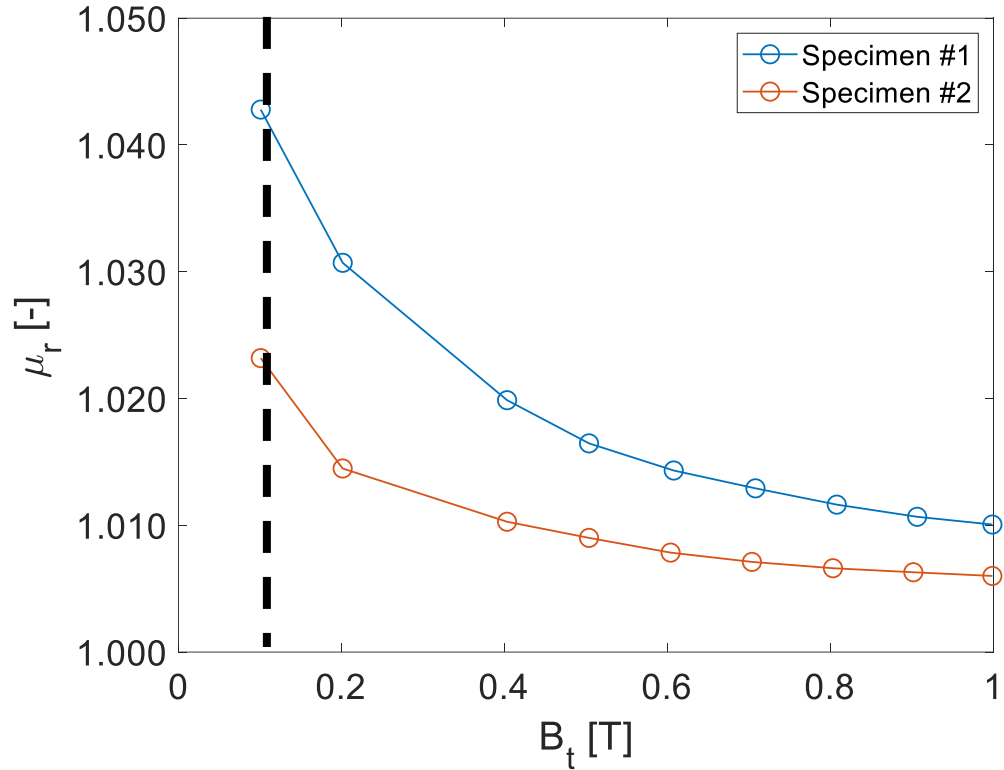


- The presence of the specimen with susceptibility  $\neq 0$  inside a magnetic field determines a **distortion of the field lines**
- By mapping the **background field** (no specimen) and the **distorted field**, the permeability can be computed by inverse-problem solving



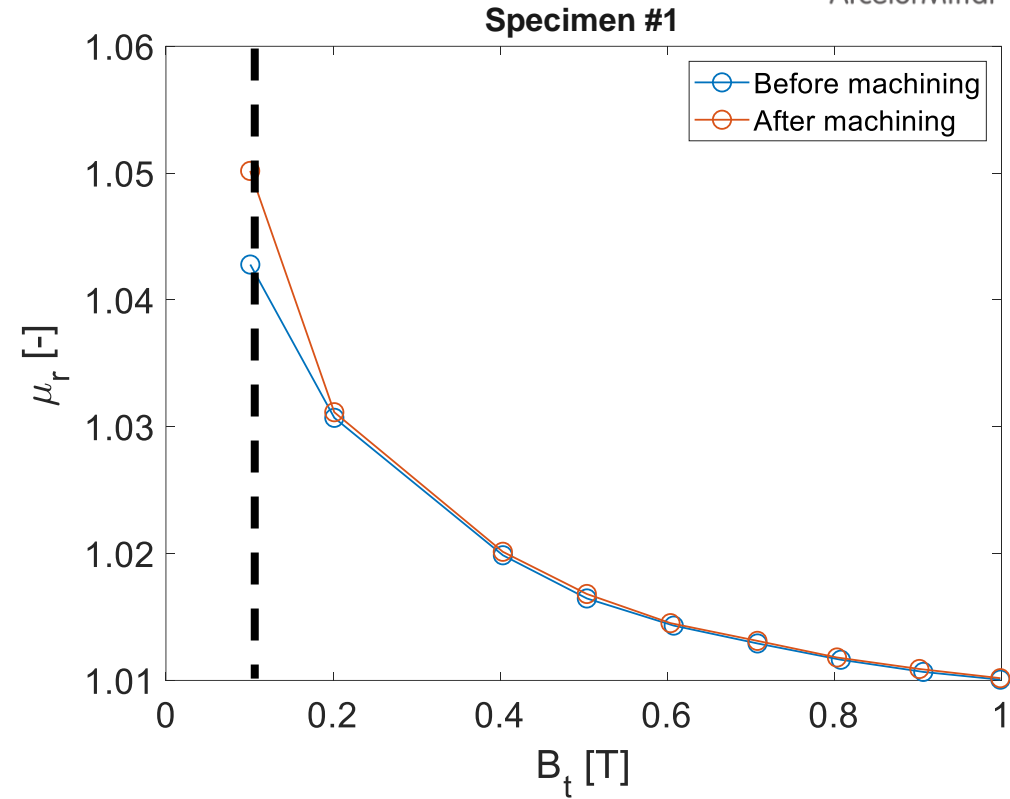


# Case study: a low permeability steel for CMS HG-CAL



## Foerster Magnetoscop 1.069

- Specimen #1: **1.0495** before machining, 1.0519 after machining
- Specimen #2: **1.0275**



## SSM

- Specimen #1 at 0.1 T: **1.04279** before machining, 1.05017 after machining
- Specimen #2 at 0.1 T: **1.02319**

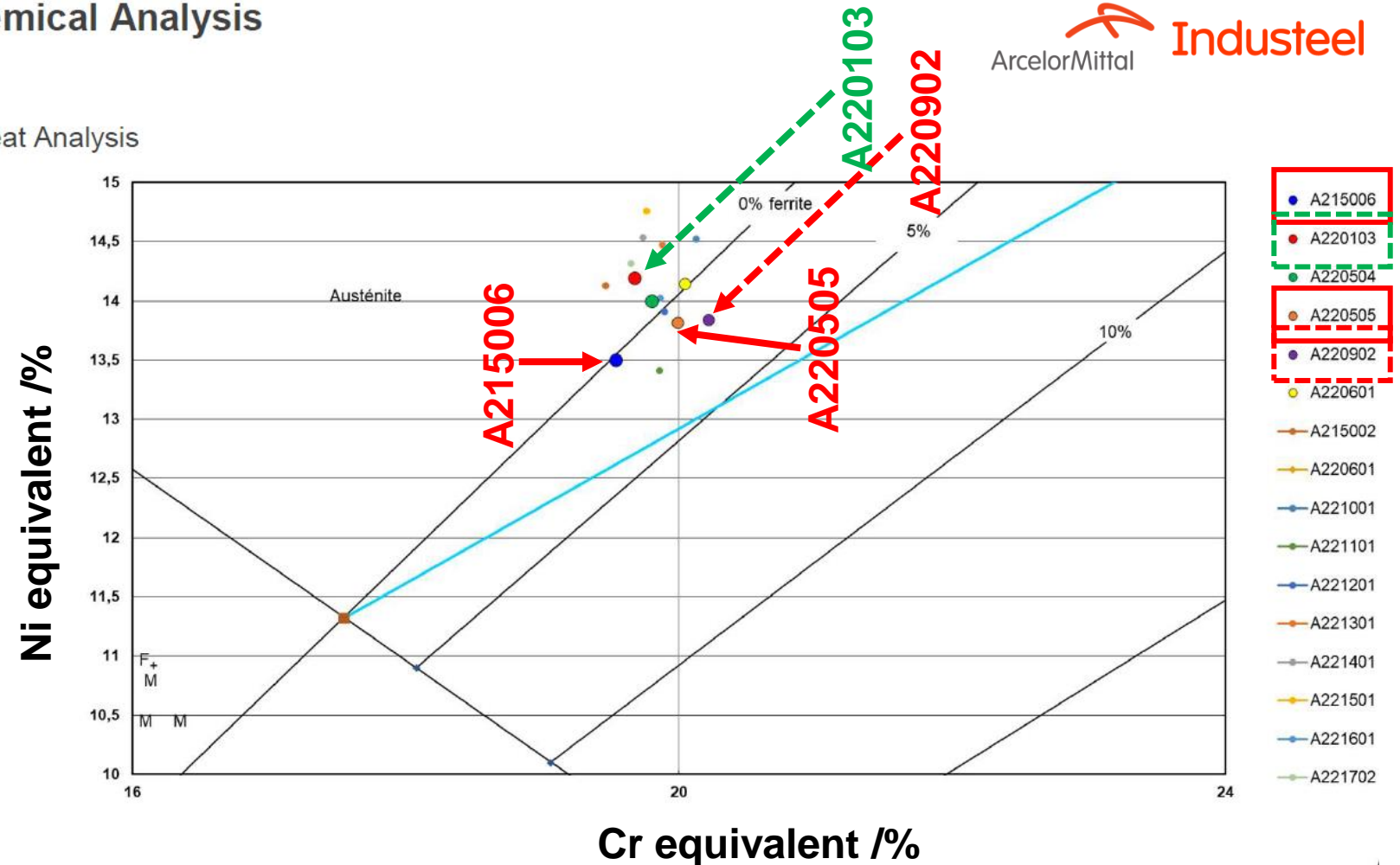
# Case study: a low permeability steel for CMS HG-CAL

