

# Comparison of neutron damage in thin FZ, MCz and epitaxial silicon detectors

**E. Fretwurst (a), L. Andricek (b), F. Hönniger (a), K. Koch (a), G. Kramberger (c), G.  
Lindström (a), H.G. Moser (b), I. Pintilie (a,d), R. Richter (b), R. Röder (e)**

- (a) *Institute for Experimental Physics, University of Hamburg*
- (b) *MPI-Semiconductor Laboratory Munich*
- (c) *Jozef Stefan Institute, University of Ljubljana*
- (d) *National Institute for Materials Physics, Bucharest*
- (e) *CiS Institut für Mikrosensorik gGmbH, Erfurt*

# Motivation

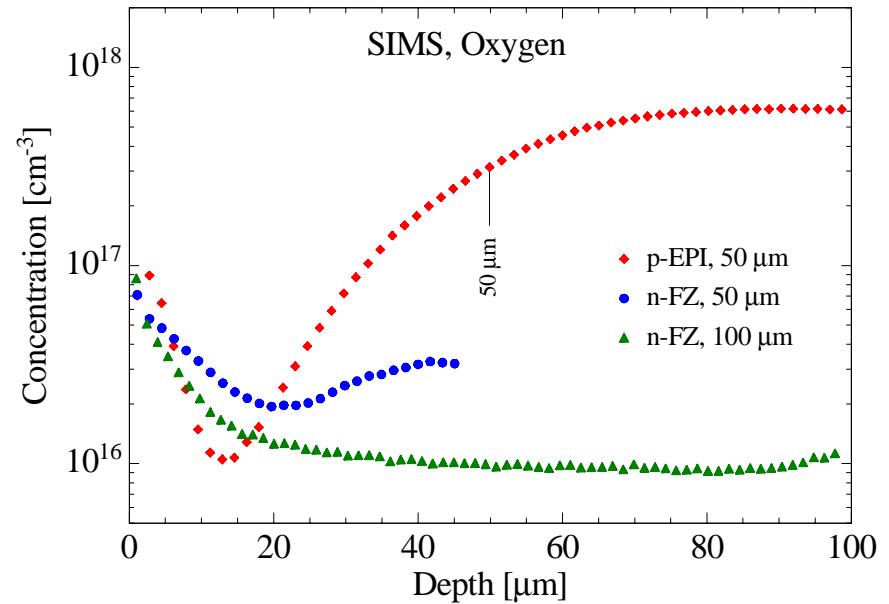
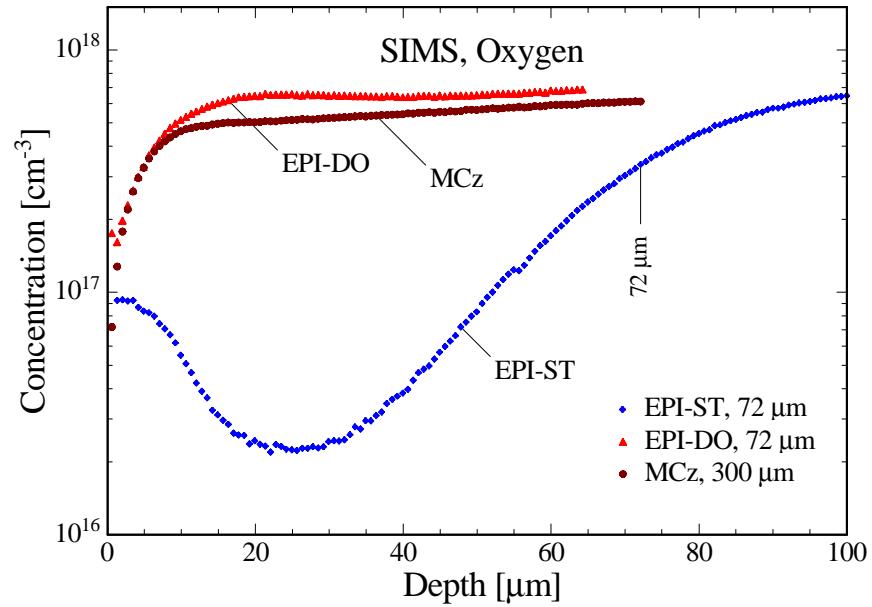
- **Findings from RD48:**  
Neutron damage is independent on Si material type  
(n- or p-type, oxygen concentration). These results were found for mainly 300 µm thick Si-devices
- **From RD50:**  
Thin n-type epi-Si (25 and 50 µm, 50 Ω·cm) show no type inversion after neutron damage
- **Questions:**
  - Is the fluence dependence of the full depletion voltage or the effective doping concentration for thin Si-detectors in general different compared to thick (300 µm) devices?
  - Which impurities play a major role (P, B, O, others)?

# Material under investigation

Material	Cond. type	Orientation	$N_{eff,0} [10^{13} \text{ cm}^{-3}]$	d [ $\mu\text{m}$ ]
N-EPI-ST(1)	N	<111>	2.6	72
N-EPI-DO(2)	N	<111>	2.6	72
P-EPI-ST	P	<111>	-9.0	50
FZ-50(3)	N	<100>	3.3	50
FZ-100	N	<100>	1.4	100
MCz-IP(4)	N	<100>	0.48	100
MCZ-DI(5)	N	<100>	0.77	100

