

The reactive Mg-Liquid infiltration to obtain long superconducting MgB₂ cables

G. Giunchi

EDISON SpA, R&D , Foro Buonaparte 31, 20121 Milano (Italy)

giovanni.giunchi@edison.it

- **Wires by Reactive Mg-Liquid Infiltration**
- **Superconducting properties**
- **Long cables issues and options**
- **Conclusions**

Reactive Liquid Infiltration (RLI): sample preparation 3

(EDISON Spa, Milano(I): EP 1390992, USP 6957480)

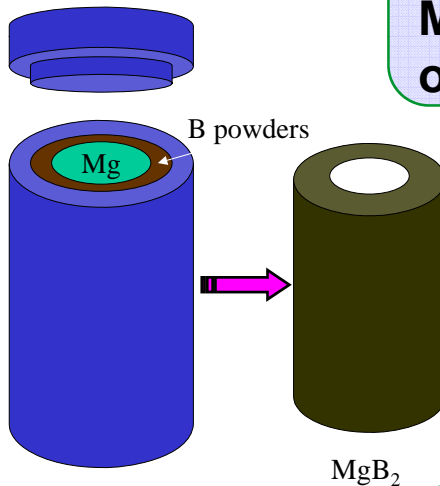
“In situ” process :

NO external pressure

Mg metal, in block form, and B powders (either crystal. or amorphous) are inserted in a SS container and sealed

Moderate thermal treatments:
(700÷900 °C x 0.5÷3 h)

The liquid infiltration proceeds several cm in deepness and is almost gravity independent



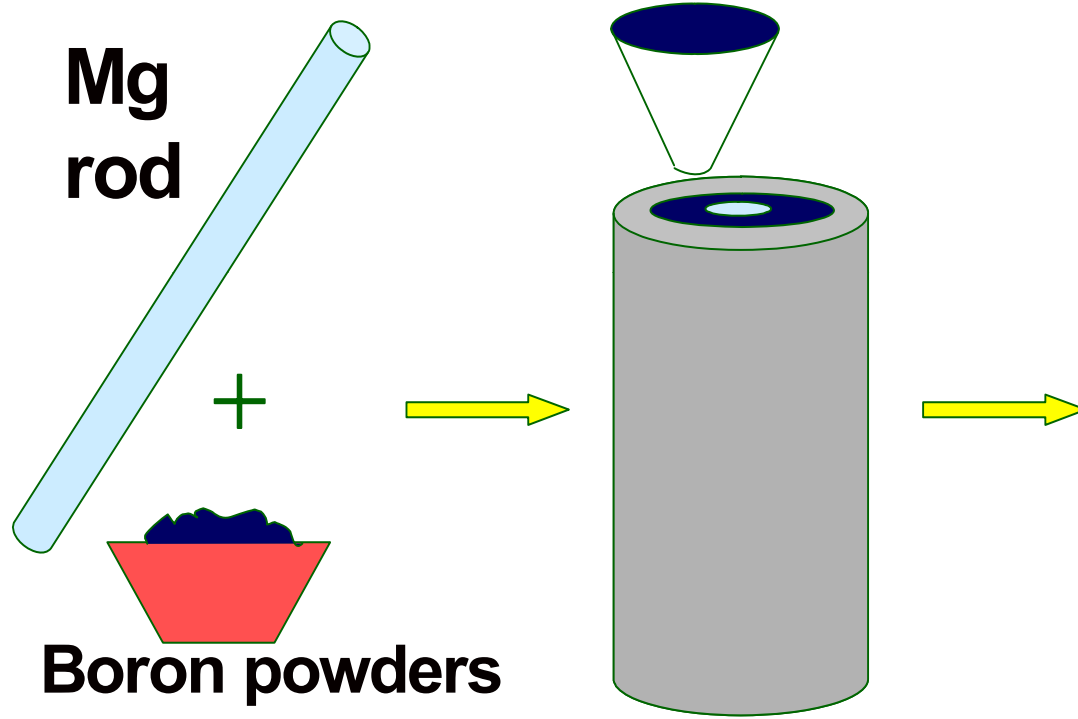
After reaction :

The B space results completely occupied by the MgB₂ and the Mg space remains almost free of products

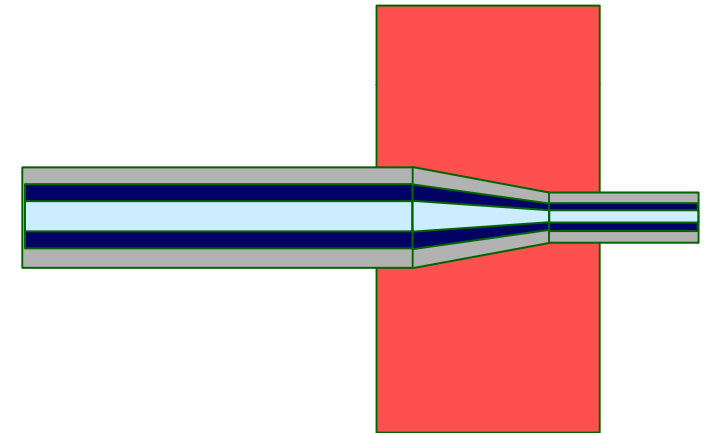
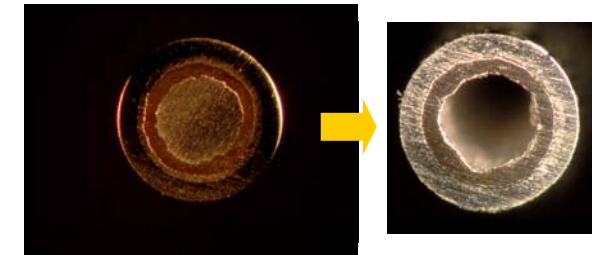
High density (~2.4 g/cm³) MgB₂ and near net shape manufactures

The morphology of the product is related to the grain size of the B powders

Precursor wire preparation



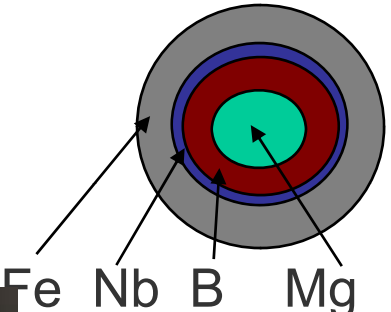
Billet packing

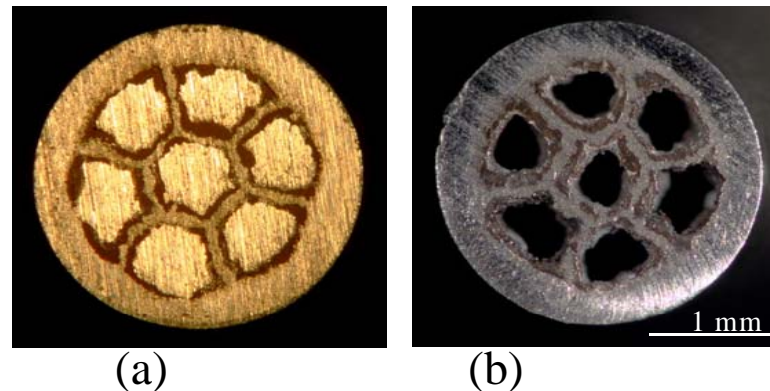


Wire drawing

The hollow wire manufacturing by Mg-RLI

5

- Cold working of a composite billet , typically made as  in Figure
- Possibility to assemble many billets for multiwire manufacturing and further cold working
- Thermal treatment



