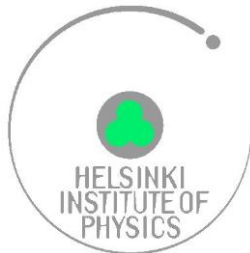


Processing and characterization of epitaxial grown GaAs radiation detectors

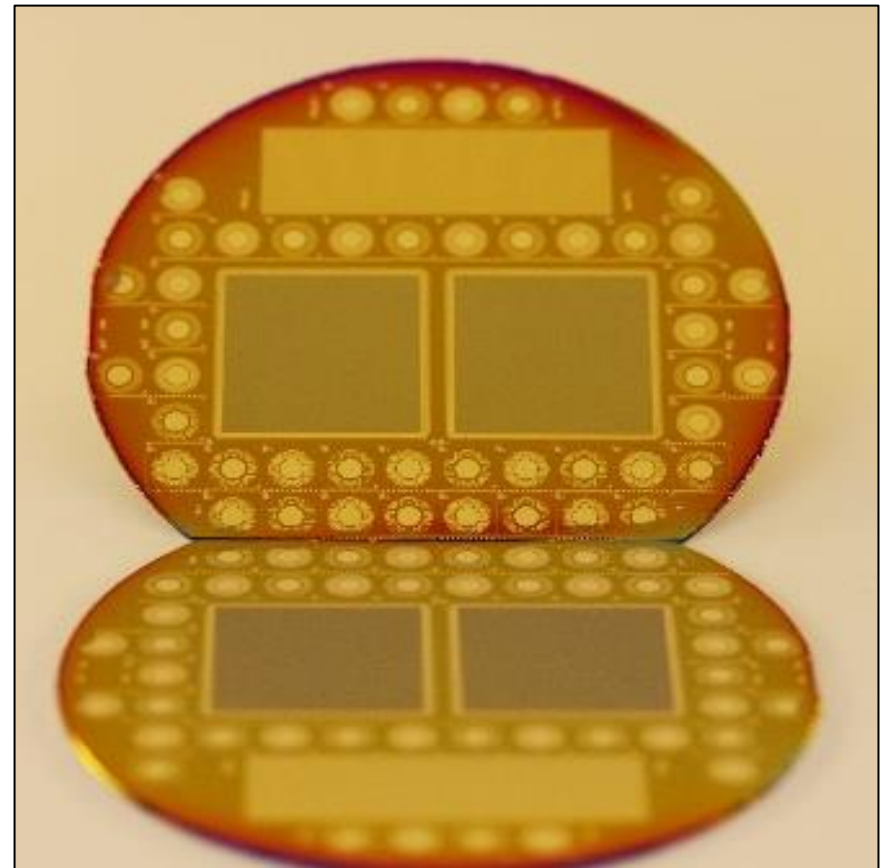
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Outline

- ❑ Motivation & Background
- ❑ Vertical CVPE reactor: Implementation
- ❑ Processing of GaAs detectors: Wafers
- ❑ Characterization
 - CV/IV
 - TCT
 - DLTS
- ❑ TCAD simulations
- ❑ Conclusions

