



# Long-term surface evolution models for accelerator materials

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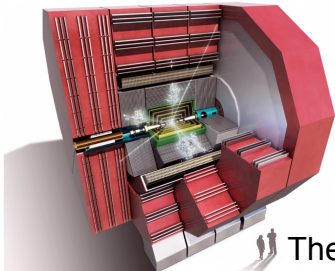
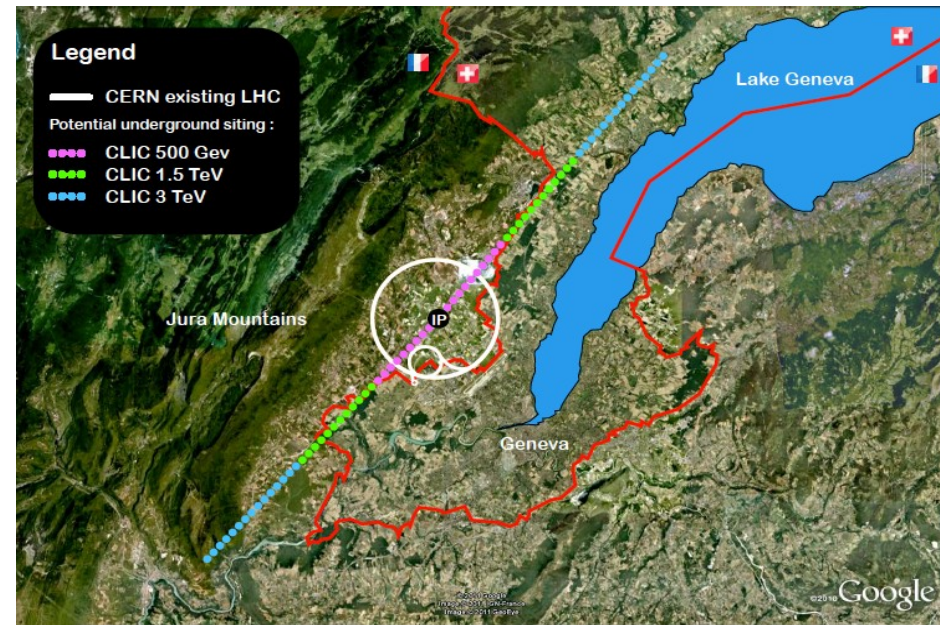
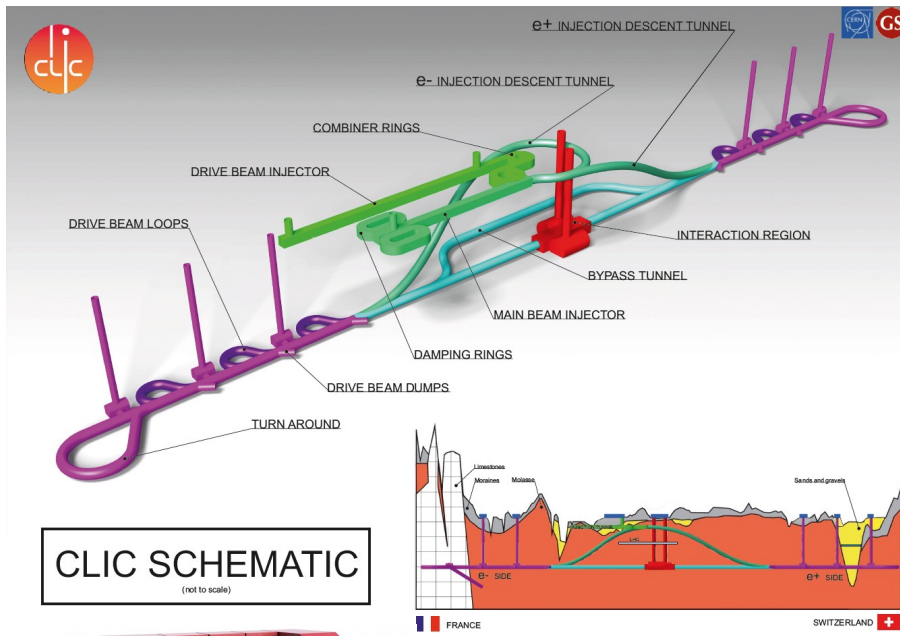


# Outline

- Background:
  - CLIC
  - Vacuum arcs
- Method: Kimocs (KMC)
- Results
  - Migration barrier calculations
  - Validation of the model
  - Stability of large tips
- Conclusions & outlook



# The Compact Linear Collider (CLIC)



The detector

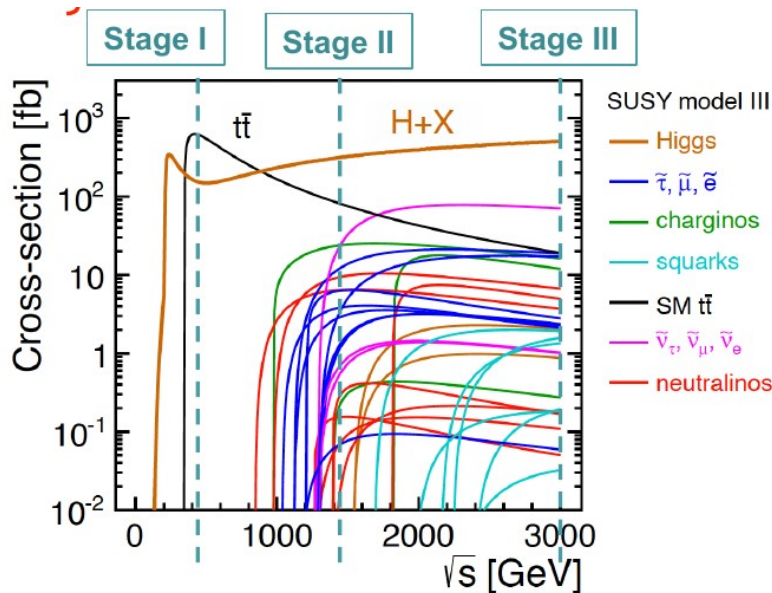
- Electron-positron collider
- Centre-of-mass energy: 380 GeV – ~3 TeV
- Length: 10—50 km
- Construction decision 2018-2019