## Chemical Analysis



### Heat Analysis

- The origin of higher permeability of some plates can be traced back to the respective heats and their position in the solidification diagram



Page 9  
10/08/2022  
Confidential



# Non-magnetic materials, phase transformations and measurements



Designation: A342/A342M – 14

## Standard Test Methods for Permeability of Weakly Magnetic Materials<sup>1</sup>

This standard is issued under the fixed designation A342/A342M; the number immediately following the designation indicates the year of original adoption or, in the case of revision, the year of last revision. A number in parentheses indicates the year of last reapproval. A superscript epsilon ( $\epsilon$ ) indicates an editorial change since the last revision or reapproval.

This standard has been approved for use by agencies of the U.S. Department of Defense.

1.2 The test methods covered are as follows:

1.2.1 *Test Method 1—Fluxmetric Method* is suitable for materials with permeabilities between 1.0 and 4.0. This method permits the user to select the magnetic field strength at which the permeability is to be measured.

1.2.2 *Test Method 2—Permeability of Paramagnetic Materials* has been eliminated as an acceptable method of test.

1.2.3 *Test Method 3—Low Mu Permeability Indicator* is suitable for measuring the permeability of a material as “less than” or “greater than” that of calibrated standard inserts with permeability between 1.01 and 6.0, as designated for use in a Low-Mu Permeability Indicator.<sup>3</sup> In this method, a small volume of specimen is subjected to a local magnetic field that varies in magnitude and direction, so it is not possible to specify the magnetic field strength at which the measurement is made.

1.2.4 *Test Method 4—Flux Distortion* is suitable for materials with permeability between 1.0 and 2.0. In this method, a small volume of specimen is subjected to a local magnetic field

that varies in magnitude and direction, so it is not possible to specify the magnetic field strength at which the measurement is made.<sup>4</sup>

1.2.5 *Test Method 5—Vibrating Sample Magnetometry* is suitable for materials with permeability between 1.0 and 4.0. This test method permits the user to select the magnetic field strength at which the permeability is to be measured.

1.3 Materials typically tested by these methods such as austenitic stainless steels may be weakly ferromagnetic. That is, the magnetic permeability is dependent on the magnetic field strength. As a consequence, the results obtained using the different methods may not closely agree with each other. When using Methods 1 and 5, it is imperative to specify the magnetic field strength or range of magnetic field strengths at which the permeabilities have been determined.

# Non-magnetic materials, phase transformations and measurements



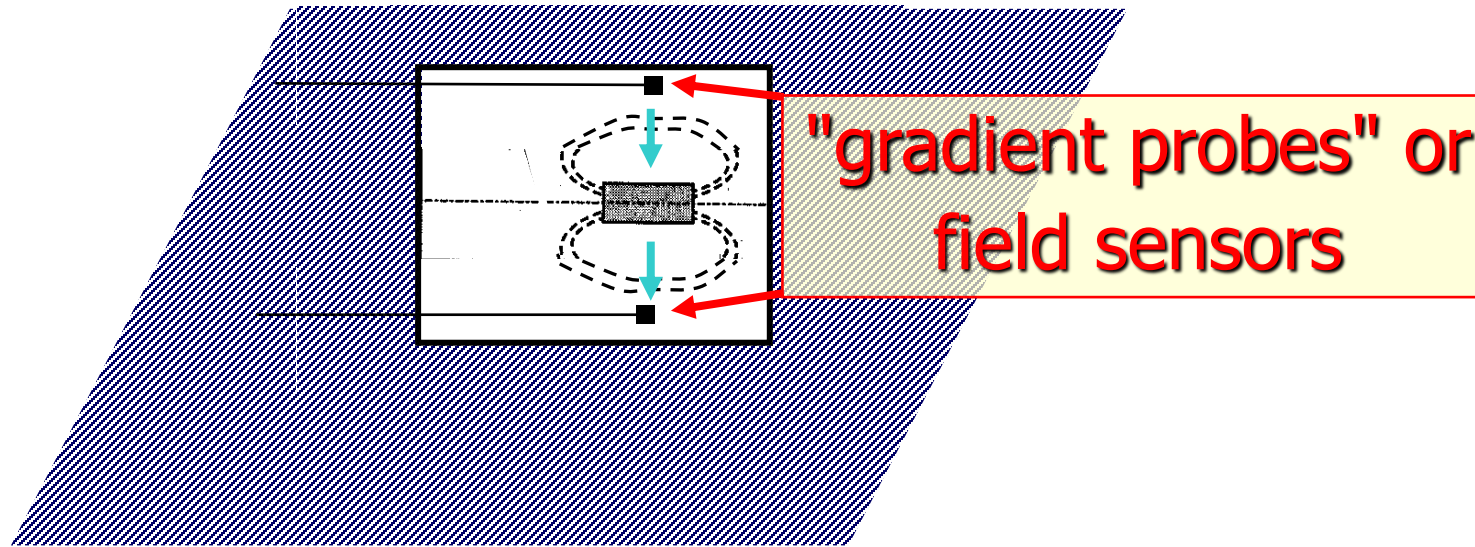
Courtesy of Foerster, MAGNETOSCOP® 1.069

- Measurement of material permeabilities
- Materials to be measured should be thicker than 8 mm
- Measurements of materials thinner than 8 mm by stacking two pieces
- Air gap between two pieces as small as possible
- Flat area not less than 20 mm in diameter
- Radius of curvature not be less than 40 mm
- Otherwise permeability less than actual value