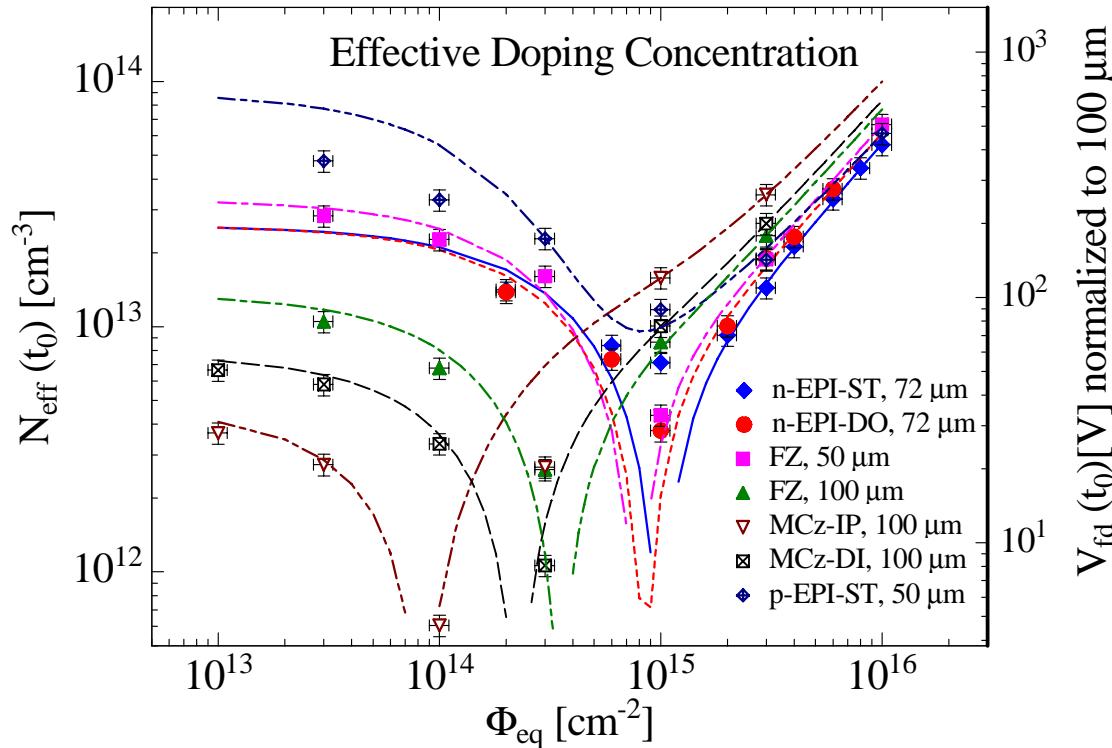
- (1) Standard detector process (*CiS*)
- (2) Oxygen enriched, diffusion for 24 h at 1100°C (*CiS*)
- (3) Produced in wafer bonding technology (*MPI*)
- (4) Rear side P implanted after thinning (*CiS*)
- (5) Rear side P diffused after thinning (*CiS*)

# Oxygen depth profiles



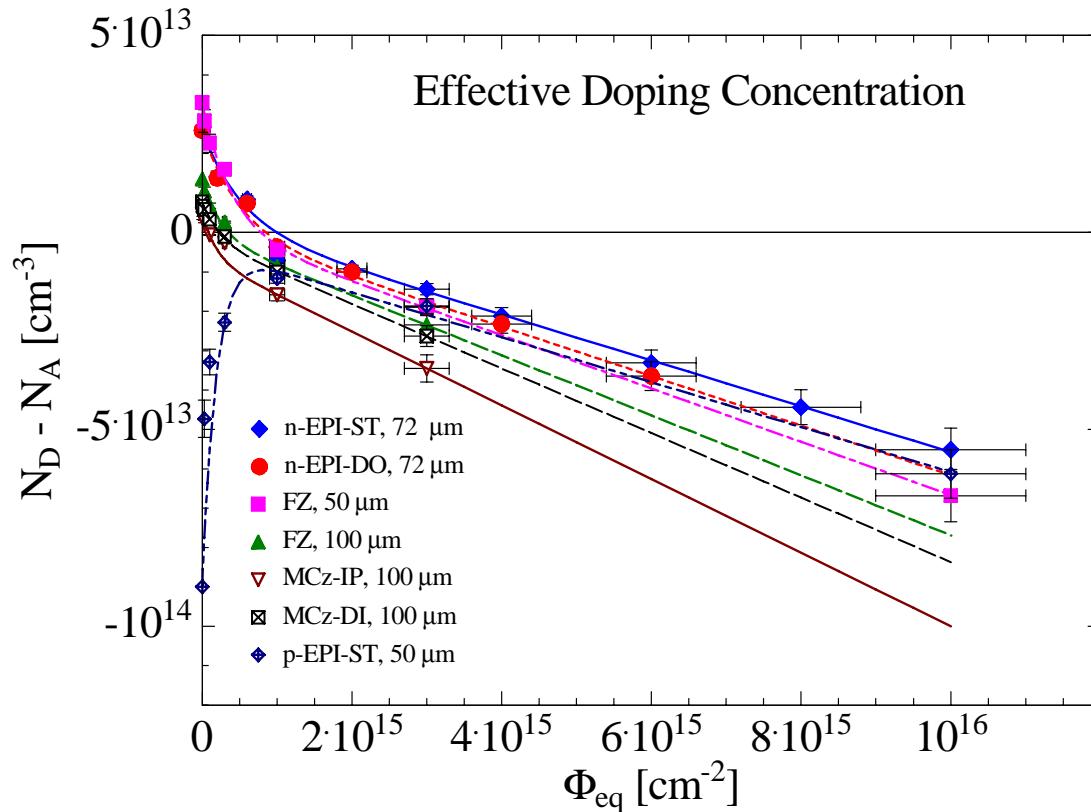
- **EPI-ST: [O] inhomogeneous,  $\langle [\text{O}] \rangle = 9.3 \ 10^{16} \text{ cm}^{-3}$**
- **EPI-DO: [O] homogeneous, except surface  $\langle [\text{O}] \rangle = 6.0 \ 10^{17} \text{ cm}^{-3}$**
- **MCz: [O] homogeneous, except surface  $\langle [\text{O}] \rangle = 5.2 \ 10^{17} \text{ cm}^{-3}$**
- **P-type EPI: [O] inhomogeneous  $\langle [\text{O}] \rangle = 1.0 \ 10^{17} \text{ cm}^{-3}$**
- **FZ 50  $\mu\text{m}$ : inhomogeneous  $\langle [\text{O}] \rangle = 3.0 \ 10^{16} \text{ cm}^{-3}$**
- **FZ 100  $\mu\text{m}$ : homogeneous, except surface  $\langle [\text{O}] \rangle = 1.4 \ 10^{16} \text{ cm}^{-3}$**