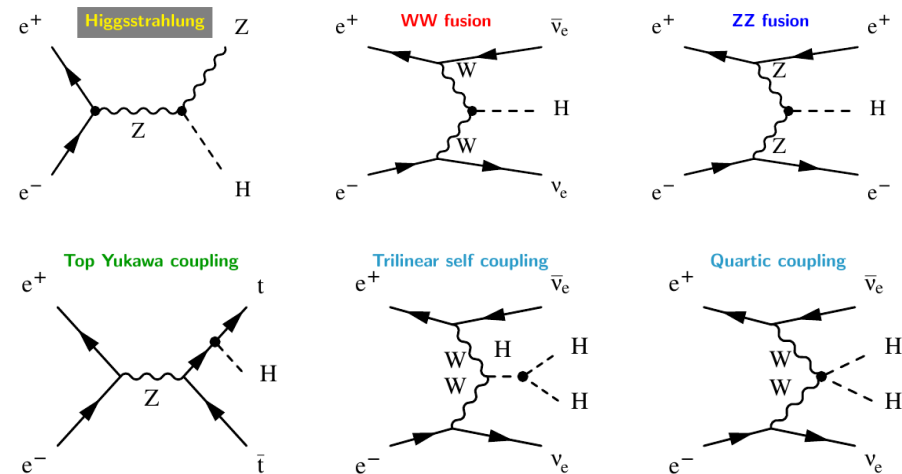
# CLIC: particle physics point-of-view

## The three stages of CLIC

- Stage I: 380 GeV, 500 fb<sup>-1</sup> (Higgs, top)
- Stage II: ~1.5 TeV, 1.5 ab<sup>-1</sup> (Higgs, gauginos, sleptons)
- Stage III: ~3 TeV, 2 ab<sup>-1</sup> (Higgs, squarks,...?)

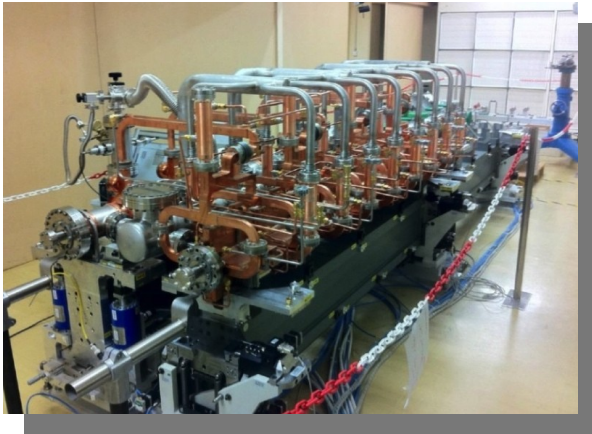


## Higgs processes of interest for CLIC

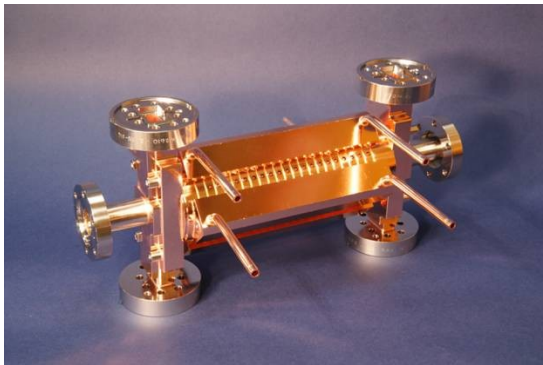
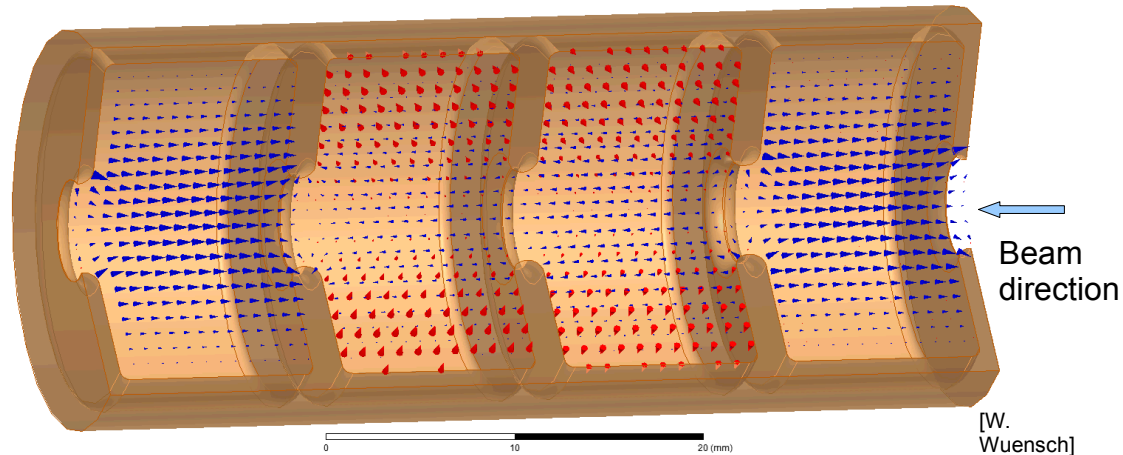