**Available in EN-MME-MM and TE-MS**



# Non-magnetic materials, phase transformations and measurements

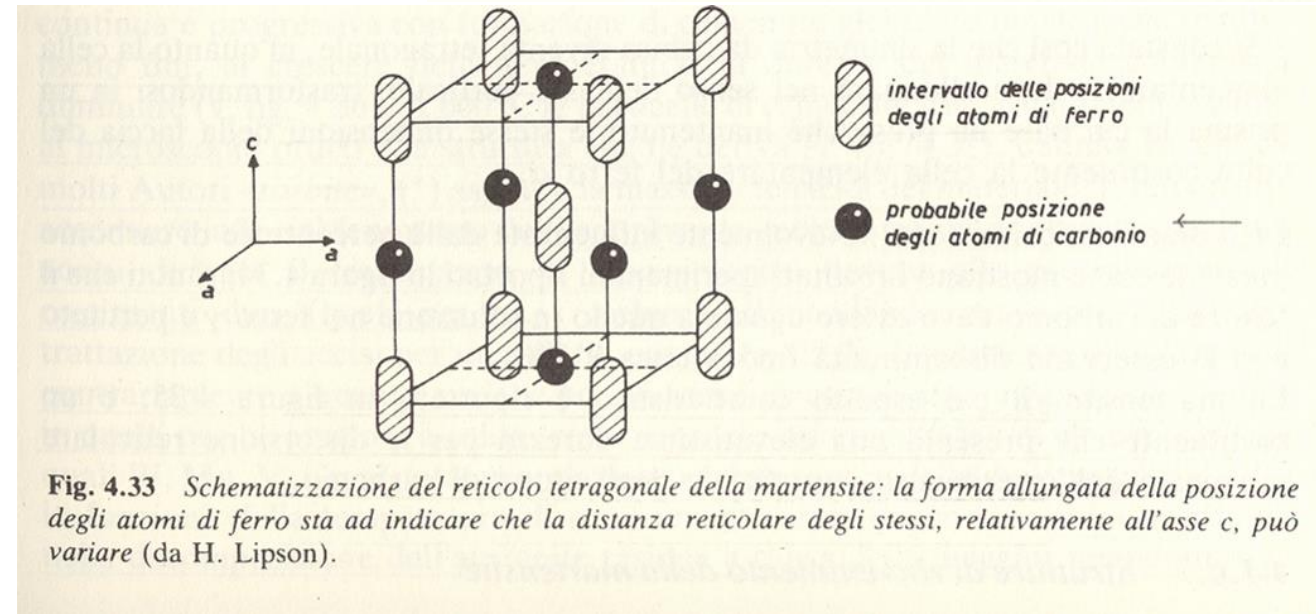
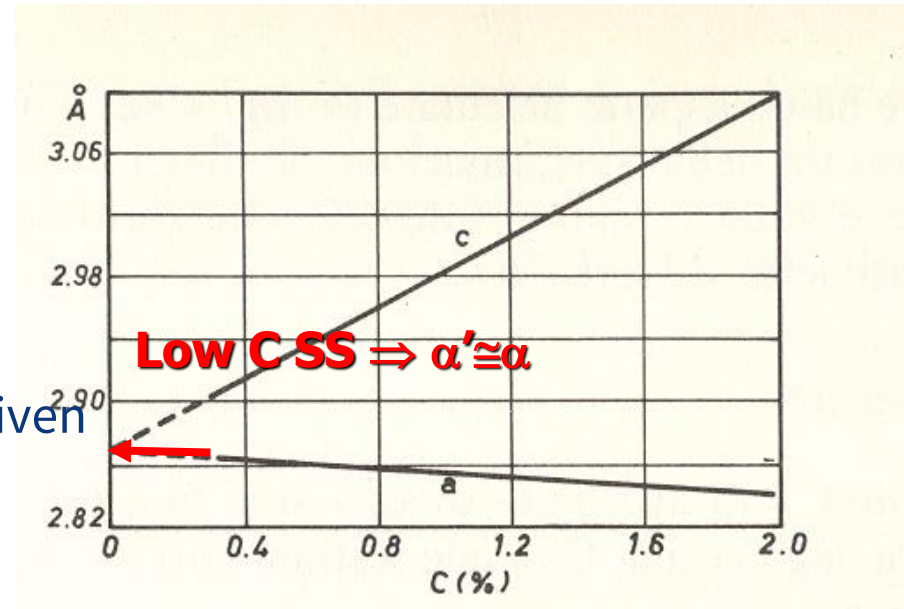


- "Based on a cylindrical permanent magnet
- A gradient probe is placed on either side of the cylindrical magnet in the plane perpendicular to the cylinder axis at the center of the permanent magnet
- If the cylindrical magnet is placed on a material whose permeability is greater than 1, there is a minute displacement of the magnetic zero of the cylindrical magnet towards the material on which the magnet has been placed
- In the lower permeability ranges this displacement is a measure of the permeability of the material"

# Non-magnetic materials, phase transformations and measurements

Martensitic transformation have two forms:

- $\gamma \Rightarrow \alpha'$ , b.c.c., relevant for magnetic purposes
- occurs spontaneously on cooling and/or is strain induced under a given temperature
- cause of loss of non-magnetism in austenitic stainless steels





# Non-magnetic materials, phase transformations and measurements

Table 9.2 Temperature equivalents for calculation of stability parameters of austenitic steels.

Investigator (Year)	Temperature Equivalent									Comments, Composition Range (wt.%)
	Base	Cr	Ni	Mn	Si	C	N	Mo	Other	
f.c.c. → b.c.c. (T <sub>ms</sub> ), cooling Eichelman, Hull (1953)										10-18Cr, 6-12 Ni, Mn, 0.3-26Si, 0.004-0.01-0.06N
Monkman, Cuff, Grant (1957)	1455	-36.7	-56.7							49 alloys: 11-19Cr, 5-13Ni,
Hammond (1963)	1105	-29	-39							
Andrews (1965)	273	-12.1	-17.7							
Hull (1973)	1755	-47	-59							
f.c.c. → b.c.c. (T <sub>md</sub> ), deformation Angel (1954) Hull (1973)										0-0.15N, 0-6Mo, Co, 0-211
Williams, Williams, Capellaro (1976)	686	-6	-25	-16	+21	-222	-222	-11		50% α' compression, 60 alloys: 12-24Cr, 0-22Ni, 0-20Mn, 0-4Si, 0-0.1C, 0-0.15N, 0-6Mo, Co 45% compression, 2.5% α', 25 alloys: 12-25Cr, 9-20Ni, 1-2Mn, 0.1-0.6Si, 0.04- 0.25C, 0.01-0.1N, 0.6-2.8Mo

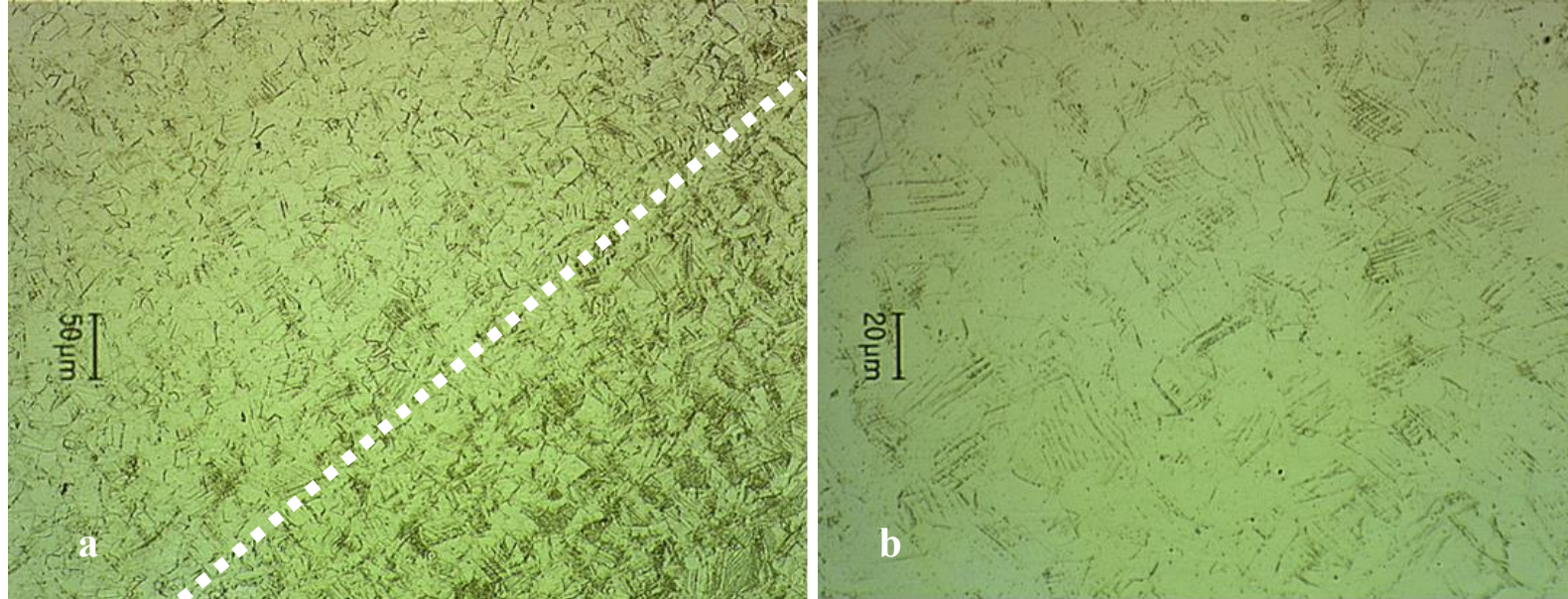
T<sub>ms</sub>, temperature of spontaneous γ ⇒ α' martensitic transformation