Cross section optical images :
a) Precursor multifilament wire,
b) Superconducting hollow wire

Monofilamentary hollow wire(2003), by Mg-RLI

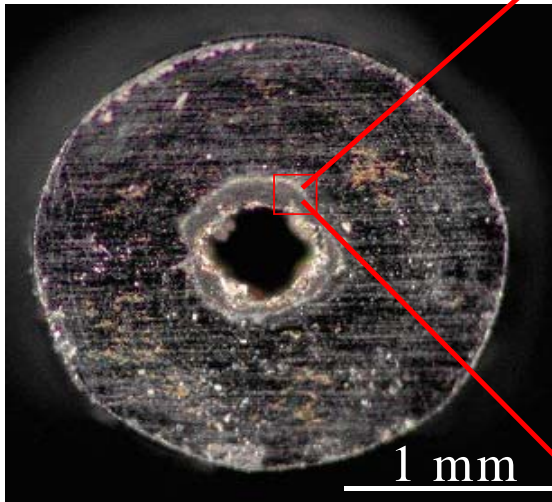
6

A precursor wire was prepared by insertion of a Mg rod, surrounded by microcrystalline B powders, in a Fe tube, internally lined with Nb foil. Thermal treatment at 900°C x 3 h

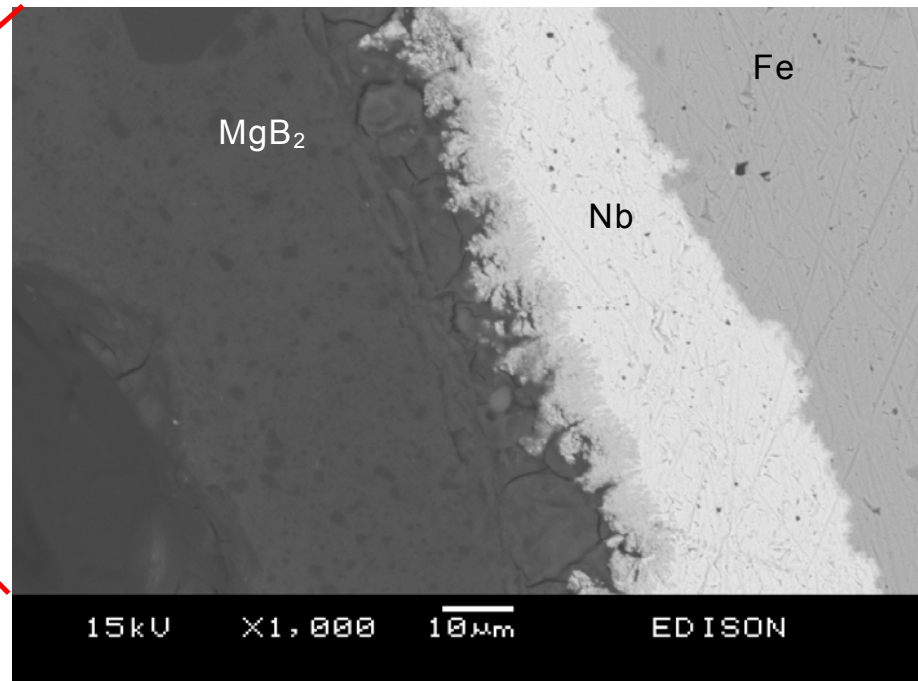
High J_c !

Low J_e !

(Cross area:
SC/tot \approx 0.026)

