

Additive Manufacturing: Fundamentals and Applications

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What is Additive Manufacturing?

Additive manufacturing is defined by the ASTM society as:

"a process of joining materials to make objects from 3D model data, usually layer upon layer, as opposed to subtractive manufacturing methodologies"















Parts are made by adding material in layers; each layer is a thin crosssection of the part derived from the original CAD data The thinner each layer is, the closer the final part will be to the original.

The major ways they differ are:

- in the materials that can be used,
- how the layers are created,
- how the layers are bonded to each other.

Development of Additive Manufacturing Technology

The Use of Layers

A key enabling principle of AM part manufacture is the use of layers as finite 2D cross-sections of the 3D model.

Almost every AM technology builds parts using layers of material added together, primarily due to the simplification of building 3D objects.

Using 2D representations to represent cross-sections of a more complex 3D feature has been common in many applications outside AM.



Classification of AM Processes

Process Technology

Raw Material Laser Fusion/Sintering Electron Beam Melting **Powder + Laser Binder Jetting** Powder Laser Polymerization Light Projection **Material Jetting** Polymerization Liquid **Fused Deposition** Wire + Electron Beam Filament / Wire Ultrasonic bonding Laser Lamination Sheet

Classification of AM Processes



Deposition Rate

AM standards

ISO/ASTM 52910 General Design Guide

ISO/ASTM 52911 PBF Design Guides

ISO/ASTM 52922 DED Design Guide

ISO/ASTM 52912 (TR) Design for functionally Graded Materials

Material Extrusion Design Guide

Design Guide for Post-Processing

ISO/ASTM 52900 AM Terminology



Additive Manufacturing Standards Structure



General Top-Level AM Standards

- General concepts
- Common requirements
- Generally applicable

Category AM Standards

Specific to material category or process category

Specialized AM Standards

Specific to material, process, or application



Powder Bed Fusion

All PBF processes share a basic set of characteristics.

These include one or more thermal sources for inducing:

- fusion between powder particles,
- a method for controlling powder fusion to a prescribed region of each layer,
- a mechanisms for adding and smoothing powder layers.

















In some cases, the build platform is also heated using resistive heaters around the build platform.

This preheating of powder and maintenance of an elevated and uniform temperature within the build platform are necessary:

- to minimize the laser power requirements of the process (with preheating, less laser energy is required for fusion);
- to prevent warping of the part during the build due to non-uniform thermal expansion and contraction (resulting in curling).



















Powders

- The powder material properties affect the further selection of all other process parameters;
- The chemical composition, thermal, optical, metallurgical, mechanical, and rheological characteristics of the material play a key role in L-PBF.
- Typically, L-PBF systems use metal powders ranging in size from 5 to 60 μm;
- Granulo-morphometric properties, such as the particle size, particle shape, elongation, roundness, specific surface area, particle size distribution (PSD), etc., affect the delivery of the powder layer, its homogeneity, and the absorption coefficient of laser radiation;





Powders

- The most suitable powders for L-PBF are those with spherical particle morphology that has a high packing density, good flowability, and are evenly deposited to the substrate;
- Powders containing a significant fraction of small particles of 1-2 μ m in size are easily agglomerated and cannot be properly deposited to the substrate.



Al-based alloys	AlSi10Mg, AlSi7Mg0.6, AlSi9Cu3
Ni-based alloys	Nickel alloy HX, IN625, IN718, IN939
Ti-based alloys	TiAl6V4, TiAl6V4 ELI, TA15, CP (commercially pure) Ti
Co-based alloys	CoCr28Mo6, CoCr28W9
Fe-based alloys	304L, 316L, 15-5PH, 17-4PH, Maraging Steel 1.2709, Maraging Steel M300, H13, Invar 36, 20MnCr5 steel, Stainless Steel CX
Cu-based alloys	CuNi2SiCr, CuSn10, CP Cu, CuCr1Zr
Precious metals	Gold (Au), Silver (Ag), Platinum (Pt), Palladium (Pd)
Refractory metals	Tungsten (W), Molybdenum (Mo)

POWDER REMOVAL







| Focus: the powder bed

- Speed of the recoating device.
- Layer thickness.
- Shape of the recoating device.



