

Oxygenation under high pressure of EuBCO and GdBCO coated conductors

**T. Prikhna^{1,2,3*}, T. Puig², A. Kethamkuzhi², R. Vlad², R. Kluge³, M. Karpets^{1,4},
S. Ponomarov⁵, V. Moshchil¹, , S. Wurmehl³, and X. Obradors²**

¹ V. Bakul Institute for Superhard Materials of the National Academy of Sciences of Ukraine,
1 2, Avtozavodska Str., Kyiv 07074, Ukraine

² Institut de Ciencia de Materials de Barcelona, CSIC, Campus UAB, 08193 Bellaterra, Spain

³ Leibniz-Institut für Festkörper- und Werkstoffforschung Dresden e. V.,
Helmholtzstrasse 20 01069 Dresden, Germany.

⁴ National Technical University of Ukraine «Igor Sikorsky Kyiv Polytechnic Institute»,
Peremogy Avenue 37, 03056 Kyiv, Ukraine

⁵ V.E. Lashkaryov Institute of Semiconductor Physics of the National Academy of Sciences of Ukraine,
41, Nauky Ave., Kyiv 03028, Ukraine

Content

1. Motivation
2. Experimental (1-160 bar oxygen pressure, 300-800 °C temperature, 3- 12 h holding time)
3. Characteristics before and after treatment (T_c , J_c , carrier density, n_H , c-lattice parameter of RE123 (RE=Gd, Eu))
4. Study of the materials structures (SEM EDS and Auger spectroscopy).
5. Conclusions

1. Motivation

- The **overdoping** for $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ (Y123) thin films (as a result of treatment via Ag layer when heated under oxygen partial pressure below 1 bar at low temperatures) was confirmed by a sudden increase of charge carrier density, n_H , associated to the reconstruction of the Fermi-surface at the Quantum Critical Point (QCP). The authors (Stangl, A., Palau, A., Deutscher, G., Obradors, X., 2021, . Sci Rep., 11, 8176.) **achieved the highest ever reported in YBCO films critical current density J_c , equal to $90 \text{ MA}\cdot\text{cm}^{-2}$ at 5 K** which corresponds to about a fifth of the despairing current.
- The Miura, M. et al. (2022, NPG Asia Materials , 14, 85) demonstrated that thermodynamic improvements of superconductors can work in parallel with already successful artificial pinning centers and that a maximum critical current density $J_c \sim 0.3 J_d$ (despairing current) appears to be the current upper limit for the enhancement in J_c . They concluded that variation in λ and ξ leads to an **intrinsic improvement in J_c via J_d** , allowing extremely high values of J_c of **$130 \text{ MA}/\text{cm}^2$ at 4.2 K**, consistent with an enhancement in J_d of a factor of 2 for **nanoparticle-doped (Y,Gd)123** coated conductors (CCs). The combination of thermodynamic and pinning optimization route for the (Y,Gd)123 CCs resulted in high **vortex-pinning force $\sim 3.17 \text{ TN}/\text{m}^3$ at 4.2 K and 18 T (H||c)** .
- The idea to apply such high oxygen pressure (**$100\text{-}160 \text{ bar}$**) and annealing temperature (**$600\text{-}800 \text{ }^\circ\text{C}$**) was caused by our experience with **melt-texturing YBCO** for which such conditions allowed essentially **decrease the time of oxygenation, increase critical current density (especially in high magnetic fields) due to increase of twin density, reduce anisotropy of critical current density, decrease micro- and macrocracking** (and thus improve mechanical properties). [3-5]

2. Experimental

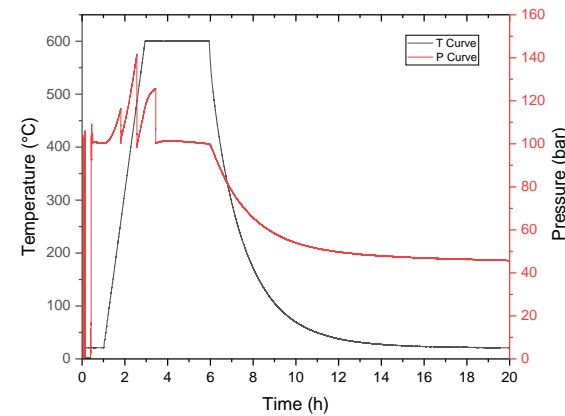
Architecture of the studied types of coated conductors:

- (1) FYSC : Ag (2 μ m)/ GdBCO (1.8 μ m)/ Al₂O₃/Y₂O₃/MgO/ CeO₂ (700 nm)/ Hastelloy (75 μ m);
- (2) FESC : EuBCO (2.5 μ m)+BHO Nanorods/ Al₂O₃/Y₂O₃/MgO/ CeO₂ (700 nm)/ Hastelloy (50 μ m);
- (3) FESC : Ag (2 μ m)/ EuBCO (2.5 μ m)+BHO Nanorods/ Al₂O₃/Y₂O₃/MgO/ CeO₂ (700 nm)/ Hastelloy 50 μ m).