# Development of $N_{\text{eff}}$ resp. $V_{\text{fd}}$ normalized to 100 μm



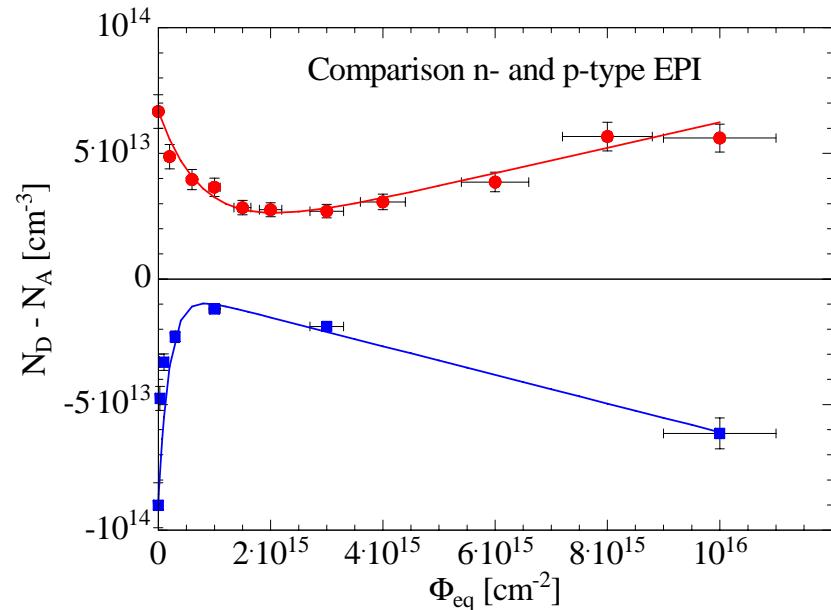
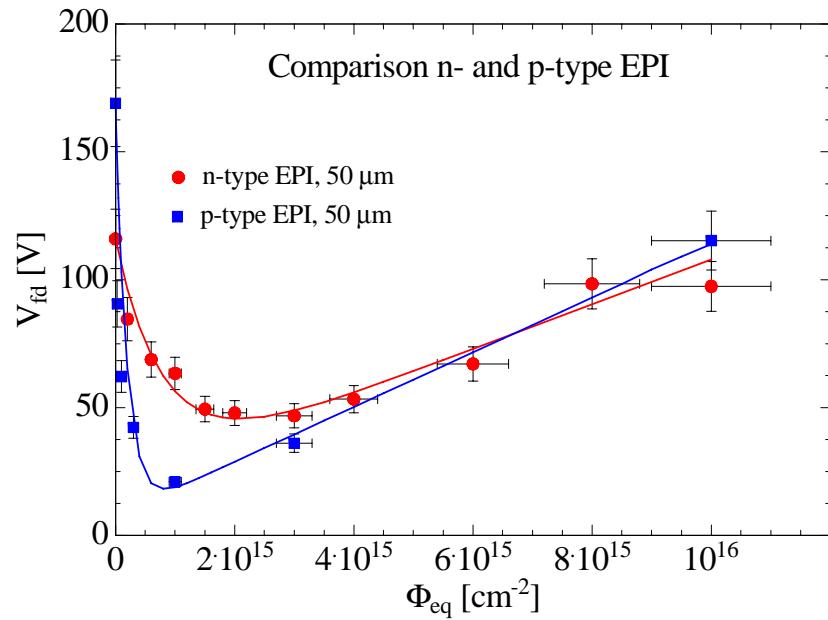
- Space charge sign inversion (SCSI) for all n-type materials, p-type EPI always negative space charge
- Fluence for SCSI depends on primary donor concentration (P and TD concentration, compare MCz-DI with TD and MCz-IP without TD)
- Development of  $N_{\text{eff}}$  for n-type and p-type material seems to be dominated by doping (P or B) removal and introduction of negative space charge although a creation of donors has to be taken into account (see I. Pintilie)

# Effective space charge concentration



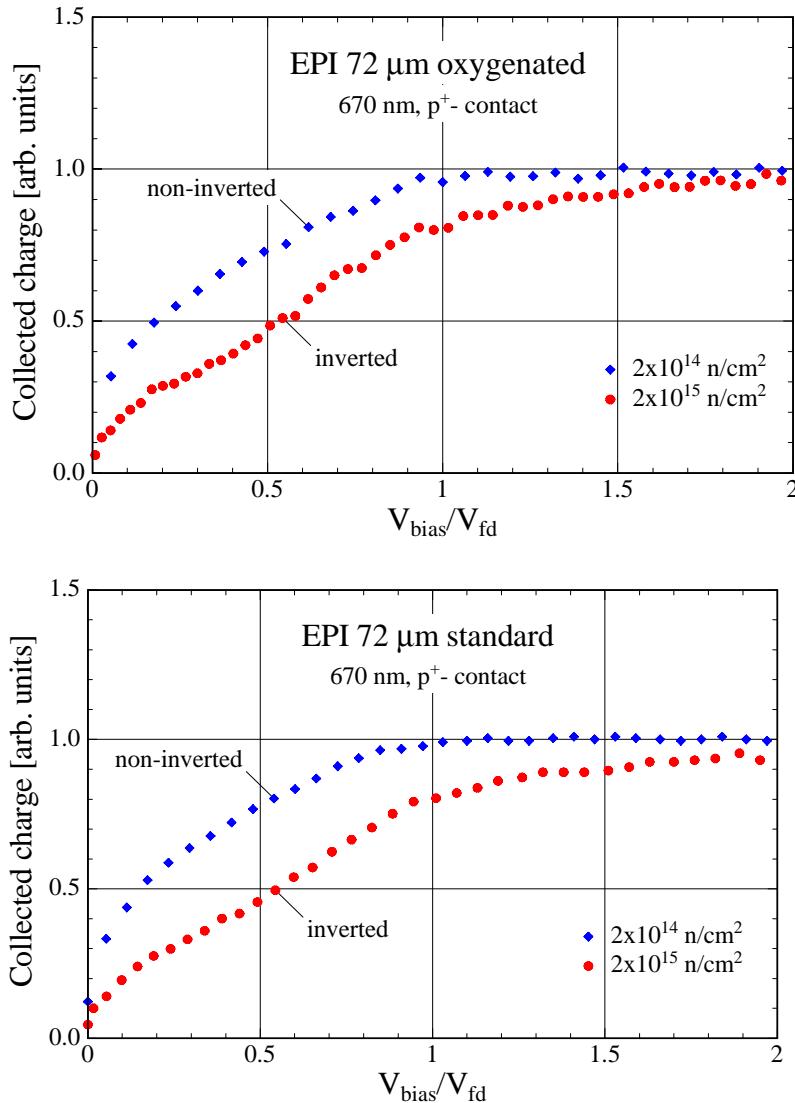
- Former studies on 50 Ωcm n-type EPI-devices revealed no SCSI contrary to these different n-type materials
- Space charge of p-type EPI stays negative
- The introduction of negative space charge is lower for low resistivity material (EPI and FZ) compared to the higher resistivity MCz material

# Comparison 50 µm n- and p-type EPI



- **N-type EPI: Donor (P) removal in low fluence range, dominant donor generation (BD) in high fluence range**
- **P-type EPI: Fast acceptor (B) removal in low fluence range, dominant acceptor generation in high fluence range**