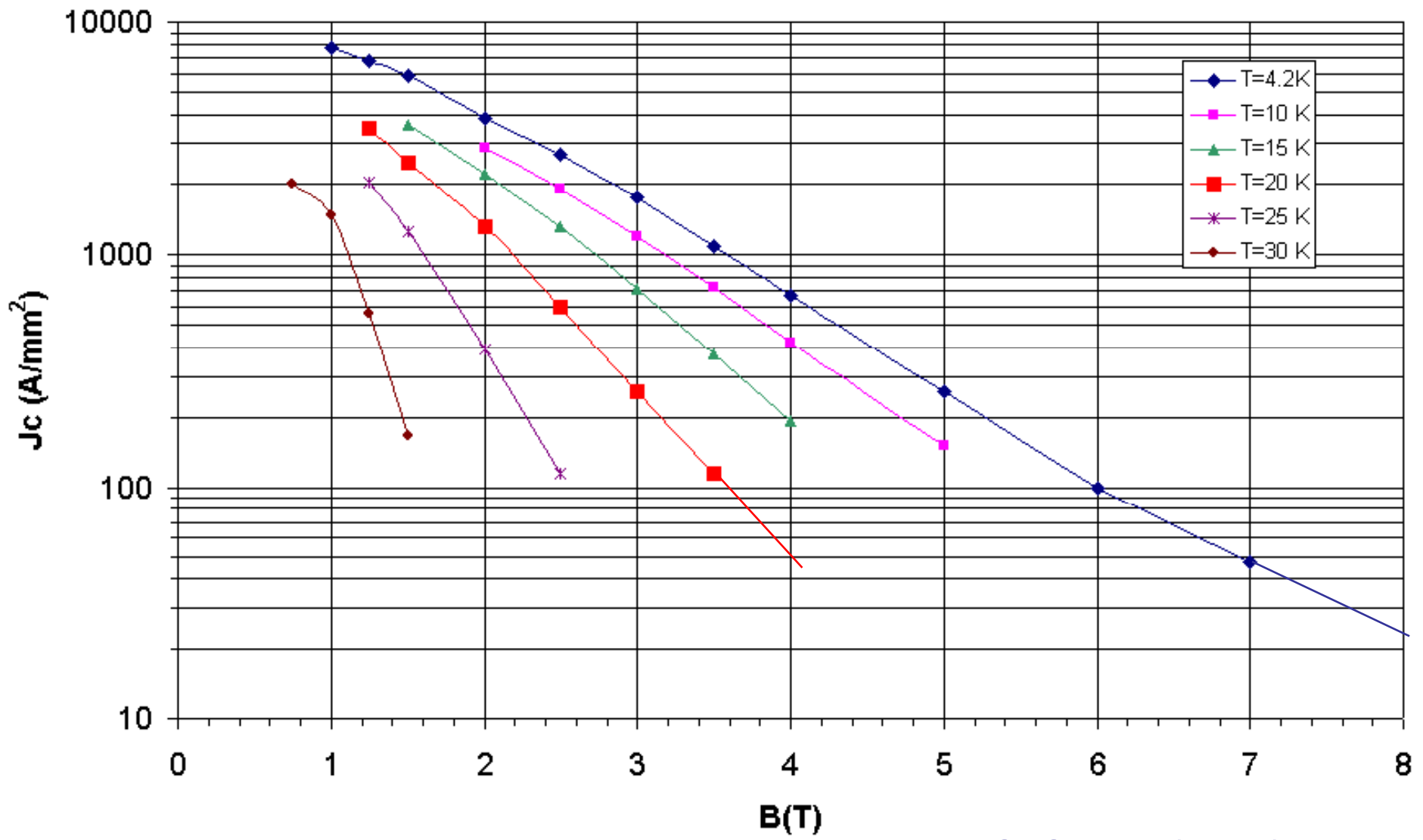


Cross section optical image



Backscattered electrons SEM image

Monofilamentary wire : J_c by transport measurements



G. Giunchi et al. *SuST*, 16 (2003) 285–291

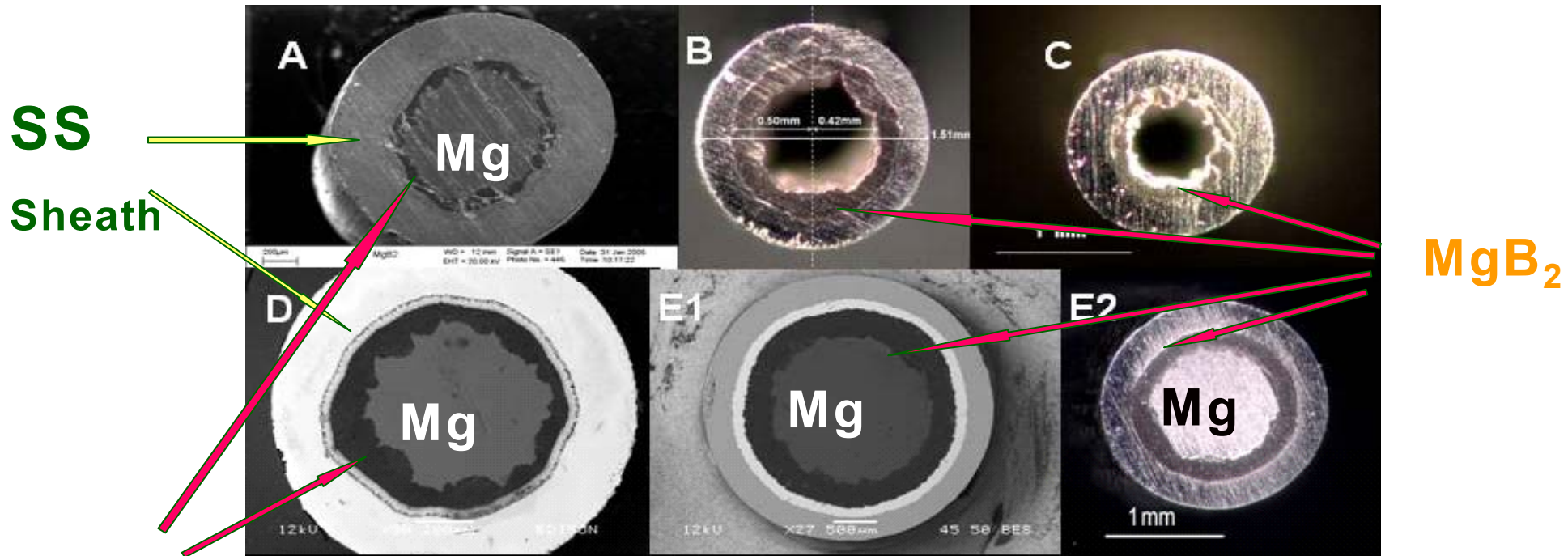
- Mg/B content (wt): **1.25 ÷ 1.59**
- boron powders quality :
amorphous /microcrystalline and purity degree
- C doping : **yes - not**
- Mg hole filling : **yes - not**
- Nb sheath interface : **yes - not**

Precursor wires characteristics

9

Wire	Mg/B (wt)	Diam. (mm)	B	C dop (6wt%)	Hole Filled	Nb
A	1.59	1.70	μ Cr	N	Y	N
B	1.36	1.51	Am	N	N	Y
C	1.25	1.51	μ Cr	N	N	N
D	1.25	1.30	μ Cr	N	Y	N
E1	1.33	3.20	Am	Y	Y	Y
E2	1.33	1.93	Am	Y	Y	Y

Cross sections of the SC MgB₂ wires



MgB₂

Wire	SC Area (mm ²)	Tot. Area (mm ²)	SC/Tot
A	0.15	2.27	0.07
B	0.23	1.80	0.13
C	0.40	1.80	0.22
D	0.37	1.31	0.28
E1	1.92	8.04	0.24
E2	0.75	2.93	0.26

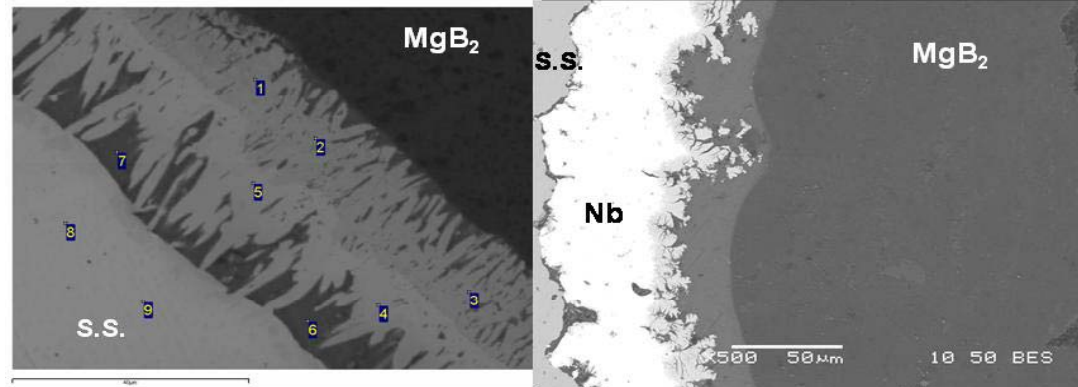
Micromorphology of the phases of MgB_2 wires by RLI

11

The interfaces between MgB_2 and the metallic sheaths

(Samples with the same thermal treatment: $900^\circ\text{C} \times 1 \text{ h}$)

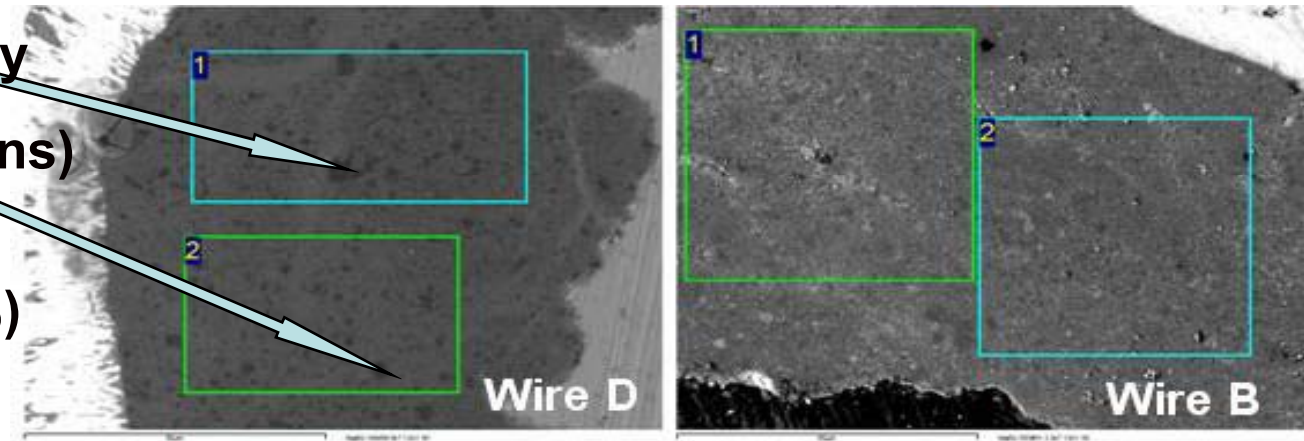
Wire D :
(SS / Boron)



Wire B :
(SS / Nb / Boron)

The MgB_2 micromorphology and impurity phase Mg_2B_{25}

Mg_2B_{25} (impurity
phase rich grains)
(averaged i.p.
comp. $\approx 7\text{mol}\%$)