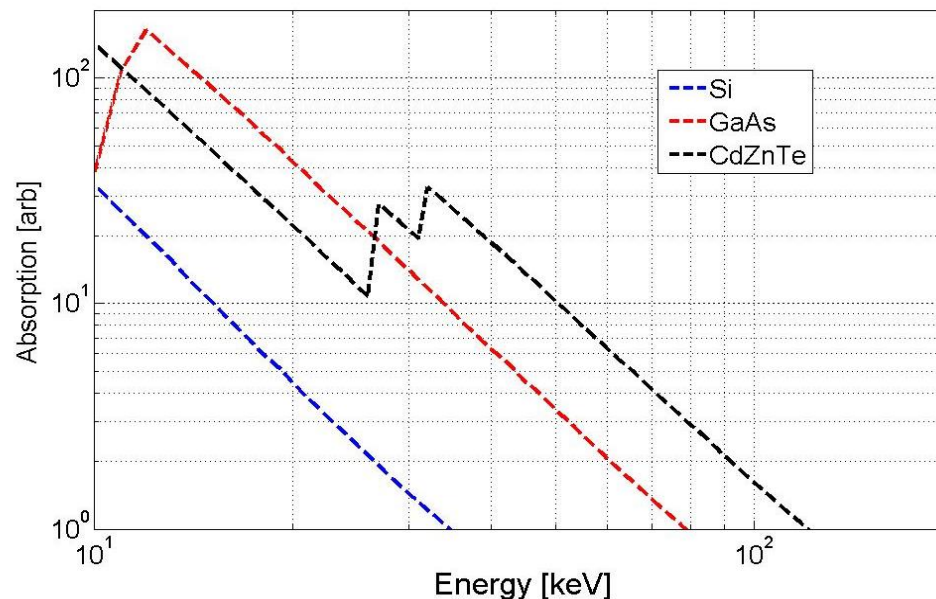


Processed GaAs wafer

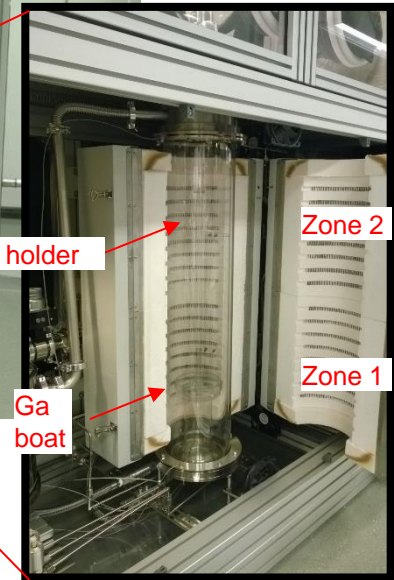
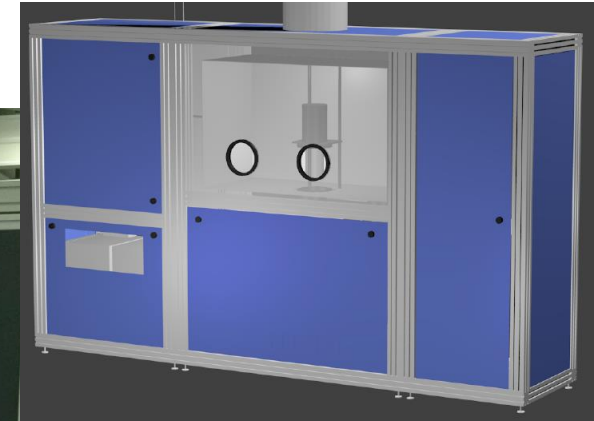
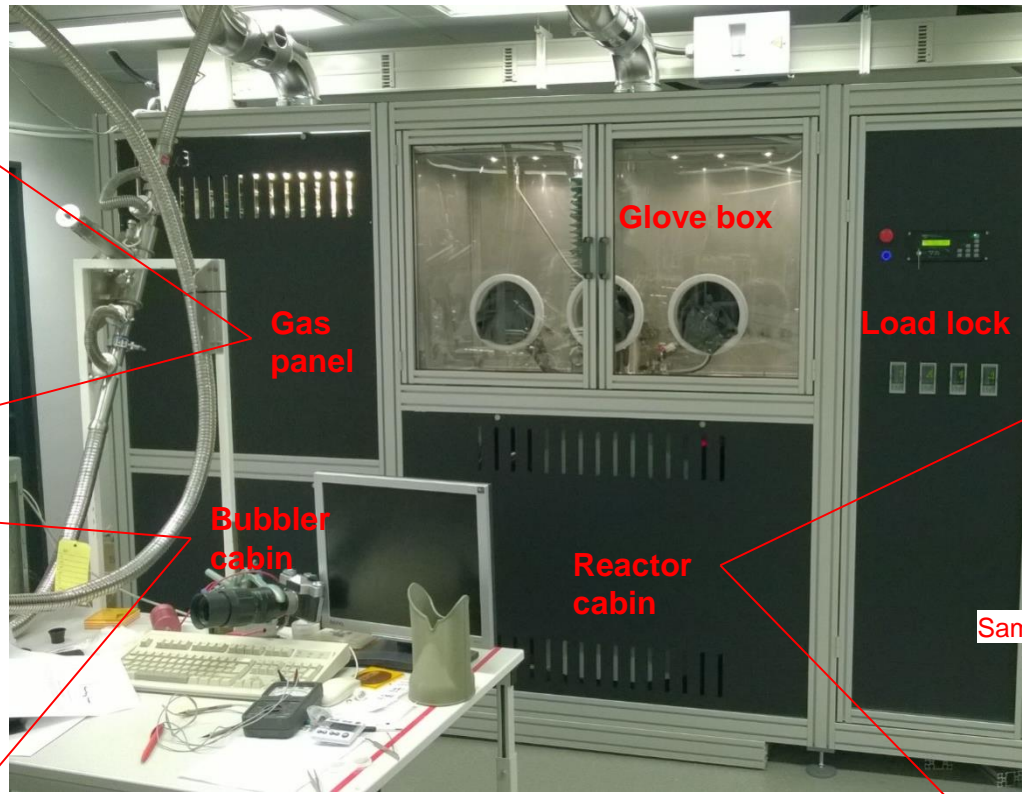
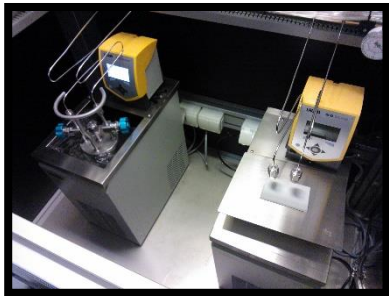
Motivation and background

- Gallium arsenide (GaAs) vs. Silicon (Si)
 - High electron mobility (~ 5.7 times higher than Si)
 - High atomic numbers ($Z=31, 33$ vs $Z=14$ in Si), direct band and wide bandgap, high absorption efficiency and low leakage current at room temperature
- Detector-grade GaAs materials: epitaxial, semi-insulating
 - Technology well-established with respect e.g. CdZnTe

Simulated absorption efficiency vs X-ray photon energy:



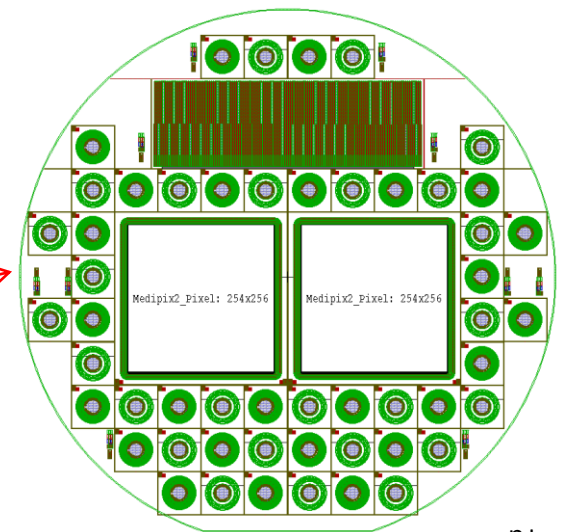
Vertical CVPE reactor: Implementation



- CVPE = Chloride Vapor Phase Epitaxy
- Growth rate $\sim 10 \mu\text{m/h} \rightarrow >100 \mu\text{m}$ layer thicknesses