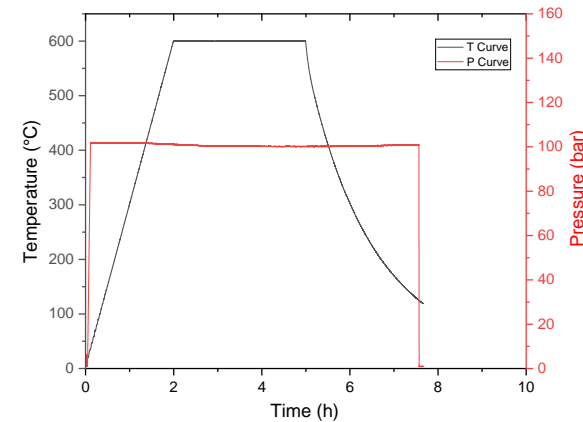


Tube furnace. 1-160 bar O₂,
300-800 °C, $v_{\text{heating}} = 5\text{K/min}$,
 $\tau = 3\text{-}12\text{ h}$

Mode 1



Mode 2



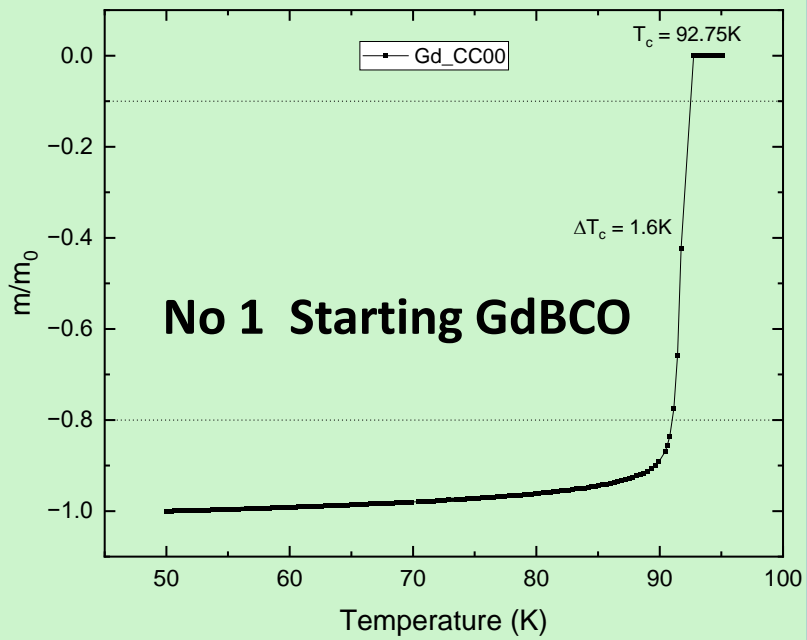
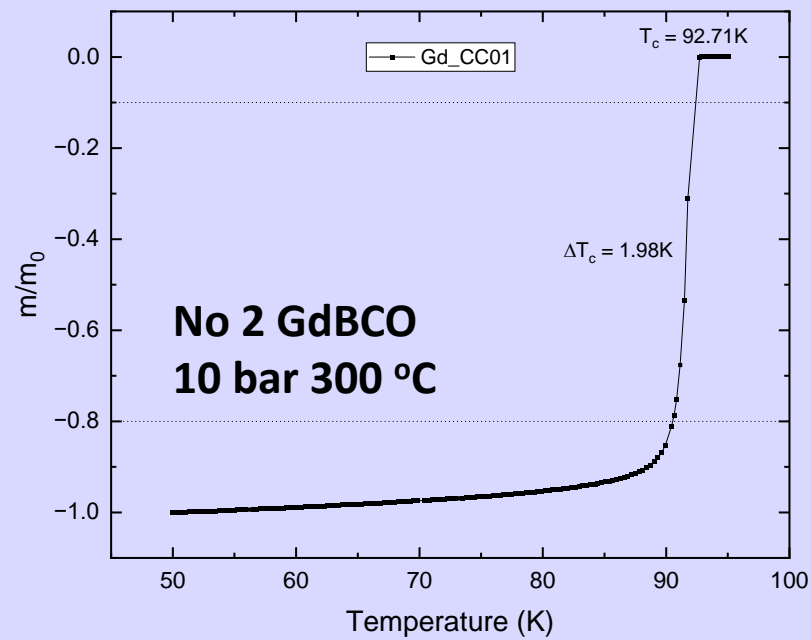
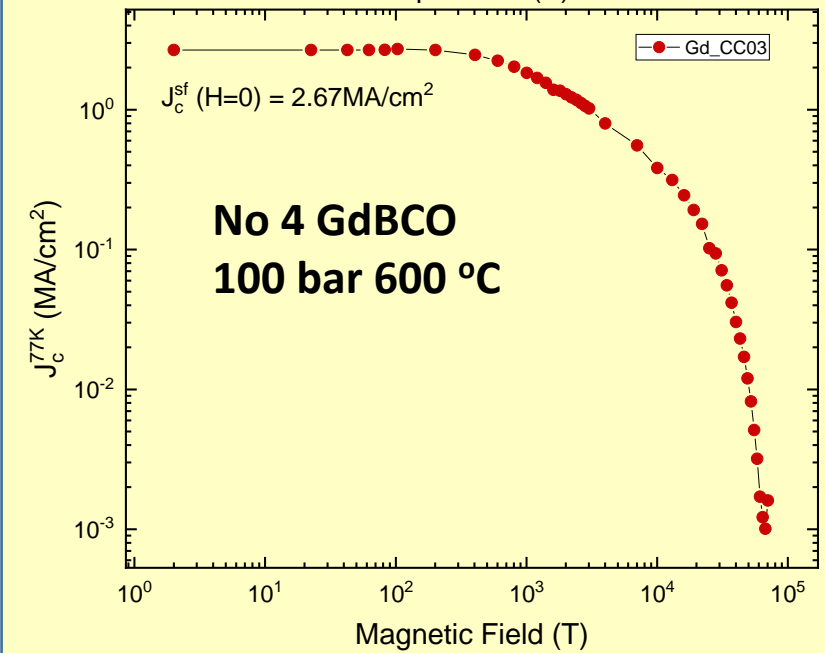
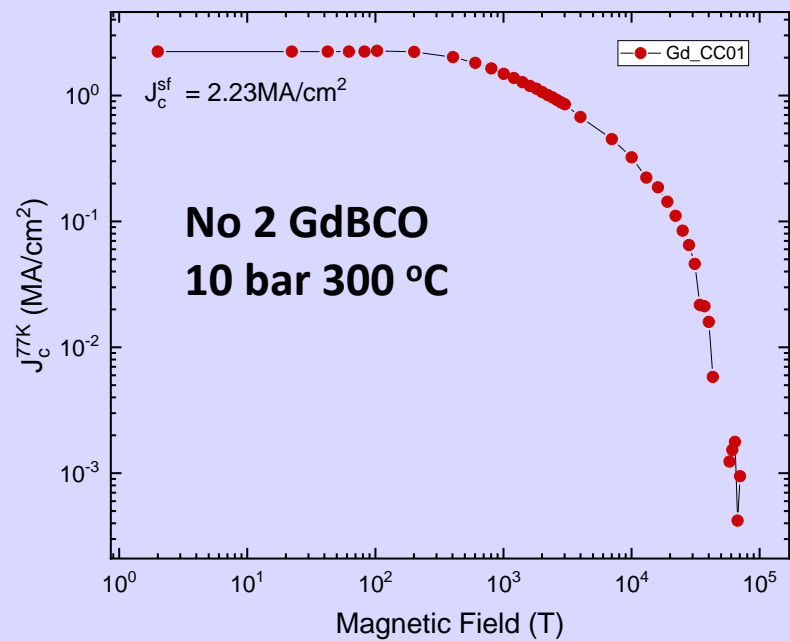