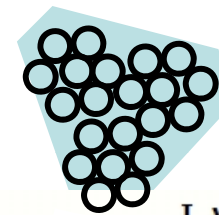
No i.p.
(averaged
i.p. comp.
 $\approx 0.3\text{mol}\%$)

Mg- RLI Stranded Wires : Inductive method to measure $I_c(B,T)$

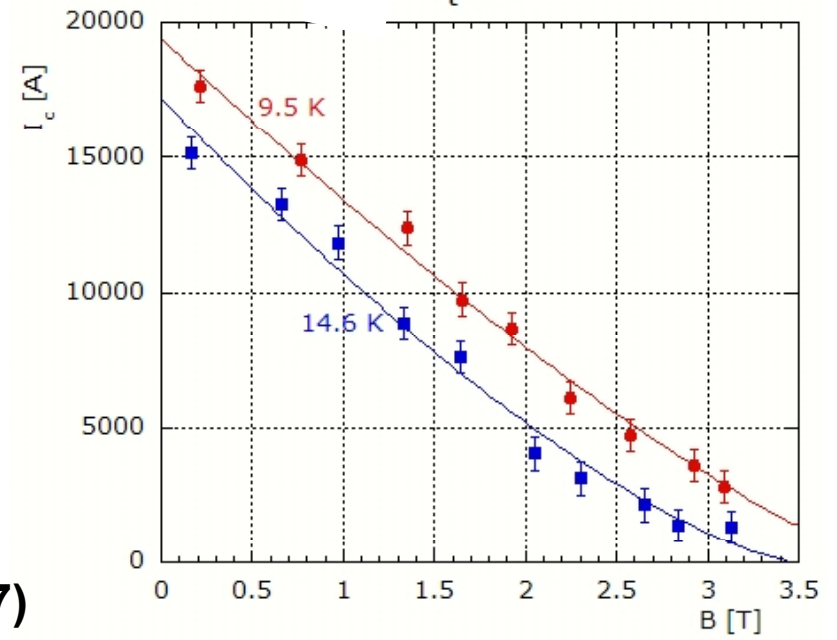
**21 Wires (#T125) ; L=2 m
Stranded (3x7)
Twist pitch $\approx 50\text{mm}$**

Tot area $\approx 40\text{ mm}^2$

SC area $\approx 2\text{ mm}^2$



I_c vs B

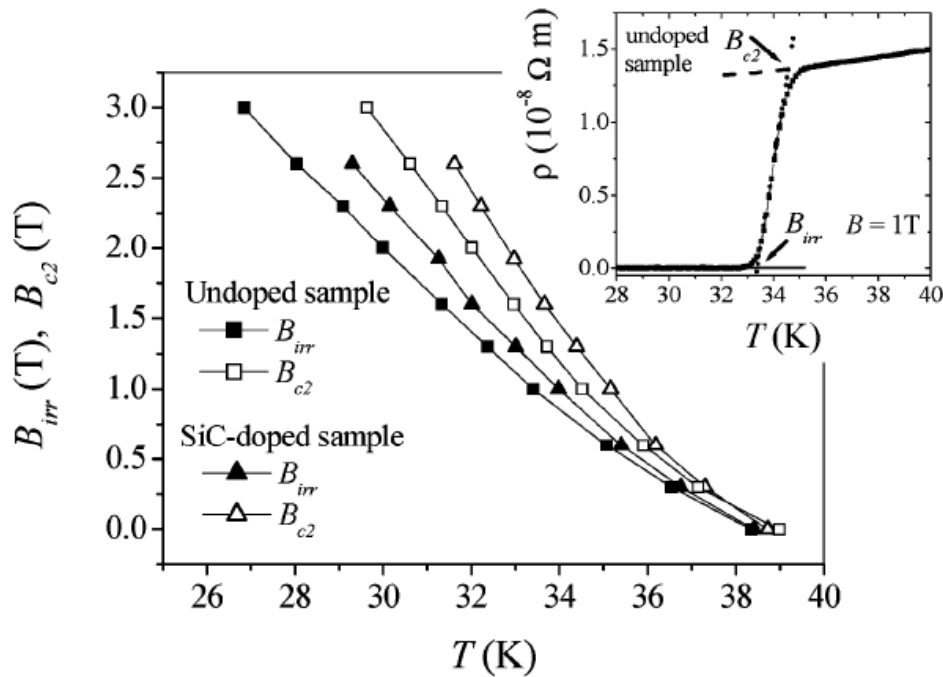


(R.Musenich, INFN, priv. comm.2007)

