



## 10<sup>th</sup> RD50 Workshop

*Radiation hard semiconductor devices for very high luminosity colliders*

*Vilnius, Lithuania, June 3-6, 2007*

### **Radiation-induced structure modification in monocrystalline silicon under high-energy ion irradiation**

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# Introduction

Surface morphology and structural modification in silicon after 1.2 GeV C<sup>6+</sup> and 21 MeV p<sup>+</sup> irradiation.

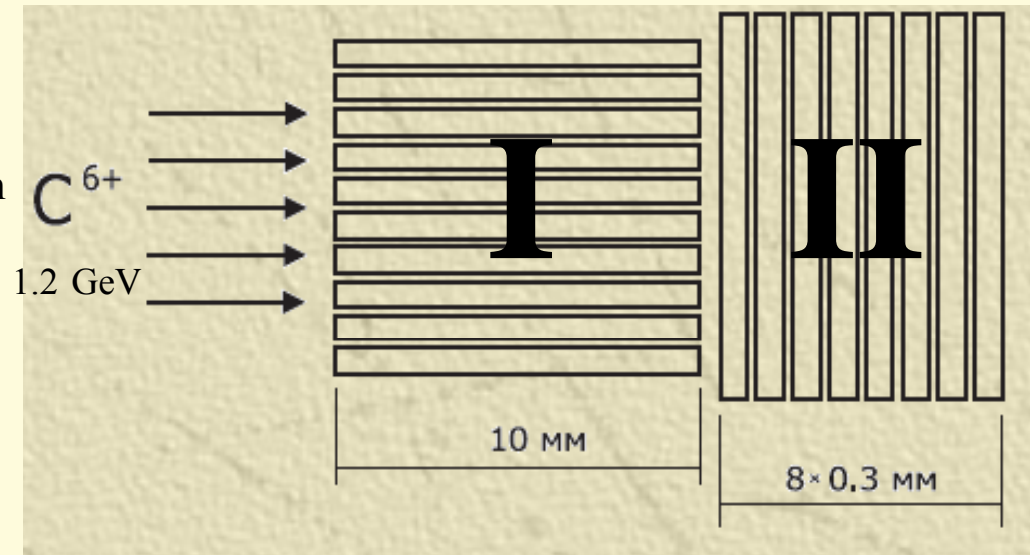
- Material: n-Si(100) kef-4.5 ( $\rho=4.5 \text{ } \Omega\text{cm}$ )
- Specimen size: 7x7x0.3 mm, 10x10x0.3 mm
- Ions: C<sup>6+</sup>, p<sup>+</sup>
- Fluences(cm<sup>-2</sup>): 6x10<sup>11</sup>, 1x10<sup>12</sup>, 5x10<sup>14</sup>, 5x10<sup>15</sup>, 1x10<sup>16</sup>
- Temperature: room, 100-300° C,
- AFM, XDS, XPS





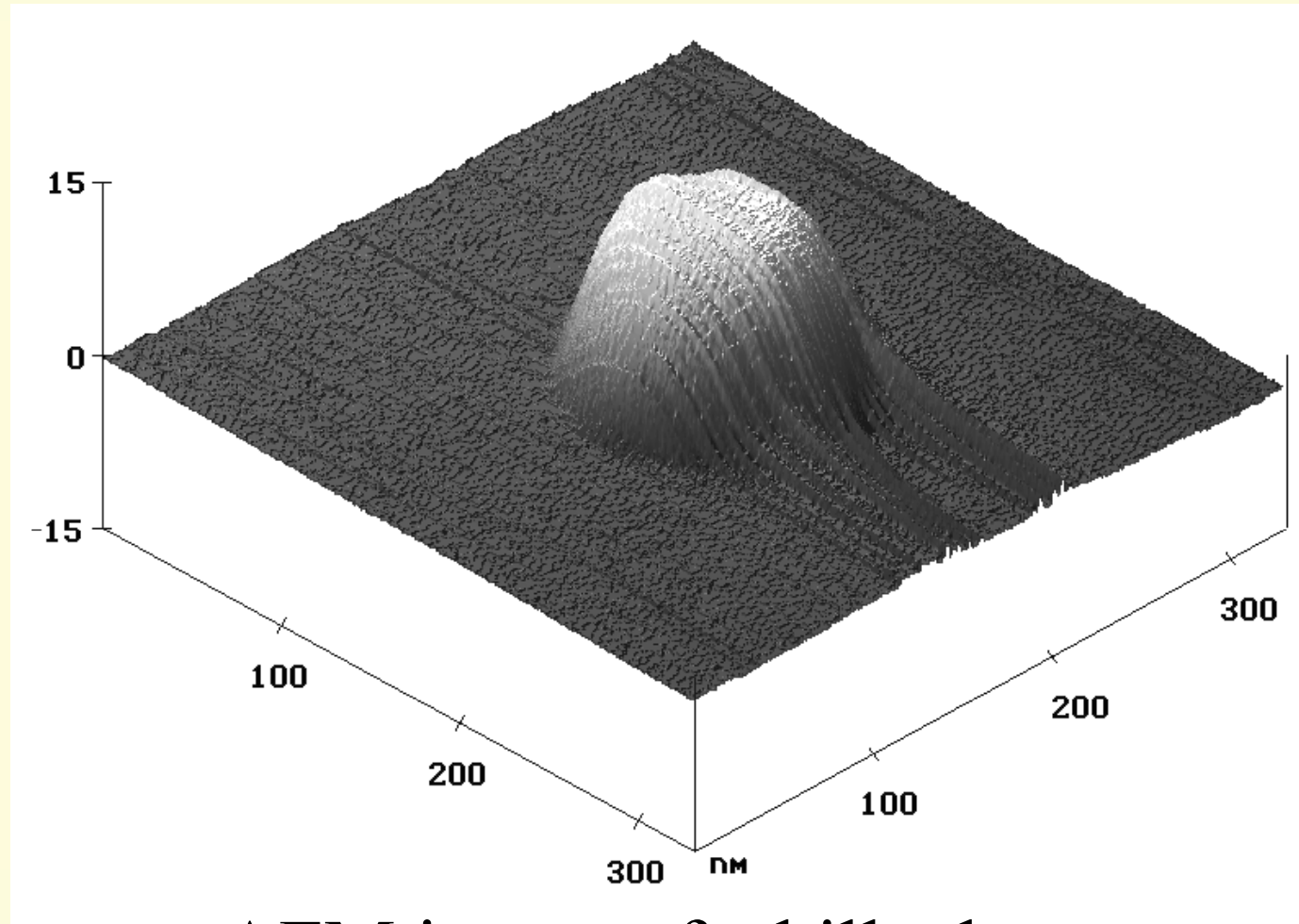
# Specimen irradiation

- Source parameters:
  - Ion  $C^{6+}$
  - Energy 1.2 GeV
  - projected range (SRIM) 14.2 mm
  - Fluence =  $1 \times 10^{12} \text{ cm}^{-2}$
- assembly parameters:
  - n-Si (100)
  - thickness 0.3 mm
  - $10 \times 8$  specimens