- Powder bed solid volume fraction.
- Effective layer thickness.
- Material/shape segregation.












Discrete element method



Force-displacement law:

$$F = (k_n \delta n_{ij} - y_n v n_{ij}) + (k_t \delta t_{ij} - y_t v t_{ij})$$

Nonlinear Hertz-Mindlin

Additional cohesive force simplified Johnson-Kendall-Roberts

www

$$k_n = \frac{4}{3} E' \sqrt{R' \delta_n}$$

F = kA

| Powder Spreading simulation

• Material segregation



• Velocity profile



- Local packing factor and density
- Local variation of the PSD
- Effective layer thickness

Experimental device





Control panel with 6 different blade's speeds and selectable temperature for the plate

Micrometric screws to adjust the layer thickness and powder feedstock

Set of portable microscopes with different leves of magnification

Experimental device





Analysis of Sem images to study variations in the PSD and characteristcs of the powders



Acquire images on situ at different levels of zoom with a portable microscope to perform analysis of the powder bed Extract samples of the powder bed to measure the local density



POWDER HANDLING: POWDER BED DEPOSITION

SPREADER GEOMETRY



Powder removal, gas supply, and filtration systems

- Since the powder has a large specific surface area, to prevent the metal material from intense oxidation, the L-PBF process takes place in an inert gas atmosphere;
- For more inert (resistant to oxidation) metal alloys (Ni-based, Co-based, Fe-based, etc.), nitrogen is used, and for more active metal alloys (Al-based, Ti-based, etc.), argon is used;
- As a result of the interaction of laser radiation with metal powder, intensive evaporation and ejection of the material occurs.
- Particles entrained by the evaporation-driven protective gas flow are dragged into the melting pool or ejected.
- The spatter ejection depends on protective gas flows, laser plume, and dynamics of the melt pool.

BALLING EFFECT



Powder removal, gas supply, and filtration systems

- To prevent contamination of the surface of the powder layer a filtration system is used.
- When changing protective gases, it is imperative to change filters:
 - o nitrogen and argon have different physical properties and different gas permeability,
 - various metals deposited on the filters can react chemically, which can cause the filters to ignite and destroy the L-PBF system.
- A directional flow of shielding inert gas, which uniformly flows directly over the surface of the powder layer, removes products of the process (metal condensate and spatter particles) from the laser-powder interaction zone.

INERT GAS FLOW



Melt-pool dynamics and track formation

• When the laser beam scans over the surface of a thin powder layer, energy absorbed from the laser beam heats the underlying material;





D=80 μm D=207 μm







Cozzolino, E., Tiley, A.J., Ramirez, A.J., Astarita A., Herderick E. (2024) Energy efficiency of Gaussian and ring profiles for LPBF of nickel alloy 718. Int J Adv Manuf Technol. https://doi.org/10.1007/s00170-024-13511-0



S.M. Thompson et al. An overview of Direct Laser Deposition for additive manufacturing; Part I: Transport phenomena, modeling and diagnostics, Addit. Manuf. 8 (2015) 36-62.



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- Molten powder particles and the substrate create a joint melt pool;
- Heating, time-evolution of the melt pool, and the solidification process depend on powder material properties, process parameters, and the build environment;
- The process parameters affect the phases, recoil pressure, surface tension and Marangoni effect, and hydrodynamics that in turn define the evolution of the melt pool, its size, and shape;



- The Marangoni effect (also called the Gibbs–Marangoni effect) is the mass transfer along an interface between two phases due to a gradient of the surface tension;
- Since a liquid with a high surface tension pulls more strongly on the surrounding liquid than one with a low surface tension, the presence of a gradient in surface tension will naturally cause the liquid to flow away from regions of low surface tension;
- The surface tension gradient can be caused by concentration gradient or by a temperature gradient (surface tension is a function of temperature).





- When the laser beam leaves the melt area, the melt pool starts to cool down and solidifies;
- To create a stable melt pool with a regular shape and geometrical characteristics, several factors have to be consistent with each other;
- Sufficient energy is needed to melt both the powder and the substrate under the laser beam and the interaction time has to be optimized to create a joint stable melt pool.