WAMSDO 2008 - CERN Geneva

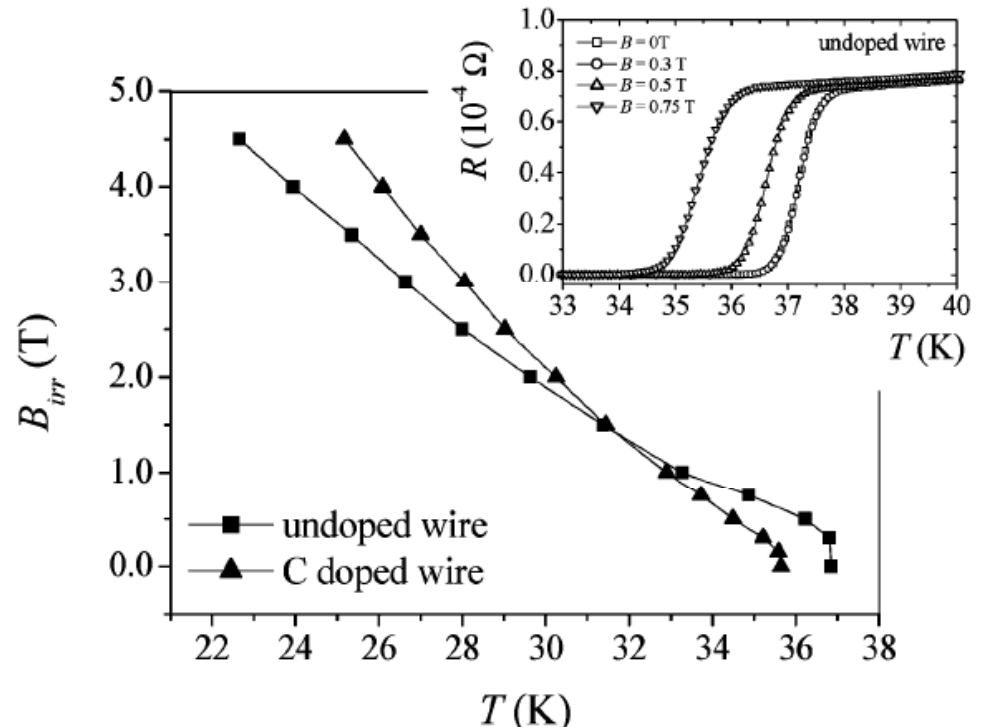
G.Giunchi May,20 2008

Doping MgB₂ by the RLI process



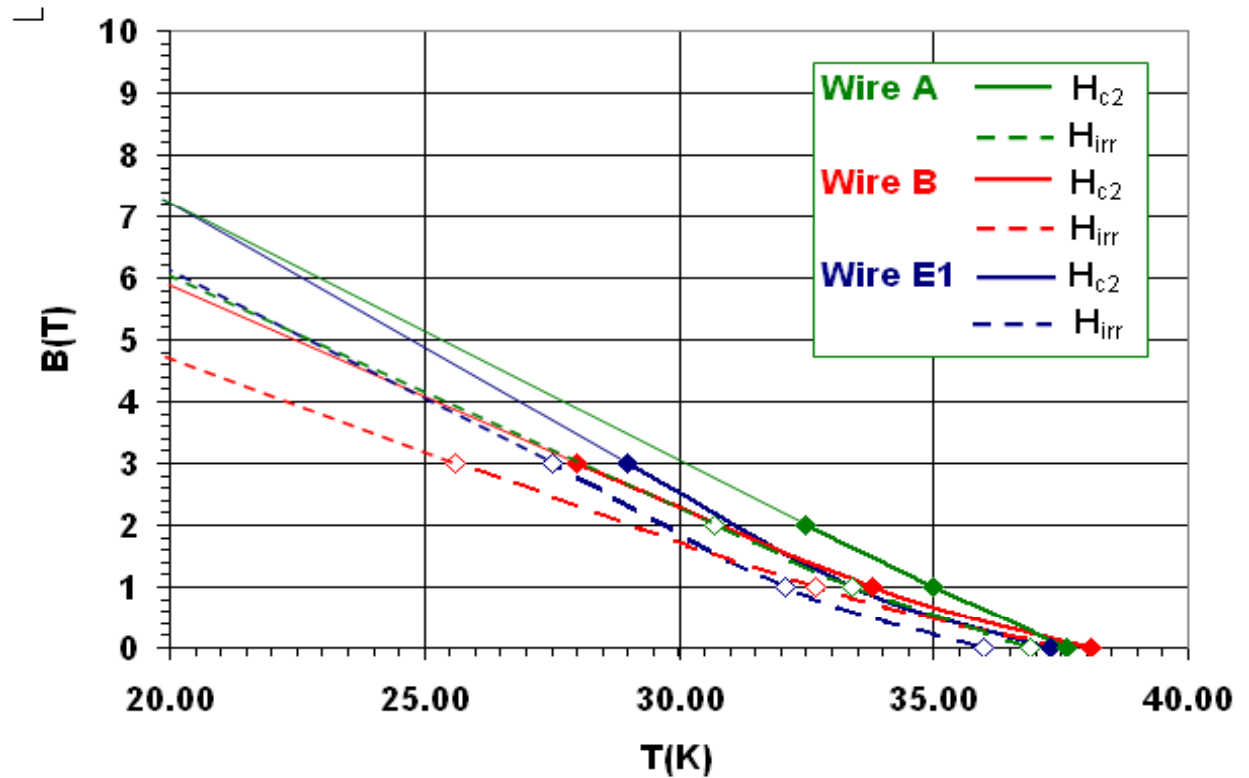
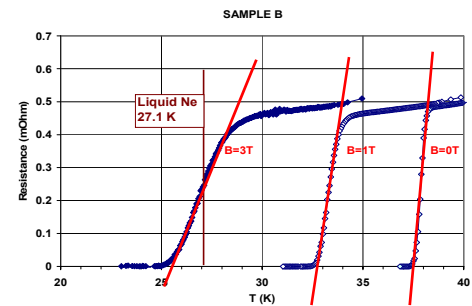
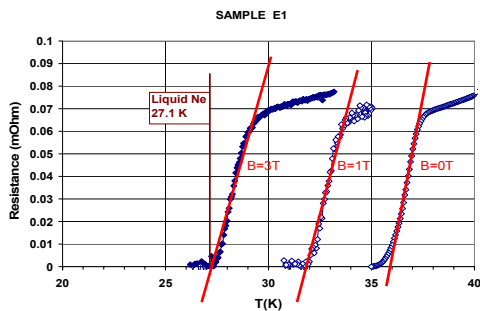
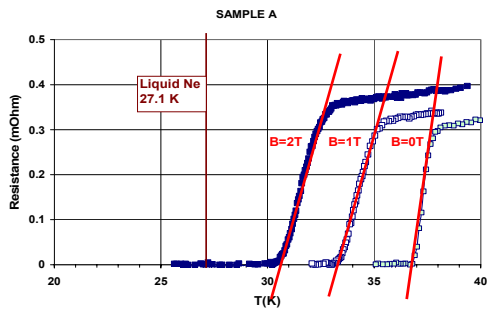
Bulk sample doped by 10% SiC nano powders

Wire doped by 6% C nano powders



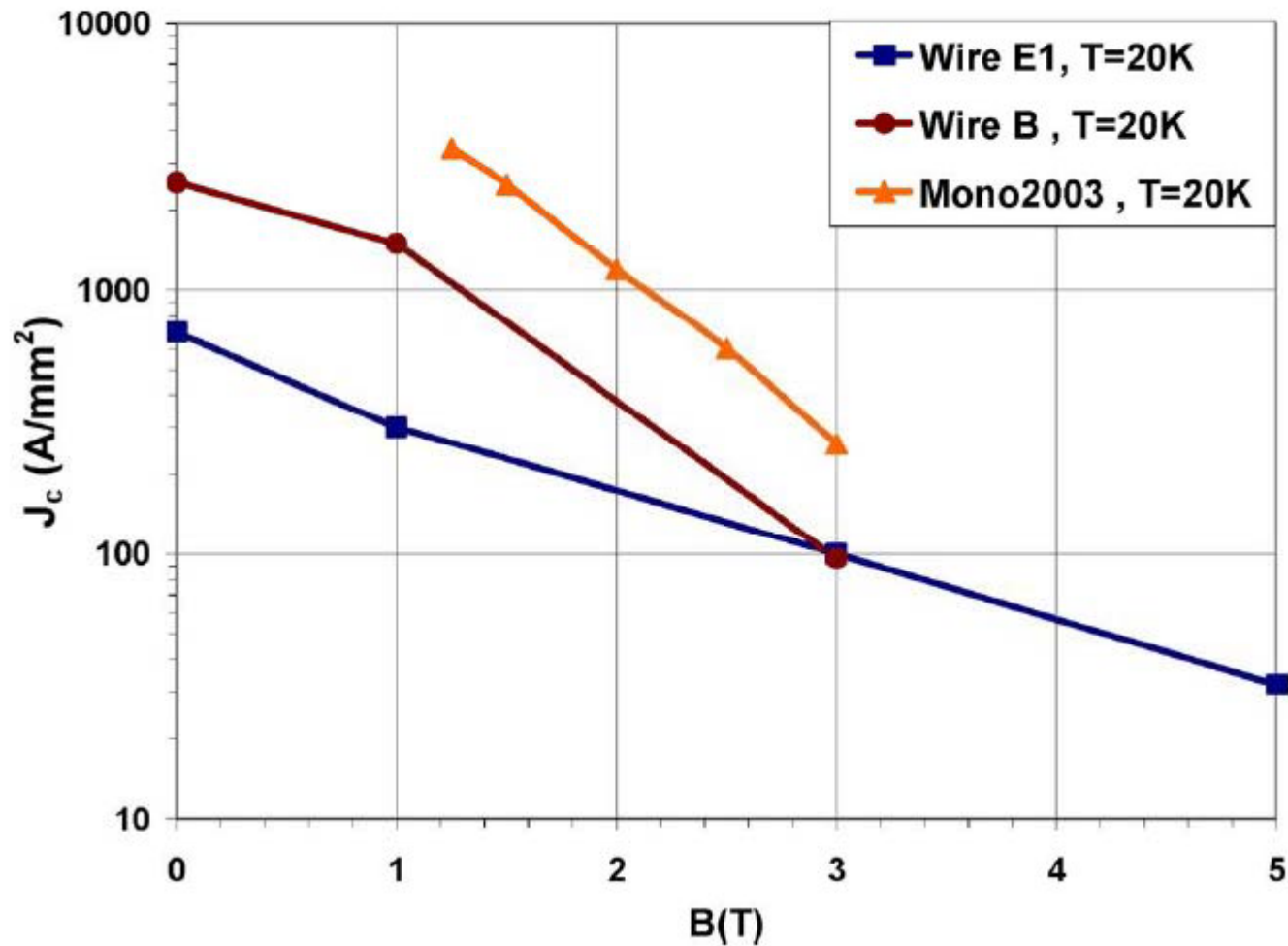
Gozzelino et al. IEEE Trans. on Appl. Supercond. 17, 2726-272(2007)