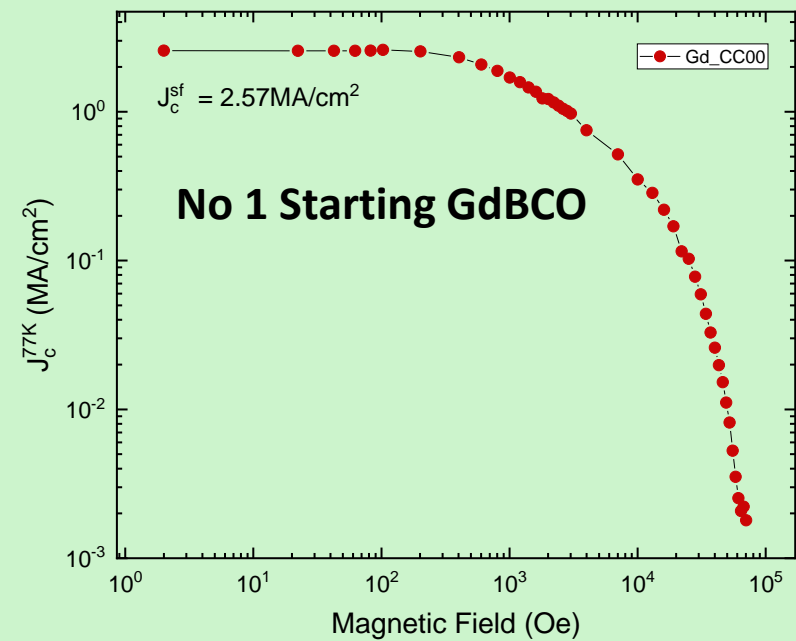
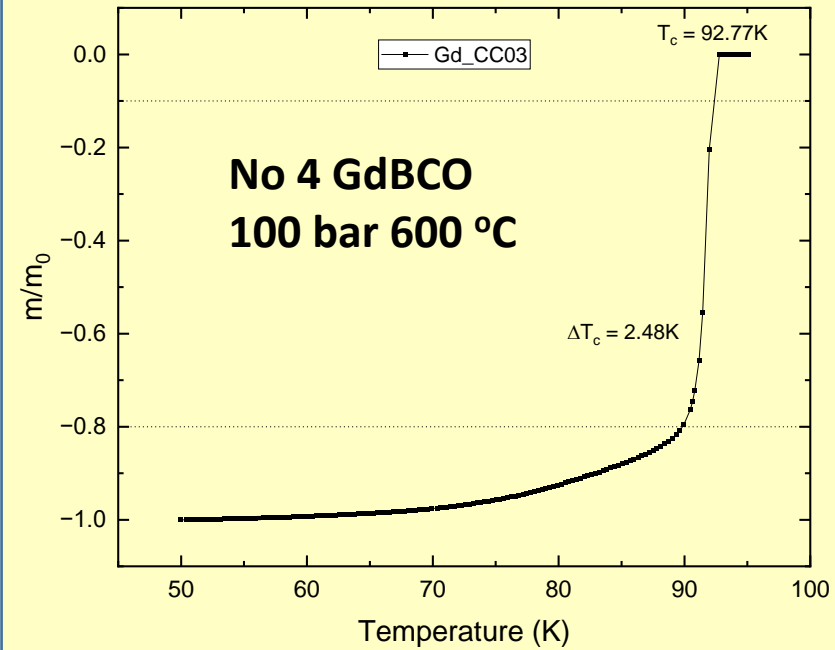
The transition temperature, T_c , and critical current density, J_c , were estimated by SQUID magnetometer and T_c for EuBCO-BHO without Ag layer were estimated using transport measurements; charge carrier density, n_H , were found from the Hall effect at 100 K.



