

The research of 4H-SiC LGAD after proton radiation

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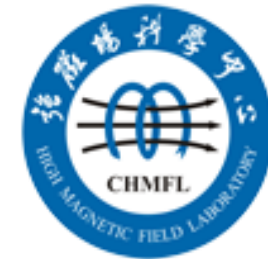
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氮化镓电子器件实验室



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Institute of High Energy Physics
Chinese Academy of Sciences



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Why do we need 4H-SiC?

Characteristic	Si	4H-SiC
E_g (eV)	1.12	3.26
Thermal conductivity	1.5	4.9
$E_{\text{breakdown}}$ (V/cm)	0.5	3
Saturated electron velocity (cm/s)	1×10^7	2×10^7
ionization energy for e-h pair (eV)	3.64	7.8
displacement energy	13	21.8



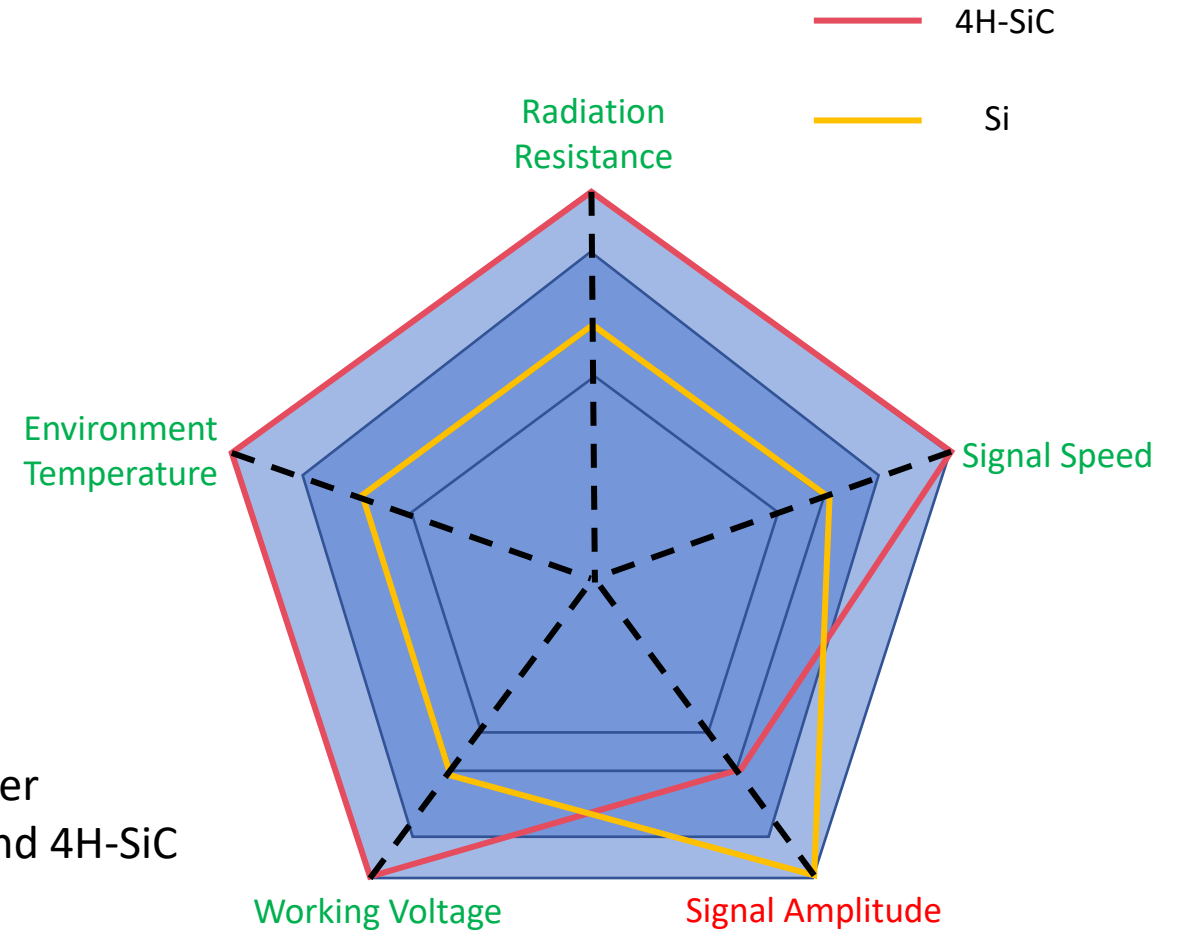
Compared with Si, 4H-SiC can standard higher radiation and work at room temperature. And 4H-SiC can generate a faster signal.



But it needs more energy to generate one e-h pair.

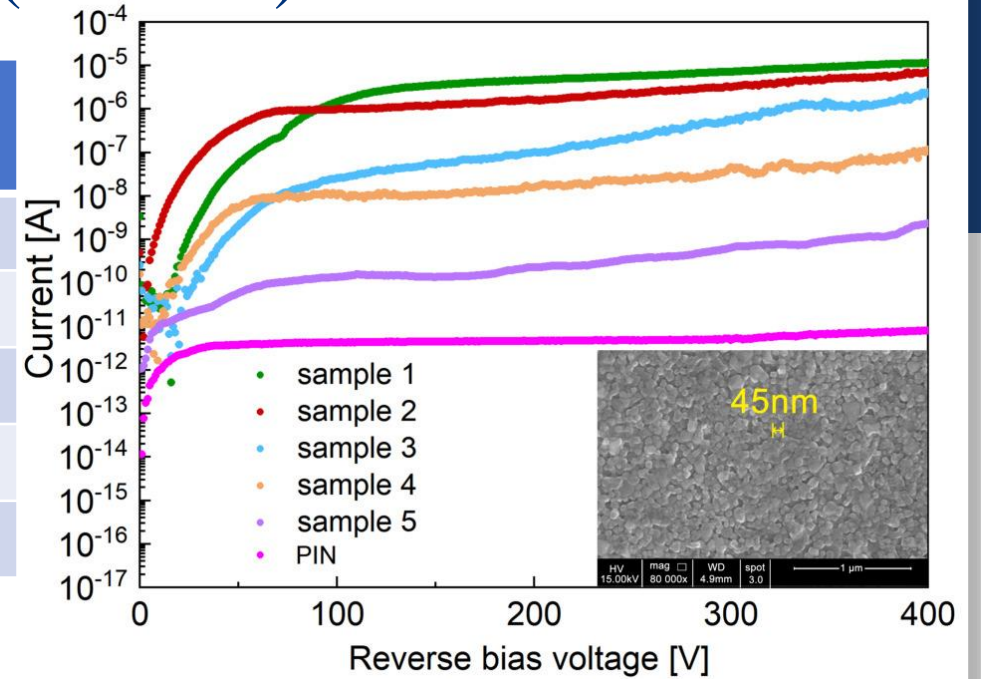


One way to close the gap: inner amplification of signal by **avalanche**.

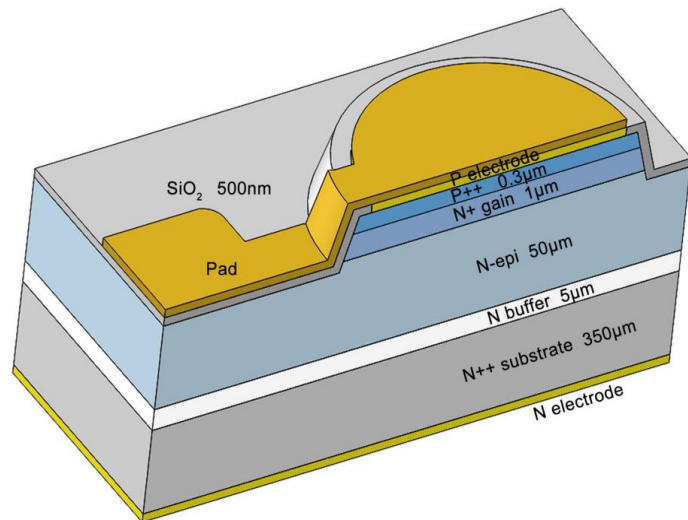


Choice of electrodes of 4H-SiC LGAD (SICAR)