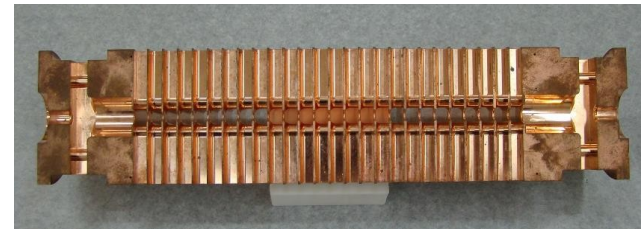
# CLIC: the accelerating structures



CLIC test model at CERN



Accelerating structure

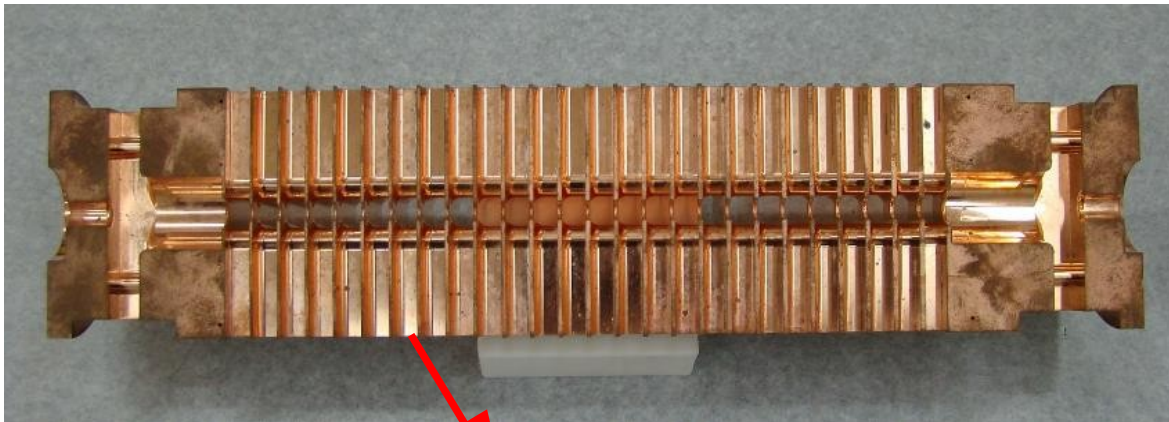


Cross section of a structure



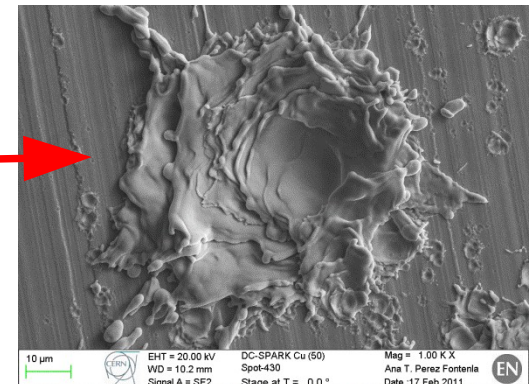
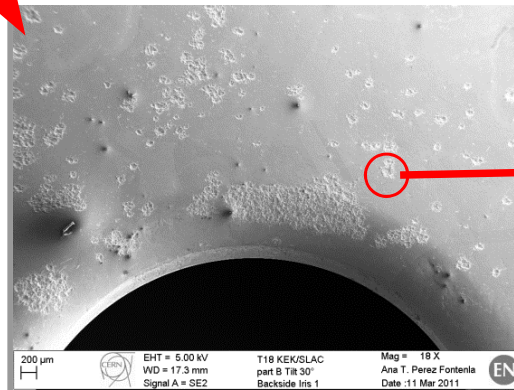


# CLIC: plasma & materials physics



**Problem:** the high fields causes frequent electric arcs (breakdowns):

- Lower performance
- Surface damage



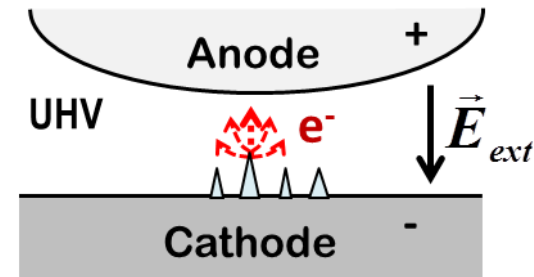


# Nanotips (cause of the arcs?)

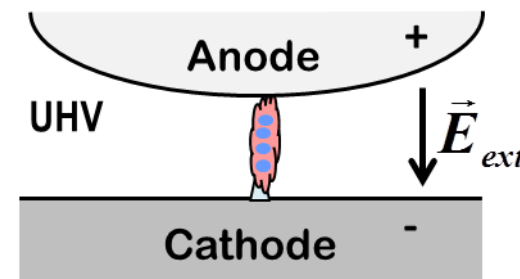
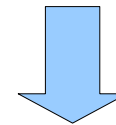
Field emission measurements suggests the existences of sharp nano-tips that would explain the  $\beta$  factor in the Fowler-Nordheim equation for field emission.