JAMP-9500F, JEOL, Japan –
Scanning Auger Microscopy which is combination
of HRSEM (secondary electron mode, resolution 3 nm)
and Local Auger-electron Spectroscopy (resolution 5 - 8 nm)

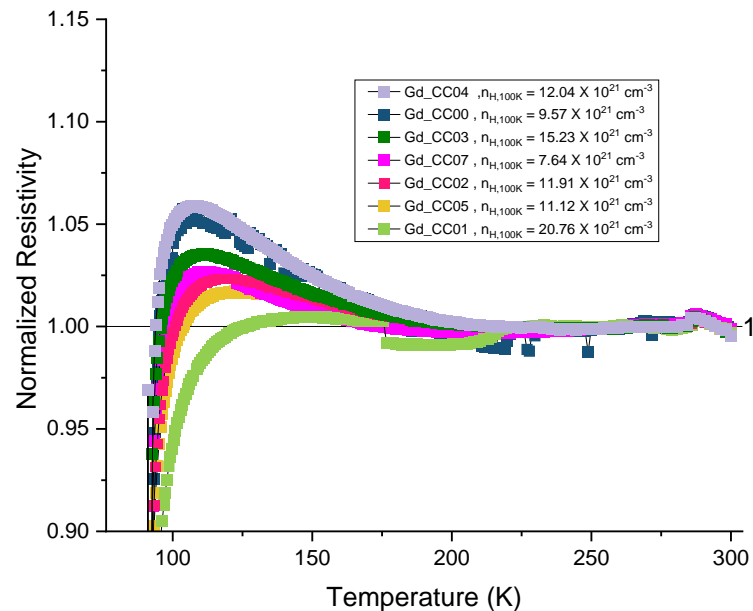
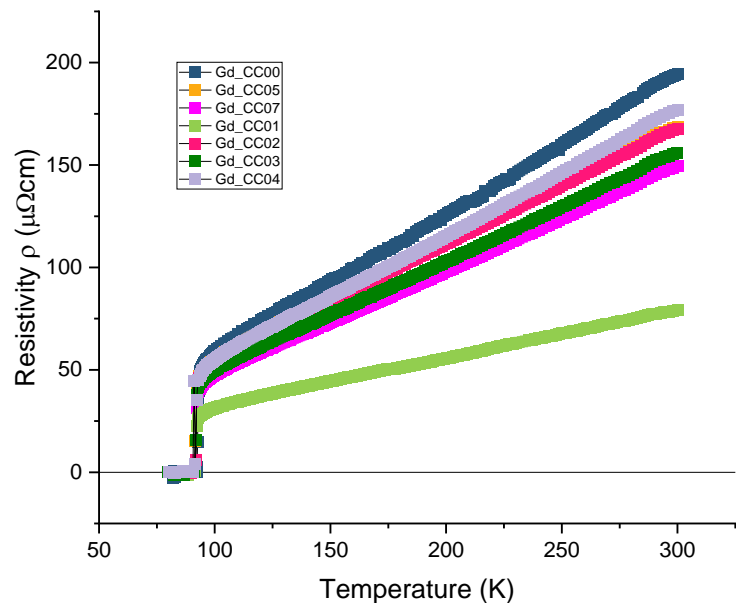
3. Characterization