Melt-pool dynamics and track formation

The energy absorbed by material is strictly dependent on laser characteristics such as:

- \circ wavelength,
- o pulse width,
- \circ frequency,
- average and maximum power,
- o intensity profile,
- \circ laser mode,
- \circ spot size,
- o irradiation time,
- \circ etc.



- The key energy parameters of L-PBF are laser power and focused beam diameter (spot size);
- The smaller the spot size, the fewer powder particles interact directly with the laser beam:



- A single track is formed not only from powder placed directly under the laser spot; adjacent particles are involved in the process and a denudation zone forms (Yadroitsev, 2009);
- The powder denudation zone defines the volume of powder involved in the track formation and spattering process.



- The dynamics of denudation depend on the geometry of the melt pool, the metal vapor flow that is induced by heating under the laser beam, and ambient gas pressure;
- At a typical pressure of about 1 atm in the L-PBF processing chamber, the dominant factor for the denudation is gas flow caused by pressure drops inside the evaporated jet that entrain powder particles surrounding the melt pool into the process;
- Powder particles from the denudation zone are pulled into the melt pool or ejected away;
- Powder particles, or their agglomerates that were partially melted, can create "satellites" at the edges of the track;
- Satellites can also occur from the spattering effect, when powder particles and melt droplets are ejected during the L-PBF process.

- Spatter particles can be divided into three main classes:
 - o particles that travel toward the vapor jet and miss the laser beam, or "cold" spatters;
 - "hot" particles that cross the laser beam;
 - ejections from the melt pool due to melt dynamics and recoil pressure (Ly et al., 2017).
- Energy input affects the size and dynamics of spattering: the general trend is that a higher laser power leads to more intense spatter behavior (Liu et al., 2015).

- Hot spatters and melt droplets are visible as bright sparks around the track when the laser beam scans the powder bed;
- The diameters of the melt droplets can be bigger than the original powder particles, thus changing the effective particle size distribution of the feedstock powder.
- When energy from the laser beam is enough to intensively evaporate underlying material, vaporized material solidifies
 rapidly in the protective gas; this is visible as a "fume" during the laser melting process and creates dark spots on the
 processed layer;
- These condensates look like a fluffy coating consisting of nano-sized particles.





- The morphology of single tracks has a complex dependence on process parameters;
- The mechanisms of distortions and irregularities in single tracks are associated with:
 - o thermophysical properties of materials,
 - o granulomorphometric characteristics of the powder,
 - inhomogeneity in powder layer thickness;
- energy input parameters such as:
 - laser power,
 - o spot size,
 - scanning speed;
- build environmental parameters.

Process stability

- Analysis of the formation of single tracks from metal powders by L-PBF showed that the process has a threshold character;
- There are continuous tracks with regular sizes and ripples (A and B), continuous tracks having periodic humps and valleys (humping effect, C), tracks with irregular flow front with many satellites (D), and irregular tracks with highly varying widths and heights (E and F) up to a chain of beads, the so-called "balling effect."



~30 µm powder layer thickness

Ti6Al4V powder

~50 μm powder layer thickness

Process stability

• The evolution of a single track from a regular shape to a chain of beads with increasing scanning speed...



316L stainless steel powder

- The evolution of a single track from a regular shape to a chain of beads with increasing scanning speed as a function of thickness layer at a constant Power level
- Droplets are formed as the result of melt pool disintegration because of a capillary instability. A long cylinder of liquid tends to break up into drops with the same volume but smaller surface. This effect is known as the Plateau-Rayleigh instability.





- One can model the melt pool as a circular cylinder of diameter D and length L. Such a cylinder is stable if its aspect ratio $L/D < \pi$ and unstable otherwise ????
- The aspect ratio of the melt pool increases with the scan speed
- The Plateau-Rayleigh instability of a circular cylinder explains the loss of stability with increasing the scan speed



- The morphology of the L-PBF Ti6Al4V track with expressed humping effect is shown in fig;
- It can be seen that the depth of penetration is sufficiently high, and the single track is continuous.