Transformation (T<sub>ms</sub>, T<sub>md</sub>, calculated):

- General purpose 304L (1.4307, X2CrNi18-9) ⇒ T<sub>ms</sub> = 280 K, T<sub>md</sub> = 346 K
- High alloy 304L (1.4306, X2CrNi19-11) ⇒ T<sub>ms</sub> = 140 K, T<sub>md</sub> = 320 K
- Prototype HG-CAL 304L (as above) ⇒ T<sub>ms</sub> = 76 K, T<sub>md</sub> = 305 K
- CERN store 316LN (1.4429, X2CrNiMoN17-13-3) ⇒ T<sub>ms</sub> = n.a., T<sub>md</sub> = 240 K
- Beam screen P506 grade ⇒ T<sub>ms</sub> = n.a., T<sub>md</sub> = 36 K

T<sub>md</sub>, temperature of strain induced γ ⇒ α' martensitic transformation

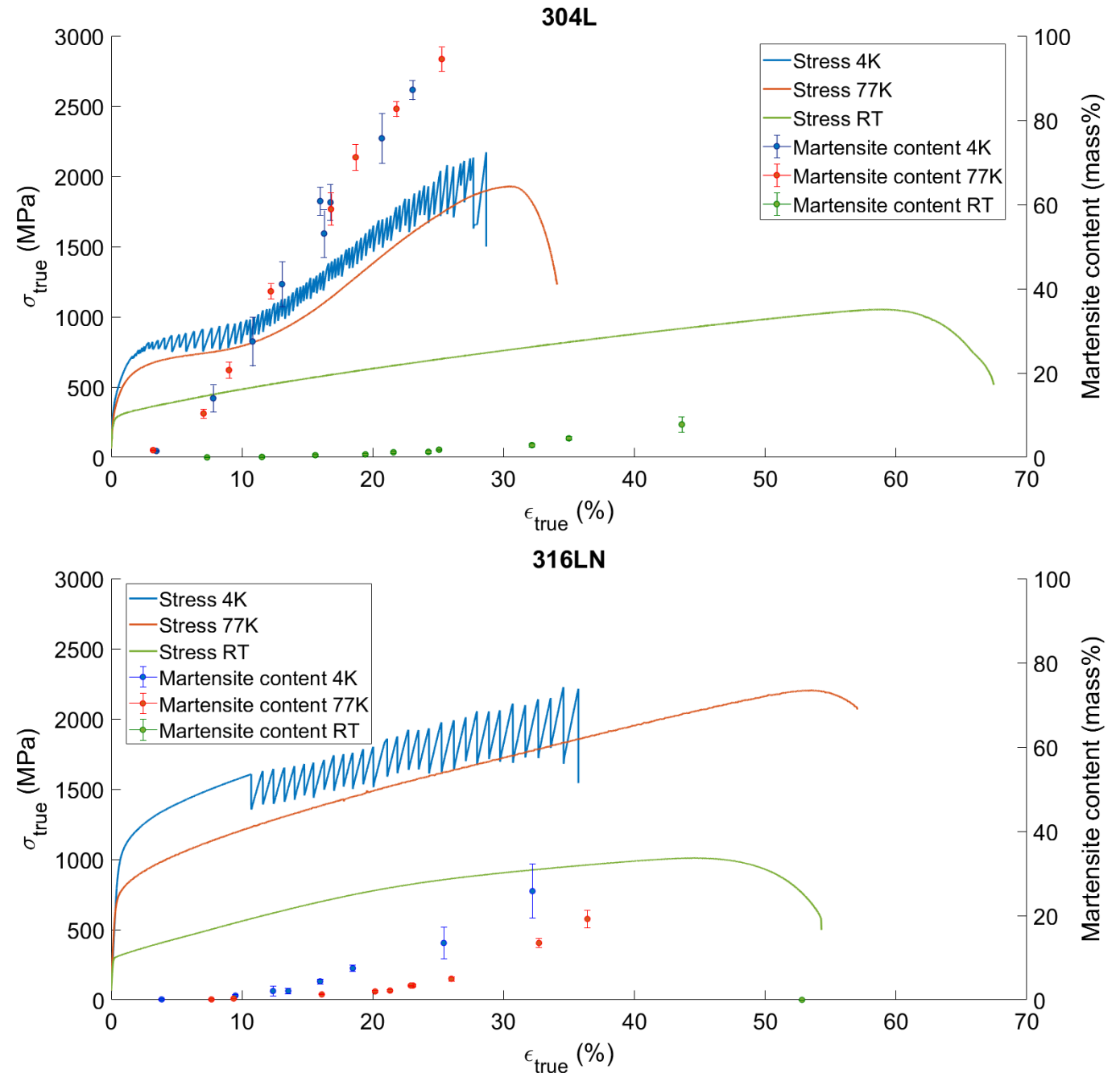
# Non-magnetic materials, phase transformations and measurements



Partially transformed austenite of an AISI 316L austenitic stainless steel sample strained 6.5% at 4.2 K. Martensite is concentrated in bands (under the white boundary in Fig. a), developing during tensile deformation. A detail of the austenite-martensite microstructure is shown in Fig. b (see also C. GARION, S. SGOBBA, B. SKOCZEN, Constitutive modelling and identification of parameters of the plastic strain-induced martensitic transformation in 316L stainless steel at cryogenic temperatures, *International Journal of Plasticity*, 22 (2006) 1234-1264)



# Non-magnetic materials, phase transformations and measurements



# Non-magnetic materials, phase transformations and measurements

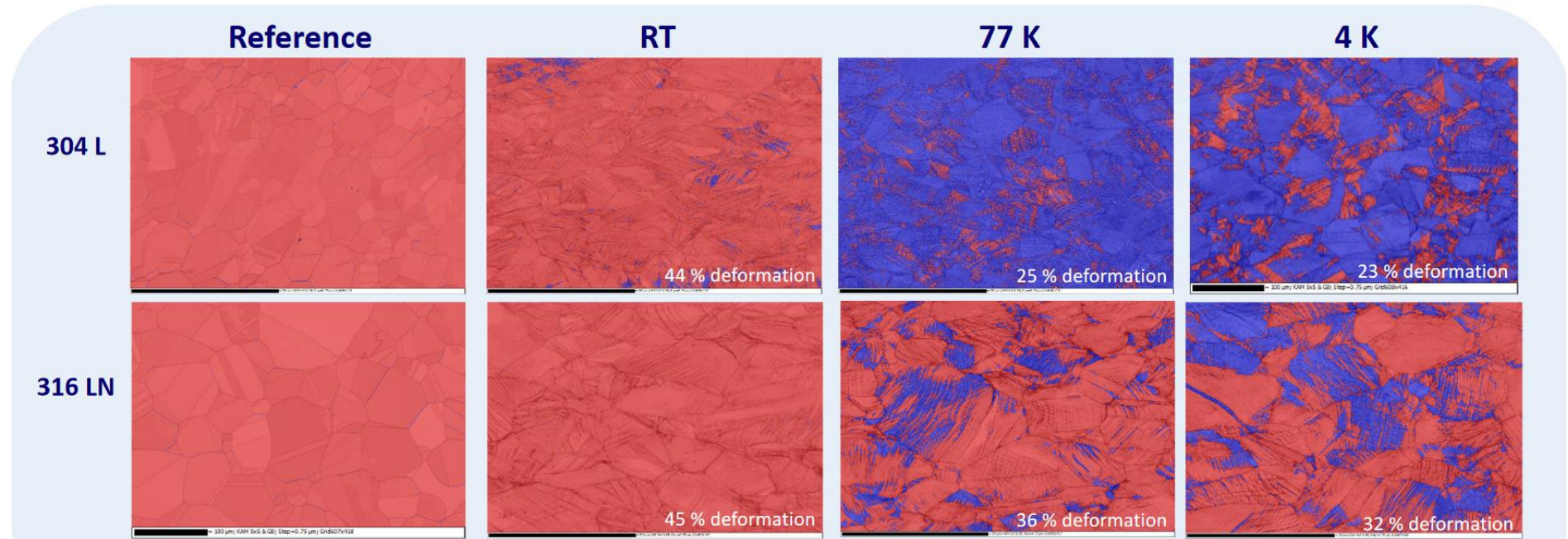


Figure 2 - EBSD phase map and band contrast map on 304 L and 316 LN samples at different temperatures. Colour code: martensite (Fe BCC) appears in blue, while the austenite (Fe FCC) is shown in red