# Surface morphology

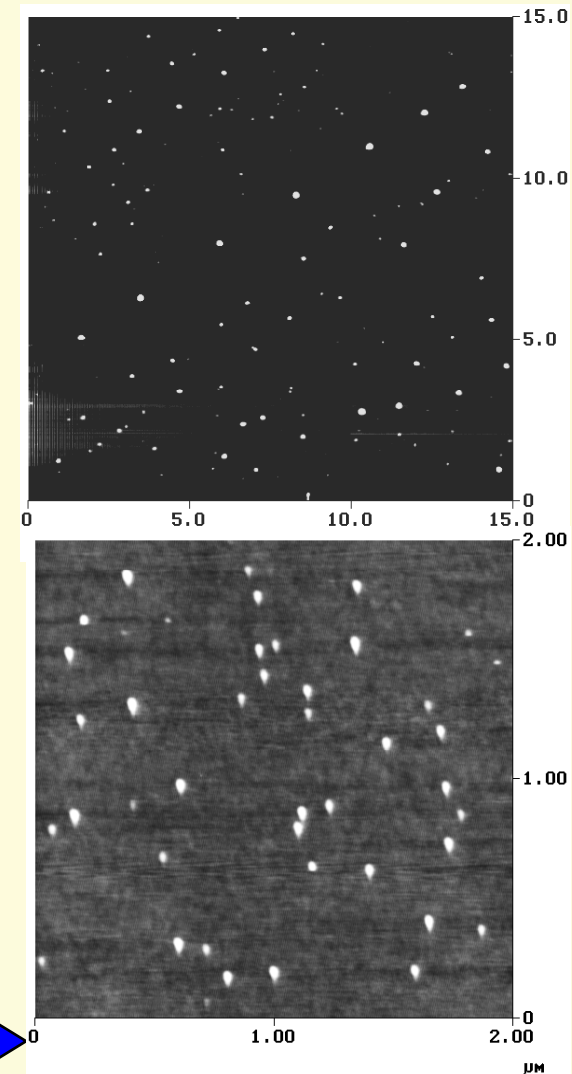


AFM image of a hillock



# Surface morphology

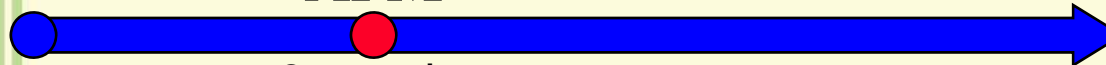
- specimens of type I (placed parallel to the beam)
  - hillocks form long straight lines
  - $d = 100 \pm 10$  nm
  - $h = 2-4$  nm
- specimens of type II (placed perpendicular to the beam)
  - hillocks placed randomly
  - $d = 200 \pm 50$  nm
  - $h = 2-4$  nm



Irr

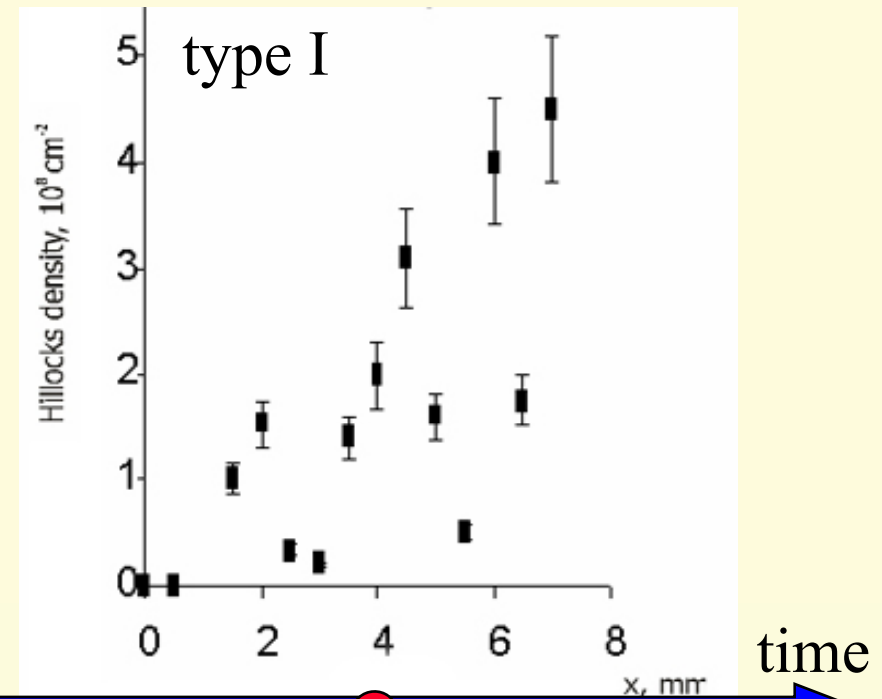
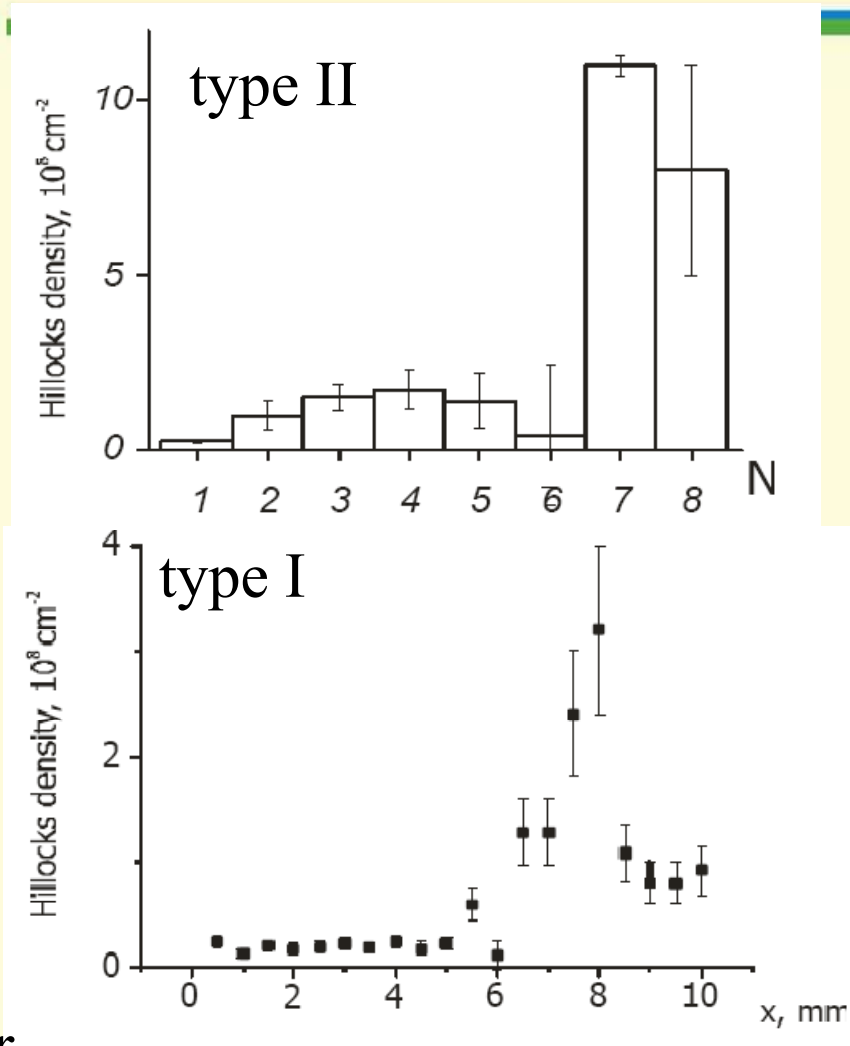
AFM