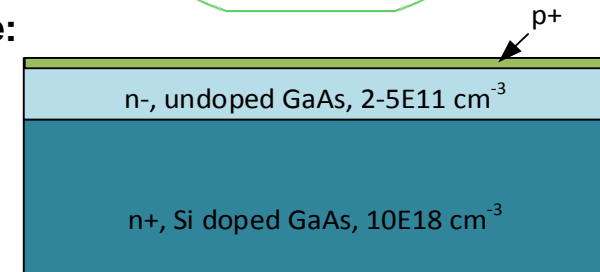
Processing of GaAs detectors: Wafers

- Five GaAs epitaxial wafers were brought from Ioffe physical technical Institute
- Two wafers were processed at VTT
 - Each wafer yields two Medipix sensors, one strip sensor and many diodes



Wafer no.	p+ layer	n-type epi-layer	n+ substrate
D167	1.5 – 2 μm	112 μm	460 μm
D168	1.5 – 2 μm	130 μm	460 μm

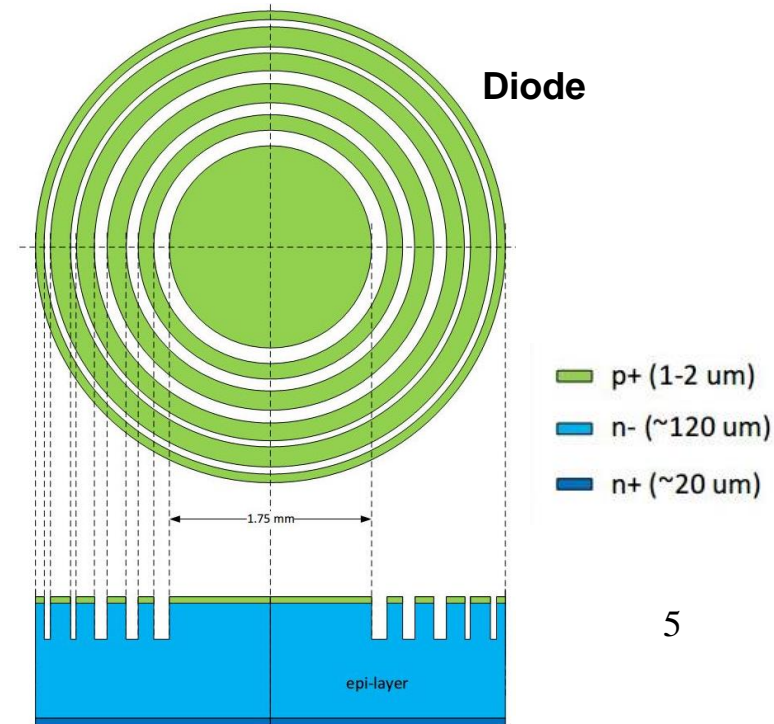
Detector structure:



Why not n⁺/p⁻/p⁺ detectors:

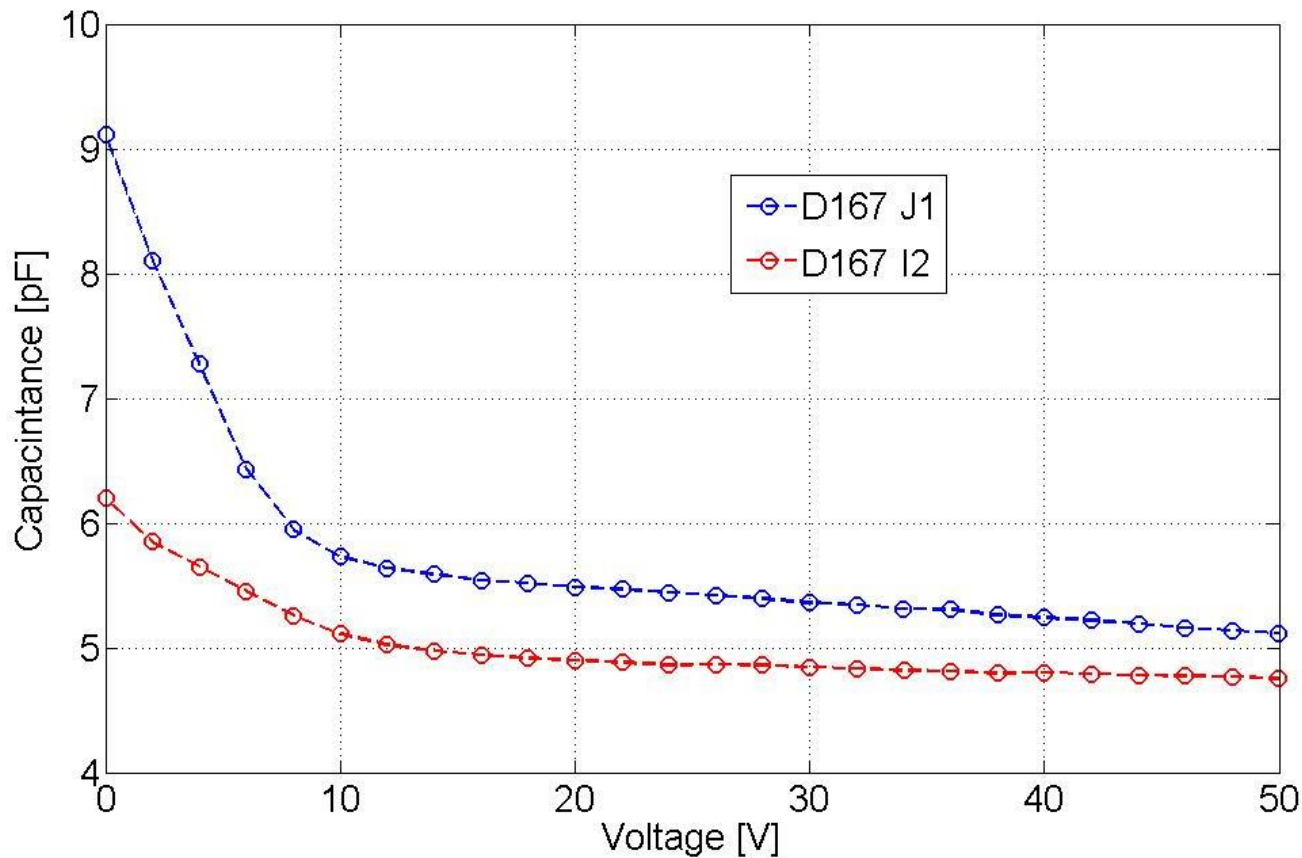
- Reason for growing n⁻ layer instead of p⁻:
 - CVPE grown GaAs is intrinsically n⁻ type
 - Controlling the growth of p⁻ layer at very low doping concentrations is difficult (or impossible)
 - Unavoidable Zn diffusion from the p⁺ substrate might dope the p⁻ layer too much
 - N-type dopant diffusion (usually Si), in contrast, is very weak.