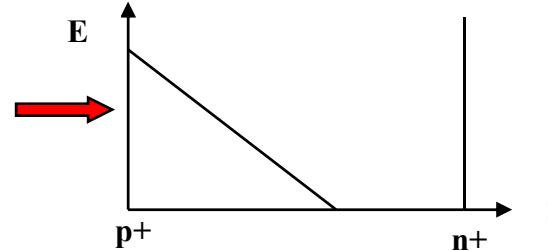
# Space Charge Sign in 72 $\mu\text{m}$ n-EPI-devices



**Illumination of p<sup>+</sup>-contact with 670 nm laser light (absorption length at RT about 3  $\mu\text{m}$ ):**

**No SCSI:**

Smooth increase of collected charge with „normalized“ bias voltage up to  $V_{\text{bias}}/V_{\text{fd}} = 1$

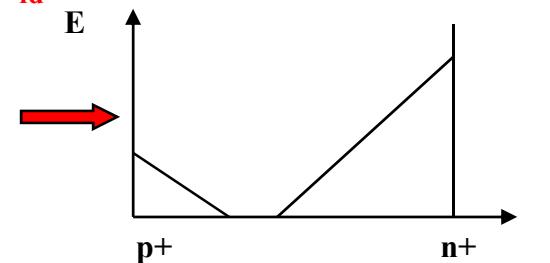


**SCSI (two E-field regions):**

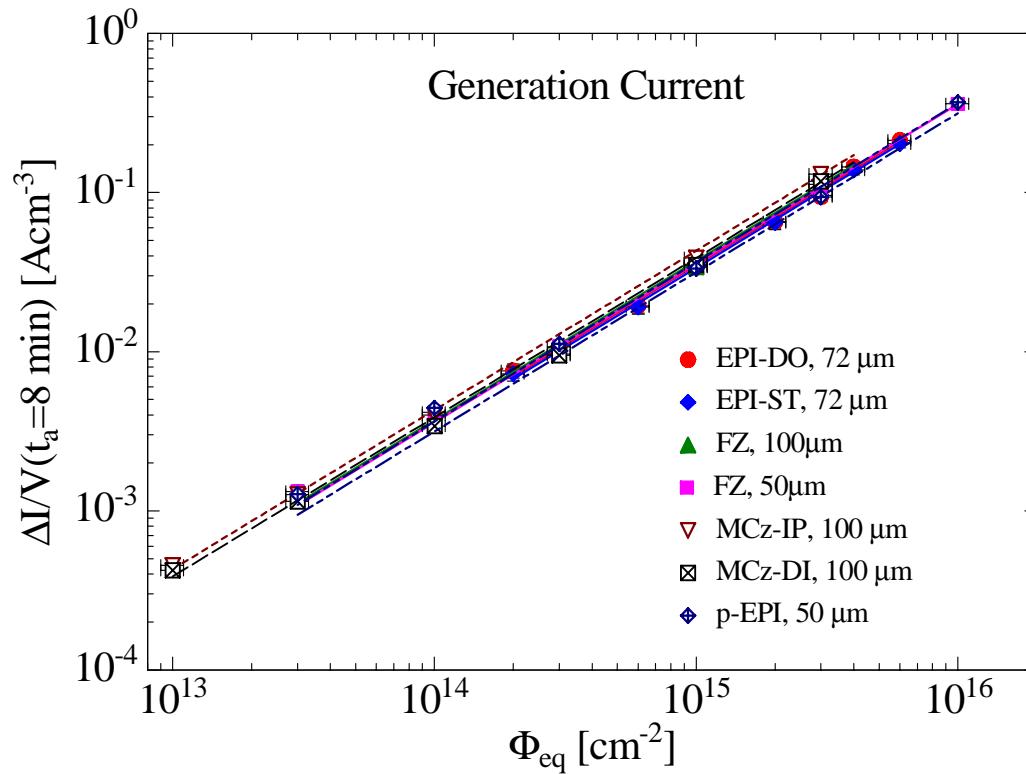
Increase of collected charge with „normalized“ bias voltage in two stages

$V_{\text{bias}}/V_{\text{fd}} = 0 - 0.5$ : small „saturating“ increase

$V_{\text{bias}}/V_{\text{fd}} = 0.5 - 1$ : linear increase



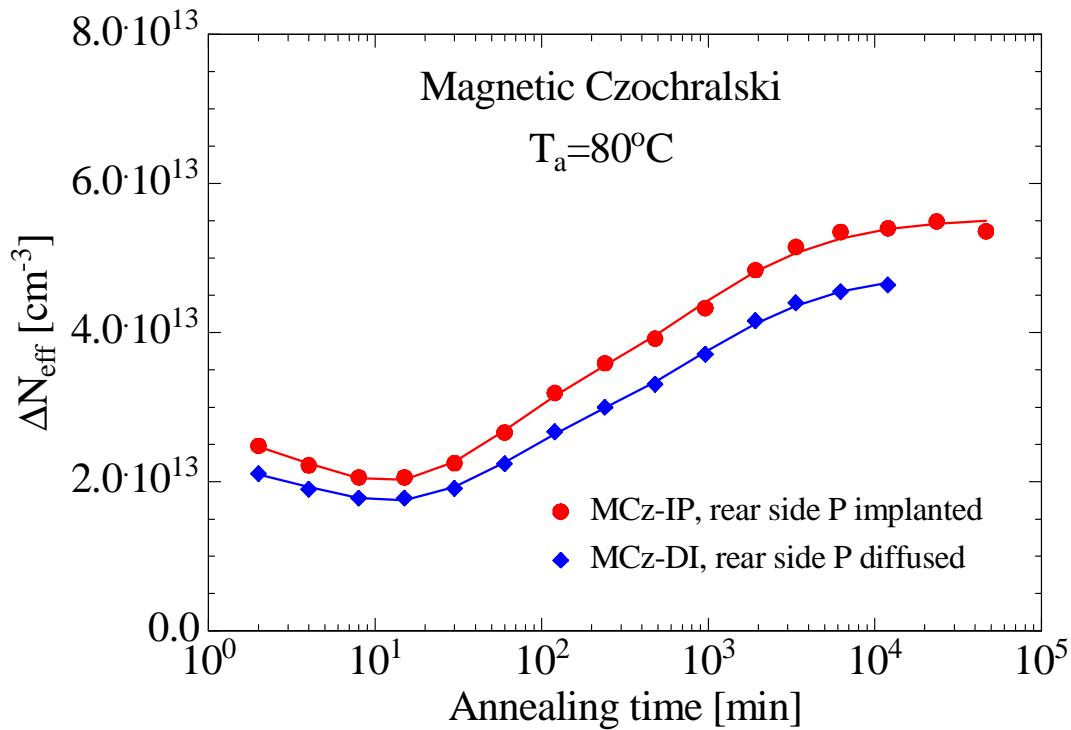
# Generation current increase



## Generation current increase after annealing for 8 min at 80°C:

- Almost linear increase between  $10^{13} \text{ cm}^{-2}$  up to  $10^{16} \text{ cm}^{-2}$   
damage parameter  $\alpha$  varies between  $3.4 \cdot 10^{-17}$  and  $4.0 \cdot 10^{-17} \text{ A/cm}$
- Nearly independent on material type