No	$p(\text{O}_2)$, bar	T_s , °C	τ , h	Mode 1 or 2	J_c (0 T, 77K), MA/cm ²	c-parameter, nm	n_H (100 K), $\times 10^{21}$ cm ⁻³	T_c , K
(1) Ag (2μm)/ GdBCO (1.8μm)/ Al₂O₃/Y₂O₃/MgO/ CeO₂ (700 nm)/ Hastelloy (75 μm)								
1			Starting		2.57	1.17351	9.57	92.75
2	10	300	3	1	2.23	1.17318	20.76	92.71
3	5	300	3	1	2.49	1.17343	11.91	92.60
4	100	600	3	1	2.67	1.17310	15.23	92.77
5	160	800	3	1	2.49	1.1,7300	12.04	92.74
6	2	300	3	1	2.47	1.17289	11.12	93.26
7	1	300	3	1	2.19	1.17320	21.90	92.76
8	100	600	6	1	2.36	-	7.64	-
9	100	600	12	1	2.65	-	43.49	-
(2) EuBCO (2.5μm)+BHO Nanorods/ Al₂O₃/Y₂O₃/MgO/ CeO₂ (700 nm)								
10			Starting		1.38	1.174024	6.05	92.35
11	10	300	3	1	0.87	1.174638	3.77	91.65
12	5	300	3	1	0.99	1.174309	3.85	-
13	100	600	3	1	0.77	1.174261	4.02	-
14	160	800	3	1	1.31	1.173633	-	-
15	2	300	3	1	1.09	1.174117	4.43	-
16	1	300	3	1	1.09	1.174015	4.33	91.65
17	100	600	3	1	0.97	1.17405	4.92	92.22
(3) Ag (2μm)/ EuBCO (2.5μm)+BHO Nanorods/ Al₂O₃/Y₂O₃/MgO/ CeO₂ (700 nm)/ Hastelloy (50 μm)								
18	160	600	3	2	1.32	1.17488	-	93.04
19	160	800	3	2	0.69	1.17471	6.69	92.09

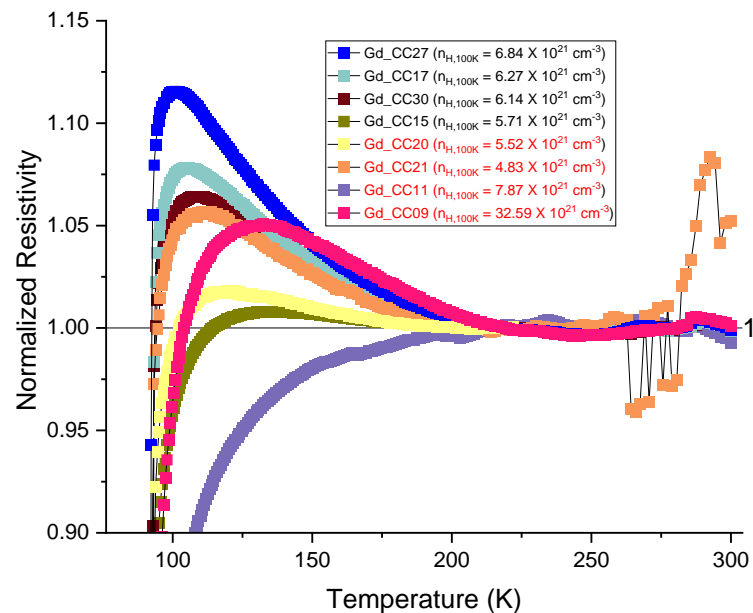
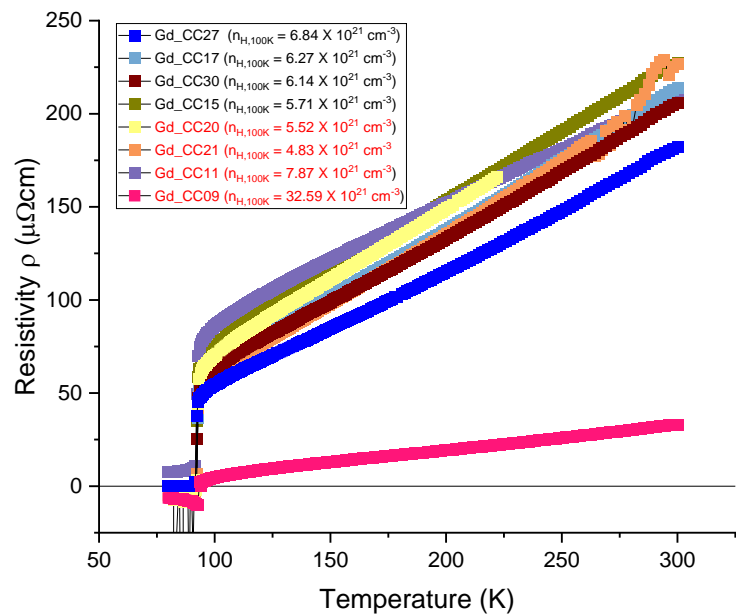
Gd_CC00**Gd_CC01****Gd_CC03**