SICAR samples	Thickness of Ni/Ti/Al (nm)	Annealing Temperature (°C)
Sample 1	60/30/80	850
Sample 2	60/30/80	950
Sample 3	60/30/80	1050
Sample 4	60/20/80	1050
Sample 5	50/15/80	1050

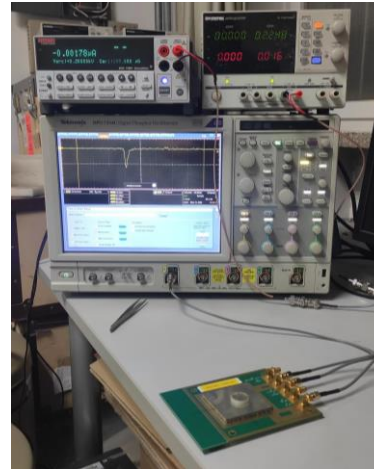
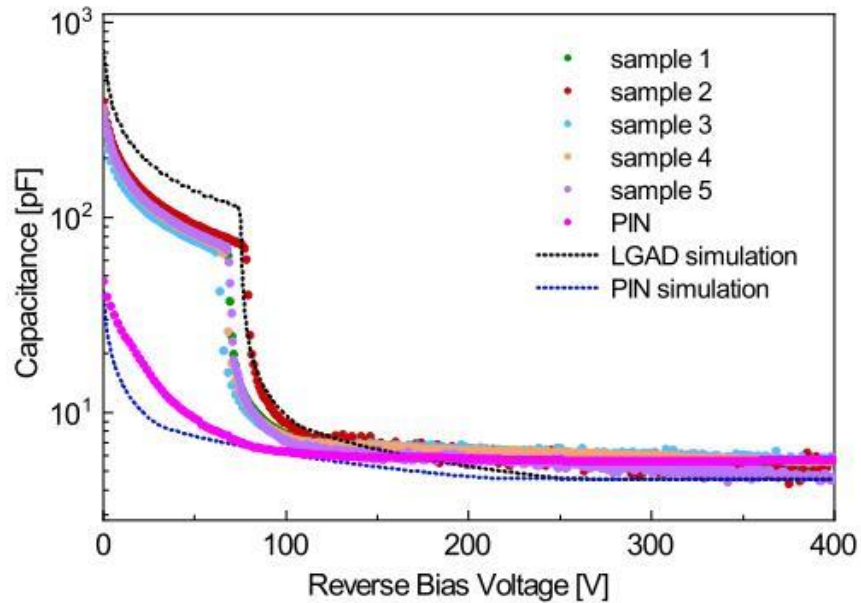


All samples will not be broken down at **400V**.



- **50/15/80** (nm) of Ni/Ti/Al
- Annealing temperature of **1050 °C**

Gain factor of SICAR

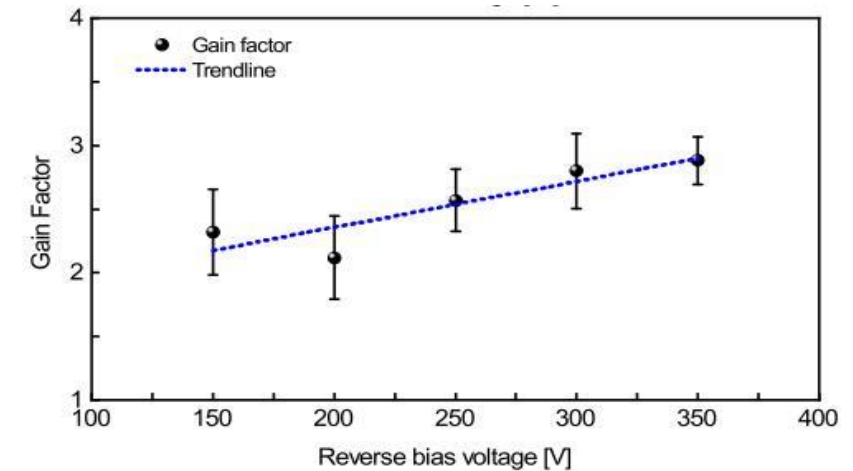
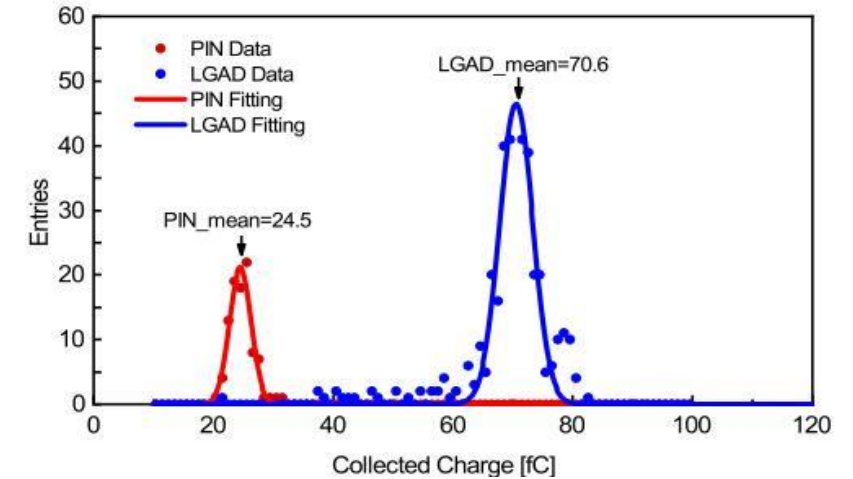


gain test setup

- The effective doping of gain layer: $8.6 \times 10^{16} \text{ cm}^{-3}$
- Depletion depth of the gain layer : $1 \mu\text{m}$
- Gain structure does exist



4H-SiC devices with a gain factor of 3 have been **successfully** prepared.

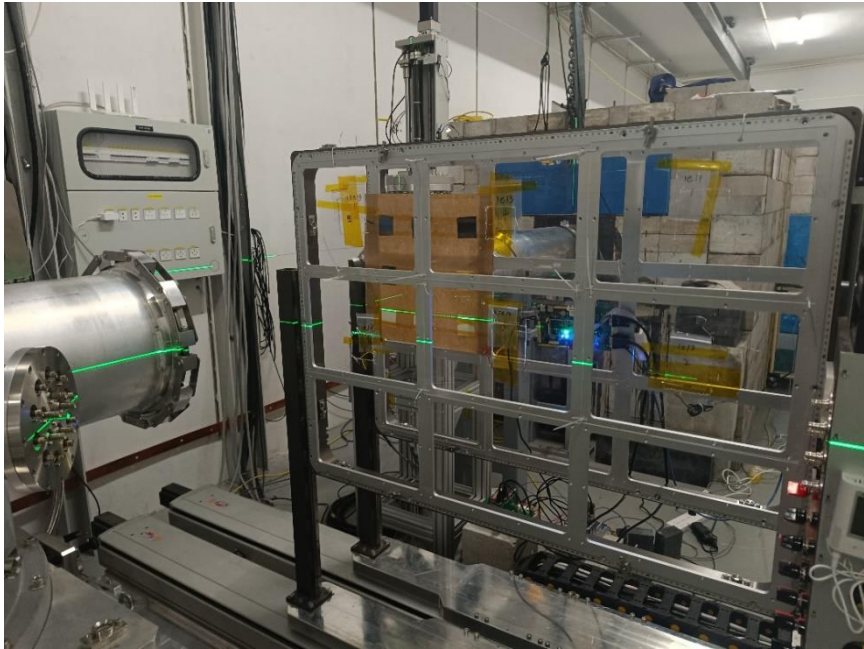


More details in

S. Zhao et al., "Electrical Properties and Gain Performance of 4H-SiC LGAD (SICAR)," in IEEE Transactions on Nuclear Science, doi: 10.1109/TNS.2024.3471863.