But tips are not likely to be seen after vacuum breakdowns

Observations of nanotips with aspect ratios of  $\sim 10$  are rare, which raises the questions: **How stable are nanotips? How do they form?**



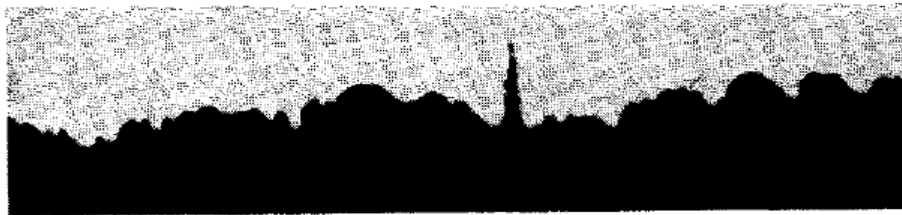
Field-emitting tips appears and a plasma is formed



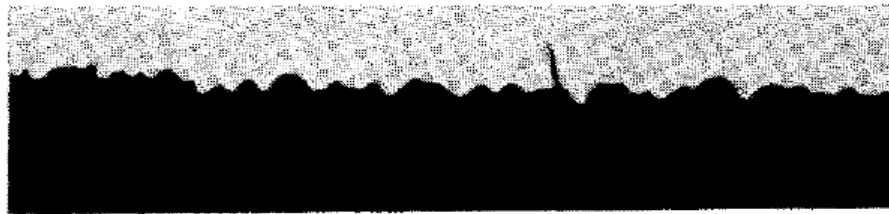
The plasma burns (breakdown) and the tips are destroyed



# Nanotips



(a)



(b)

FIG. 4. Microprojections in the emission areas on optically polished cathode surfaces. (a) 304 stainless steel, (b) aluminum, magnification approximately  $\times 2000$ .

$\sim 2 \mu\text{m}$  tips with an aspect ratio of  $\sim 10$  were observed already 1963 on Cu and other metal surfaces by Little & Whitney (JAP), using transmission electron microscopy.

They used a field of  $10 \text{ MV/m}$  and a  $10^{-5} \text{ Pa}$  vacuum.



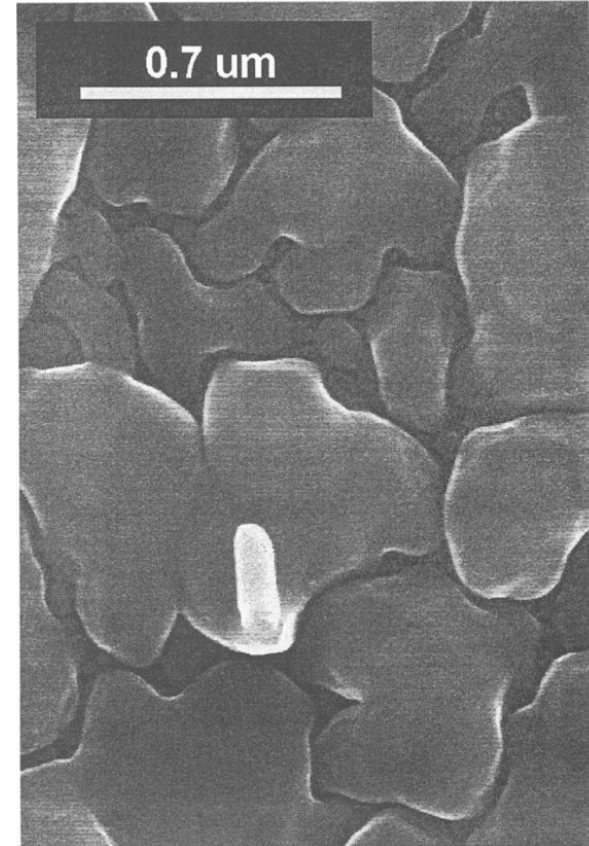


# Nanotips

Copper tips with a radius  $<120$  nm and height  $>300$  nm, like the one in the SEM figure, have been found to be **stable for more than a month** at room temperature

The tip was produced using metal organic chemical vapor deposition of Cu on quartz

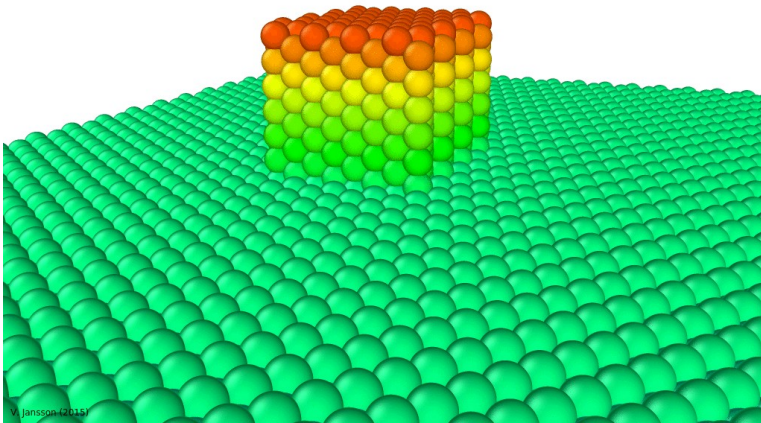
**In lack of good observations of tips, modelling can provide more insights.**