3 months





# Distribution of hillocks number density versus ion penetration depth



Irr.



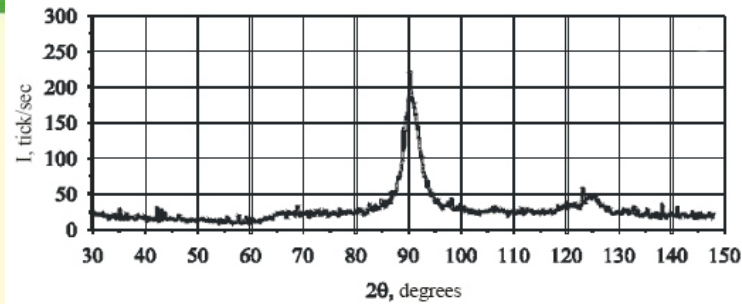
3 months

9 months

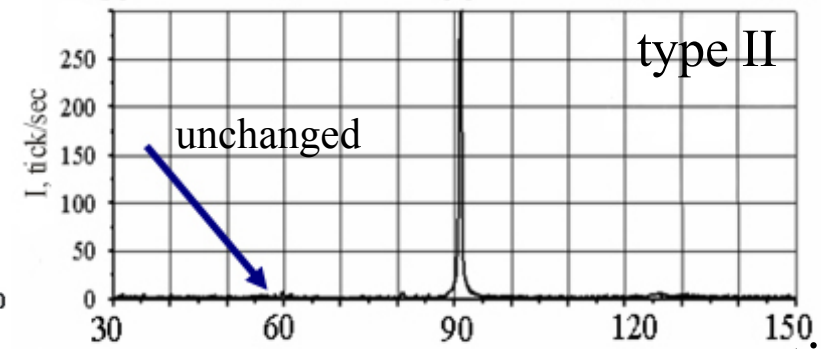
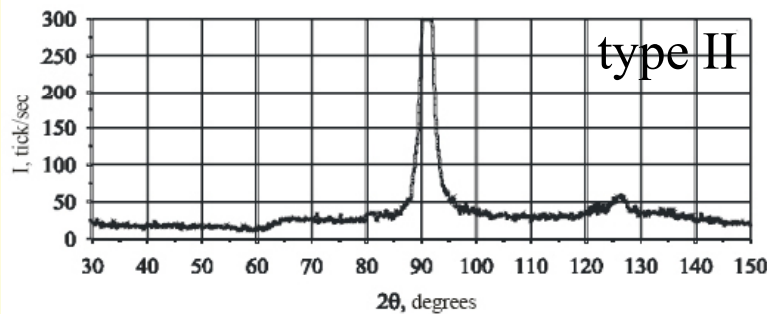
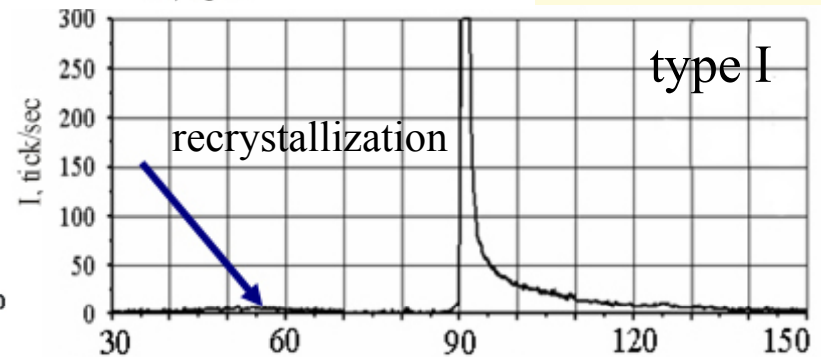
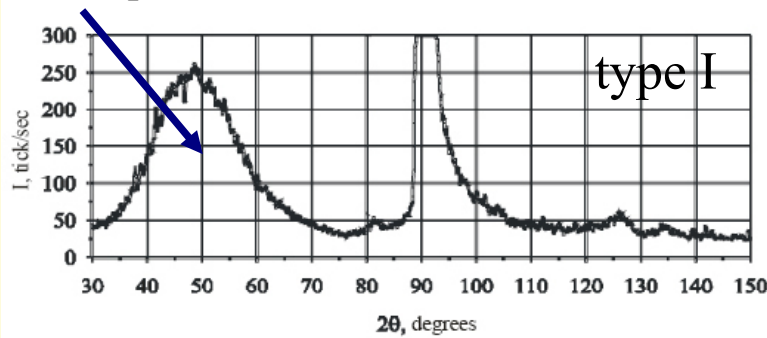