# Annealing of $\Delta N_{\text{eff}}$ at 80 °C



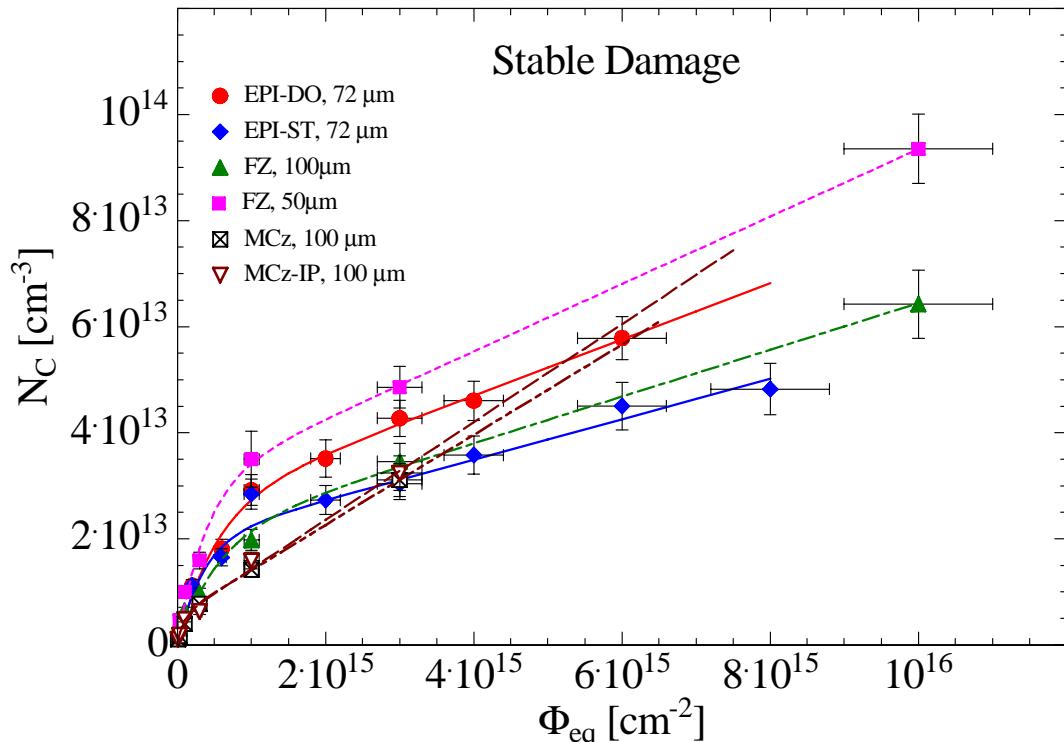
$$\Delta N_{\text{eff}}(\Phi, t) = N_a(\Phi, t) + N_C(F) + N_Y(\Phi, t)$$

with

$$N_Y(\Phi, t) = N_{Y,1}(\Phi, t) + N_{Y,2}(\Phi, t)$$

- Typical annealing behavior:  
**short term annealing**  $N_a$   
**+ stable component**  $N_C$   
**+ reverse annealing**  $N_Y$
- **reverse annealing best described by 2 components:**  
**1. order + 2. order process**
- **Difference between MCz-IP (rear side P implanted) and MCz-DI (rear side P diffused) due to process induced TD's in MCz-DI**
- **TD's are stable at 80°C**