Quantitative assessment through EBSD techniques associated to SEM

Flow and fracture of austenitic stainless steels at cryogenic temperatures, P. Fernández-Pisón, J.A. Rodríguez-Martínez, E. García-Tabarés, I. Avilés-Santillana, S. Sgobba (Dec, 2021), Engineering Fracture Mechanics 258 (2021) 108042, <https://doi.org/10.1016/j.engfracmech.2021.108042>

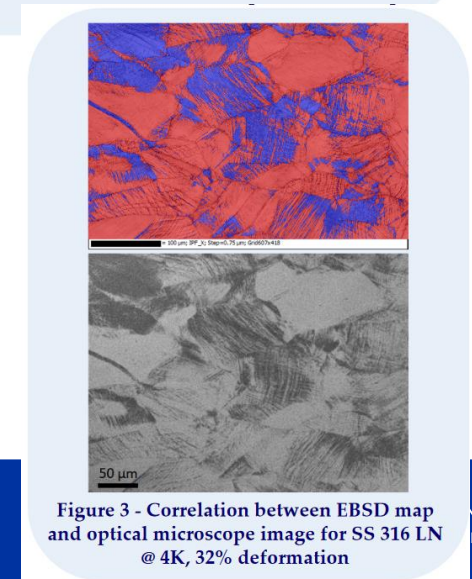
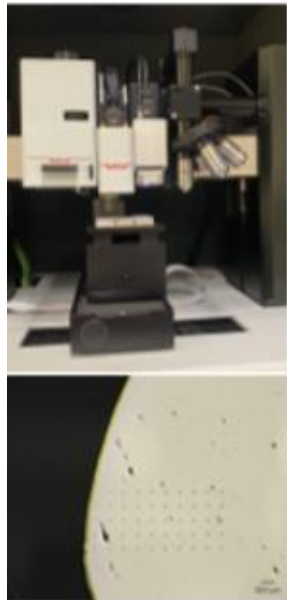


Figure 3 - Correlation between EBSD map and optical microscope image for SS 316 LN @ 4K, 32% deformation



# Non-magnetic materials, phase transformations and measurements

## New Technologies

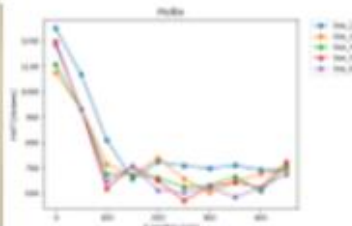


### Instrumented indentation testing

Anton Paar Step 700:

- Nanoindentation (1 – 100 mN)
- Microindentation (up to 30 N)
- Atomic force microscope

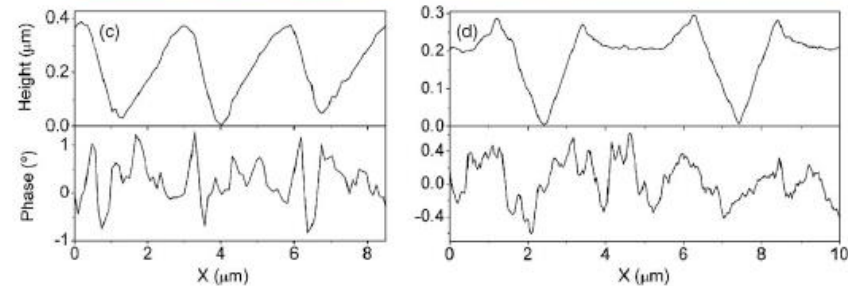
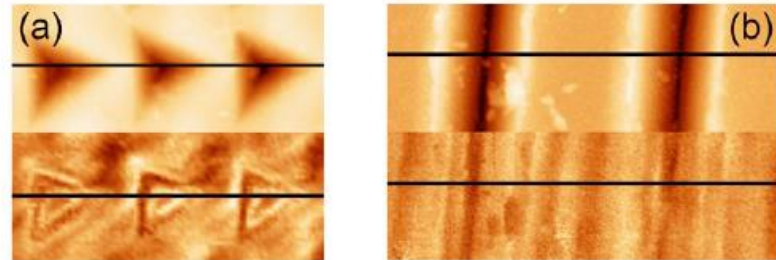
30/09/2022