Radiation setup

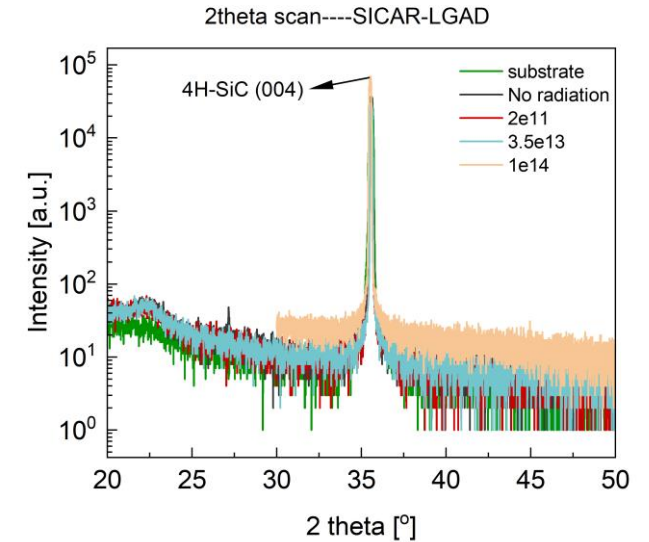
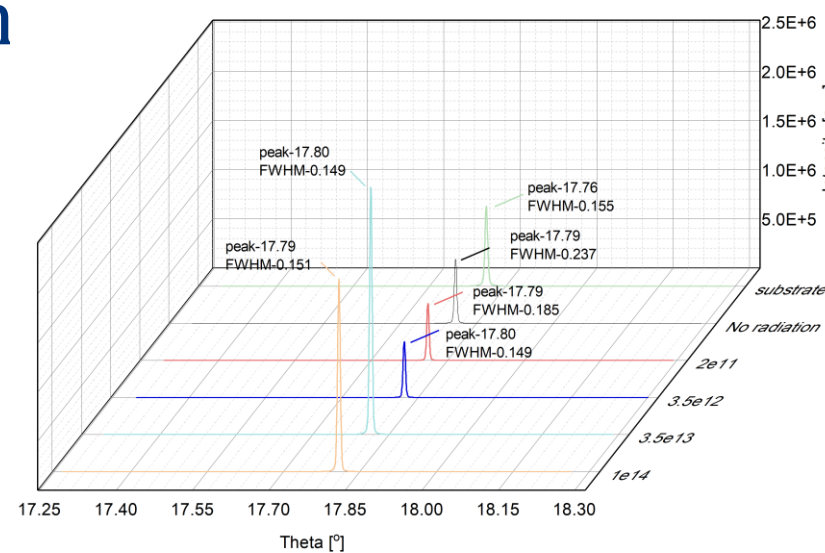
- 80 MeV proton radiation
- Flux: $2 \times 10^{11} \text{ n}_{\text{eq}}/\text{cm}^2$, $3 \times 10^{12} \text{ n}_{\text{eq}}/\text{cm}^2$, $3.5 \times 10^{13} \text{ n}_{\text{eq}}/\text{cm}^2$, $1 \times 10^{14} \text{ n}_{\text{eq}}/\text{cm}^2$
- Size: rectangle of $2 \text{ cm} \times 2 \text{ cm}$



China Spallation Neutron Source

Defect Characterization

----XRD test

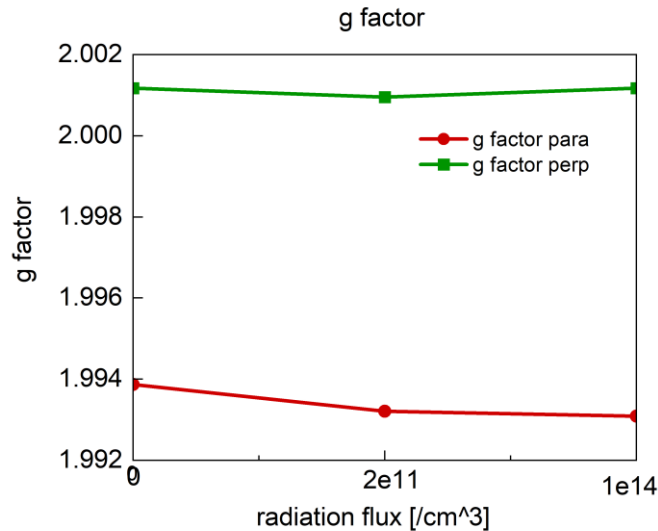


	Substrate	No radiation	2e11	3.5e13	1e14
peak(2θ) [°]	35.5188	35.587	35.589	35.597	35.587
FWHM(β) [°]	0.155	0.237	0.185	0.149	0.151
Size (D) [Å]	6.076	7.7839	9.1137	9.5365	9.2887

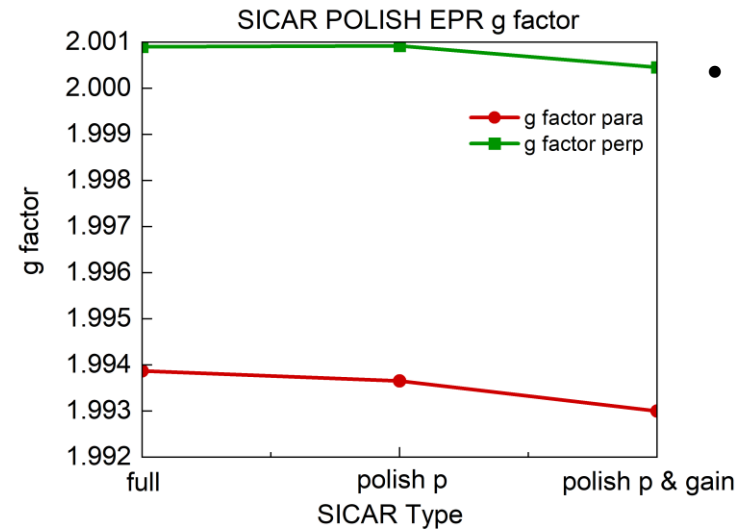
- **Good quality** of single crystal
- No out-of-plane lattice **distortion** and **twinning** due to proton irradiation
- Crystallinity even **improved** with the increase of irradiation

Defect Characterization

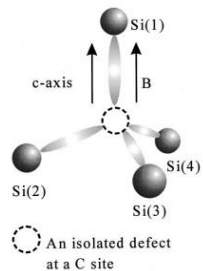
----EPR test



- The defects exhibit obvious anisotropies, and the symmetry of the unit cell changes



- The type of defect is generally consistent within the device



Symmetry from C_{3v} to C_{1h}

$$H = \mu_B \mathbf{B} \cdot \mathbf{g} \cdot \mathbf{S} + \sum_{j=1-4} \mathbf{S} \cdot \mathbf{A}^{(j)} \cdot \mathbf{I}_{Si_j}$$

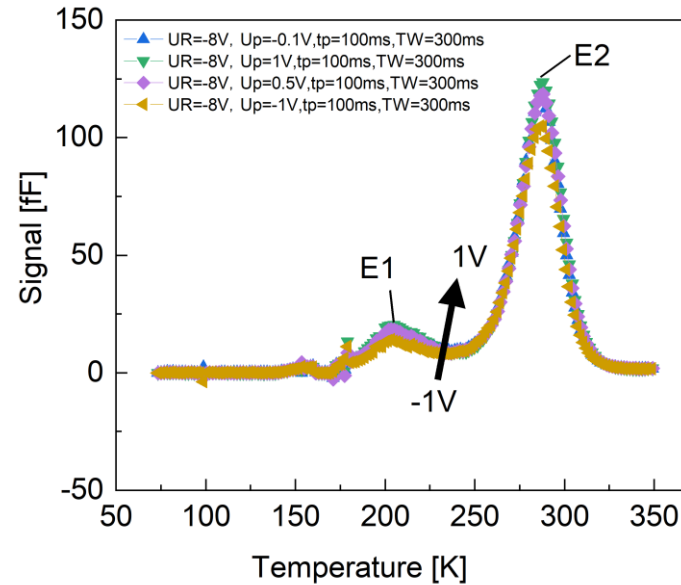
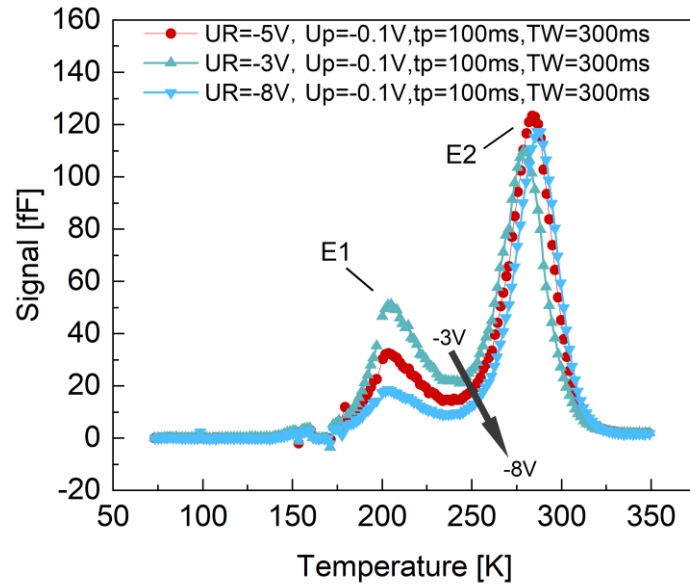
The tensor \mathbf{A} can be used to describe the anisotropy of the vacancy