[F. Atamny, A. Baiker. Surf. Sci. (1995)]



# Kinetic Monte Carlo (KMC)



- Simulation method for slow physical processes (nanoseconds – seconds for atomic processes)
- Thermally activated processes, such as atom jumps, have probability rates of the form

$$\Gamma = \nu \exp\left(\frac{-E_m}{k_B T}\right)$$

where

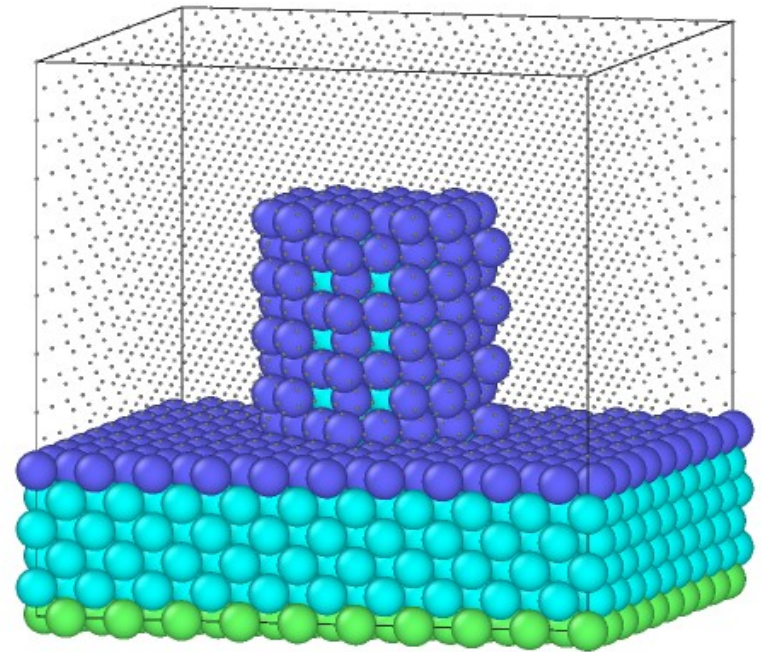
- $\nu$  is the attempt frequency
- $E_m$  the migration (or activation) energy
- $\nu$  and  $E_m$  need to be known for all processes



# Kimocs

A KMC code for the surface

- Designed to simulate the surface evolution on the atomic scale
- Only surface atoms (blue) can migrate by jumping to vacant lattice sites on a rigid 3D lattice
- Bottom atoms (green) are usually fixed
- Different surfaces are possible: (100), (110) and (111), but not different grains
- Only fcc and bcc are implemented

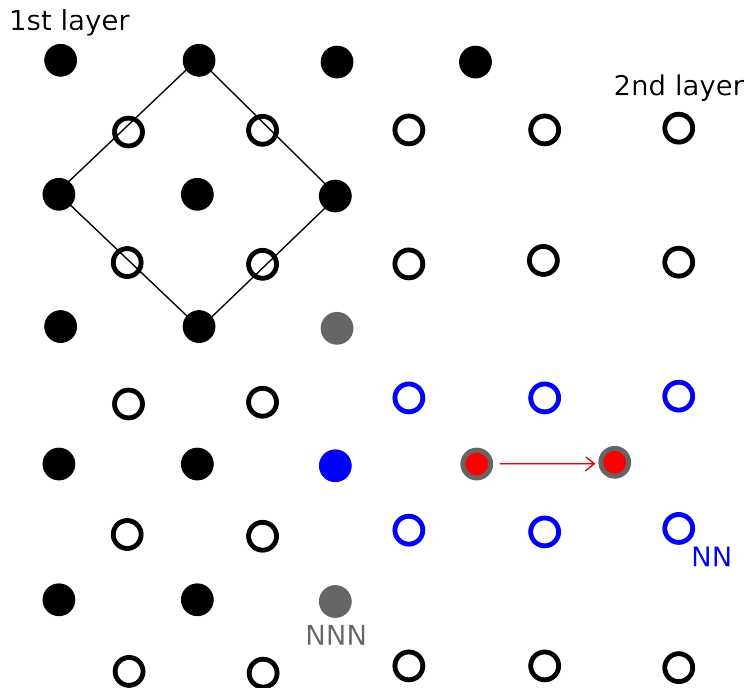




# Kimocs

A KMC code for the surface

## (100) surface

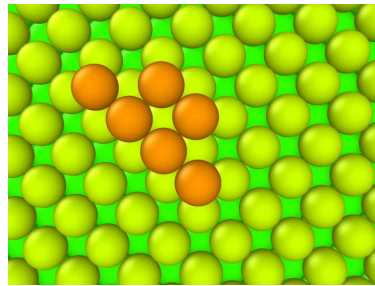
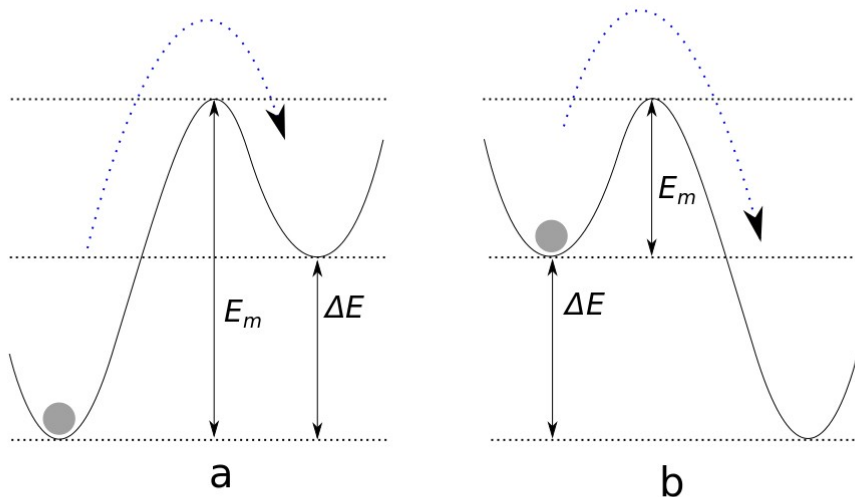