H_{irr}, H_{c2} evaluation from resistivity measurements



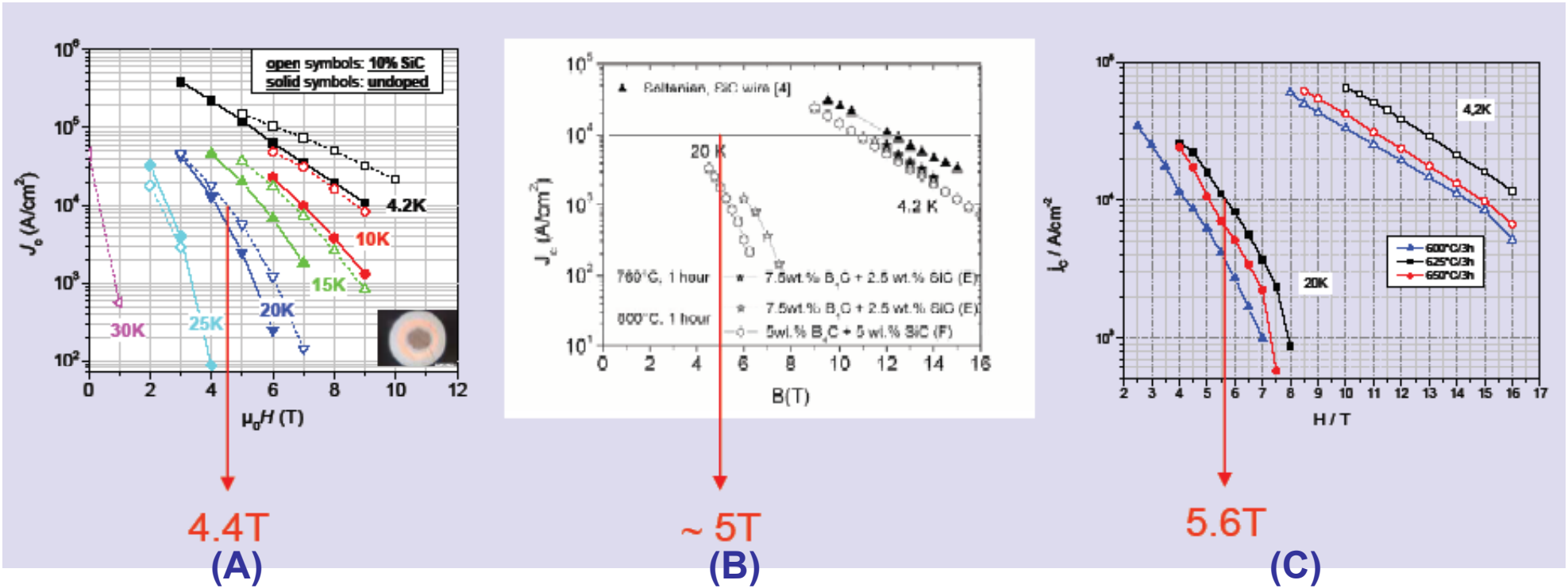
The in-field behavior goes in the order:
Wire E1(C dop.) > Wire A(i.p. dop.) > Wire B(w/o i.p.)

Mg-RLI wires: critical current density at 20K



E1 \rightarrow $B_{am} + 6at\%C$; **B** \rightarrow B_{am} ; **Mono2003** \rightarrow $B_{\mu cr}$

MgB₂ wires : recent J_c comparison



A) SiC doped MgB₂-FZK Karlsruhe: J_c (20K,4T) = 20 kA/cm² ; J_c (4.2K,10T) = 20 kA/cm²

B) B₄C+SiC doped, Un. Geneva: J_c (20K,4T) = 7.5 kA/cm² ; J_c (4.2K,10T) = 15 kA/cm²

C) HE mill+nanoC doped, IFW Dresden: J_c (20K,4T) = 24 kA/cm²; J_c (4.2K,10T) = 40 kA/cm²

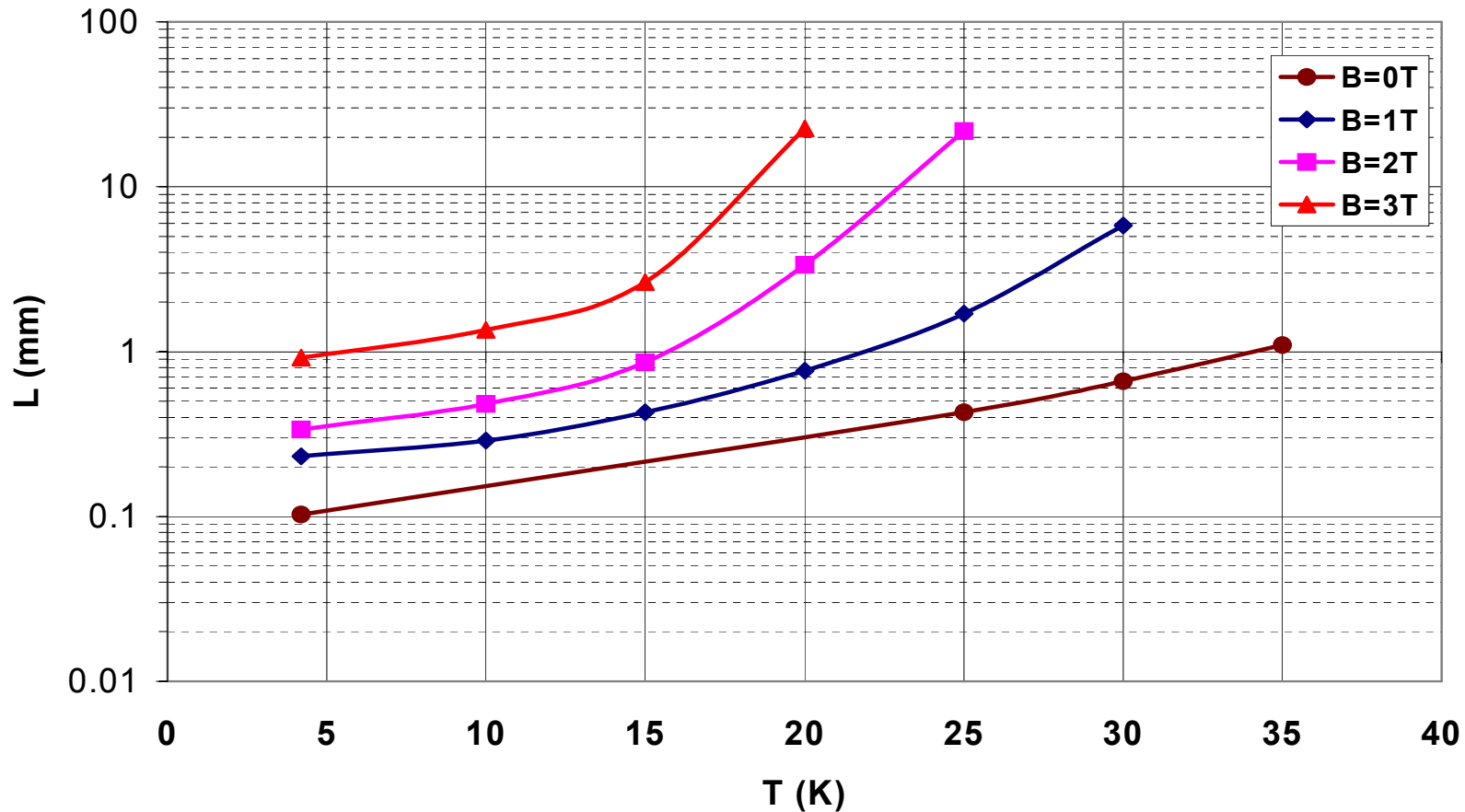
EDISON Mg-RLI +6% C doped : J_c (20K,4T) ≈ 6 kA/cm² ; J_c (4.2K,10T) ≈ 8 kA/cm²