It should be guessed that the symmetry contains the vacancy at the C_{1h} position

A portion of this work was performed on the Steady High Magnetic Field Facilities, High Magnetic Field Laboratory, CAS

Defect Characterization

----DLTS test : No radiation

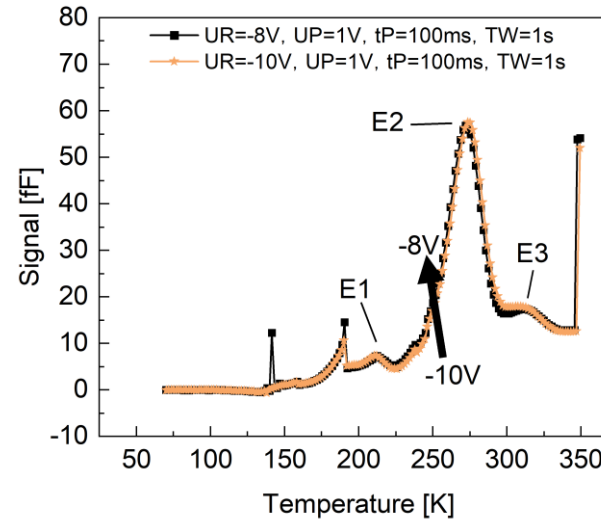
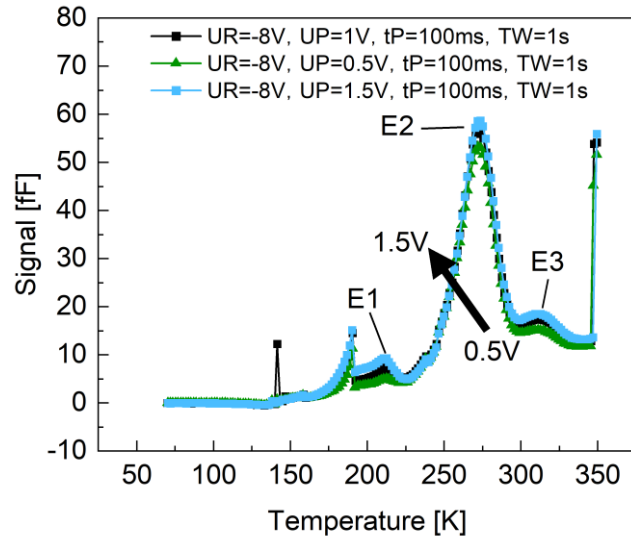


- E2 defect is sensitive to **electric field** (position of E2 peak move to the higher temperature)
- After the E1 emits carriers, the electrical properties of the trap are neutral.

	Trap	Activation energy(eV)	Capture cross-section (cm ²)	Trap concentration NT (cm ⁻³)
EH ₁ ←	E1	0.44	1.88E-14	1.01E+12
Z _{1/2} ←	E2	0.63	1.07E-14	6.74E+12

Defect Characterization

----DLTS test : $2e11$



- E3 is a **new trap after irradiation.**

	Trap	Activation energy(eV)	Capture cross-section (cm ²)	Trap concentration NT (cm ⁻³)
EH ₁ ←	E1	0.46	1.66E-15	1.38E+12
Z _{1/2} ←	E2	0.61	1.27E-15	1.07E+13
EH ₃ ←	E3	0.70	1.30E-15	3.48E+12

Defect Characterization

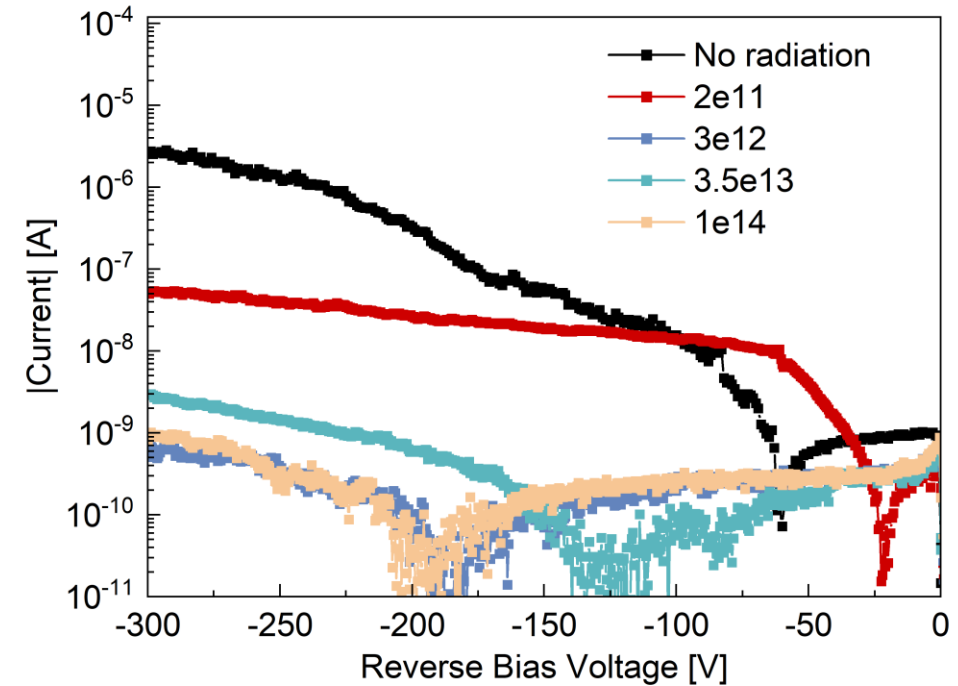
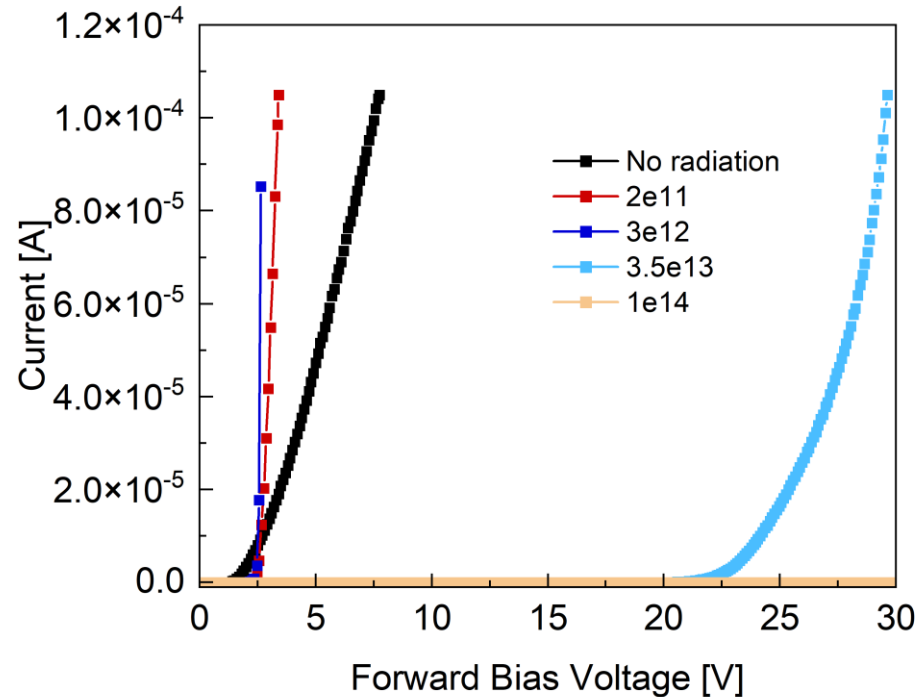
Conclusion

	Trap	Activation energy(eV)	Origin
EH_1 ←	E1	0.46	$\text{V}_{\text{Si}}(3-/=)$
$\text{Z}_{1/2}$ ←	E2	0.61	$\text{V}_{\text{C}}(=/0)$
EH_3 ←	E3	0.70	$\text{V}_{\text{Si}}(=/-)$

- E1 and E2 were **intrinsic defects** introduced during the 4H-SiC growth process.
E3 was introduced by **80MeV proton radiation**.
- The **capture cross-sections** of E1 and E2 were reduced by an order of magnitude after irradiation.
- The trap concentration of **E1 irradiation was almost unchanged**
The **trap concentration of E2 irradiation increased**, indicating that the radiation has a greater impact on carbon than silicon atoms.