$$E_m(5,3,5,1)$$

- Atom jumps are characterized by
  - The number of nearest neighbour (1nn) atoms
  - The 2<sup>nd</sup> nearest neighbour (2nn) atoms
 of the initial and final lattice sites
- $E_m$  for all processes are calculated with Molecular Dynamics

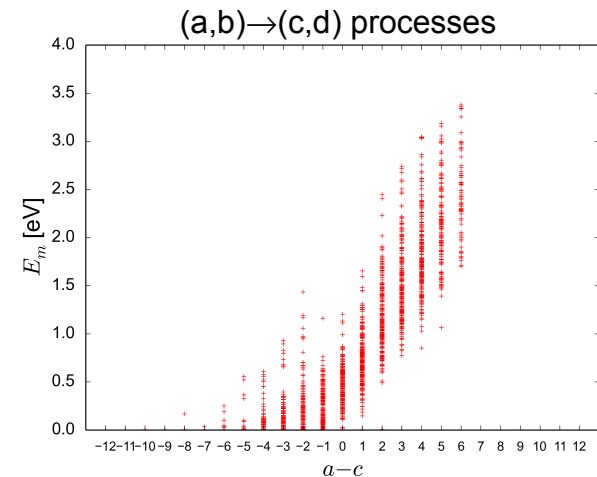


# Calculations of the migration energies, $E_m$



(6,2)→(7,2)

- $E_m$  for a process depends on the local atom configuration and can be described as (1nn,2nn)→(1nn,2nn)
- Thousand of configurations has to be considered
- The Molecular Dynamics method Nudged Elastic Band (NEB) is used



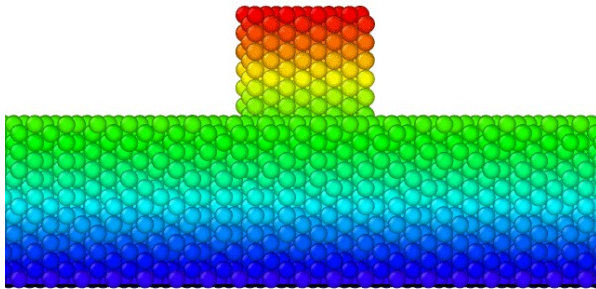
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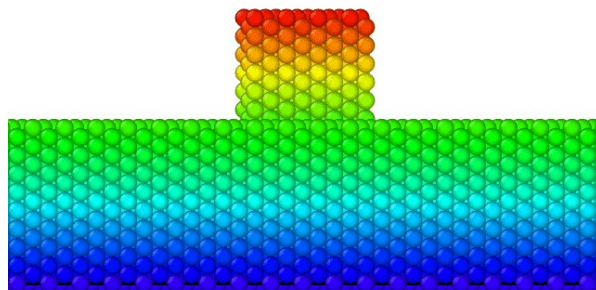


# The attempt frequency $\nu$

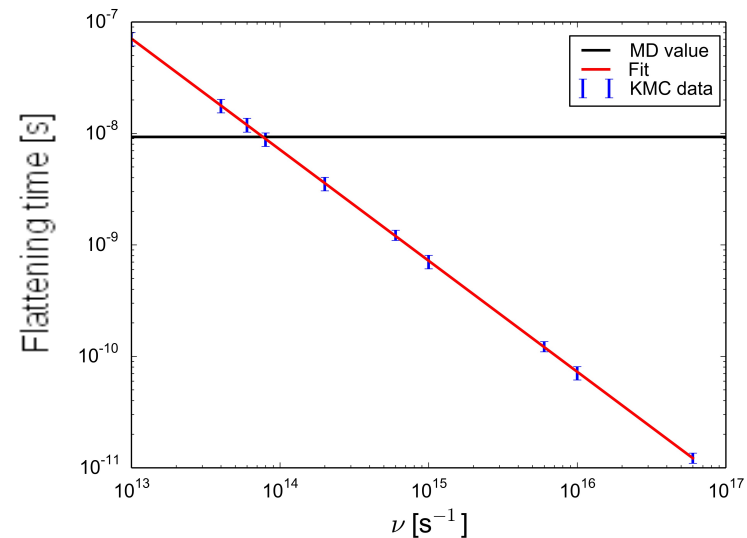
MD



KMC



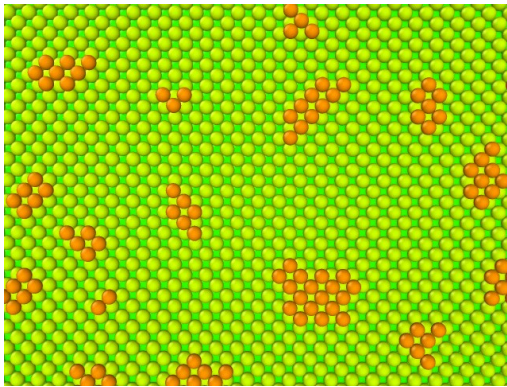
- Since all processes are atom jumps,  $\nu$  can be considered a constant near to the Debye frequency of Cu,  $4.5 \cdot 10^{13} \text{ s}^{-1}$
- Our value  $\nu = 7 \cdot 10^{13} \text{ s}^{-1}$  is obtained by fitting to MD simulation results of the Cu tip flattening process



