

Present state-of-the-art dpa models





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INTRODUCTION / MOTIVATION

- Neutrons/Ion will damage the reactor
- The primary damage is formed as the incoming particle hits a lattice atom and gives it recoil energy
- This can/will then cause macroscopic changes to the material



Straalsund J L, et al. J. Nucl. Mater. 108/109 299-305 (1982)

INTRODUCTION / MOTIVATION

- The primary damage will then evolve and ultimately cause macroscopic changes
- But it is clear that to understand the macroscopic behavior we need to understand the primary damage
- Also, the "dose" we use to quantify the damage at large scale is related directly to the primary damage



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DISPLACEMENTS PER ATOM / DPA NRT-dpa

- If radiation damage would be only formed in a sequence of binary collisions, it should scale linearly with damage energy T_d
- This leads to the NRT-dpa equation for the number of defects N_d

$$N_{\rm d}(T_{\rm d}) = \begin{bmatrix} 0 & , & T_{\rm d} < E_{\rm d} \\ 1 & , & E_{\rm d} < T_{\rm d} < \frac{2E_{\rm d}}{0.8} \\ \frac{0.8T_{\rm d}}{2E_{\rm d}} & , & \frac{2E_{\rm d}}{0.8} < T_{\rm d} < \infty \end{bmatrix}$$

M. J. Norgett et al. Nuc. Eng. Des. 33 (1975) 50-54



K. Nordlund et al. J. Nucl. Mater. 512 (2018) 450-479

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DISPLACEMENTS PER ATOM / DPA Problems with NRT-dpa

• The PKA event is not a binary collision sequence, but a many-body problem



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DISPLACEMENTS PER ATOM / DPA Why a improved analytical model is needed

- The MD simulations are easily doable thanks to modern computers, and can in principle replace NRT-dpa
- However, for parameterizing higher-scale models (neutronics damage models, FEM, ...) it is not really feasible to do MD simulations for every recoil
- If a simple analytical model can be fit to the data, huge advances are gained:
 - Analytical equation: nanosecond on one core
 - MD simulation: **one week on thousand cores**

DISPLACEMENTS PER ATOM / DPA arc-dpa

 Considering the heat spike as two spherical zones with perfect recombination inside *R*, and a shell *R-L* with surviving interstitials, the surviving defect fraction:

$$\xi_{\text{survive}} = \frac{V_{\text{outer}} - V_{\text{inner}}}{V_{\text{outer}}} = \frac{\frac{4\pi R^3}{3} - \frac{4\pi (R-L)^3}{3}}{\frac{4\pi R^3}{3}} = 3\frac{L}{R} - 3\frac{L^2}{R^2} + \frac{L^3}{R^3} \approx 3\frac{L}{R}$$

- The cascade radius is $R \propto T_d^x$ where x < 1
- Then $N'_{\rm d}(T_{\rm d}) \frac{0.8T_{\rm d}}{2E_{\rm d}} \xi_{\rm survive} = \frac{0.8T_{\rm d}}{2E_{\rm d}} 3\frac{L}{R} \propto \frac{0.8T_{\rm d}}{2E_{\rm d}} 3\frac{L}{T_{\rm d}^{\rm x}} \propto T_{\rm d}^{1-x}$
- At high energies cascades split into sub-cascades
- -> linear, and demanding consistency with the

NRT-dpa at the threshold, one arrives at the final form

$$\xi_{\text{arcdpa}}(T_{\text{d}}) = \frac{1 - c_{\text{arcdpa}}}{\left(2E_{\text{d}}/0.8\right)^{b_{\text{arcdpa}}}} T_{\text{d}}^{b_{\text{arcdpa}}} + c_{\text{arcdpa}}$$

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K. Nordlund et al. Nat. Commun. 9 (2018) 1084

DISPLACEMENTS PER ATOM / DPA rpa

- Consider now the atomic mixing (replacements)
- It is actually a very good approximation that all atoms within the core are replaced. Hence simply N∝R³
- Combining this with the previously noted $R \propto T_d^x$ and demanding linearity:

$$\xi_{\rm rpa}(T_{\rm d}) = \left(rac{b_{
m rpa}^{c_{
m rpa}}}{(2E_{
m d}/0.8)^{c_{
m rpa}}} + 1
ight) rac{T_{
m d}^{c_{
m rpa}}}{b_{
m rpa}^{c_{
m rpa}} + T_{
m d}^{c_{
m rpa}}}$$

 We need four parameters in addition to the threshold displacement energy per material