Electrical performance of SICAR after proton irradiation

Current vs Voltage

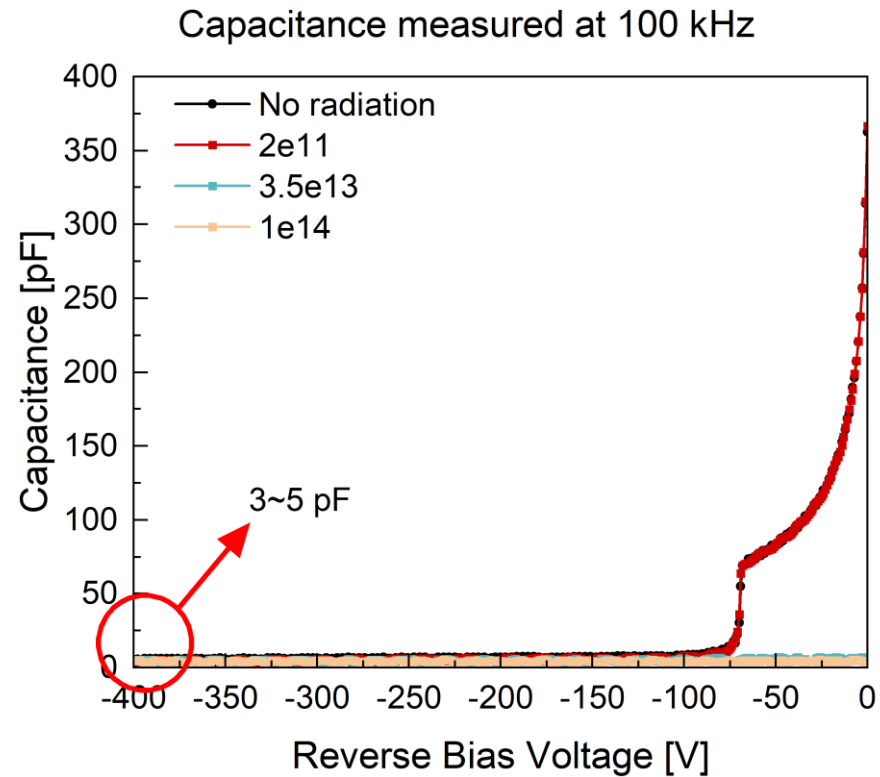


- Threshold voltage **increases** with radiation flux

- The leakage current **reduced by 2-4 orders of magnitude**

Electrical performance of SICAR after proton irradiation

Capacitance vs Voltage



- Capacitive properties disappear at irradiation flux of **3.5e13 and 1e14**.
- The total miscellaneous capacitance of the entire test system is approximately **3 to 5 picofarads (pF)**.

Defect concentration

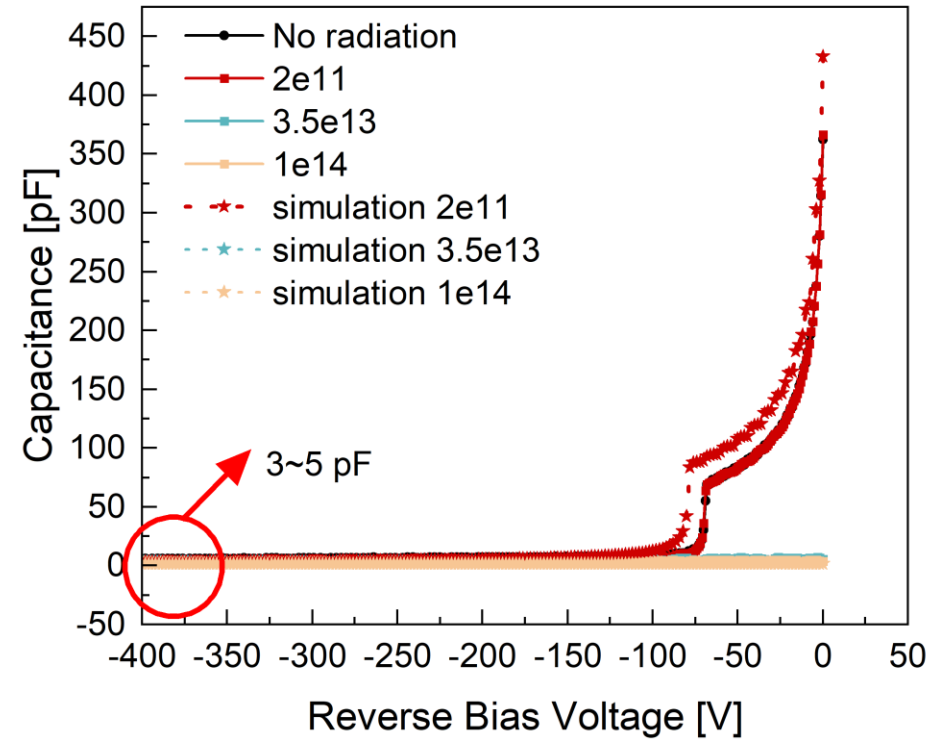
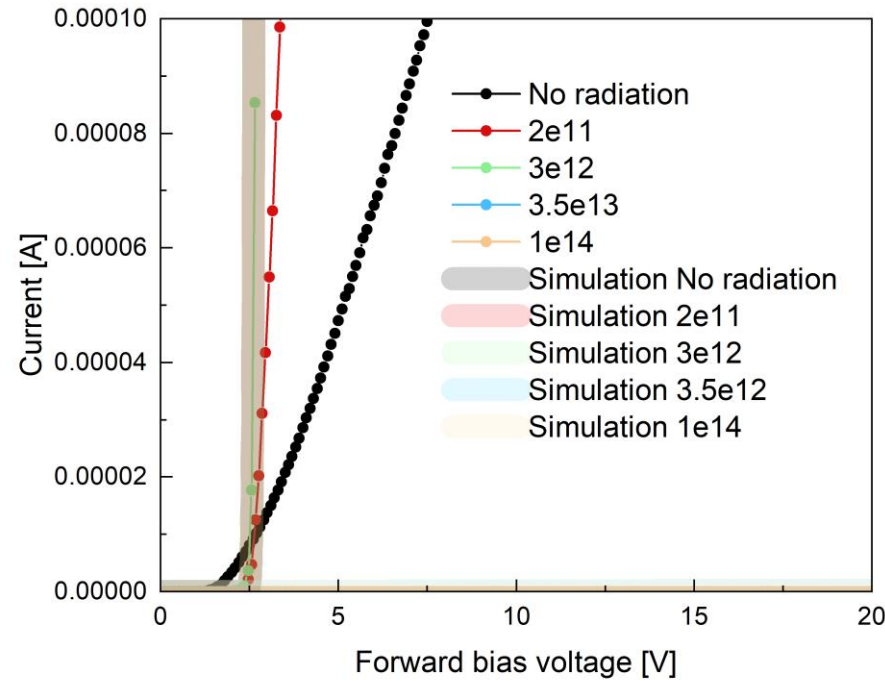
Assuming the concentration of defects changes linearly with the radiation flux:

- $N_{EH3} = g_{EH3} * \Phi_{flux}$ ($g_{EH3} = 17.4 \text{ cm}^{-1}$)
- $N_{Z1/2} = g_{Z1/2} * \Phi_{flux}$ ($g_{Z1/2} = 19.8 \text{ cm}^{-1}$)