Thermal stability of the MgB₂

17

Limitation on the thickness

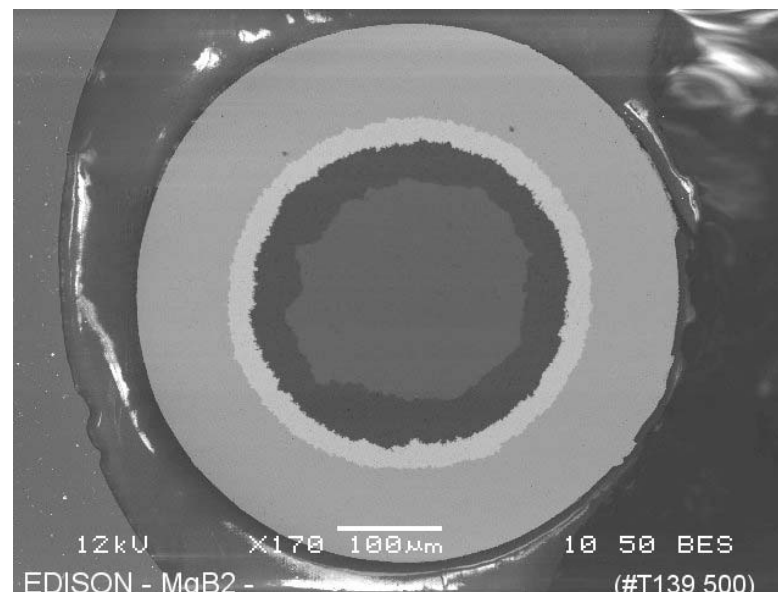
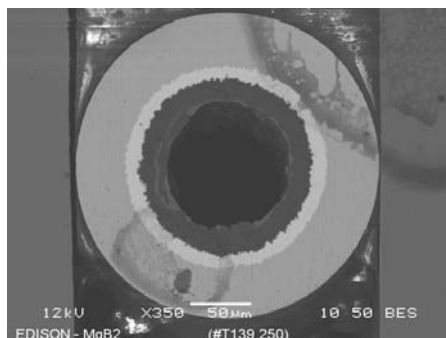
$$\mu_0 J_c^2 L^2 < 12 \gamma C (T_c - T)$$



G.Giunchi, *Adv. Cryo. Eng.Trans. of ICMC*, vol 52 813-819 (2006)

- #T139 precursor wire (Monel/Nb/B/Mg) where successfully drawn to long length (6m) and **0.25 mm** diameter (*)
- ITER Barrel test on **0.52 mm** intermediate MgB_2 wire, W&R, confirms the $J_c(B)$ at 4.2K obtained for larger diameter (1 mm)

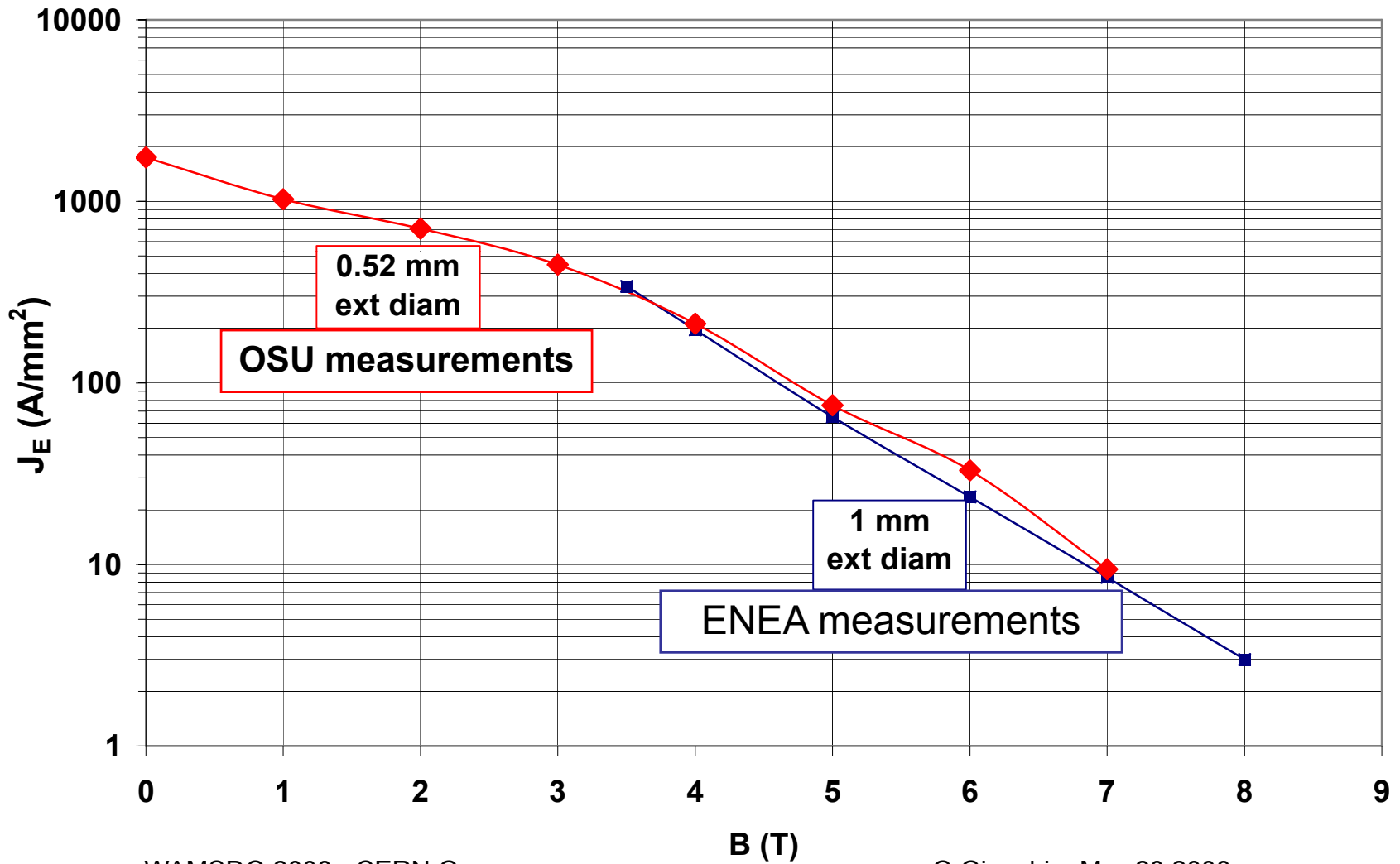
These wires have a SC fill factor = 0.16