K. Nordlund et al. Nat. Commun. 9 (2018) 1084



DISPLACEMENTS PER ATOM / DPA Limitation of arc-dpa -> full energy range dpa

- The hard cut-off at the E_d
 - E_d an average energy needed, not the minimum B
 - In an alloy, this is also PKA specific
 - Linear correction to the minimum added

$$N_{d}(E) = \begin{vmatrix} 0 & E < E_{d} \\ 1 & E_{d} < E < 2E_{d} / 0.8 \\ \frac{0.8E}{2E_{d}} \xi(E) & 2E_{d} / 0.8 < E < \infty \end{vmatrix}$$

$$N_{d}(T_{d}) = \begin{cases} 0, & T_{d} < E_{d}^{\min} \\ \frac{0.8T_{d}}{2E_{d}^{\operatorname{avr}}}, & E_{d}^{\min} < T_{d} < \frac{2E_{d}^{\operatorname{avr}}}{0.8}, \\ \frac{0.8T_{d}}{2E_{d}^{\operatorname{avr}}} \xi(T_{d}), & \frac{2E_{d}^{\operatorname{avr}}}{0.8} < T_{d} \end{cases}$$



Q. Yang and P. Olsson, Phys. Rev. Mater. 5 (2021) 073602

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DISPLACEMENTS PER ATOM / DPA Limitation of arc-dpa -> crc-dpa

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inear collision cascade, ~ 0.1 ps Heat spike, $\sim 1 - 10$ ps Defect mobility: taking into account Passing high-energy particle Knocked-on atoms Knocked-on atoms Primary, PKA Primary, PKA correlated movement of interstitials and Secondaries Secondaries Tertiaries Tertiaries Quatemaries Heat spike atoms vacancies, separately. **Defect Production Efficiency (Surviving Defect Fraction) in Irradiated Copper** surviving defects (NRT dpa fraction) ARC-DPA **CRC-DPA CRC-DPA** (T>stage Id) (T>stage III) Electron Remaining primary damage ~ 100 ps Defect mobility, ~ 100 ps to years 0.8 Defects Defects before migration 🕶 Interstitial Interstitial O Vacancy Vacancy 0.6 Interstitial position acancy position Annihilation event 0.4 Fast Electron Fast neutron Fast neutron 0.2 neutron Electron 0 No diffusion After correlated recom- After correlated recombination by vacancies bination by interstitials (T_{irr}~4 K) **HELSINGIN YLIOPISTO** (T_>45 K) (T_>250 K) S. J. Zinkle and R. E. Stoller, J. Nucl. Mater. 577 (2023) 154292 HELSINGFORS UNIVERSITET **UNIVERSITY OF HELSINKI** Faculty of Science / Fredric Granberg / RADSUM 25 www.helsinki.fi/yliopisto 15-17.01.2025 10

DISPLACEMENTS PER ATOM / DPA Current status

- With current models we can take into account the:
 - Defects produced from binary collisions (NRT-dpa) [1]
 - Account for the heat-spike [2,3]
 - Corrects real damage (arc-dpa)
 - and replacements (rpa)
 - Correction to low energy regime, full energy range dpa [4]
 - Account for thermal correlated movement, crc-dpa [5]
- What is still missing:
 - Defect production a range, not exact value/morphology
 - High energies, stable defect clusters can form

[1] M. J. Norgett et al. Nuc. Eng. Des. 33 (1975) 50-54
[2] K. Nordlund et al. Nat. Commun. 9 (2018) 1084
[3] K. Nordlund et al. J. Nucl. Mater. 512 (2018) 450-479
[4] Q. Yang and P. Olsson, Phys. Rev. Mater. 5 (2021) 073602
[5] S. J. Zinkle and R. E. Stoller, J. Nucl. Mater. 577 (2023) 154292
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- The dpa models can account for single PKAs in pristine material, however...
 - The dose is micro- to milli-dpa, maybe not relevant



J. Byggmästar et al. J. Phys. Condens. Matter 31 (2019) 245402



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LIMITATIONS Cascade overlap/High-dose simulations

 The dpa estimate only applies when there is practically no cascade overlap



E. Levo et al. J. Nucl. Mater. 490 (2017) 323-332



F. Granberg et al. J. Nucl. Mater. 556 (2021) 153158

EXAMPLE: Multiscale modelling Need for "cascade overlap"

- Utilize DFT-based ML potentials to carry out MD cascades and utilize OKMC to evolve the time to experimental ones.
- Initial studies, on W, show that utilizing only OKMC with a MD database is not giving the same evolution as the full combination



J. Wu et al. https://arxiv.org/abs/2409.15856



CONCLUSIONS

- The NRT-dpa value is still the most used measure for dose, used for comparison
- Better models do exist to predict the real damage or displacements
 - Considering heat-spike/recombination and long-term diffusion
- In single elements replacements are quite useless, however, in complex materials the replacements can be the real defects
- Current limit is high energies (material specific), where complex defects can form
- Still, the limit is when the PKA is not anymore in a pristine material
 - This happens quickly as the dose increases
 - Practically only solution is full simulations instead of analytical models
 - → We still need full-scale high-energy and high-dose simulations!



THANK YOU FOR YOUR ATTENTION



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