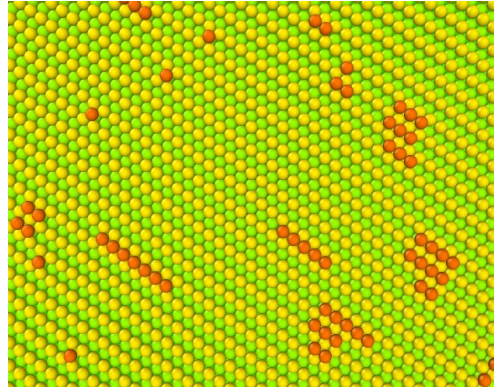


# **(100), (110) and (111) surfaces**

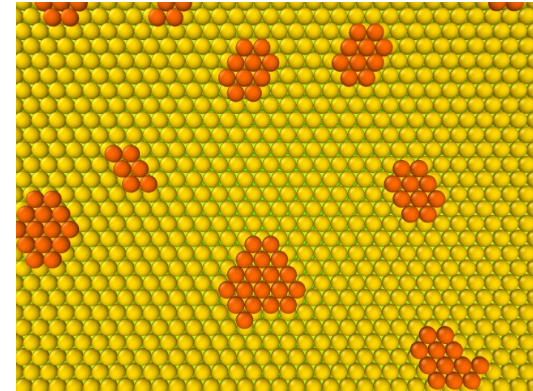
Kimocs simulations: Adatom islands



(100)



(110)



(111)

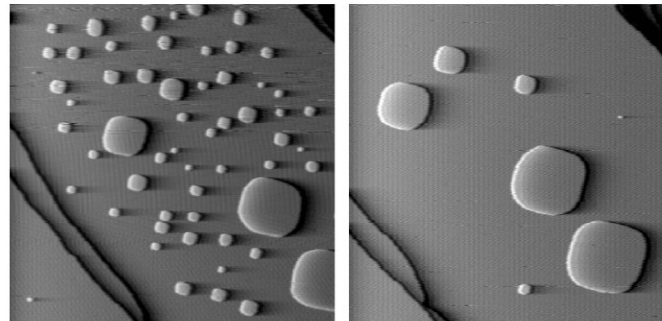
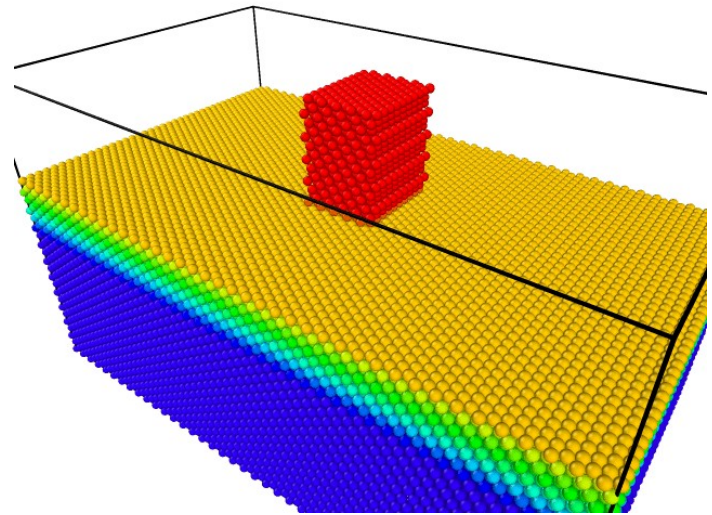
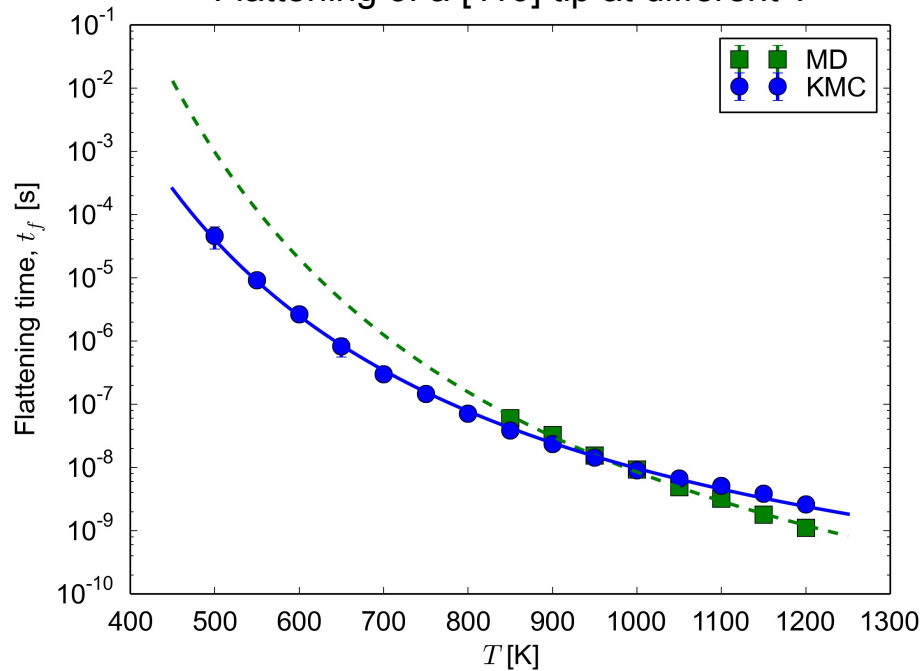