Trap concentration	No radiation	2e11	3e12	3.5e13	1e14
Z _{1/2} ($\eta=19.8$)	6.74e+12	1.07e+13	6.6e13	7.0e14	1.98e15
EH1 ($\eta=1.75$)	1.01e+12	1.38e+12	6.26e12	6.23e13	1.75e14
EH3 ($\eta=17.4$)	0	3.48e+12	5.2e13	6.09e14	1.7e15

Simulation

Add the defects to the simulation

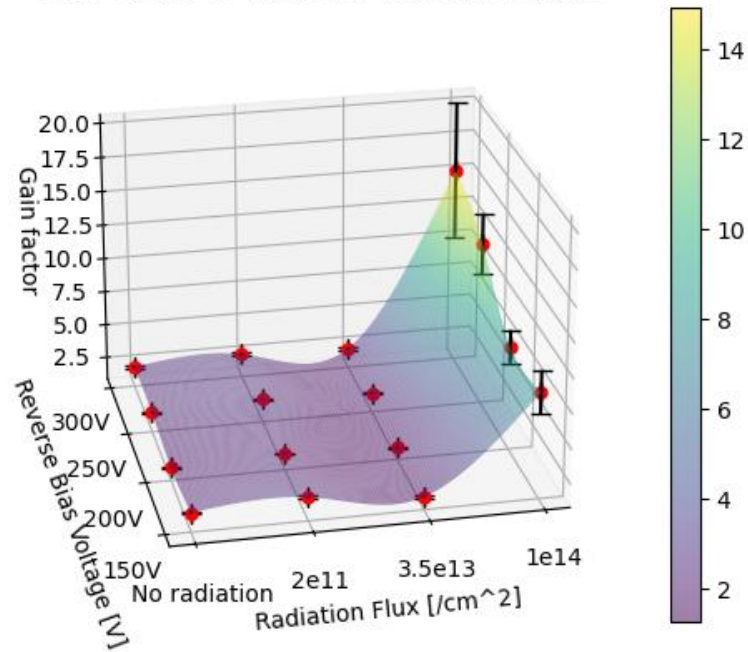


The simulation results are in good agreement with the experimental results.

$Z_{1/2}$ and EH_3 are the critical reason of the impact on IV and CV curves.

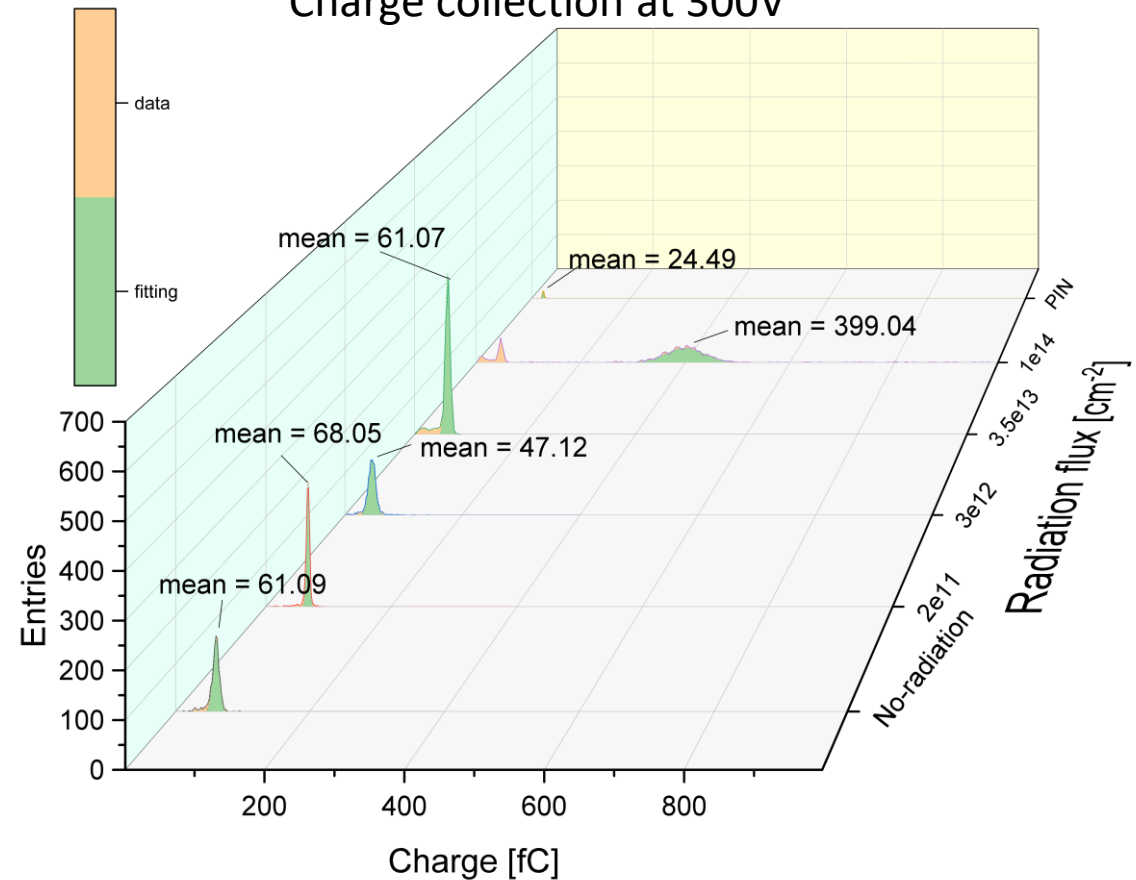
Gain factor

Gain Factor vs radiation flux and voltage



- Gain factor increases slowly with the reverse bias voltage
- The gain factor increases at a radiation flux of $1e14$

Charge collection at 300V



- We speculate that the uneven distribution of defects within the high-dose irradiated device changes the gain path of the carriers, leading to the appearance of higher charge collection.

Summary and Plan

Summary

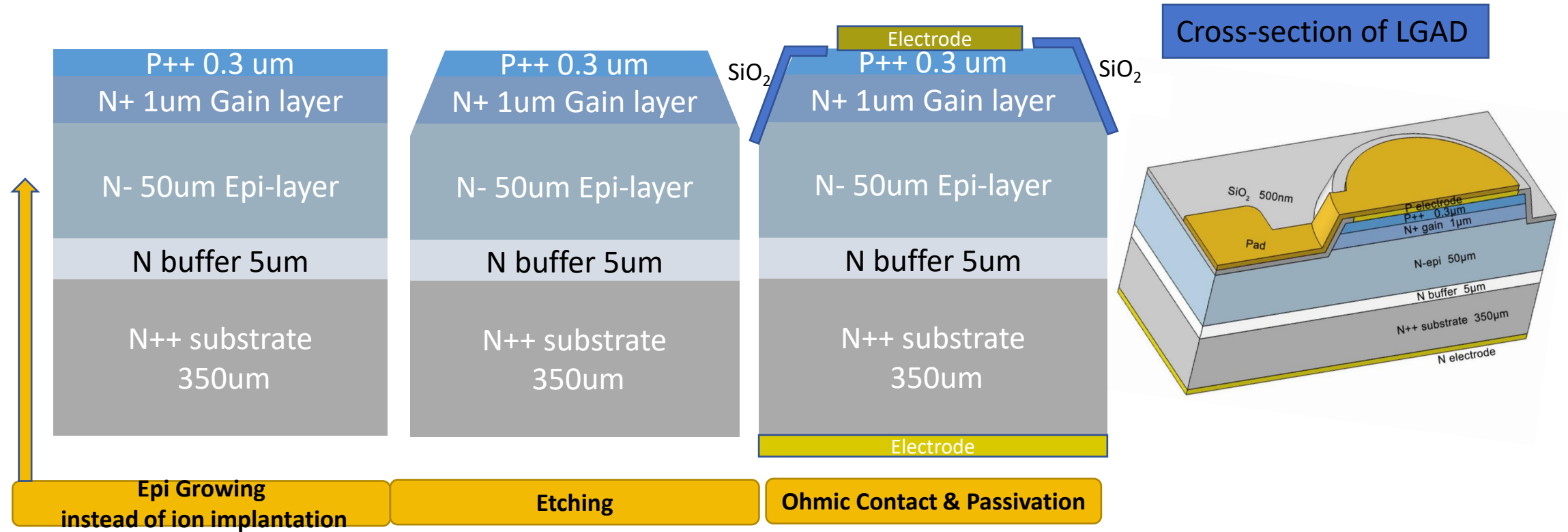
- Proton irradiation introduces **additional vacancy defects** that are key points of the influence of IV and CV.
- At high irradiance levels, the gain factor of the SICAR device increases from 2.3 to 15.