Commercial AISI 316 austenitic stainless steel, annealed at 1400 K

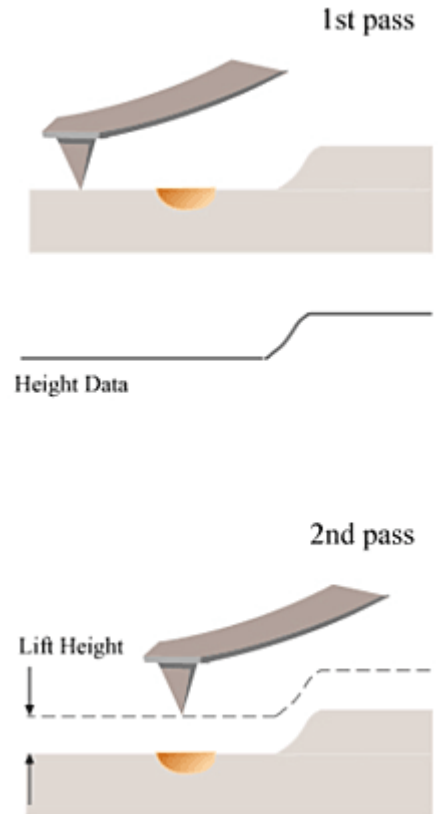
AFM  $\Rightarrow$

MFM  $\Rightarrow$

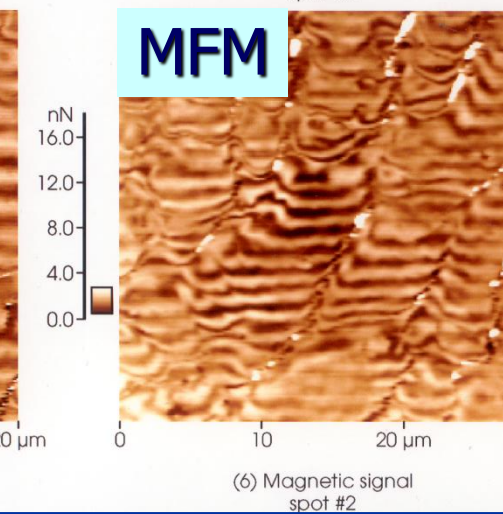
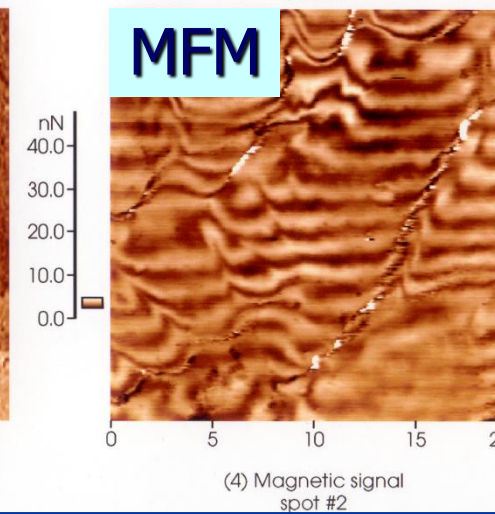
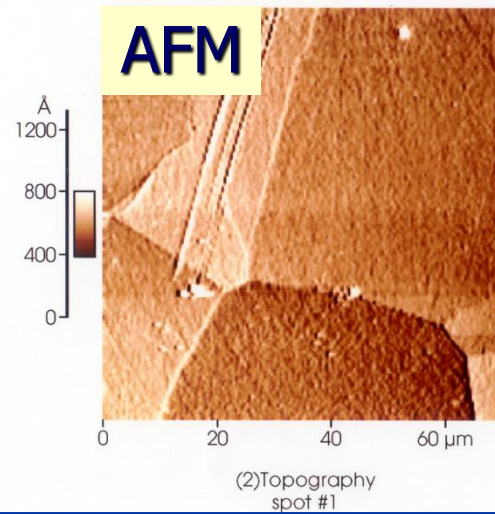
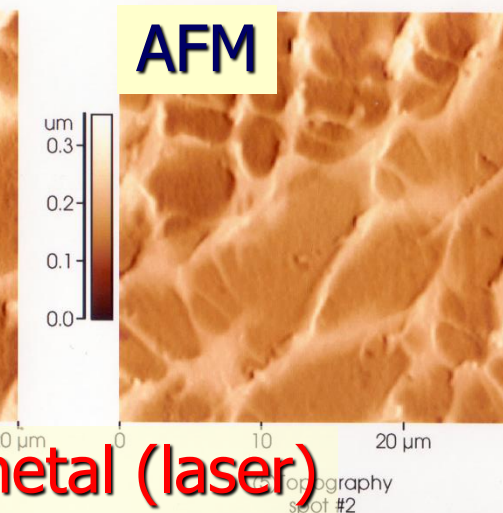
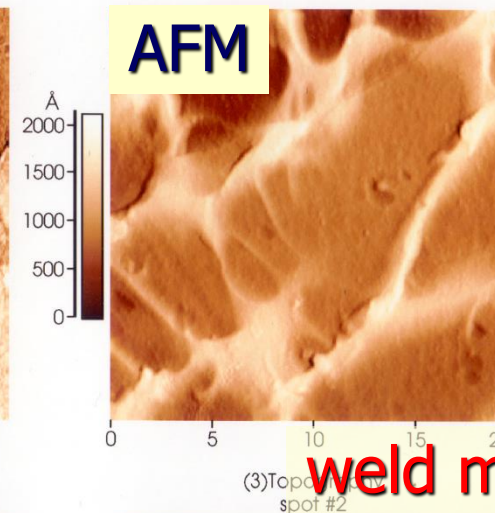
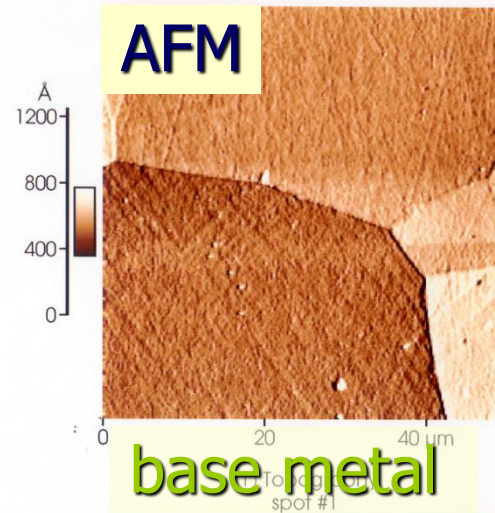
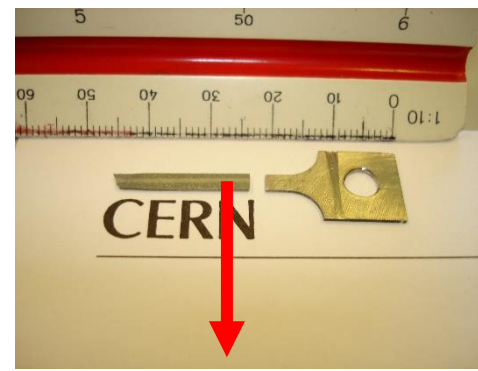


J. Sort et al., Applied Physics Letters **89** (2006) 032509

Now available in EN-MME-MM!



# Non-magnetic materials, phase transformations and measurements





# 6. Conclusions

- Materials for an accelerator or fusion magnet part is not a mere "chemical composition" or a designation
  - specification
  - steelmaking
  - definition and extent of the controls
  - certification
  - quantity
  - price
- Application of extensive “state of the art” NDT techniques
  - stainless steels for vacuum, cryogenic and/or structural applications 100 % examined during production and at reception
- Low T and/or non-magnetism of components require special care and extensive cross check by advanced testing
- Irreprocheable production route, starting from steelmaking
- Advanced grades, even if standardized, imply extensive prior R&D. However, several grades already qualified for cryogenic structural applications

Concutiat motus, quis fulgura ducat hiatus,  
 Unde torrens nubes, quo lumine floreat arcus;  
 Hæc mihi quaerenti, si quid deprendere veri  
 Mens valet, expediat. Lapis est cognomine Magnes,  
 Decolor, obscurus, vilis: non ille repexam  
 Cæsariem regum, nec candida virginis ornat  
 Colla, nec insigni splendet per cingula morsu:  
 Sed nova si nigri videas miracula saxi,  
 Tum pulchros superat cultus, et quidquid Eois  
 Indus litoribus rubra scrutatur in alga.

demens de la terre, et, déchirant la nue, en fait jaillir  
 l'éclair, et fait gronder la foudre; quelle lumière colore  
 l'arc d'Iris: moi aussi, je cherche à résoudre ces grands  
 problèmes; et, si votre esprit peut entrevoir la vérité,

**There is a stone called the magnet; black, dull, and common. It does not adorn the braided hair of kings nor the snowy necks of girls, nor yet shine in the jewelled buckles of warriors' belts. But consider the marvellous properties of this dull-looking stone and you will see that it is of more worth than lovely gems and any pearl sought of Indian amid the seaweed on the Red Sea's shores.**

## C. L. F. PANCKOUCKE.

Exegi monumentum ære perennius.  
 (Hon., *Od. lib. iiii, ode 30.*)



TOME SECOND

PARIS

C. L. F. PANCKOUCKE, ÉDITEUR

OFFICIER DE L'ORDRE ROYAL DE LA LÉGIION D'HONNEUR

RUE DES POITEVINS, N. 14

M DCCC XL



# Spare slides

# Case study: a low permeability steel for CMS HG-CAL

Plate	Averaging method	Localised measurement (0.1 T)	Volumetric measurement (0.1 T)
281142T – 120 mm (original thickness)	Layers in thickness, including rolled faces	1.0495	
	Mid thickness x 3 and rolled faces	1.0522	1.04279
	Mid thickness x 1 and rolled faces	1.0306	
281142T – 100 mm (machined down to)	Layers in thickness, including rolled faces	1.0519	
	Mid thickness x 3 and rolled faces	1.0545	1.05017
	Mid thickness x 1 and rolled faces	1.0343	
281146 – 60 mm	Layers in thickness, including rolled faces	1.0275	
	Mid thickness x 3 and rolled faces	1.0508	1.02319
	Mid thickness x 1 and rolled faces	1.0343	



# Case study: a low permeability steel for CMS HG-CAL

Two heats affected by ferrite precipitation resulting in 3:2 average  $\mu_r > 1.05$  on some plates

Plate n°	Heat n°	Thickness / mm	Average permeability 3:2	Average permeability 1:2
293526-11	A215006	60	1.065	1.040
293525-11	A215006	60	1.052	1.032
293525-21	A215006	60	1.044	1.027
293526-21	A215006	60	1.060	1.037
281146.11	A220902	60	1.038	1.025
281148.11	A220902	60	1.029	1.019
287355.11	A220103	80	1.006	1.008
287354.11	A220103	80	1.009	1.007
281088-11	A220505	80	1.088	1.057
280472.11	A215006	80	1.036	1.022
280470.11	A215006	80	1.021	1.014

A215006

A215006 A220505

# Case study: a low permeability steel for CMS HG-CAL

Plate n°	Heat n°	Thickness / mm	Average permeability 3:2	Average permeability 1:2
293526-11	A215006	60	1.065	1.040
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281146.11	A220902	60	1.038	1.025
281148.11	A220902	60	1.029	1.019
287355.11	A220103	80	1.006	1.008
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280472.11	A215006	80	1.036	1.022
280470.11	A215006	80	1.021	1.014