# X-Ray Diffraction

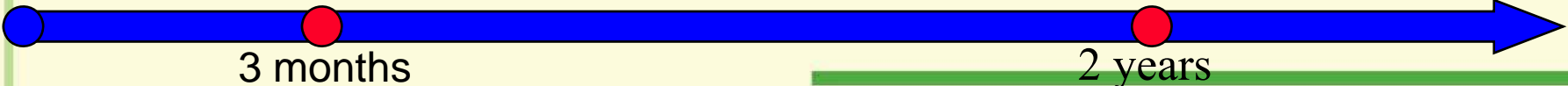
Unirradiated  
silicon



amorphization

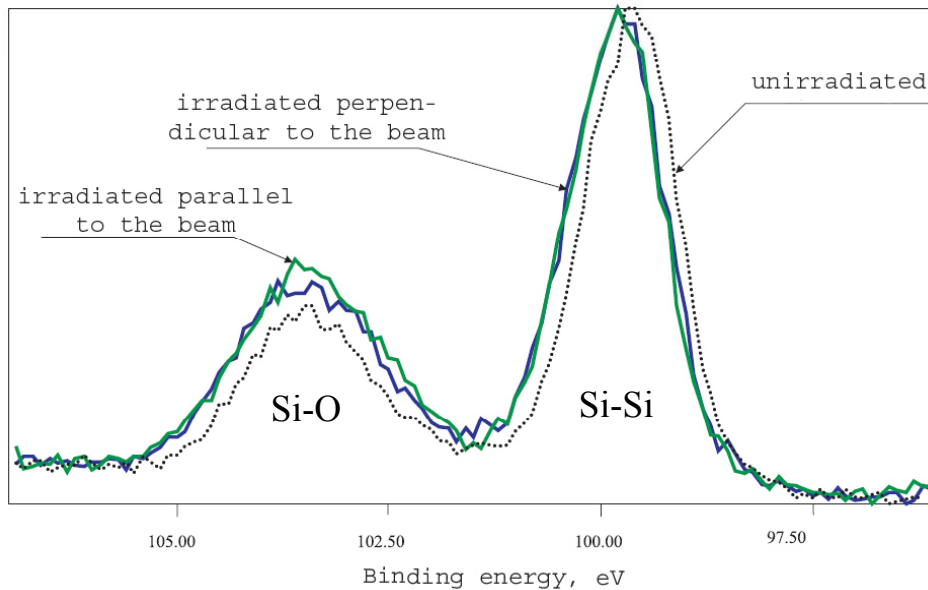


Irr.





# X-Ray Photoelectron Spectroscopy (XPS)



Uncompensated charge in bulk Si dissolves after irradiation  
Increasing of oxidized silicon concentration

## Results

- hillocks consists of oxidized silicon
- hillocks properties are the same (different ion energies)

Irr.







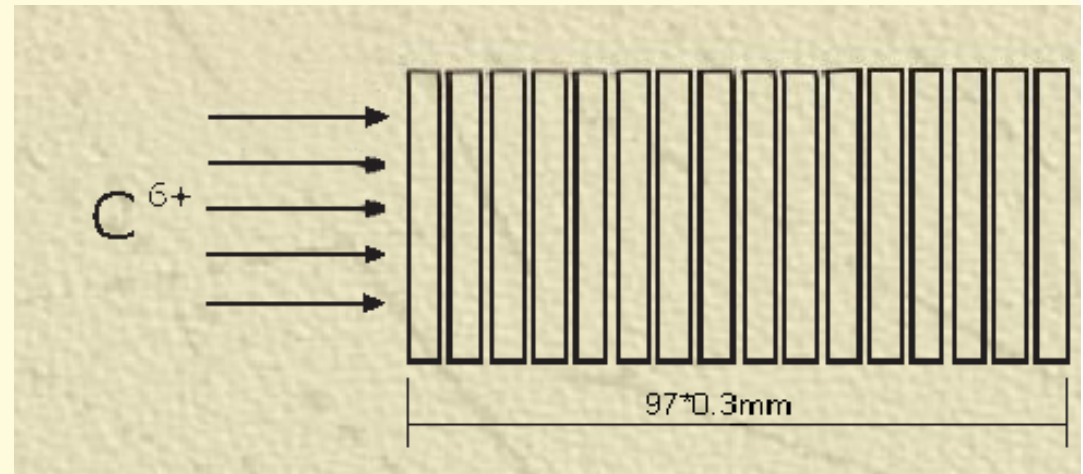
# Intermediate conclusions

- ✓ After irradiation a number of hillocks emerged on the surface.
- ✓ Hillocks density increases with time.
- ✓ If the ion beam is perpendicular to the specimen plane, only surface defects were found (no amorphization), but in the case of a parallel beam both changes in a surface and bulk can be observed.
- ✓ In a year amorphization dissolves (re-crystallization occurred)
- ✓ Nature of hillocks (oxides)
- Arisen questions:
  - Influence of irradiation parameters (energy, fluence) on hillocks formation
  - The role of temperature in hillocks formation.
  - Typical hillocks nucleation time
  - The role of “tracks” in hillocks formation



# Second experiment

- Source parameters:
  - Ion  $C^{6+}$
  - Energy 1.2 GeV
  - projected range  $14.2 \pm 0.6$  mm (approximately 47 plates)
  - Fluence =  $6 \times 10^{11}$   $cm^{-2}$
- Assembly parameters:
  - n-Si (100)
  - thickness 0.3 mm
  - 97 perpendicular specimens