Diode



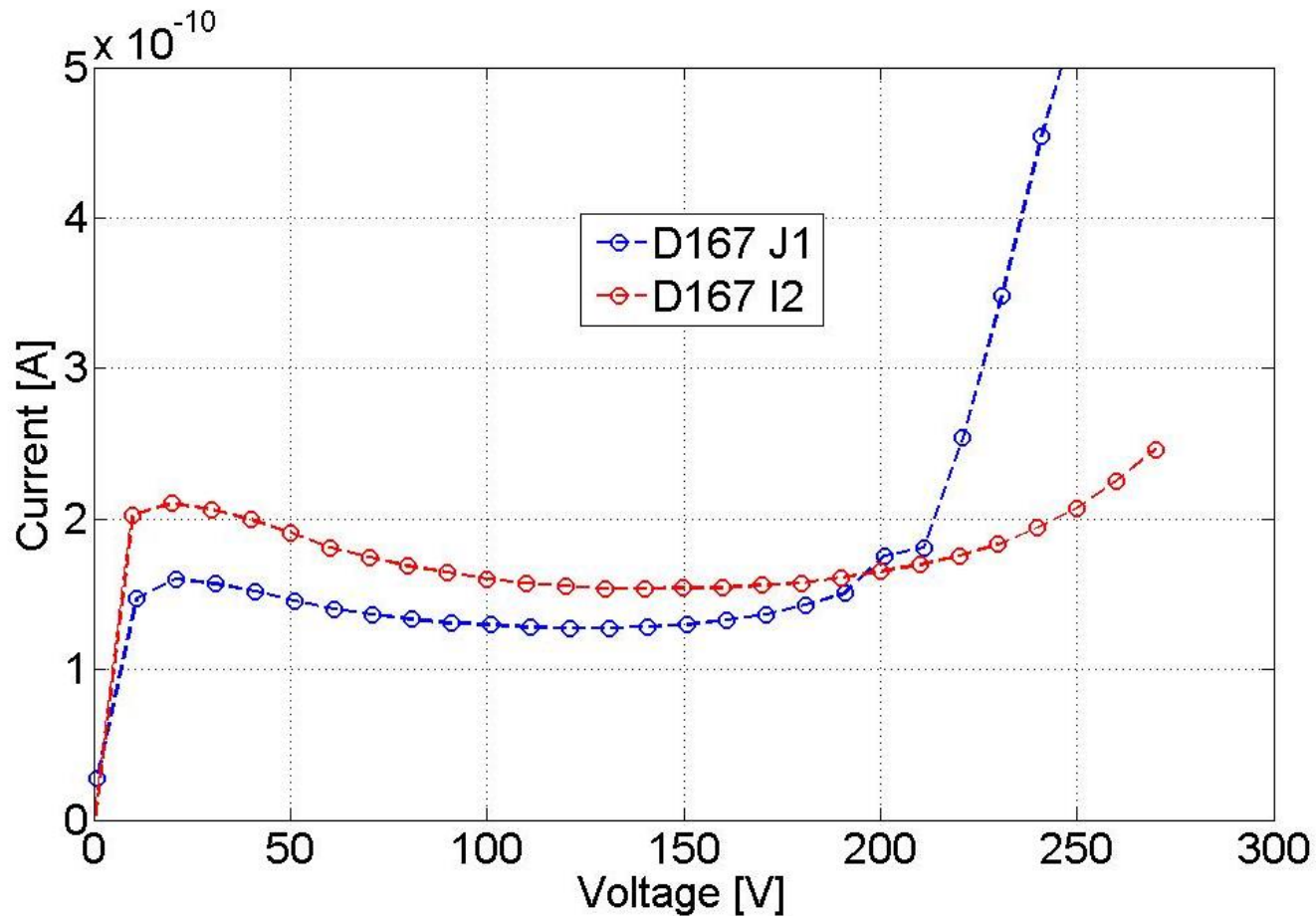
Characterization – CV/IV

- $V_{fd} \sim 20 - 25V$
- Clear capacitance saturation not apparent
 - Most likely due to device structure (no guard rings)