Plan

- Higher flux of proton radiation
- Electron and neutron radiation
- Physical analysis of EPR and DLTS
- Modeling and Physical Interpretation of Irradiation for LGAD Devices

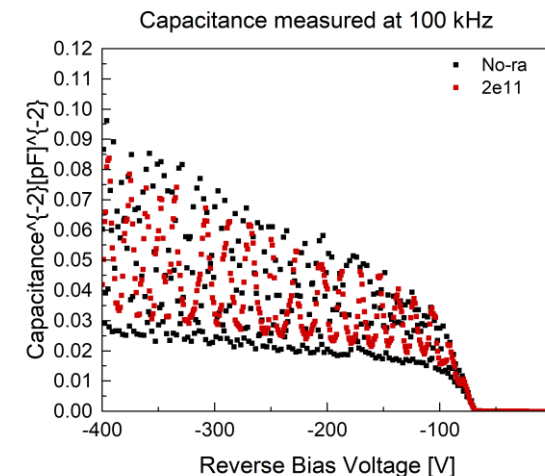
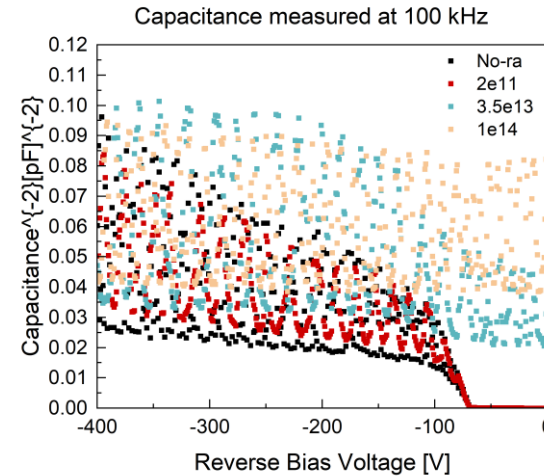
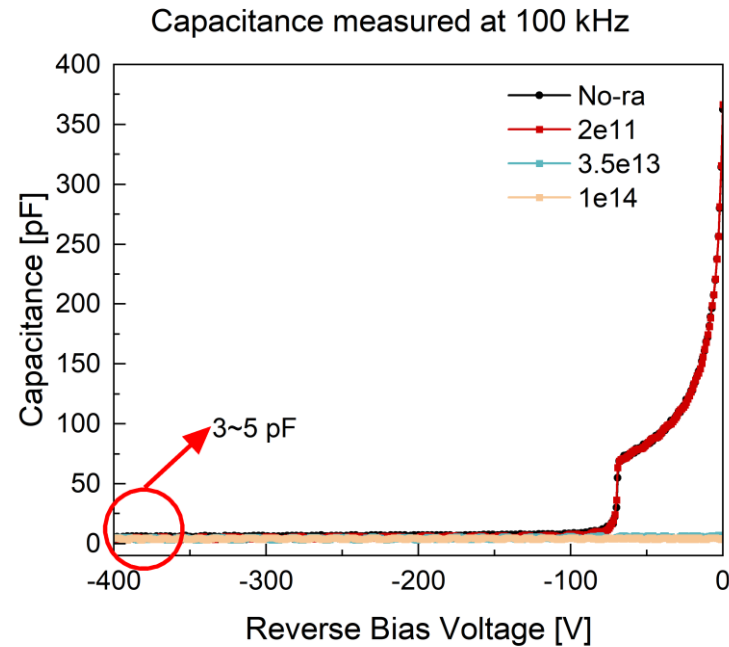
Backup

Process of 4H-SiC LGAD



Electrical performance of 4H-SiC LGAD after proton irradiation

Capacitance vs Voltage

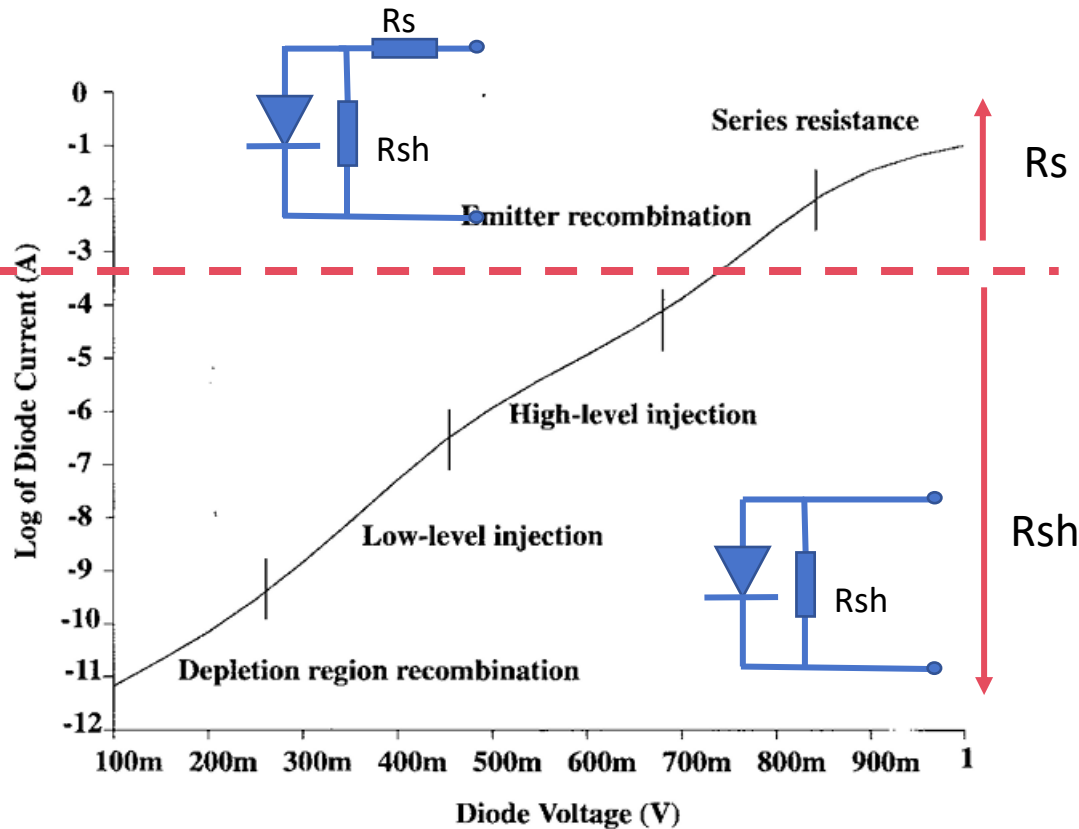


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- The total miscellaneous capacitance of the entire test system is approximately **3 to 5 picofarads (pF)**.

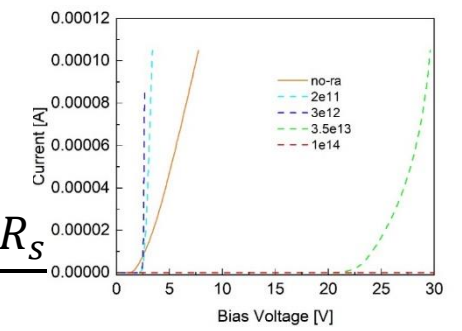
- The capacitance of radiation flux at **3.5e13 and 1e14** change to be constant.