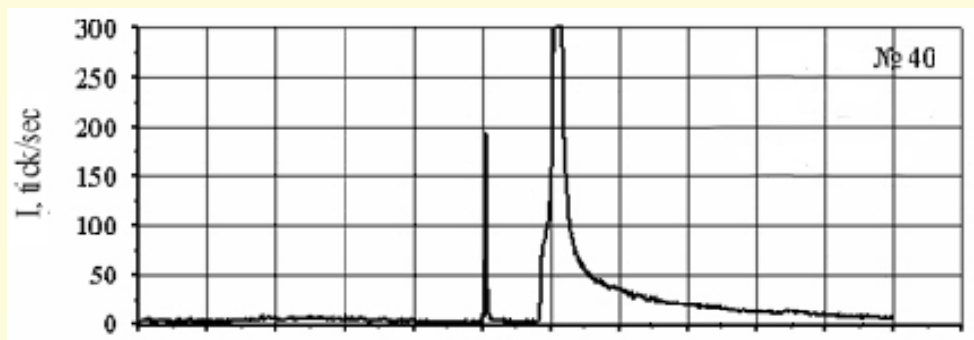


# Surface morphology disturbance

Specimens placed perpendicular to the beam

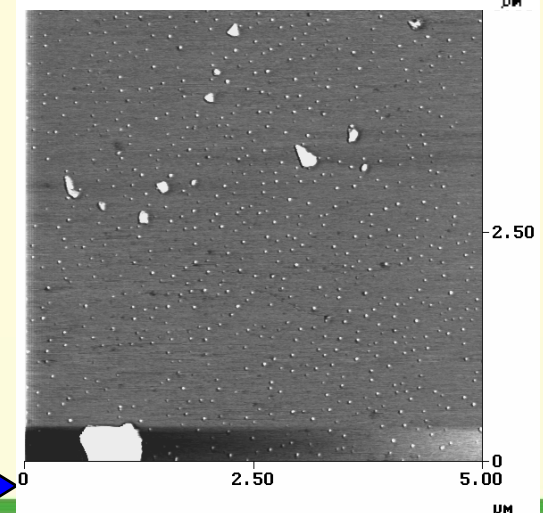
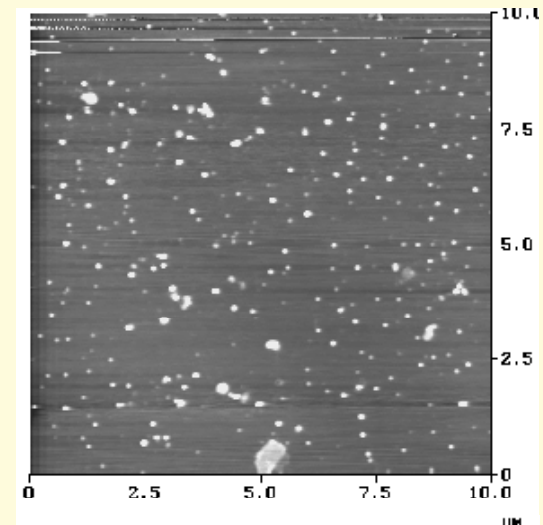
- hillocks
  - randomly placed
  - $d = 100$  nm
  - $h = 8-10$  nm
  - $n = 2 \times 10^9$  cm<sup>-2</sup>

- no amorphization occurred



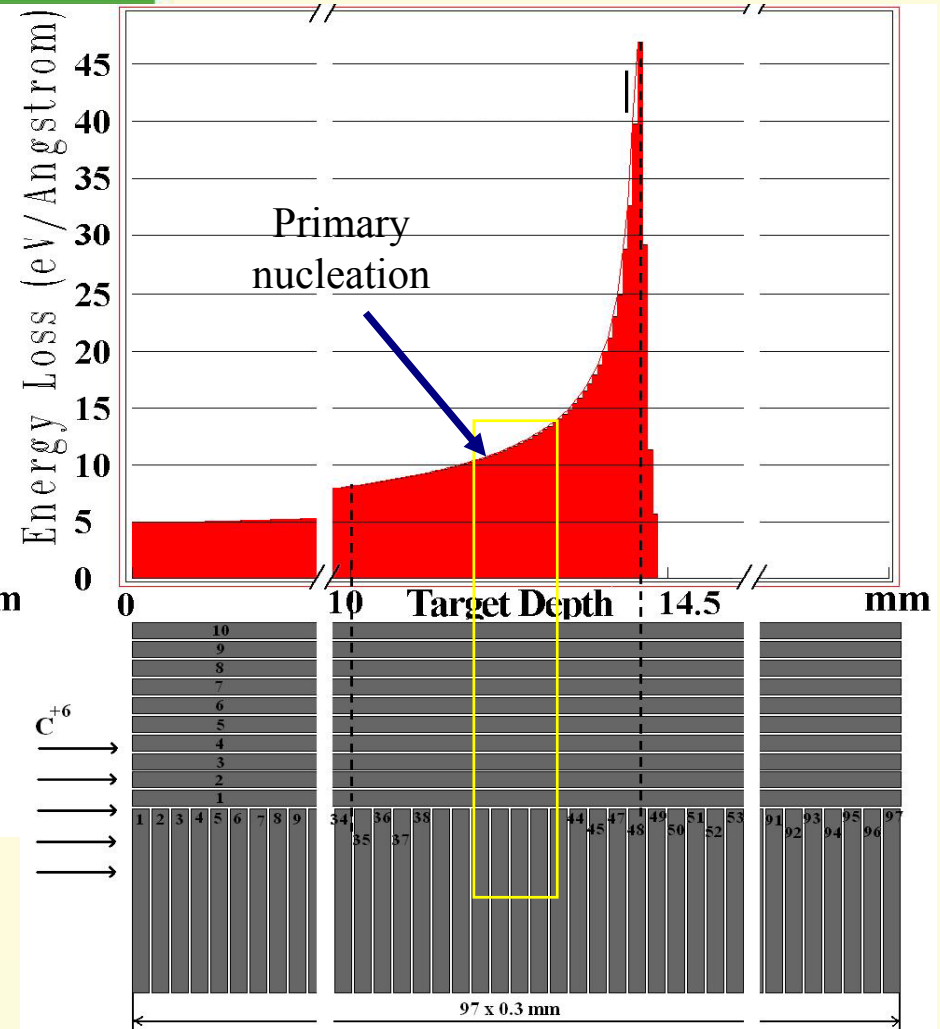
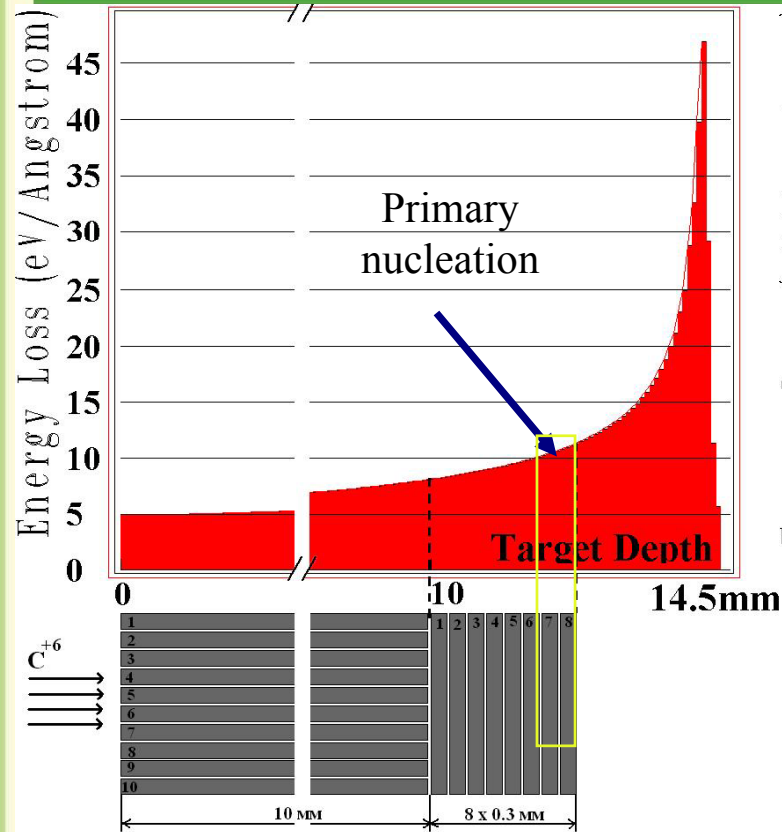
8 month