# Stable damage component $N_C$

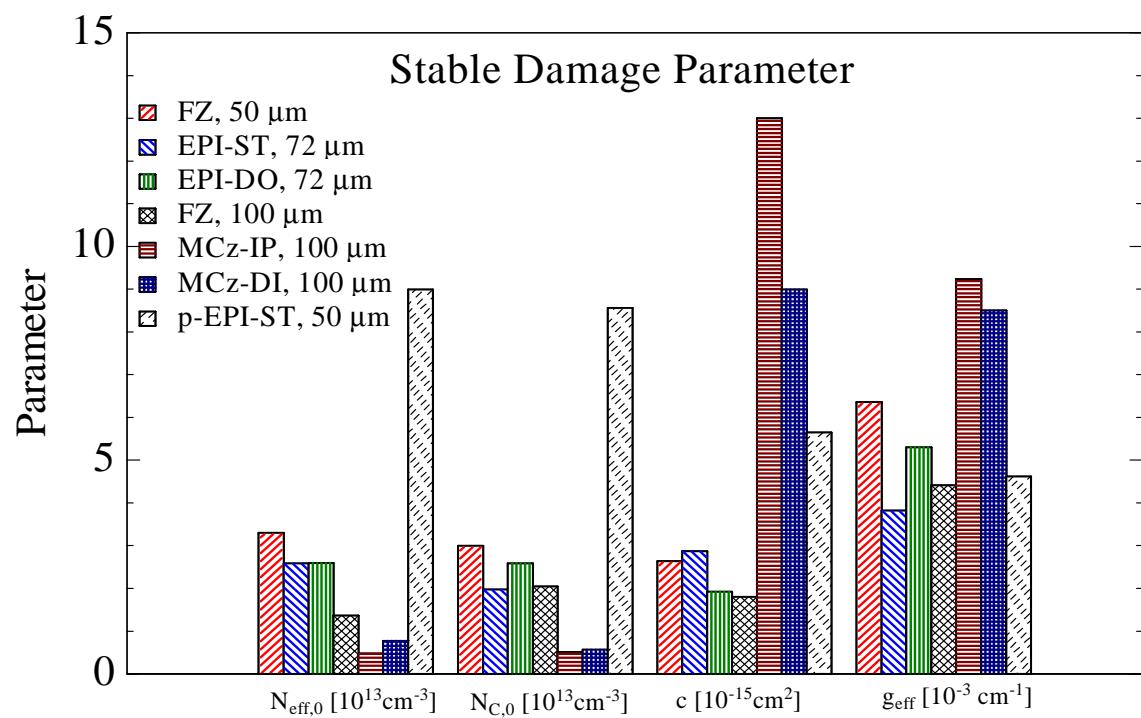


$$N_C(\Phi) = N_{C,0} \cdot \{1 - \exp(c \cdot \Phi)\} + g_C \cdot \Phi$$

Donor or acceptor removal + introduction of charged acceptors and donors

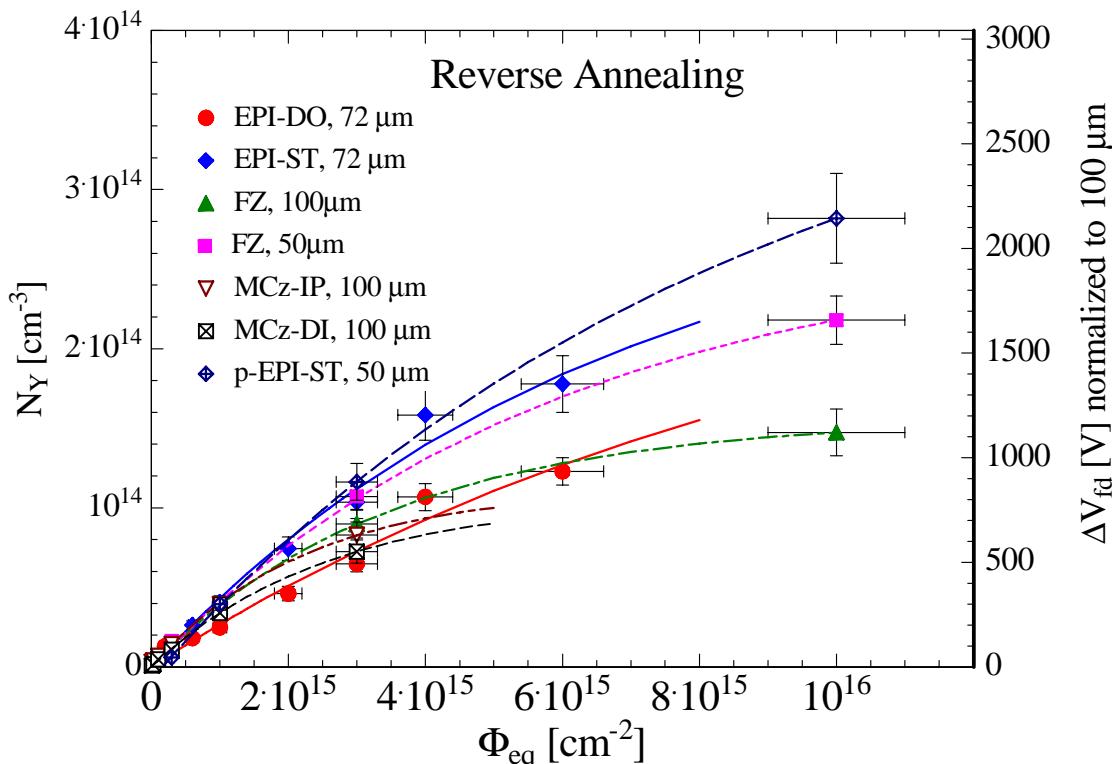
- **Similarities:**
  - in high fluence range  $N_C$  linear in  $\Phi$  for all materials ( $g_C$  term)
  - $g_C$  is nearly the same for low resistivity material
  
- **Differences:**
  - the offset ( $N_{C,0}$ ) seems to be correlated with the primary doping
  - the introduction rate  $g_C$  for the MCz material is larger compared to all other materials

# Stable damage parameter



- Doping removal  $N_{C,0}$  correlates with primary doping concentration  $N_{eff,0}$   
 $N_{C,0} \approx N_{eff,0}$
- Removal parameter  $c$  depends on  $N_{C,0}$  resp.  $N_{eff,0}$ .  
The “removal rate”  
 $c \times N_{C,0}$  is nearly constant  
 $(5.6 \pm 1.5) \cdot 10^{-2} \text{ cm}^{-1}$   
for n-type material.  
For p-type the value is:  
 $48 \cdot 10^{-2} \text{ cm}^{-1}$
- Introduction rate  $g_{eff}$  ( $=g_C$ ) for high doping:  
 $(4.9 \pm 1.0) \cdot 10^{-3} \text{ cm}^{-1}$   
For low doping (MCz)  
 $(8.9 \pm 0.5) \cdot 10^{-3} \text{ cm}^{-1}$