Comparison of normalized resistivity

Mode 1



Mode 2



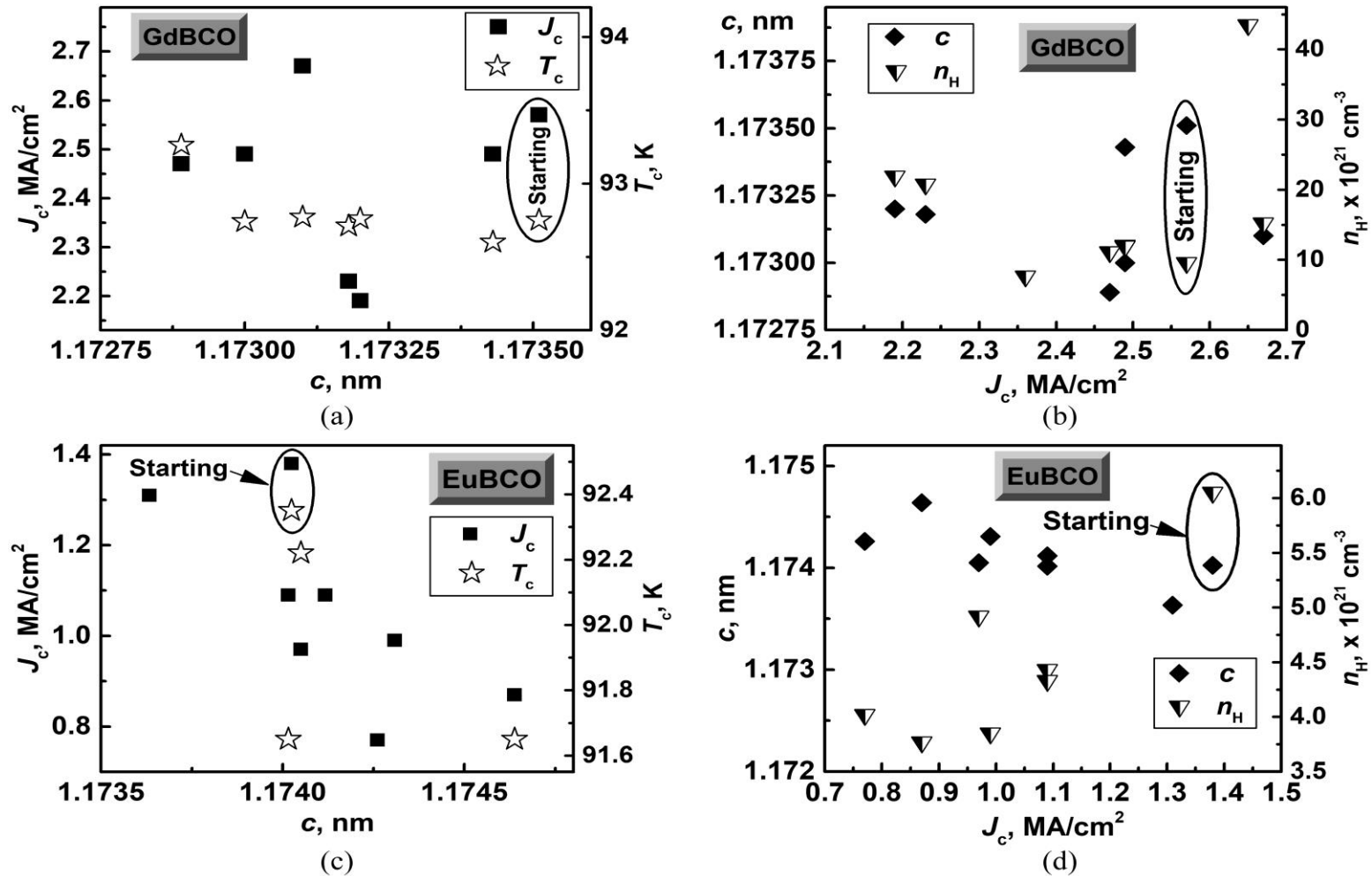
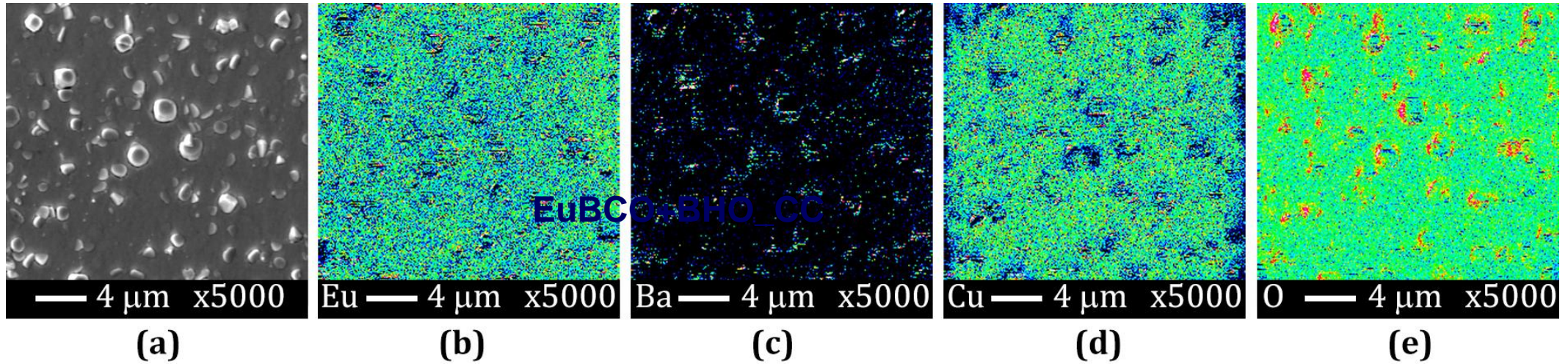