- The Humping effect is a phenomenon which is observed approximately since 50 years in various welding procedures and is characterized by droplets due to a pileup of the melt pool. It occurs within a broad range of process parameters;
- It is necessary to clearly distinguish the balling and humping effects since they have different origins and both influence the final part quality.



- Similar to welding, the humping effect was found in the L-PBF process and results in continuous tracks which can have periodic waviness of the profile and undercuts.
- It was found that in L-PBF, the humping effect is very pronounced at high laser power and high scanning speed.
- The low surface tension and positive surface tension gradient, recoil pressure, and viscous shear stress contribute to the humping effect.



- The deep penetration of the melt pool into the substrate observed during the humping effect can make it possible to build nonporous parts from irregular (humping) tracks by reducing the hatch distance (shift between center of tracks).
- With the balling effect, when the penetration depth is very small, lack of fusion porosity will be very pronounced in 3D objects.
- Humping as well as balling effects are undesirable processes in L-PBF, since they can lead to inhomogeneity of the following powder layer or to collision with the recoater/roller that deposits the powder layer.
- An impact can deform the L-PBF part, recoater, or even the whole system.



Process stability: OXYGEN

- Different materials show different and sometimes peculiar behavior in the process of single track formation.
- A high content of oxides on the surface of the substrate is undesirable: oxides change the absorption coefficient of laser radiation and increase the melting temperature of the scanned material.
- This affects the wettability of the substrate by molten material, and can lead to the balling effect.
- Oxides also contribute to the formation of cracks.


Process stability: ROUGHNESS

- With high roughness it is impossible to deliver a thin homogeneous powder layer.
- Inversely, particles on top of a mirror-polished substrate surface can roll easily and move during deposition leading to an apparently low powder density.
- The roughness of the substrate surface therefore influences the morphology and geometry of single tracks.
- It is recommended to use machined substrates with an average roughness and wavelength of the same order of magnitude as the powder particle size.



Process stability: SINGLE TRACK CHARACTERIZATION

- Key features that are used for single tracks' characterization are the shape of the tracks and their geometric dimensions.
- From the top view without cross-sectioning, tracks can be evaluated as having one or more of the following characteristics:
 - a) continuous and uniform,
 - b) transitional (continuous but with necking or irregularities),
 - c) having expressed swelling and depression zones (humping effect),
 - d) consisting of a chain of beads (balling effect),
 - e) cracks,
 - f) satellites,
 - g) droplets near the sintered track.

Process stability

• The main geometrical characteristics of single tracks are:

o width,

- \circ remelted depth (or penetration depth),
- o height.



Process stability

- The most influencing factors on the remelted depth of single tracks are laser power density and interaction time.
- With a constant layer thickness, penetration depth increases with laser power density.



to keyhole mode

Schema for conduction transiton - keyhole modes

Keyhole mode

Process stability



Process stability





- Each solidified L-PBF layer is a superposition of single tracks.
- Its surface morphology depends on the morphology and the geometrical characteristics of individual single tracks, the scanning strategy, and the hatch distance, which is the shift between tracks in the plane of the laser beam scanning.
- The start-and-stop effect in the hatched area leads to a specific shape of the edge; the attached powder particles and irregularities deteriorate accuracy and surface quality and so contouring is often used.





- Different laser power and scanning speeds can be used for hatching and contour areas;
- Scanning strategies represent the manner of scanning of a cross-section;
- In a layer, different hatching patterns can be realized; the more frequently used methods are scanning by stripes, islands (chess-board), or the whole cross-section is scanned without partitioning by elementary hatching patterns;
- In-layer patterns can vary in size and can be done in a different order:
 - Randomly or sequentially,
 - stripe-by-stripe or island-by-island.
- Hatching patterns always overlap to avoid porosity;
- Laser beam scanning inside each pattern can be done in one direction, zigzag (back-andforth), spiral, or other programmed laser beam movements.



- The beginning-end effect of single tracks and the changing direction of the laser beam or path (when the laser beam accelerates/decelerates and turns around) has an influence on the melt-pool size and morphology of single tracks.
- In single layers and 3D parts this leads to edge and corner effects where rough and irregular surfaces may occur.
- To decrease edge ridges and corner effects it is recommended optimizing laser power and scanning speed, as well as to scan top layers several times with lower linear energy input.
- In a mirror-based laser scanning system for L-PBF, mirrors accelerate and decelerate in turning points which can be one possible reason for overheating and keyhole porosity.
- A skywriting option, incorporated in L-PBF EOS systems, shuts off the laser beam when the scanner is positioning the beam for scanning so that powder material does not melt during positioning.