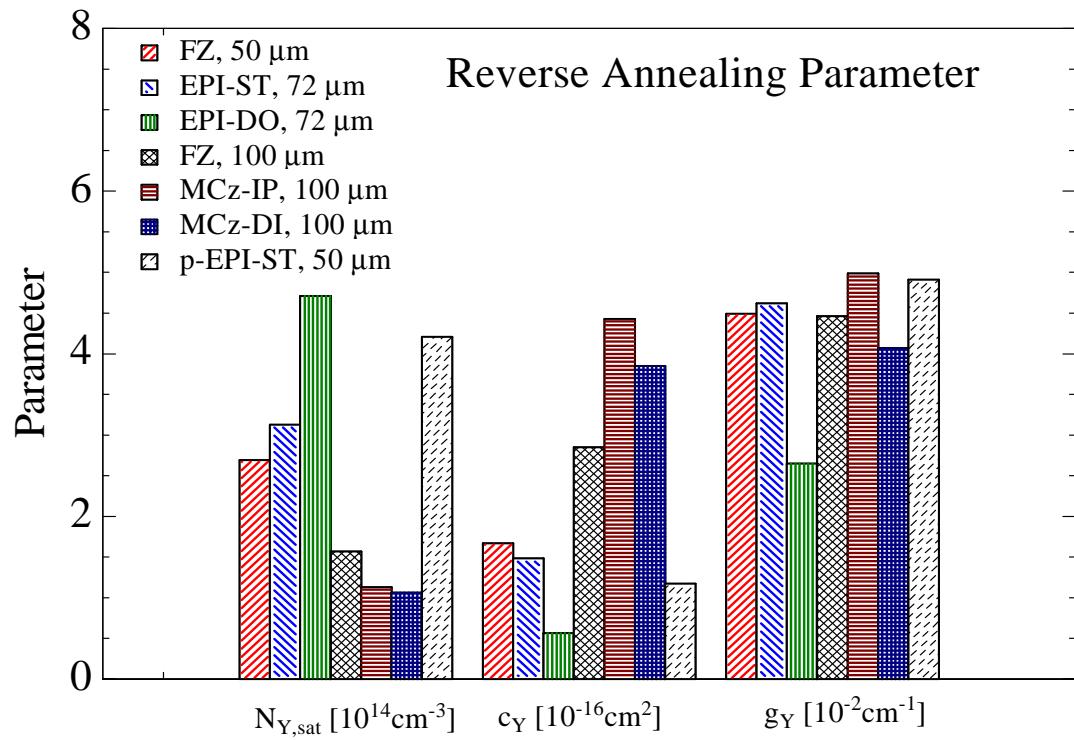
# Reverse annealing amplitude $N_Y$



- **Saturation:**  
Saturating development holds for all materials
- **Oxygen effect?**  
A dependence of  $N_Y$  on the oxygen concentration can so far not be stated,  
e.g. FZ 100 μm material has extremely low [O] and a very low  $N_Y$  at  $10^{16} \text{ cm}^{-2}$
- **Introduction rate  $g_Y$ :**  
see next page

$$N_Y(\Phi) = N_{Y,S} \cdot \{1 - \exp(-c_Y \cdot \Phi)\}$$

# Reverse annealing parameter



- Saturation amplitude N<sub>Y,sat</sub>:  
strong variation between different materials
- Rate parameter c<sub>Y</sub>:  
also strong variation between different materials but seems to be anti-correlated with N<sub>Y,sat</sub>

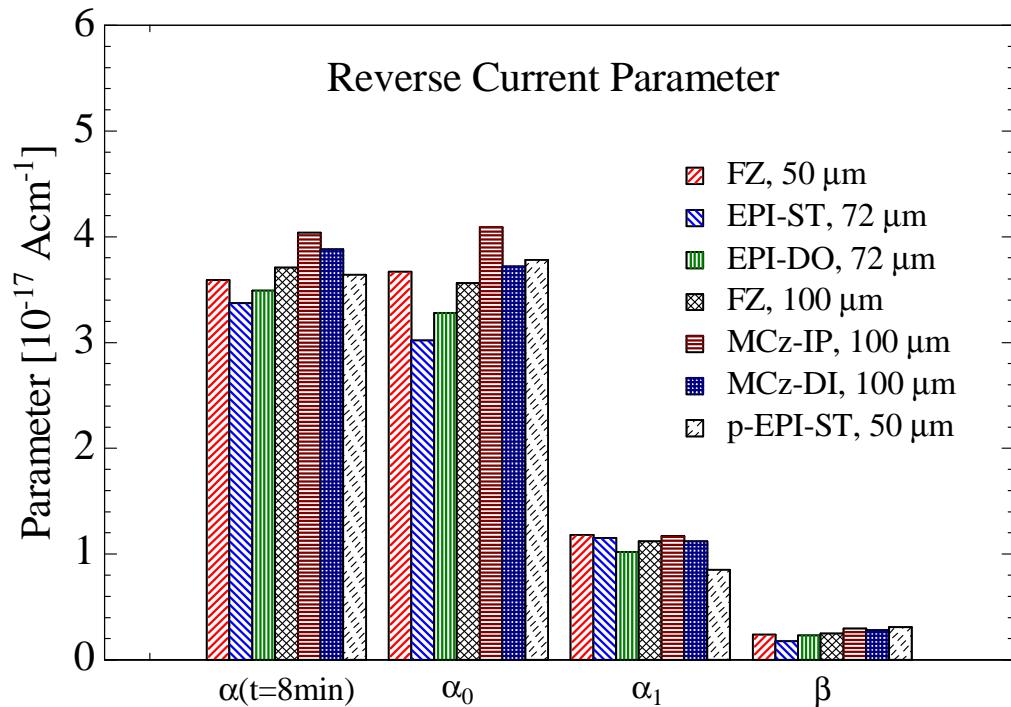
↓
- Introduction rate g<sub>Y</sub>:  
defined by  

$$g_Y = c_Y \cdot N_{Y,sat}$$
**excluding EPI-DO**

$$\langle g_Y \rangle = (4.6 \pm 0.3) \cdot 10^{-2} \text{ cm}^{-1}$$

EPI-DO:  $g_Y = 2.7 \cdot 10^{-2} \text{ cm}^{-1}$

# Reverse current annealing parameter



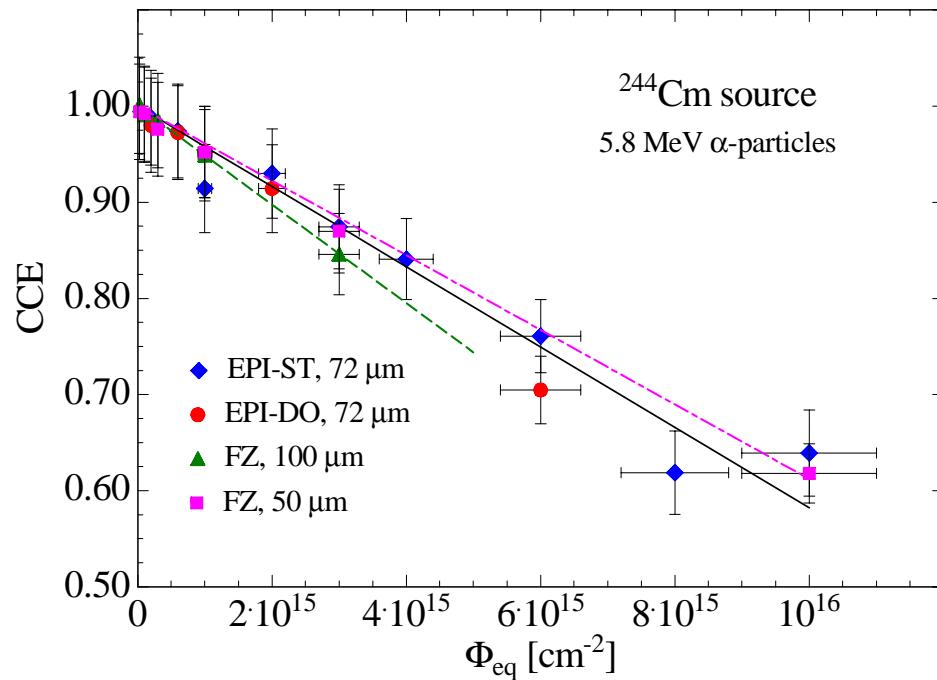
$$\alpha(t) = \alpha_1 \cdot \exp(-t / \tau_I) + \alpha_0 - \beta \cdot \ln(t / t_0)$$