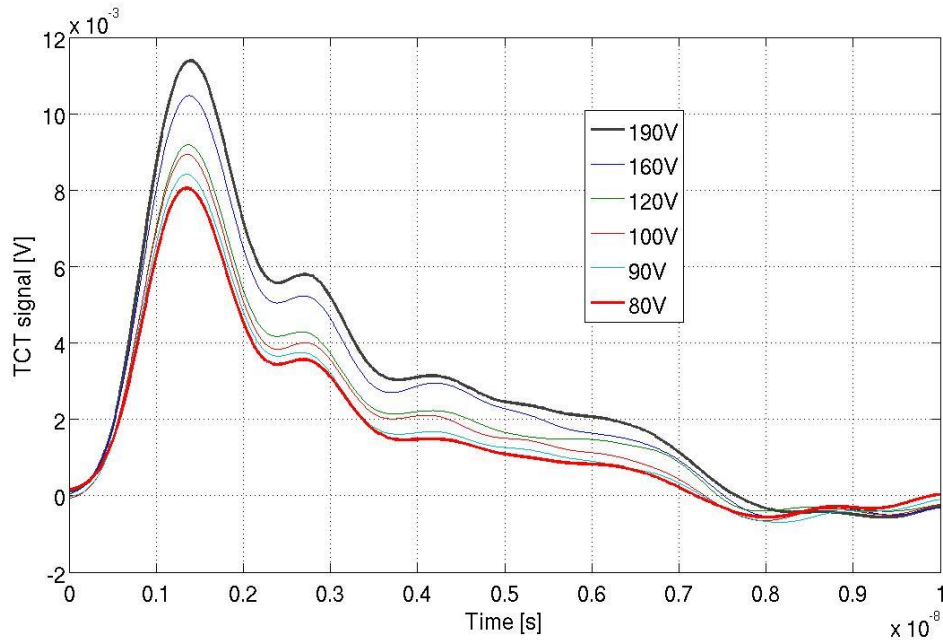


Characterization – CV/IV

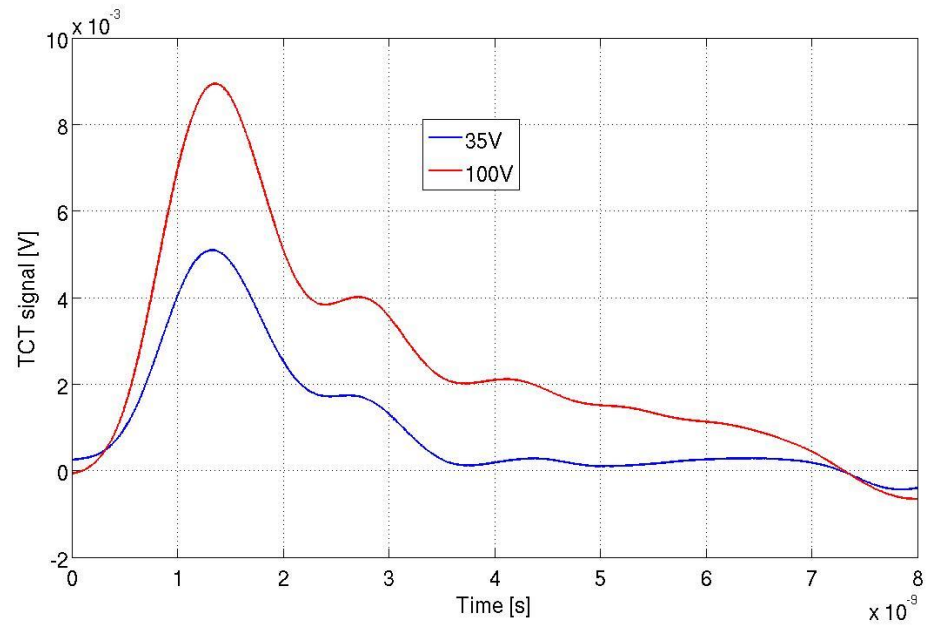
- Leakage current $I_{\text{leak}} \sim 100 - 200 \text{ pA}$
- Current density $J \approx 10 \text{ nA/cm}^2$



Characterization – TCT



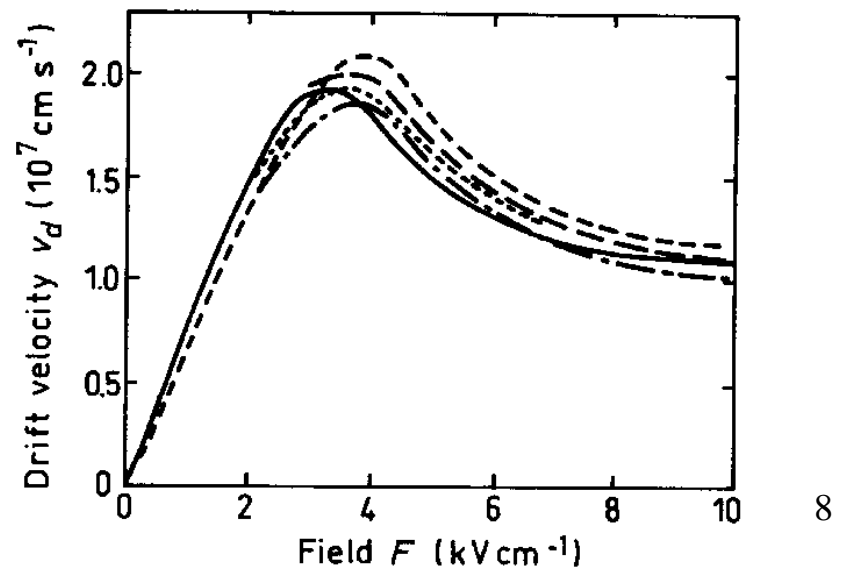
High electric field, $t_{transit} \approx 8$ ns