time





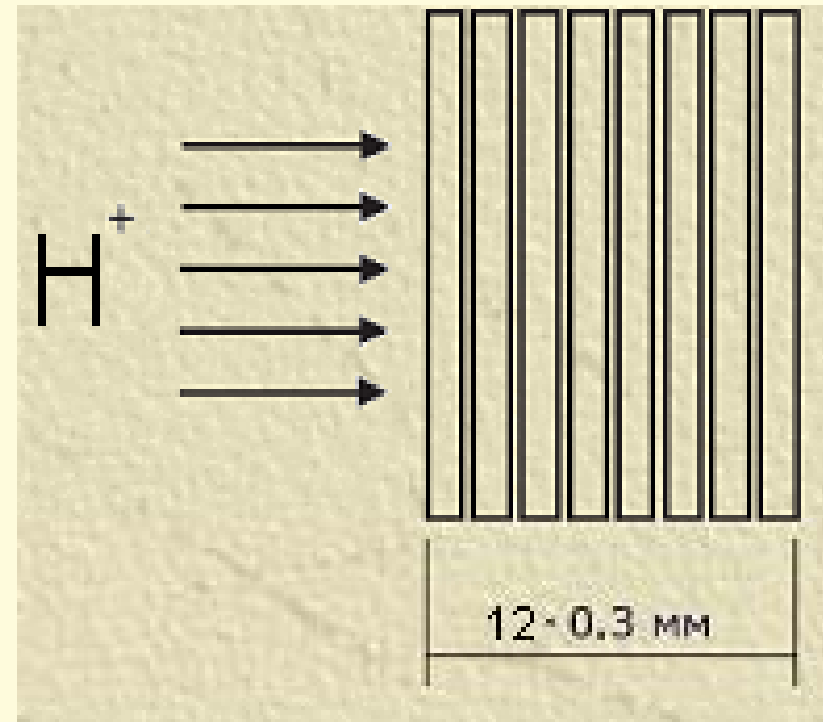
# Comparison of two experiments





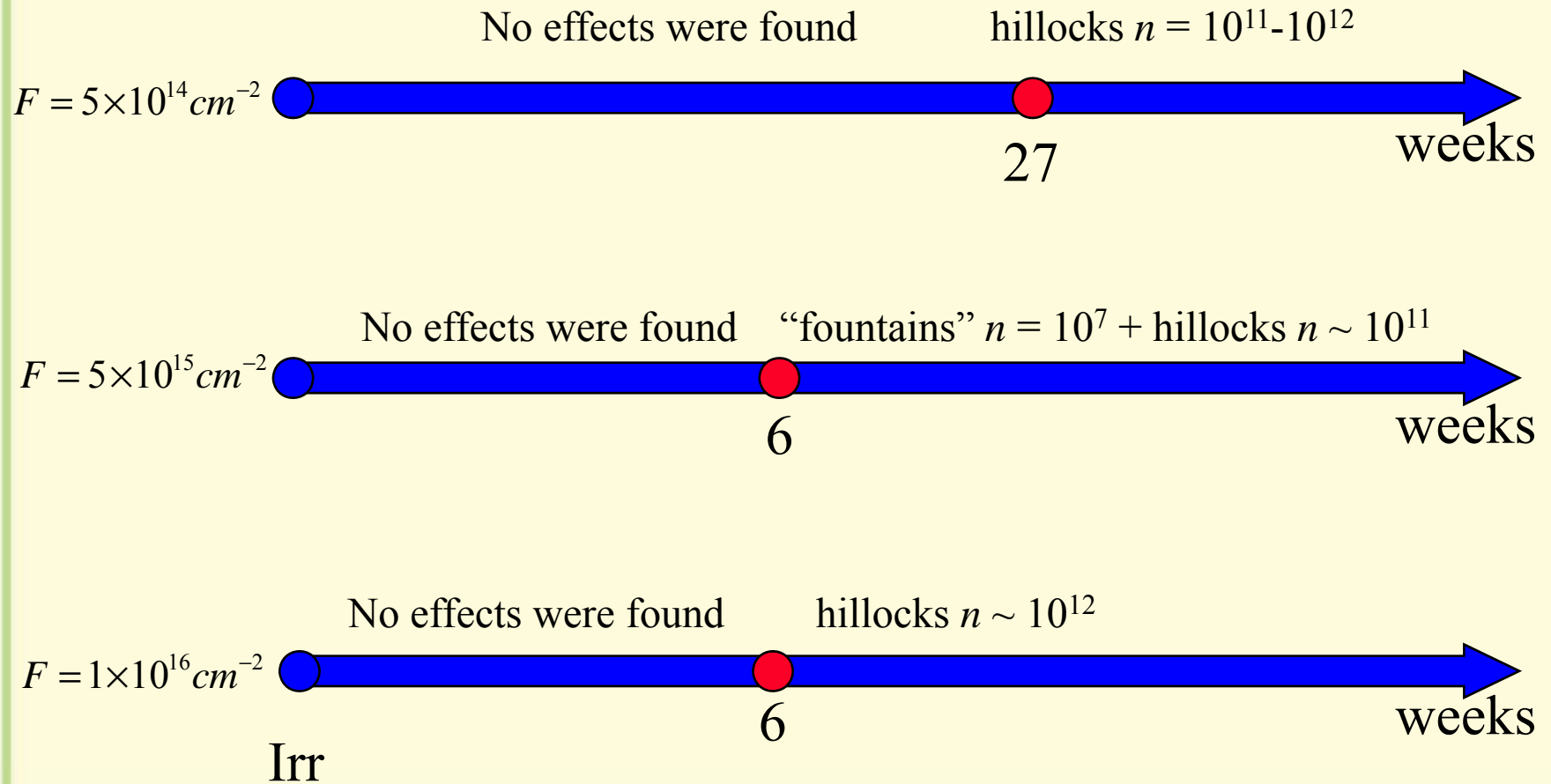
# Proton irradiation

- Source parameters:
  - Ion  $H^+$
  - Energy 21 MeV
- Fluences:
  - $5 \times 10^{14} \text{ cm}^{-2}$
  - $5 \times 10^{15} \text{ cm}^{-2}$
  - $1 \times 10^{16} \text{ cm}^{-2}$
- Assembly parameters:
  - n-Si (100)
  - specimen thickness 0.3 mm
  - 12 plates in the assembly





# AFM Investigation





# Inhomogeneous nucleation

- Round shaped areas filled with hillocks