- Pre-contouring (scanning the contour of the part before hatching) or post-contouring (after hatching) can be used to improve in-plane (XY) accuracy and surface roughness
- Different scanning parameters can be used for contouring.
- Since the melt pool is bigger than the laser spot size, the laser beam has to be offset from the edges of the scanned cross-section to compensate for this difference and to provide accurate dimensions of the part to be manufactured.



Single layer formation

• A special "power profile strategy" that adjusts the laser power depending on the position of the laser scanning was proposed by Martin et al. (2019) to mitigate keyhole defects near the edges.

Characterization of a single layer

- Nonuniform thickness of the deposited powder layer can also influence the morphology of single tracks and single layers.
- It could be critical for the density of the 3D object: if there is insufficient energy to remelt the thickest powder layer, the balling effect starts, which provokes porosity formation.



Characterization of a single layer

• The surface morphology after laser melting includes peaks and valleys, attached powder particles and droplets, i.e., spatters.





Characterization of a single layer

• For the characterization of L-PBF parts, the surface roughness, waviness, deviation from prescribed dimensions, presence of spatters and surface pores have to be analyzed.



Characterization of a single layer

- If a single layer was built with nonregular tracks or with nonoptimal hatch distance, the surface has irregular morphology.
- For characterization of a single layer, SEM images, optical 3D measurement techniques, and CT scans are often used.
- The balling effect, cracks, and overlapping can be identified on the top view; CT scans and cross-sections help to find internal porosity and other defects.





Thin wall formation

- A thin wall can be defined as an object consisting of single tracks superimposed on each other in the vertical direction (Z-axis).
- A thin wall can be considered as a single layer manufactured in the vertical direction.
- Thin walls may also act as indicators of the manufacturing quality for fine features, for a given set of process parameters.

Thin wall formation

 the surface of single-pass thin walls fabricated at a laser power of 50 W and a spot size of 70 μm, with a gradual increase in the layer thickness from 40 to 80 mm with a step of 10 mm for each of 20 layers



Thin wall formation

- Powder particles attached to the surface of thin walls are highly influential on surface roughness and dimensions of the walls.
- Scan length has an influence on the accuracy of thin-wall production. If the thin wall consists of more than one pass, its orientation relative to the scan tracks has an influence on the topology of the thin walls.



Thin wall formation

- When a complex specimen consists of thin walls, the surface topology is quite different: there are smooth areas and extremely rough regions: the protective gas flow and turbulence can be responsible for these phenomena.
- When a part has a rough surface, the recoating blade can start to make contact with the specimen that increases the risk of distortion of the specimen, delamination is possible, and this can lead to damage of the powder deposition system.
- There are limitations in accuracy and surface roughness when fine components are produced such as thin walls.
- This has been demonstrated in numerous studies on lattice structures, where fine microwalls and struts are decisive factors in perfecting L-PBF structures

- L-PBF provides freedom of design and allows the manufacturing of complex structures such as lattice structures, topology optimized parts, graded structures, and parts with integrated functions.
- However, there are some limitations and specific features typical of the powder bed fusion process, for example, the dimensional accuracy and surface finishing of the parts.
- Fine structures and minimum feature sizes are limited by powder material and process parameters, such as spot size, laser power, scanning speed, layer thickness, scanning strategy, etc.

- A 3D component can be divided into three parts: the core part, upskin, and downskin regions.
- Areas that have no upper layers are called upskin; conversely, downskin has no underlying solidified layer.
- The inner region of the component is called the core part.
- For all these areas, the scanning strategy needs to be optimized.









- There are many possible ways of scanning:
 - scanning the whole component with similar process parameters or scan by stripes or islands;
 - o scanning with different patterns, such as in one direction, zigzag, spiral, etc.;
 - rescanning of the solidified layer;
 - rescanning only specific areas; changing the scanning direction for each subsequent layer, etc.
- Each of the scanning strategies can be applied to achieve specific goals: to improve density, surface quality, and manufacturing accuracy; to decrease residual stress; to achieve a specific microstructure, etc.