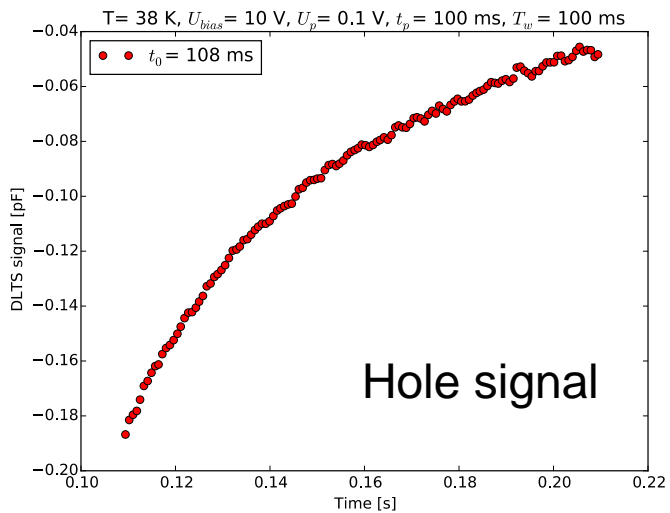
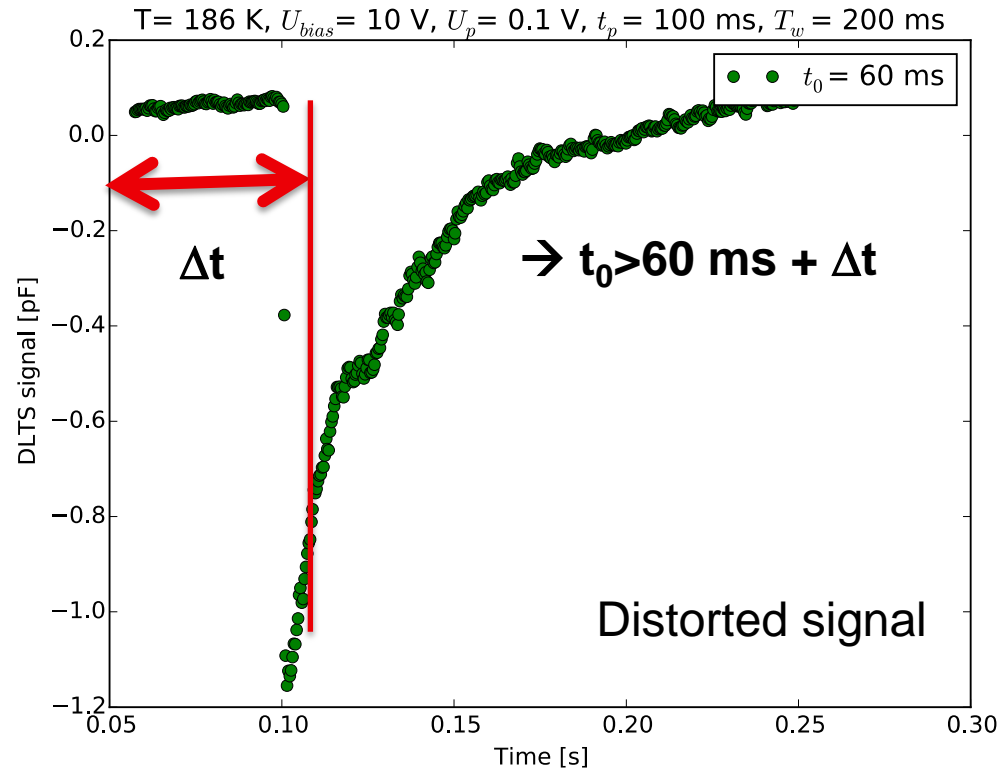
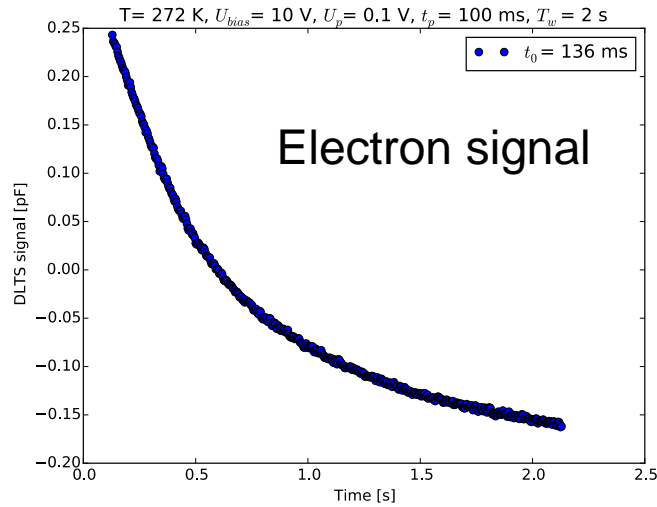
Low electric field, $t_{transit} \approx 4$ ns

Field dependences of the electron drift velocity (Blakemore[1982])

- Solid curve calculated by (Pozhela and Reklaitis[1980])
- Dashed and dotted curves are measured data, 300 K