had emerged on the plates, irradiated up to  $F=5 \times 10^{15} \text{ cm}^{-2}$ , after 5 weeks.

- Parameters of this “fountains”:

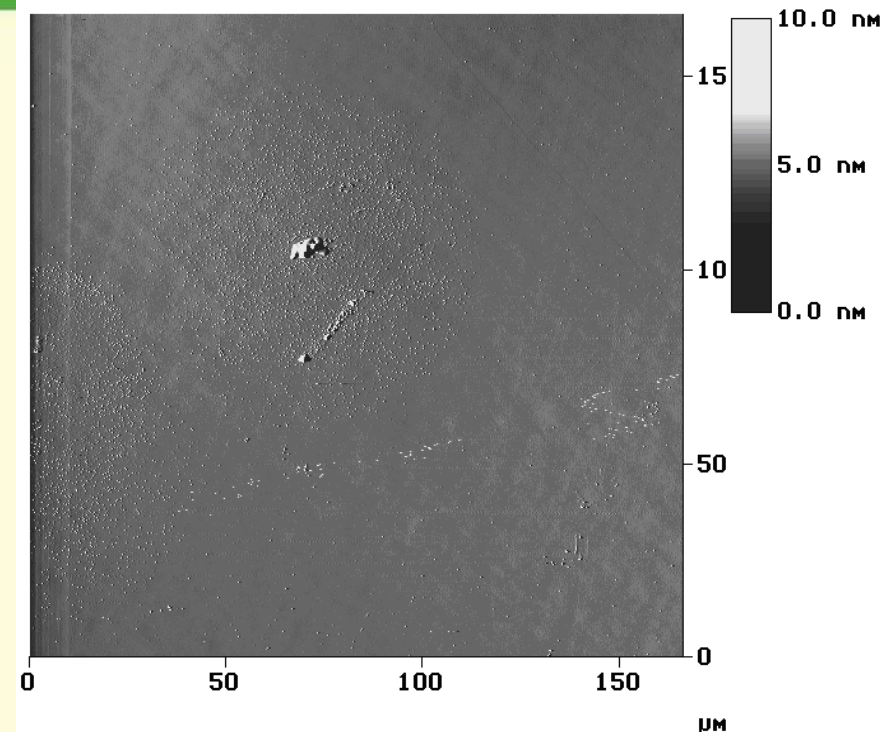
- $d = 10 - 100 \mu\text{m}$

- $h = 5 - 10 \text{ nm}$

- $n = 10^7 - 10^8 \text{ m}^{-2}$

- $n_h \sim 3 \times 10^{10} \text{ m}^{-2}$

- the rest of the surface was covered with hillocks  $n \sim 10^{10} \text{ m}^{-2}$ .



Two types of “fountain” were found:

- with round areas with homogeneous distribution of hillocks (they are the biggest)