- The capacitance of radiation flux at **2e11** even has no change.

n and Rsh analysis



$$I = I_S * \left(e^{\frac{qV - I * R_s}{nkt}} - 1 \right) + \frac{V - I * R_s}{R_{sh}}$$

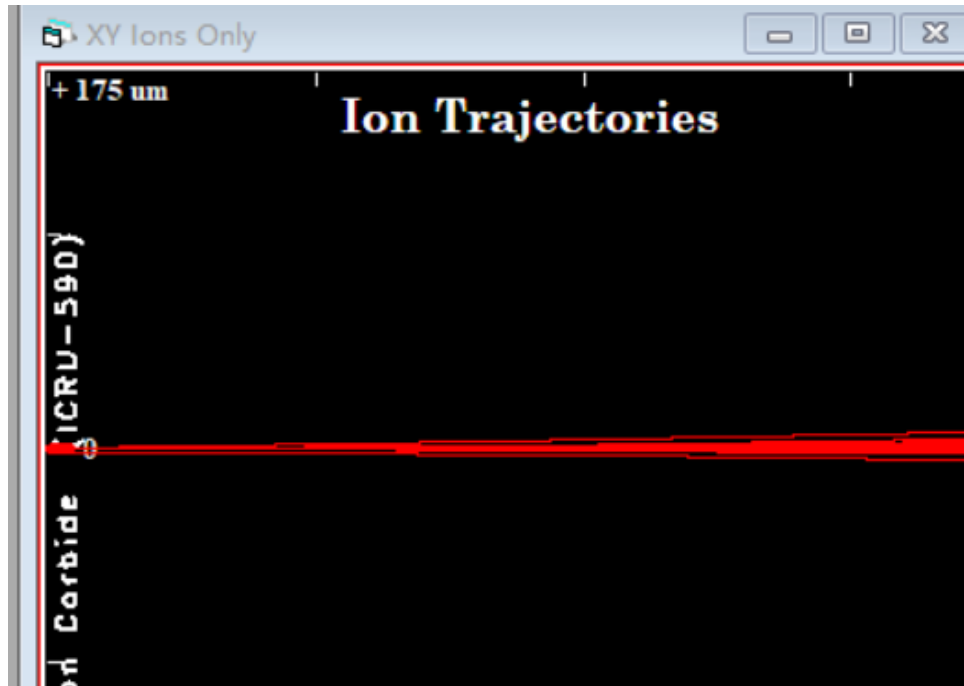


$$I = I_S * \left(e^{\frac{qv}{nkt}} - 1 \right) + \frac{V}{R_{sh}}$$

- Higher $n \rightarrow$ low conductivity: dominated by recombination in the depletion region
- Lower $n \rightarrow$ high conductivity: dominated by diffusion in the depletion region
- **Need higher current to find out the value of R_s**

H. A. Mantooth and J. L. Duliere, "A unified diode model for circuit simulation," in *IEEE Transactions on Power Electronics*, vol. 12, no. 5, pp. 816-823, Sept. 1997, doi: 10.1109/63.622999.

What happened inside the LGAD



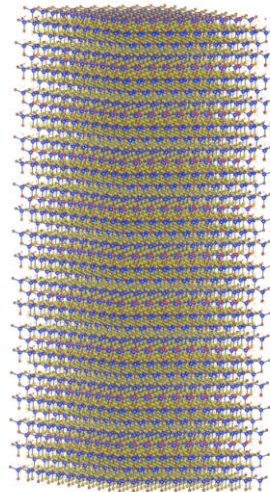
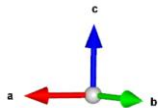
Simulation of the proton injection by SRIM

- Only **vacancies by injection**

- We tested **XRD**, **EPR**, and **DLTS** to find out whether which defect is in the LGAD.

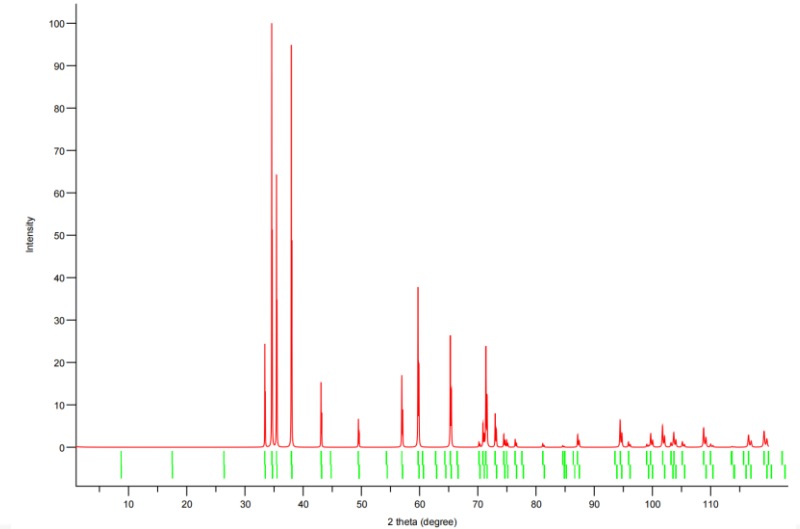
What happened inside the LGAD

----XRD test



The structure put into the simulation.

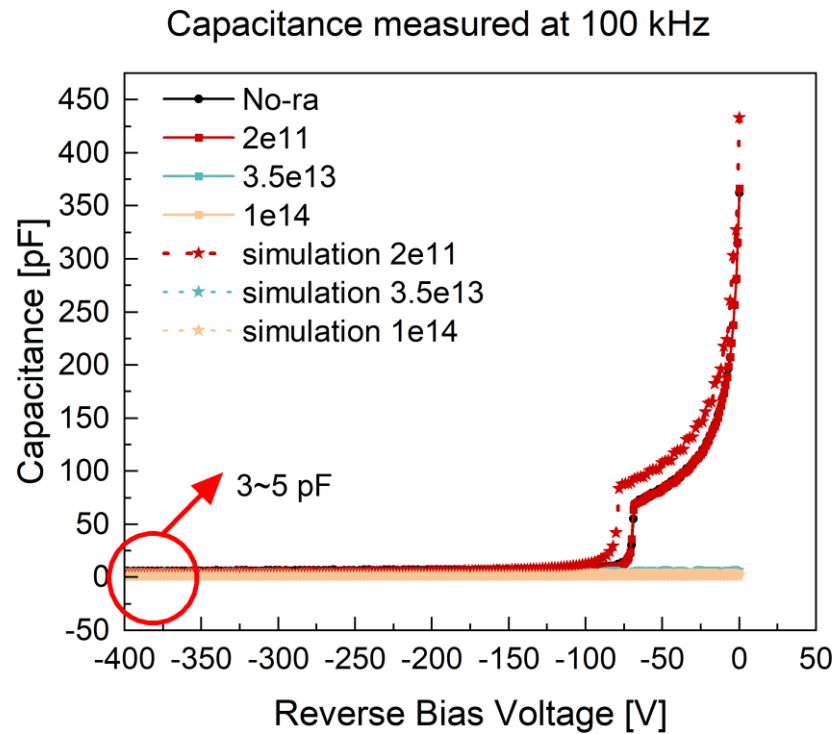
The **internal microstructure changes.**



1	0	1	2.59036	-12.4569	-16.3344	20.5423	34.6859	11.9891
-1	1	-1	2.59036	13.52	-15.8652	20.8445	34.6859	12.3444
-1	1	1	2.59036	-12.4568	-16.3344	20.5422	34.6859	11.9890
1	-1	1	2.59036	12.4568	16.3344	20.5422	34.6859	11.9890
1	-1	-1	2.59036	-13.52	15.8652	20.8445	34.6859	12.3444
0	0	-4	2.53195	13.4846	39.8791	42.0973	35.4237	96.1726
0	0	4	2.53195	16.1278	-39.7819	42.9268	35.4237	100.0000
0	0	-4	2.53195	13.4791	39.8818	42.098	35.5123	47.8258
0	0	4	2.53195	16.134	-39.7841	42.9311	35.5123	49.7374
0	-1	-2	2.36837	-9.74576	-21.2255	23.356	37.9606	25.4275
0	1	2	2.36837	-11.1412	19.8712	22.7813	37.9606	24.1916
-1	0	-2	2.36837	-9.74576	-21.2255	23.356	37.9606	25.4275

The simulation value of 2θ is different from test due to defects.

Simulation: Capacitance vs voltage



Assuming the concentration of defect changes linearly with the radiation flux:

- $N_{EH3} = \eta_{EH3} * \Phi_{flux}$ ($\eta_{EH3} = 17.4 \text{ cm}^{-1}$)
- $N_{Z1/2} = \eta_{Z1/2} * \Phi_{flux}$ ($\eta_{Z1/2} = 19.8 \text{ cm}^{-1}$)

When radiation flux reaches $3.5e13$ or higher, the effective doping becomes 0.

$$n_{eff} = n_0 - \eta \times \Phi_{flux}$$

$Z_{1/2}$ and EH_3 are the critical reason of the carrier removal effect.