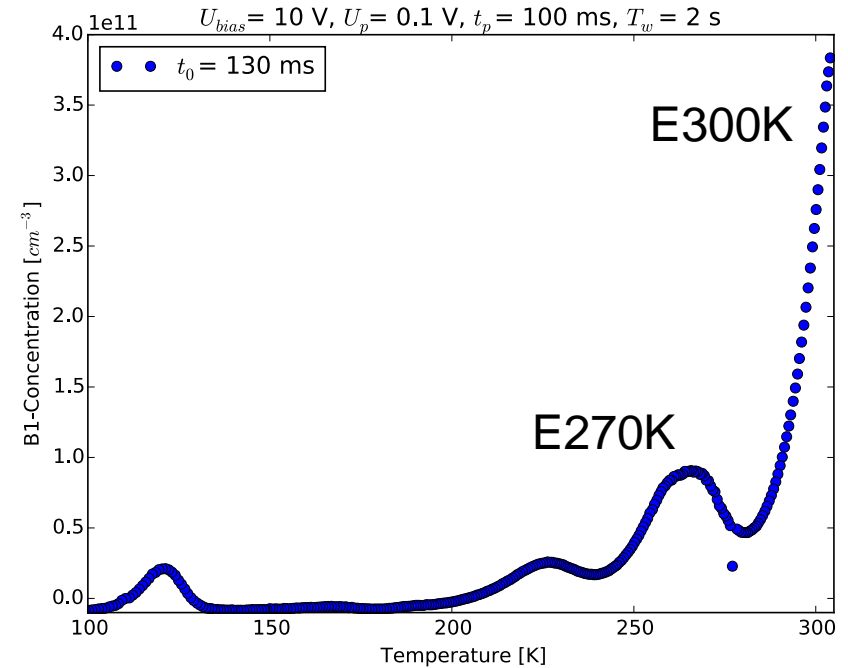
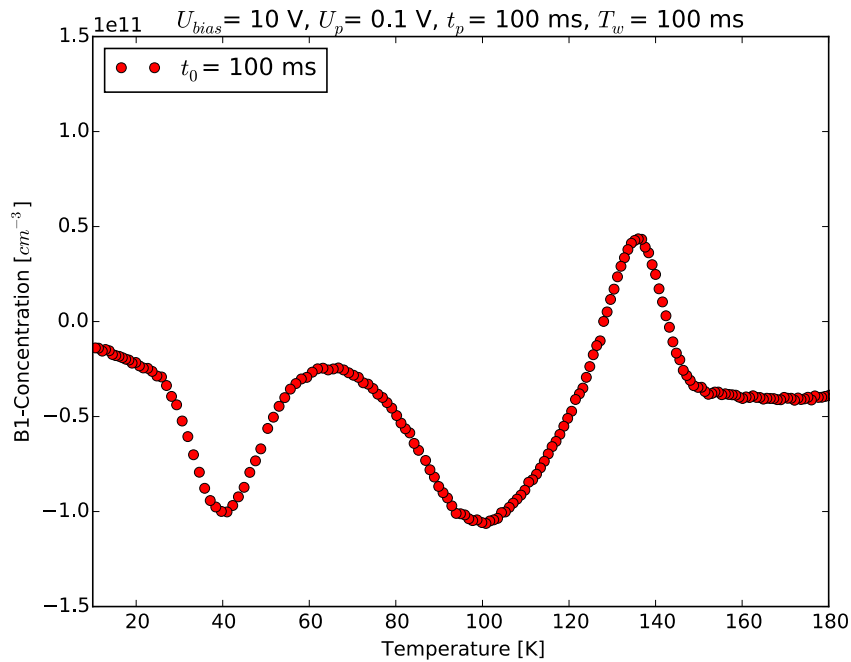


DLTS - Parameters



“Charge up effect”?
Prolong bridge recovery time (t_0) for stable measurement
 \rightarrow Careful adjustment required or cut into signal !

Traps in the band gap



Shallow hole traps
 Not visible for high t_0
 → Fast emission
 → Concentration few $\times 10^{10} \text{ cm}^{-3}$

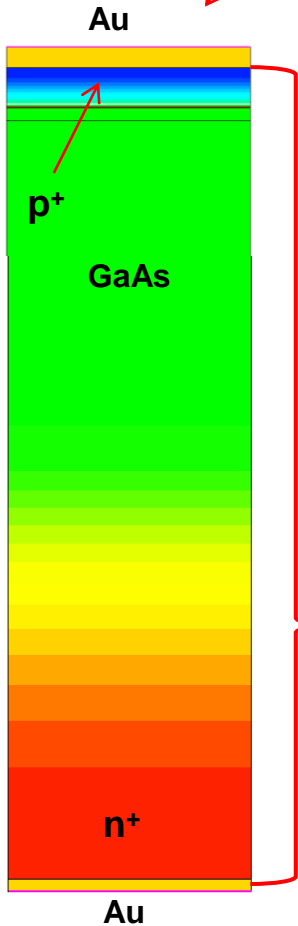
Deep electron traps
 Concentration up to few $\times 10^{11} \text{ cm}^{-3}$
 → Need to go to 350 K for full peak

❑ Deep electron traps could also be lattice distortions, further analysis needed for confirmation