FIG. 2. Two  $300 \text{ nm} \times 300 \text{ nm}$  STM images, separated in time by 20 000 s, showing island ripening on Cu(001) at 343 K. [J.B. Hannon et al. PRL 1997]



# Tip flattening: KMC vs MD

Flattening of a [110] tip at different  $T$

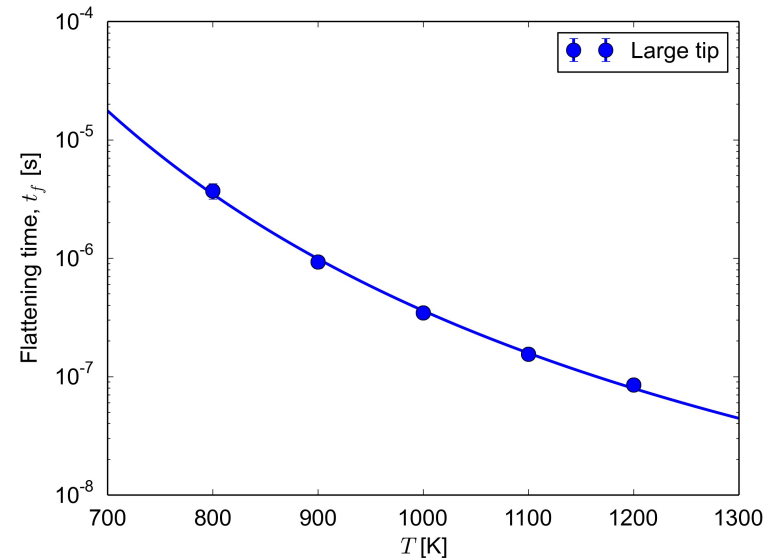
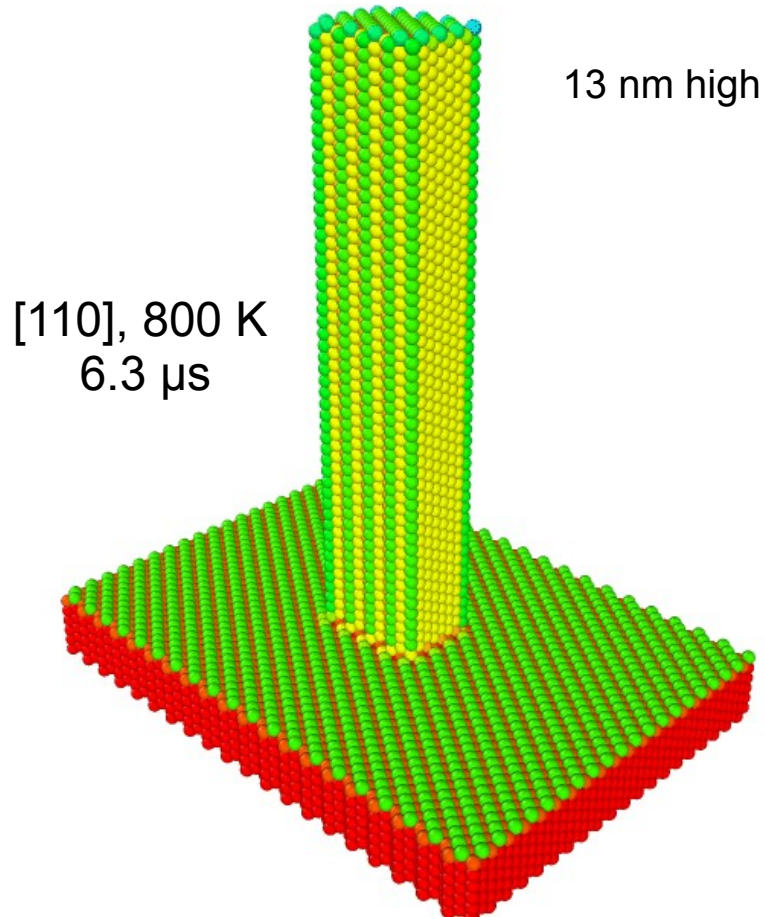


- Good agreement at 850—1150 K
- Strong temperature dependence
- KMC is  $\sim 100$  times faster than MD



# Stability of large tips

Kimocs simulations



- $\langle 110 \rangle$  tips are the most stable ones
- At 300 K, no change was seen after 10  $\mu$ s
- Extrapolation gives a flattening time of 3.1 h at 300 K

<http://arxiv.org/abs/1508.06870>



# Summary & conclusions

- Our Kinetic Monte Carlo surface model
  - Allows us to study surface evolutions at ns to ms
  - Works for Cu, Fe and W (Au in progress)
- Cu tips can be stable for hours at room temperature
  - High dependence on the temperature
- Inclusion of the electric field in progress
- Long term goal: understand the arc problem in CLIC





# Acknowledgement

V. Jansson thanks

**Academy of Finland  
and  
Waldemar von Frenckells Stiftelse**

E. Baibuz is supported by a **CERN** K-contract

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Cu(100), 5.5 ps, 700 K