Figure. (a, b) – Critical current density, J_c , at 77K in 0 T and transition temperature T_c vs c -lattice parameters of GdBCO with Ag layer on the top and of EuBCO+BHO nanorods without Ag layer on the top, respectively; (c, d) - c -lattice parameters and charge carrier density, n_H (100K) vs critical current density, J_c , at 77K in 0 T of of GdBCO with Ag layer on the top and of EuBCO+BHO nanorods without Ag layer on the top, respectively.

GdBCO_CC (starting)



EuBCO+BHO_CC (starting)

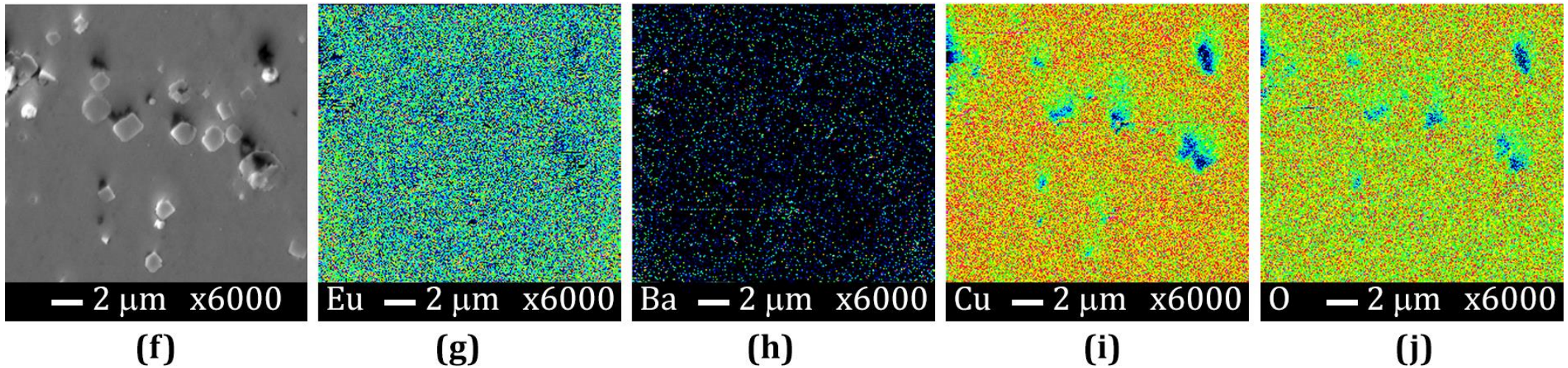
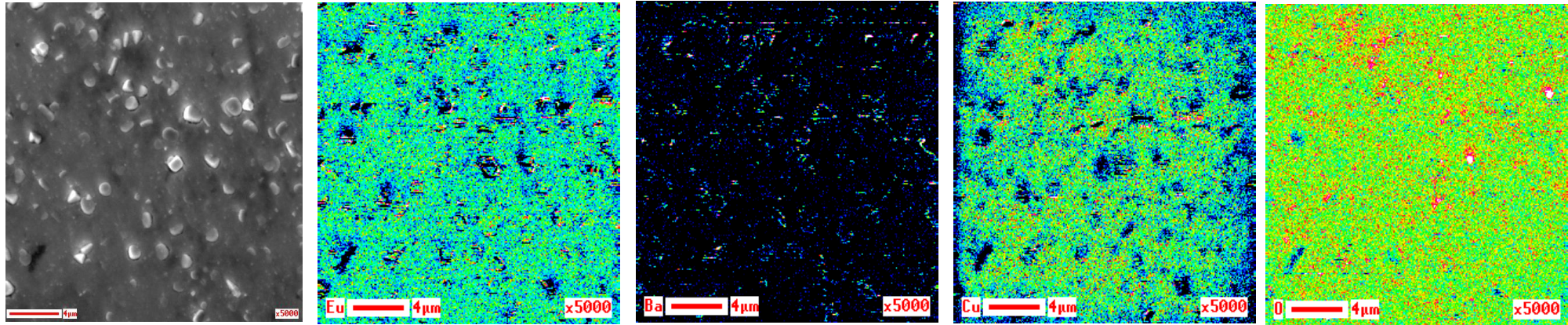
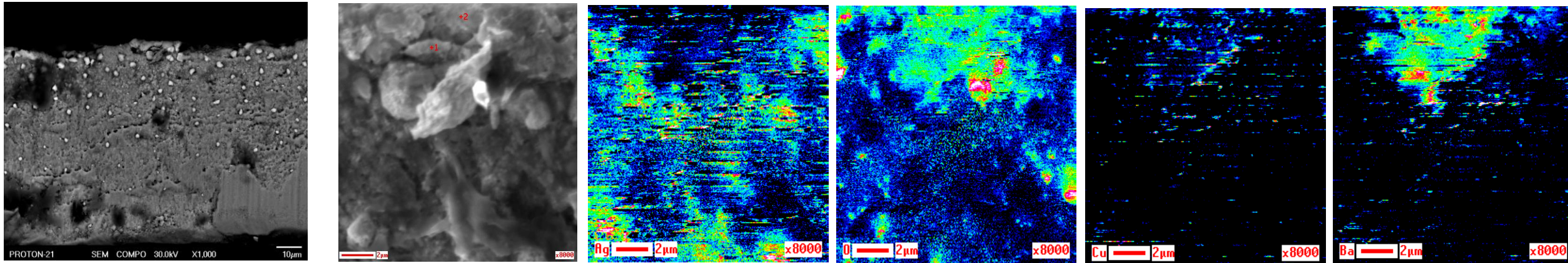