- All parameters are independent of the material type.
- This holds also for the p-type EPI material
- The generation current is dominated by intrinsic defects like vacancy clusters

*this work*    *M. Moll*

$\alpha(t=8\text{min})$ [ $10^{-17}\text{A/cm}$ ]	$3.7 \pm 0.2$	4.01
$\alpha_0$ [ $10^{-17}\text{A/cm}$ ]	$3.6 \pm 0.3$	4.2
$\alpha_1$ [ $10^{-17}\text{A/cm}$ ]	$1.1 \pm 0.1$	1.1
$\beta$ [ $10^{-17}\text{A/cm}$ ]	$0.25 \pm 0.05$	0.28
$\tau$ [min]	$14.8 \pm 3.1$	9

# Charge Collection Efficiency



$^{244}\text{Cm}$   $\alpha$ -source,  $E_{\alpha} = 5.8 \text{ MeV}$

Collected charge measured by TCT voltage scan  
Integration time window 10 ns

Collected charge taken at about  
2 x full depletion voltage

Estimate of damage parameter  $\beta_{\alpha}$   
from linear fits

$$\langle \beta_{\alpha} \rangle = (4.5 \pm 0.6) \cdot 10^{-17} \text{ cm}^2$$

Trapping is independent on material type

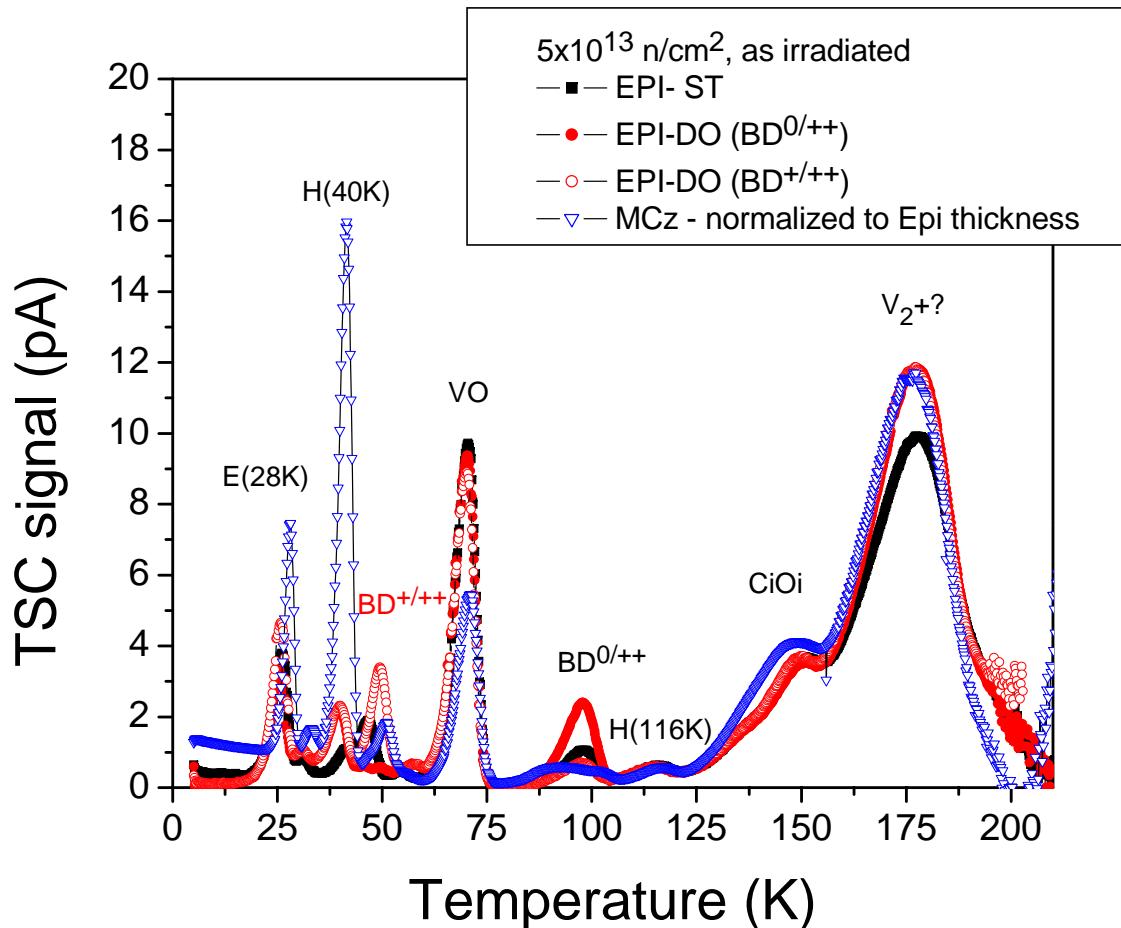
# Conclusions



**Comparison of thin Si-detectors processed on different materials (n- and p-type EPI, FZ and MCz) after neutron irradiation shows:**

- Neff development dominated by doping removal (P,B; low fluence) and introduction of negative space charge (acceptors)
- Oxygen effect not seen
- SCSI observed for all n-type materials contrary to former results on  $50 \Omega\text{cm}/50 \mu\text{m}$  EPI-material
- No SCSI for p-type EPI
- Introduction of shallow donors (BD) could be detected via TSC in n-type EPI and MCz (see I. Pintilie) but the introduction rate is too small for an overcompensation of the induced negative space charge
- Reverse current increase independent on material type
- No difference is seen in the charge collection properties of n-type EPI and FZ materials

# TSC Studies on Neutron Irradiated Devices



## Main defects:

- V<sub>2</sub>, clustered
- C<sub>i</sub>O<sub>i</sub>
- VO
- Bistable donor:  
BD<sup>(0/++)</sup>
- BD<sup>(+/++)</sup> first time observed
- Several shallow hole and electron traps (H(40K), E(28K))

## Main differences:

- BD(+/++) only in EPI-DO?
- BD(0/++) dominant in EPI-DO, but also detected in EPI-ST and MCz
- [VO] identical in EPI-DO and EPI-ST, lower in MCz