(*) performed by Matt Rindfleisch (Hypertech,OH)

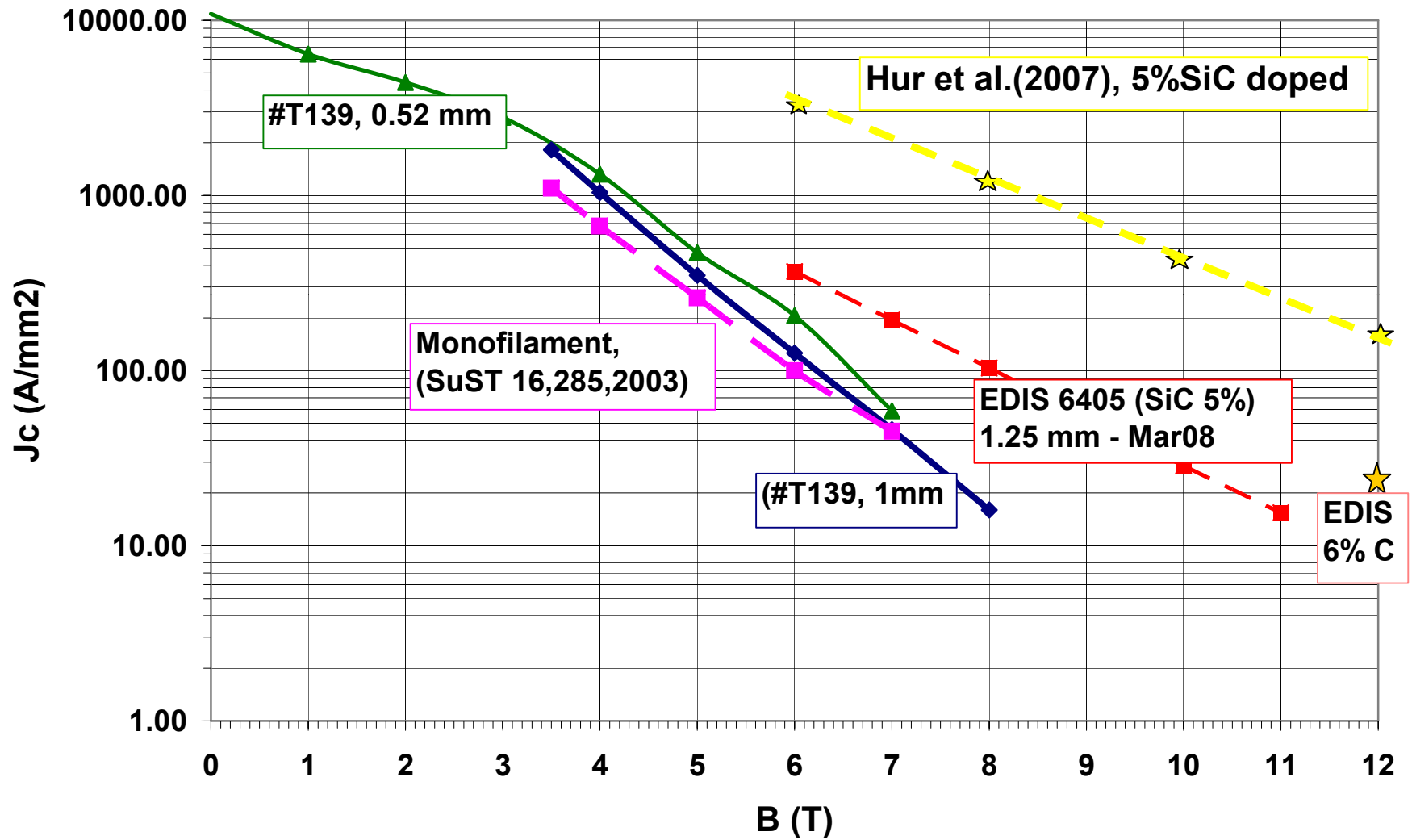
Engineering current density of MgB₂ wire by Mg-RLI

#T139 wire (1 m long) - T=4.2 K
ITER Barrel



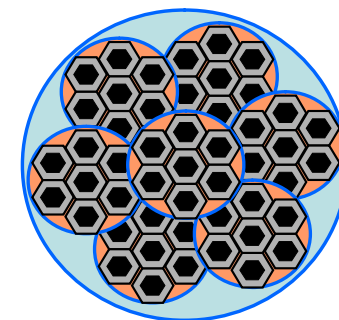
Mg-RLI wires : Jc comparison

MgB₂ wires by Mg-RLI T=4.2K

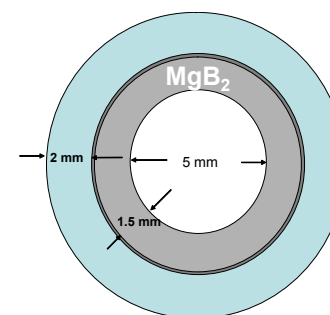


A1) Thin hollow multiwires (0.25 mm) to perform stranded cables. Typical diameter =3 mm

A2) Multiwires by hollow monofilament of 1 mm. Typical diameter 12 mm



B) Thick monofilament hollow cable. Typical diameters from 10 to 15 mm



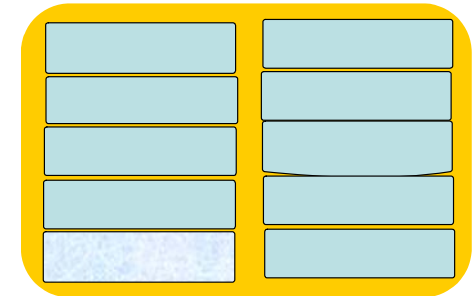
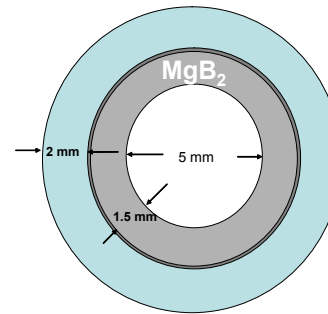
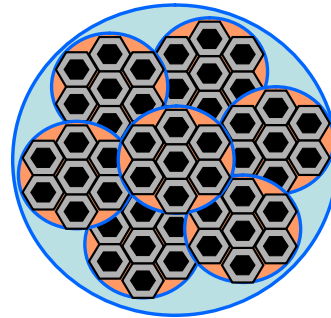
C) Thick massive cable. Typical dimensions will be (10÷20) x (50÷100) mm²



Large magnets: Superconducting cables vs Copper

Reference winding:

$I_{total} = 870 \text{ kA (@20K,4T)}$
 Area = (10x20) cm²
 Diameter = 500 cm



Cable	Lenght (m)	Turn number	I(20K,4T) (kA)	diameter (mm)	Fill factor (%)	Current Density (A/mm ²)	Winding weight (kg)
multiwire 49	1809	120	7.2	12	15	420	1750
monowire	1809	120	7.2	12	30	210	1100
Massive	150	10	87	90x19	60	85	830
Copper	1809	120	7.2	12	100	60	2700

- The Mg-RLI technique allows to obtain MgB_2 round wires ,thin and long.
- The actual performances depend on the quality of the B powders and on the thermal treatment optimization.
- A substantial improvement in magnetic field is expected from the doping.
- The distinguished characteristics are lower weight ,a more easy cooling and high mechanical strength
- The cost of the cable manufacturing should be very low
- The magnet manufacturing will use mainly W&R , but also R&W will be accessible to the thinner wires.