- Side surfaces of L-PBF parts always make contact with powder material, so these surfaces often show many attached powder particles and pronounced layered structure.
- The layer-by-layer L-PBF process results in the surface quality being different in the vertical (between layers, Z-direction) and horizontal directions (in-layer, XY direction).
- The powder material and the track-by-track, layerwise nature of L-PBF govern the surface topology of L-PBF parts.



- Upward (upskin) and downward (downskin) surfaces are different in L-PBF, and frequently downskin is rougher and has poorer surface quality in comparison with upskin.
- To improve surface quality, upward surfaces can be rescanned several times with a special pattern similar to laser polishing, while side and internal surfaces require special post-processing surface finishing.
- Powder from external surfaces of a 3D object can be removed with compressed air, ultrasonic bathing, mechanically, with chemical reagents, plasma and electrochemical methods, etc.
- However, for lattice structures and parts with small channels with complex shape, powder cleaning presents a real problem that limits applications of powder bed fusion manufacturing.









- Objects of complex shapes can contain elements that are at an angle to the base plate, when the critical angle axis exceeds 45 degrees, it needs to be supported.
- During manufacturing, supported overhanging parts have different cooling rates compared with unsupported components; this influences the microstructure and mechanical properties.
- Areas with high residual stress can be determined by simulations of the process and an optimal support type with sufficient strength can be chosen for further manufacturing.




Anytime material is deposited in AM needs to be attached to something else by a physical connection.

In addition, in a process which deposits molten material, that material must be supported by something underneath it.

These reasons, plus the need to mitigate the effects of residual-stress-induced and shrinkage-induced warpages means that support structures are needed in many AM processes.



Most AM processes need supports if printed surface is less than 45° from the horizontal.

This is particularly true for MEX and for PBF of metals with respect to the interaction of dynamic shear force in the liquid phase and gravity forces in the solid phase.



When using supports in AM processes, differences in cooling rates between the bulk of the material and contact points with supports increases the risk of the cracks.



In addition, supports must be removed after the build process by post-process operations such as machining.



- Optimization and minimization of support structures improve process efficiency, reduce deformation, and improve quality of L-PBF components.
- Supports in L-PBF is a system of thin walls, pins, and cellular structures that serve several purposes:
 - o heat dissipation,
 - $\circ\;$ to fix the part,
 - to stiffen the structure,
 - o to resist deformation and bending of the parts during the manufacturing process,
 - to provide a convenient and simple separation of the finished part from the base plate.
- L-PBF makes use of specialized software tools that can generate different types of supports and allows for the selection of certain configurations to change the type of supports and size of their contact zone with the part.







Porosity in AM

Similar to cast metals, porosity in AM occurs in specific forms and these are related to specific mechanisms of the additive process used

In L-PBF there are numerous pore formation mechanisms

The presence of porosity in L-PBF is widely attributed to process parameters

The two most well-known and widely occurring forms of porosity in L-PBF are lack of fusion porosity and keyhole porosity:



Porosity in AM

lack of fusion porosity

keyhole porosity



Porosity in AM

- Lack of fusion (LoF) porosity occurs when insufficient melting occurs, either due to too high scan speed or too low laser power for the selected powder layer thickness
- This type of porosity is irregular in shape and may contain unmelted powder particles
- These pores may occur in different sizes and, due to the irregular shape, typically have sharp edges.
- These sharp edges act as stress concentrators under load, causing a significant effect on mechanical properties



Porosity in AM

The keyhole porosity occurs when the laser power is high and scan speed is low

Keyhole mode melting occurs when the laser causes sufficient material vaporization to create a vapor cavity which creates a depression in the surface

This depression or keyhole cavity may penetrate deeply into the melt pool, and may become unstable and collapse as the melt pool moves, leaving a trapped vapor cavity (or keyhole pore) in the track as it solidifies

This porosity type is more rounded in shape and has a lower influence on mechanical properties if its presence is in low quantities or in small size