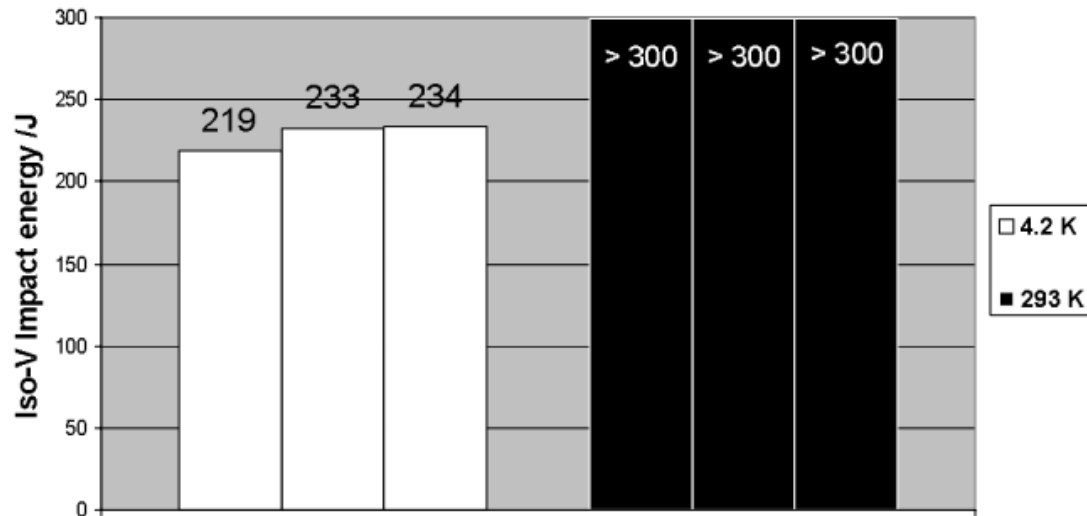
A220103  
Low  $\mu_r$  heat

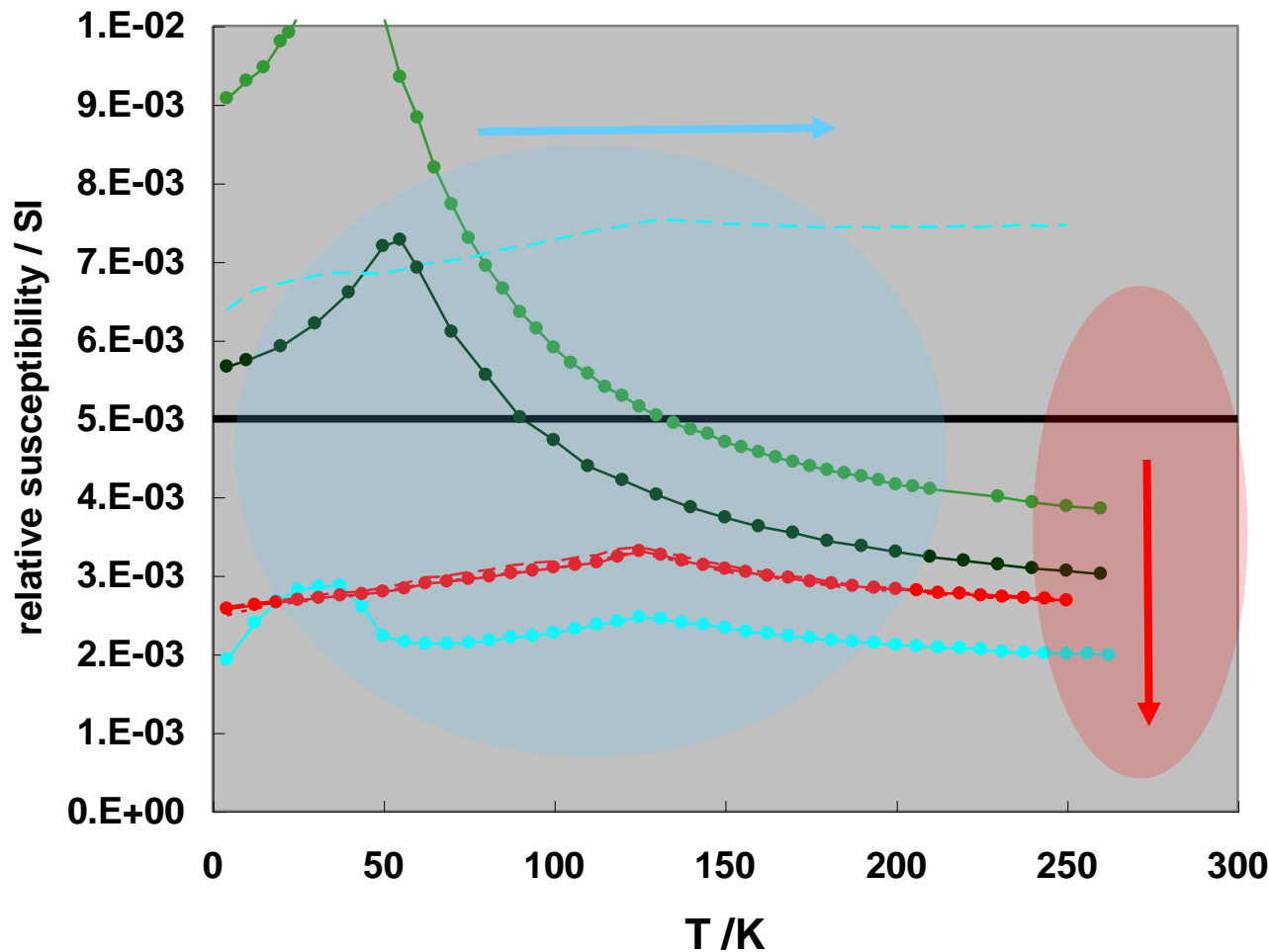


# Materials for magnets: austenitic stainless steels

Temperature [K]	direction	$\sigma_{y0.2}$ [MPa]	$\sigma_r$ [MPa]	$\epsilon_r$ [%]
293	Long.	$450 \pm 10$	$835 \pm 20$	$61 \pm 3$
	Transv.	$440 \pm 10$	$825 \pm 5$	$45 \pm 4$
77	Long.	$1180 \pm 10$	$1760 \pm 20$	$57 \pm 4$
	Transv.	$1120 \pm 50$	$1715 \pm 100$	$45 \pm 3$
4.2	Long.	$1620 \pm 50$	$2115 \pm 60$	$18 \pm 3$
	Transv.	$1700 \pm 100$	$2105 \pm 90$	$15 \pm 4$

P506: remarkable combination of strength, ductility and impact toughness at 4.2 K





## Materials for magnets: austenitic stainless steels

**1. Lowest possible magnetic susceptibility at RT**

**2. Highest possible temperature of antiferromagnetic transition**

**3. Total absence of  $\delta$ -ferrite in parent metal and welds  $\Rightarrow$  high purity of the parent metal**



# Materials for magnets: austenitic stainless steels

P. Libeyre et al., "Addressing the Technical Challenges for the Construction of the ITER Central Solenoid", IEEE Trans. Appl. Superconductivity, Vol. 22, (2012), p. 4201104

**Tie plates, LKB and upper structures:**  
**FXM-19 (Nitronic®50)**

**Yield strength required at RT when using JK2LB for the jackets: 230 MPa**  
→ F316LN excluded (205 MPa)  
→ FXM-19 (Nitronic®50) selected (345 MPa)

- Initially considered both **forged or welded solution**
- Steelmaking process → ESR or VAR as needed, 3D forging
- Microinclusion cleanliness → As per ASTM E 45 method D  $\leq 0.5$  (A and C),  $\leq 1.0$  (B and D)
- Macroinclusions not permitted → ESR generally applied to meet this and other requirements
- Fineness of microstructure → grain size number  $\geq 3$  as per ASTM E112
- Homogeneity of microstructure → grain size within the range of  $\pm 1$  number
- **Limited ferrite content → less than number 1.0**
- **Limited magnetic permeability →  $\leq 1.03$**
- Internal soundness → 100 % UT
- Mechanical strength and fracture toughness specified, including at cryogenics →

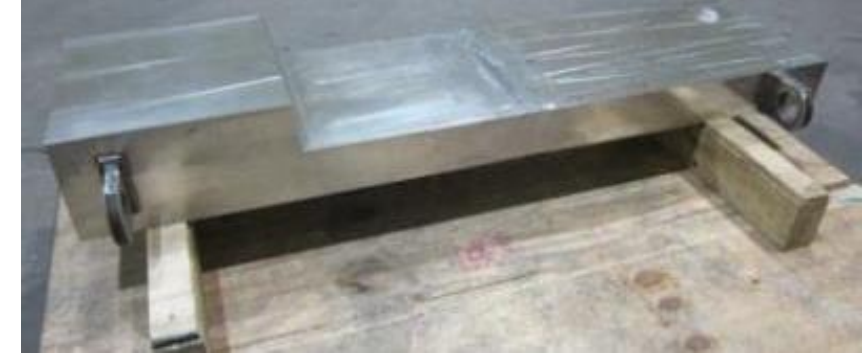
S.A.E. Langeslag et al., "Extensive Characterisation of Advanced Manufacturing Solutions for the ITER Central Solenoid Pre-compression System," Fusion Eng. Des. (2015), <https://doi.org/10.1016/j.fusengdes.2015.06.007>