Figure (a, f) – Images in SEI (secondary electrons) mode of GdBCO₂ and EuBCO₂+BHO₂ (materials without Cu and Ag layers, which were removed by etching in acids), respectively, and **Auger maps** of elements distribution over these images: (b-e) – Eu, Ba, Cu, O; (g-j) – Gd, Ba, Cu, O.

No 14 EuBCO+BHO_CC (160 bar O₂, 800 °C, 3 h)



No 4 GdBCO_CC (100 bar O₂, 600 °C, 3 h)



S1

Elements	Back	Peak	RSF	Intensity	at.%
C	-77	45	0,08	122	12,3
O	-753	626	0,365	1379	30,4
Cl	-294	257	0,7	551	6,3
Cu	-63	41	0,21	104	4,0
Rh	-32	39	0,5	71	1,1
Ag	-131	164	0,78	295	3,0
Ba	-115	116	0,085	231	21,9
Gd	-21	44	0,025	65	20,9

S2

Elements	Back	Peak	RSF	Intensity	at.%
C	-136	66	0,08	202	14,2
O	-1255	1063	0,365	2318	35,7
Cl	-103	104	0,7	207	1,7
Cu	-87	63	0,21	150	4,0
Rh	-22	51	0,5	73	0,8
Ag	-211	279	0,78	490	3,5
Ba	-201	166	0,085	367	24,3
Gd	-15	55	0,025	70	15,8

Table 2. Approximate stoichiometry (according EDS study) and amount of oxygen, at. % (on 1 atom of Cu estimated by Auger quantitative analysis) in GdBCO and EuDCO+ BHO. The numbering is the same as in Table 1.

No	Material	Approximate stoichiometry according SEM EDS		Amount of oxygen according to Auger analyses, at.%	
		Matrix	Inclusion	Matrix	Inclusion
1	GdBCO_00_CC	$\text{GdBa}_{1.8}\text{Cu}_{2.6}\text{O}_7$	$\text{GdBa}_{1.8}\text{Cu}_{2.6}\text{O}_{7.8}$	60.8	62.2
2	GdBCO_01_CC	$\text{GdBa}_2\text{Cu}_{2.8}\text{O}_x\text{Ag}_y^*$	Did not estimated	30.4	Did not estimated
4	GdBCO_03_CC	$\text{GdBa}_{1.4}\text{Cu}_{2.7}\text{O}_{8.9}\text{Ag}_y^*$	Did not estimated	52.2-58.2	Did not estimated
10	EuBCO_00_CC	$\text{EuBa}_2\text{Cu}_3\text{O}_8$	$\text{EuBa}_{2.1}\text{Cu}_{2.9}\text{O}_{9.6}$	59.8	60.6
13	EuBCO_03_CC	$\text{EuBa}_{2.1}\text{Cu}_{2.9}\text{O}_{7.8}$	$\text{EuBa}_{2.1}\text{Cu}_{2.9}\text{O}_{8.5}$	55.2	56.2
14	EuBCO_04_CC	$\text{EuBa}_2\text{Cu}_{2.9}\text{O}_{7.1}$	$\text{EuBa}_2\text{Cu}_{2.9}\text{O}_{8.3}$	54.0	52.2

* The compositions of GdBCO_01_CC and GdBCO_03_CC could not be analyzed properly due to the presence of an Ag top layer; The analysis of the layered structure was carried out from the cut side partially crushed as a result of the cut. When studying GdBCO_00_CC, the Ag layer was removed and the top side was analyzed.

Conclusions

- Treatment of GdBCO_CC under 100 bar of O₂ at 600 °C for 3 h (cooling Mode 1) led to an increase in J_c (77K, 0 T) by 6% and a decrease in the c-parameter of Gd123 to 1.17310 nm, which may be associated not only with overdoping with oxygen, but also with silver diffusion into Gd123.
- No correlations was observed between J_c , T_c , c-parameter of RE123 (RE=Eu, Gd) and carrier density n_H of EuBCO_CC and GdBCO_CC treated at 300-800 °C, 1-160 bar O₂ for 3-12 h.