Porosities











Single layers



Pore formation mechanisms and porosity types

Single layers



Pore formation mechanisms and porosity types

Single layers

The hatch scanning of the core of the part is usually followed by contour scanning

Contour scanning may use different process parameters to improve surface properties

The overlap between the hatch core tracks and contour tracks needs to be sufficient, as this region may be particularly prone to porosity formation



Pore formation mechanisms and porosity types

Scan Strategy Surface (IFM) Porosity (CT) Top (SEM) Bottom (SEM) Antiparallel Lines α' α Island Stripes • 5 µm

Island Modelled

Pore formation mechanisms and porosity types

Multiple layers

This remelting process results in interesting 3D distributions of the remaining pores in different types of checkerboard patterns of porosity



Checkerboard pattern resulting from LoF tracks remelted at 90° layer rotations

LoF regions are between tracks

Pore formation mechanisms and porosity types

Summary of pore types

Porosity name	Mechanism	Shape, distribution, and size
Low energy lack of fusion (LoF)	Insufficient melting, can be due to various reasons:	Irregular shaped pores with sharp edges, can be aligned perpendicular to or parallel to the build
	 Low energy causing insufficient penetration to previous layer Low energy causing insufficient overlap between tracks (or too large hatch spacing) Insufficient track overlap between hatch and contours near surfaces of material 	direction, depending on the LoF with previous layers or between tracks. Large LoF pores may contain unmelled powder.
Keyhole	Excessive energy creating rounded pores inside the solidified tracks, can occur:	Rounded pores all along tracks or at end of scar tracks. At intermediate powers this also occurs, the pores are simply smaller.
	 All along the track Only at turnaround points where laser slows down to turn around or switches off leaving keyhole to collapse Size of pores depends on keyhole size which scales with laser power and spot size 	
Track instability LoF	Instability and uneven tracks lead to variations in powder layer thickness on the next layer, and in critical cases balling or humping occurs, creating special forms of LoF on the next layer due to varying thickness of the powder layer	Irregular and sometimes large LoF pores.
Turbulent melt-pool induced LoF	Conditions of melt-pool turbulence creating entrapment of pores, particle and spatter ejection, denudation, and combinations of pore formation mechanisms	Large irregular LoF pores.
Powder porosity	Pores inside powders trapped inside melt pool and do not get time to escape as the solidification is too fast	Rounded and small (smaller than powder particles which are generally about 30 µm).

Pore formation mechanisms and porosity types

Summary of pore types

Checkerboard pores	Stop-start pores		Local heating induced keyhole porosity	Uneven powder bed LoF	Gas entrapment
When LoF occurs with some scan strategies, remelting on subsequent layers closes some pores. May occur between tracks or between islands or stripes—regions where porosity often occurs	When system stops and restarts, part cooling and shrinkage creates a thicker powder layer and insufficient melting of new layer	 At supported areas but between supports At solid areas where thermal dissipation is lower, e.g., in corners of parts 	Local heating creates different melt-pool dynamics creating conditions for pore formation and trapping in solidified material; this can be due to lack of solid material under it (overhang regions) for thermal dissipation	Uneven powder bed due to scraper damage, local powder clumping, or inregular powders (e.g., re-used powder). Uneven powder bed affects melt-pool stability and indirectly causes pores. In extreme cases, for example, a large particle will not melt sufficiently creating lack of fusion under the region	Powder packing density or shielding gas related. Pores are entrained into the melt pool and move inside the melt pool but cannot escape before solidification. Entrapment depends on melt-pool conditions as well as shielding gas and powder packing density
Small pores in regular grid-patterns.	Layered pores in build plane.		Small pores at down-skin regions and at narrow features.	Elongated pores in build plane (pore tracks), or similar to layered LoF.	Small pores trapped into melt pool, found randomly all over part—typically much smaller than melt pool (also in region of 30 µm).

Porosity measurement

Optical microscopy

The process of sectioning, polishing, and etching parts is already in wide use for microstructural analysis; therefore, the analysis of porosity is often done in tandem.











Effect of defects

Porosity is widely present in different forms and in different extents.

But at what point is this a problem?

Is there a threshold value that can be defined as being "acceptable"?