R. Walsh et al., "Welded tie plate feasibility study for ITER central solenoid structure", Adv. Cryo. Eng, AIP Conference Proceedings 1574, 16 (2014); <https://doi.org/10.1063/1.4860598>

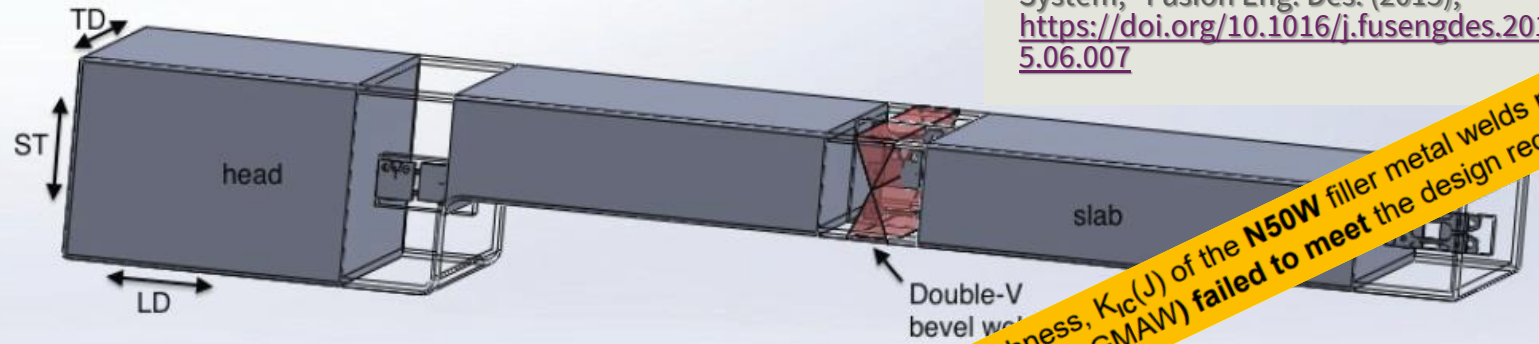
# Materials for magnets: austenitic stainless steels

## Requirements:

- Very high strength and toughness at 4 K:  $R_{p0.2} > 1200 \text{ MPa}$ ;  $R_m > 1600 \text{ MPa}$ ;  $K_{IC} > 130 \text{ MPa}\sqrt{\text{m}}$
- Fatigue resistance at cryogenic temperature
- Larger thermal contraction than central solenoid jackets (JK2LB) for an effective pre-compression at 4 K



## Single piece welded and forged solution

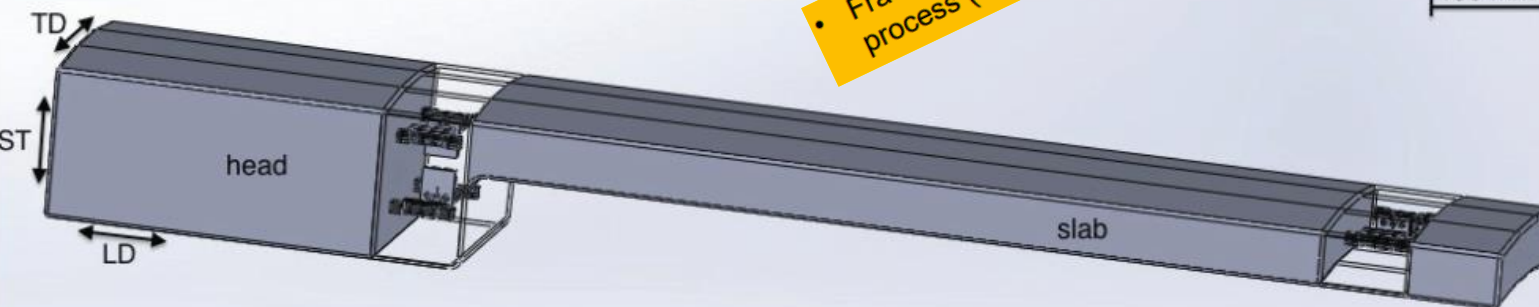


Head- and slab-forgings, joined together via a Gas Metal Arc Welding (GMAW) process.

- The cryogenic mechanical properties of the weld did not fulfil the requirements

R. Walsh et al., "Welded tie plate feasibility study for ITER central solenoid structure", Adv. Cryo. Eng, AIP Conference Proceedings 1574, 16 (2014); <https://doi.org/10.1063/1.4860598>

• Fracture toughness,  $K_{IC}(J)$  of the N50W filler metal welds made with either process (GTAW or GMAW) failed to meet the design requirements.



Produced by applying successive forging steps, followed by a final machining operation.

- Higher mechanical properties, including cryogenic fracture toughness and lower



# Our process

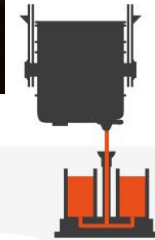
Courtesy Dr. N. Pauze,  
ArcelorMittal Industeel

Industeel

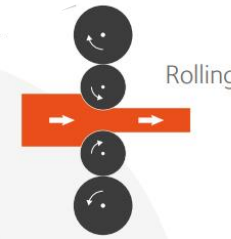
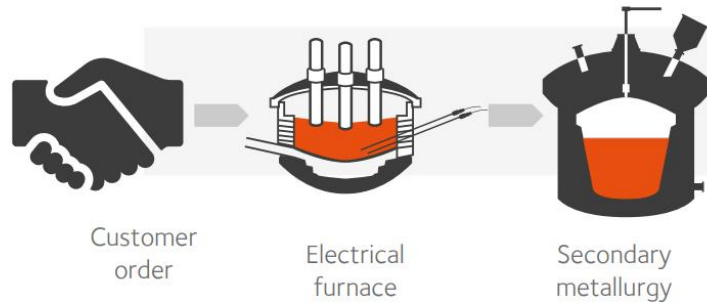
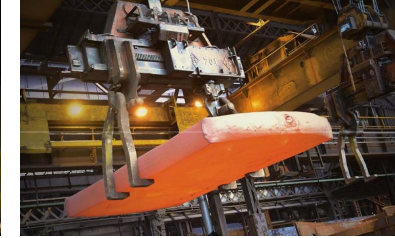
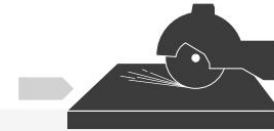
ArcelorMittal  
Industeel is an ArcelorMittal company



Casting ingot



Ingots  
preparation



Heat  
treatment



Shipment



Certification



Inspection and  
testing



Cutting



# Non-magnetic materials, phase transformations and measurements

All coefficient are negative:

- good rule: "the more alloying elements one uses (and can afford!), the more stable the austenite will be"
- 304L is the least stable among the alloys used at CERN.
- **1.4306** generally specified by CERN and preferred to **1.4307** (general purpose)
- total stability requires a specific alloy selection or design, see the (HL-) LHC beam screen example

Transformation ( $T_{ms}$ ,  $T_{md}$ , calculated):

- |   |  |
|---|--|
| • General purpose 304L (1.4307, X2CrNi18-9) $\Rightarrow$   | $T_{ms} = 280 \text{ K}, T_{md} = 346 \text{ K}$ |
| • High alloy 304L (1.4306, X2CrNi19-11) $\Rightarrow$       | $T_{ms} = 140 \text{ K}, T_{md} = 320 \text{ K}$ |
| • Prototype HG-CAL 304L (as above) $\Rightarrow$            | $T_{ms} = 76 \text{ K}, T_{md} = 305 \text{ K}$  |
| • CERN store 316LN (1.4429, X2CrNiMoN17-13-3) $\Rightarrow$ | $T_{ms} = \text{n.a.}, T_{md} = 240 \text{ K}$   |
| • Beam screen P506 grade $\Rightarrow$                      | $T_{ms} = \text{n.a.}, T_{md} = 36 \text{ K}$    |