- with increasing hillock number density toward the center



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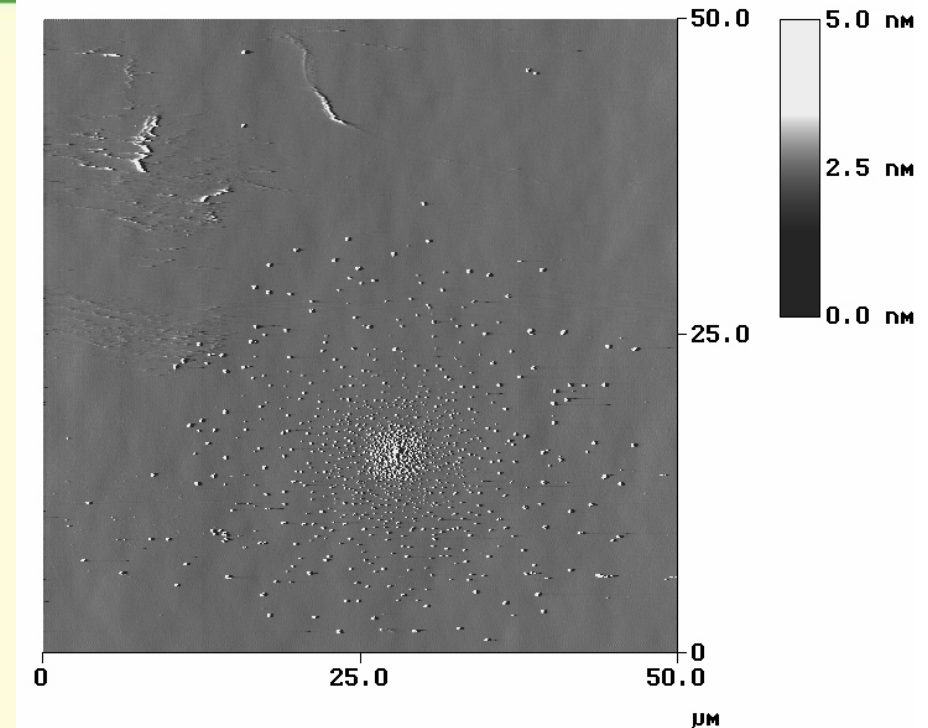
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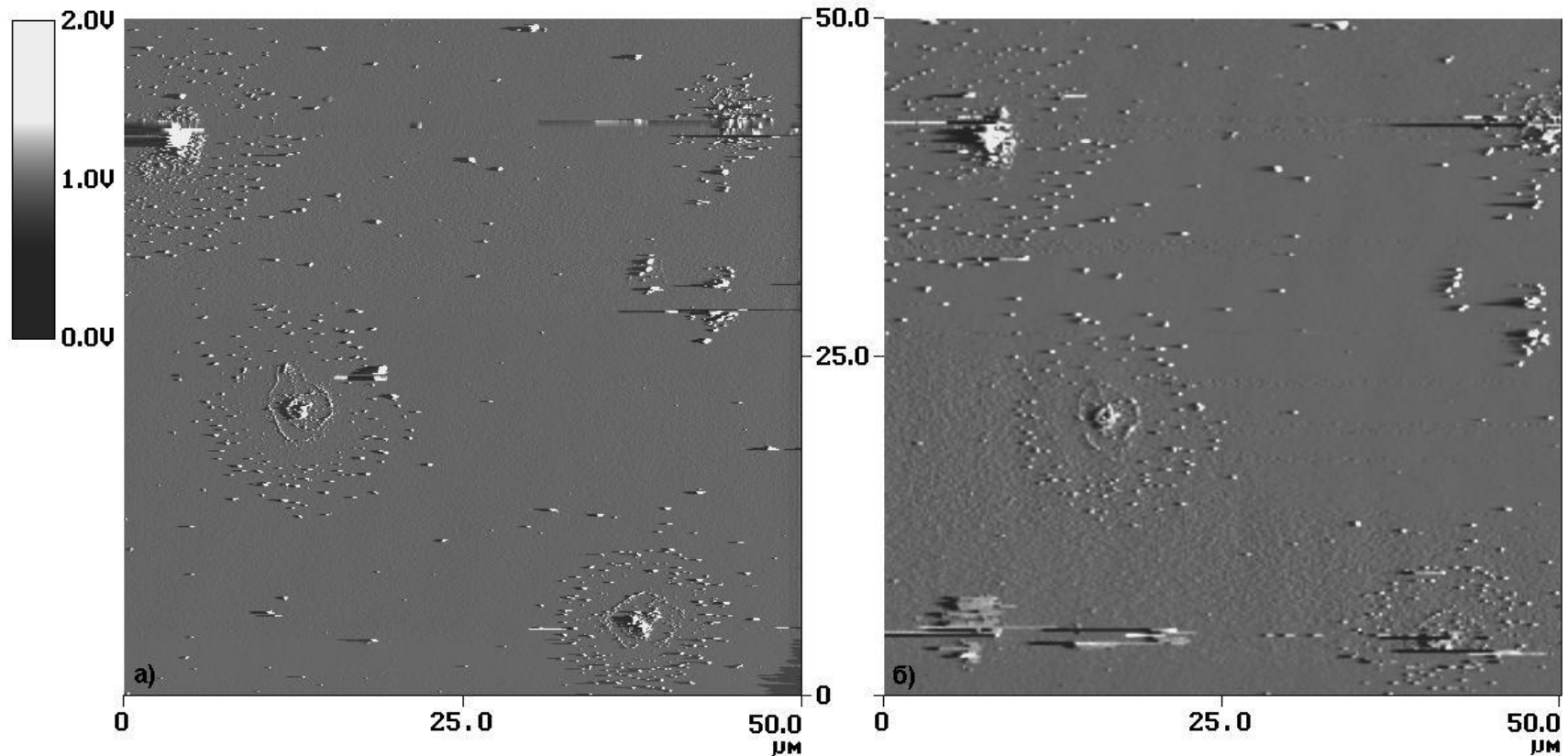




# Annealing

protons, 21 MeV,  $F=5 \times 10^{15} \text{ cm}^{-2}$

Preliminary annealing at  $T=100^\circ \text{ C}$ ,  $t=20 \text{ min}$     Further annealing at  $T=300^\circ \text{ C}$ ,  $t=20 \text{ min}$



Annealing leads to “fountains” formation on earlier stages (accelerates nucleation)



# Conclusions

- High energy ion irradiation results in surface modification (hillocks formation)
- These precipitates are considered to consist of silicon dioxide
- Hillocks nucleation reveals a considerable delay (weeks, months) which depends on dose.
- Hillocks formation is a thermal activated process. Heating is increasing this process rate.
- Presence of the considerable hillock nucleation delay indicates that “tracks” influence is minimal (if any) in this process.
- Inhomogeneous hillock nucleation on structural defects (dislocations, etc.) becomes possible at intermediate fluences. As result fountain-like structures emerge.



# Discussion

- ✓ Hillocks nucleation processes: nucleation is considered to be connected with recombination of defects that have been created inside irradiated material (during irradiation).
- ✓ Limiting factors (hillocks nucleation and growth):  
room temperature irradiation: defects mobility  
 $D_V \sim 10^{-8}-10^{-9} \text{ cm}^2/\text{s}$ ,  $D_I \sim 10^{-4} \text{ cm}^2/\text{s}$ , etc.
- ✓ For binding of defects recombination and hillocks nucleation experiments with simultaneous (parallel) investigation of surface and volume defects concentration are required.