Is there a critical pore size?

How does porosity affect the properties of the parts?

A case study

A @ 90° B @ 0° C @ 60°



A case study



A case study



Fracture surface images, at different magnifications, of a type A specimen

Low stressed and lasted over 1 * 10⁶ cycles

A case study



Fracture surface images, at different magnifications, of a type A specimen

Low stressed and lasted over 1 * 10⁶ cycles

A case study



Fracture surface images, at different magnifications, of a type A specimen

Low stressed and lasted over 1 * 10⁶ cycles

A case study



Fracture surface images, at different magnifications, of a type C specimen

Low stressed and lasted slightly less than 1 * 106 cycles

A case study



Fracture surface images, at different magnifications, of a type C specimen

Low stressed and lasted slightly less than 1 * 106 cycles

Pore closure and mitigation

Hot isostatic pressing



Pore closure and mitigation

Hot isostatic pressing

This high temperature and high pressure process consolidates pores entirely and closes all forms of pores in L-PBF <u>except those very near the surface in isolated cases</u>

Internal pores are easily closed but near-surface pores connected to the surface make the HIP process ineffective and are not closed

Pore closure and mitigation

Hot isostatic pressing

HIP is performed not only for pore closure but it also coarsens the microstructure removing anisotropy and improving the ductility of the material

This typically improves the fatigue strength, and a partial role is played by the closure of pores in this improvement

Near-surface pores may remain problematic which also highlights the need for surface processing or strategies to improve the surface finish in the L-PBF process itself
Porosity in laser powder bed fusion

Pore closure and mitigation

Hot isostatic pressing





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____1.5 mm

Rapid Tooling

SL resins expand more than wax and crack the ceramic shell.

The QuickCast[™] pattern style was added automatically to hollow part STL files by SL machine pre-processing software.

The QuickCast[™] lattice struts were designed to flex, collapse inward, break, but not transfer high loads to the part skins which could crack the shell.



Rapid Tooling

Boeing, Northrop Grumman, and other aerospace companies have used Material Extrusion technology to fabricate tooling.

They developed tooling designs for composite part layup



Rapid Tooling

Other reported tooling applications included drill guides and various assembly tools.



The Use of AM to Support Medical Applications

The Use of AM to Support Medical Applications

- Orthopedics
- Surgical and diagnostic aids
- Prosthetics and implants
- Tissue engineering
- Software tools and surgical guides



The Use of AM to Support Medical Applications

Orthopedics



The Use of AM to Support Medical Applications

Surgical and diagnostic aids



The Use of AM to Support Medical Applications



The Use of AM to Support Medical Applications



The Use of AM to Support Medical Applications



The Use of AM to Support Medical Applications





The Use of AM to Support Medical Applications

Five Fraunhofer institutes have developed a bespoke additively manufactured joint implant, intended to restore the mobility of fingers with damaged joints

The goal was to create an automated process which would allow personalised finger joint implants to be produced from metal or ceramic in a rapid, safe and certified process.



The Use of AM to Support Medical Applications

Tissue engineering



The Use of AM to Support Medical Applications

Surgical guides



The Danish Technological Institute has partnered with coffee supplier Bentax to showcase the use of Additive Manufacturing in the replacement part sector.

The company replaced the previous, expensive milk pumps with additively manufactured spare parts, reducing production waste and expense





Airflight, a startup aerospace company based in Brønderslev, Denmark, is working to develop multi-rotor aircraft, including large drone-based flying cranes.

The company is reported to have already benefitted from the technology, reducing the weight of brackets used to hold the arms of the drone by some 67%.



Shell has successfully installed its first additively manufactured leak repair clamp in service.

Clamps are engineered solutions used to encapsulate and restore pipelines against pipeline defects or wall thinning arising through erosion and corrosion mechanisms.







Researchers from Stanford University, California, USA, have designed a new metal/polymer composite for nanoscale Additive Manufacturing capable of absorbing twice the energy of materials with a similar density.















WHAT IS THE FUTURE OF ADDITIVE MANUFACTURING?



WHAT IS THE FUTURE OF ADDITIVE MANUFACTURING?



Q&A (hopefully)

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