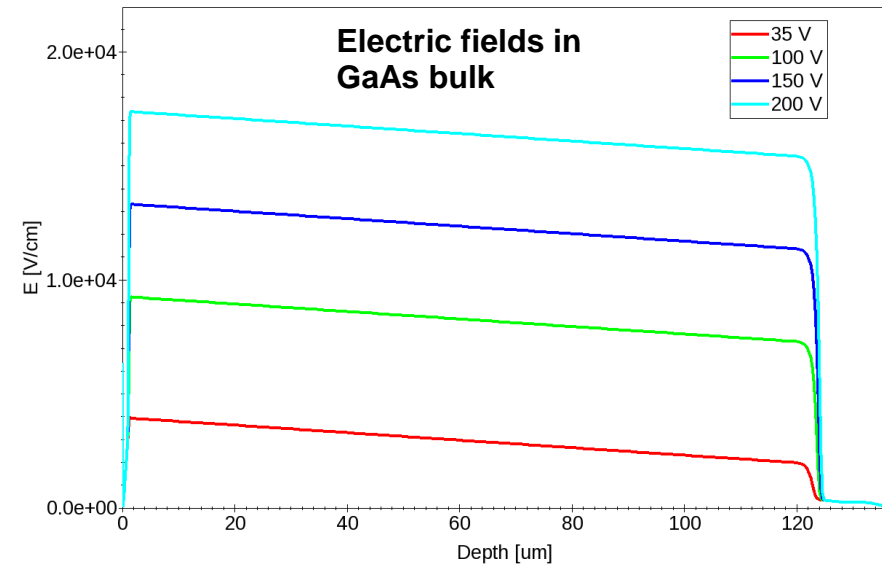
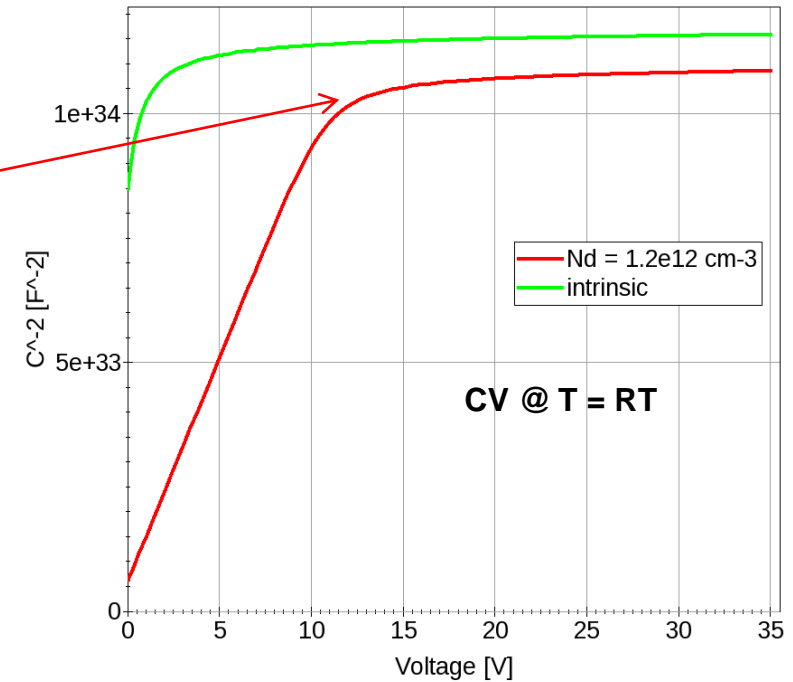
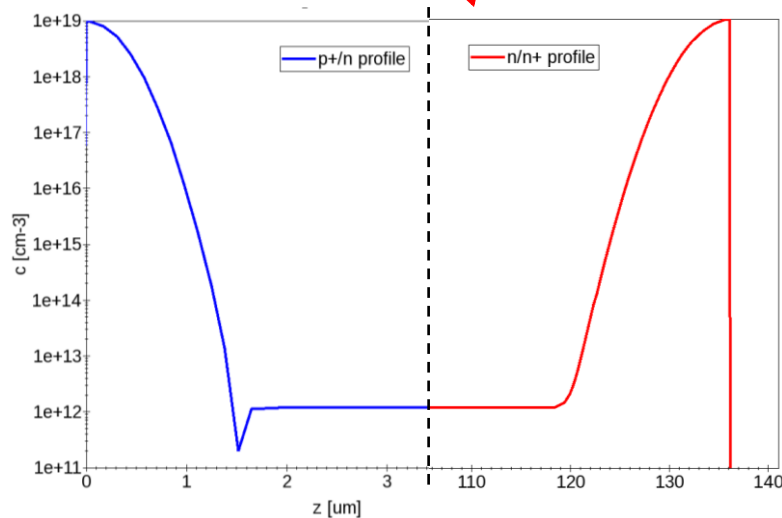
Trap	E_a [eV]	σ [cm^2]	N_t [cm^{-3}]
E270K	0.700	2×10^{13}	1×10^{11}
Very preliminary: E300K (from fit)	0.970	1×10^{13}	5×10^{11}

TCAD simulations - structure & parameters

- 10x1x136 μm^3 diode structure
- Dc-coupled gold contacts $d = 500 \text{ nm}$
- Initial structure p-i-n $\rightarrow V_{\text{fd}} \approx 15 \text{ V}$ reached by arsenic doping $1.2 \cdot 10^{12} \text{ cm}^{-3}$
- Epi-layers modelled by varying the depth of p⁺ and n⁺ doping profiles



Simulated diode structure (not in scale)



TCAD simulations - Gunn effect

❑ **Gunn effect:** E-field in the material reaches a threshold level → mobility of electrons starts to decrease with higher E-field due to transferred electrons from one valley in the conduction band to another (small → large effective mass & very high → very low mobility)

❑ Charge injection from front surface $d \approx 4 \mu\text{m}$
→ transient signal from electron drift

❑ Gunn effect reproduced

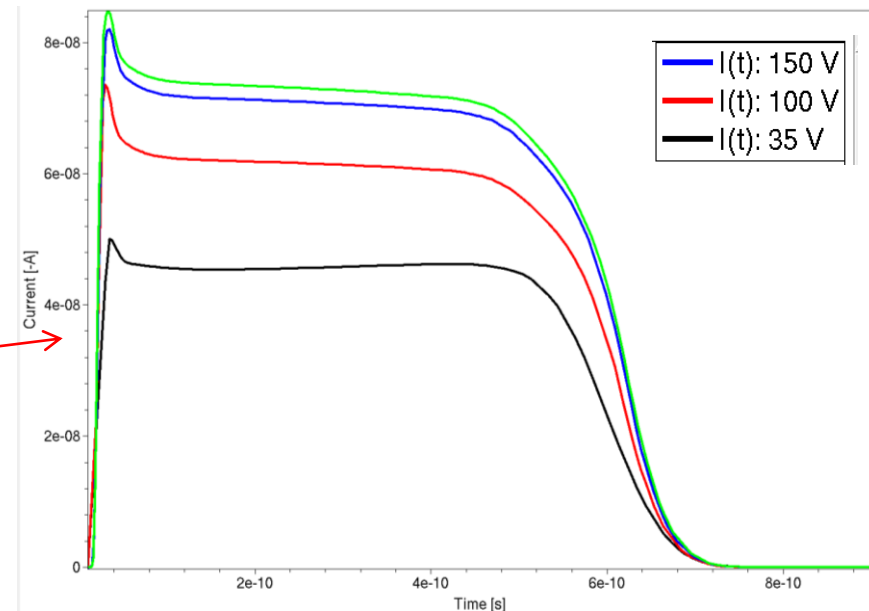
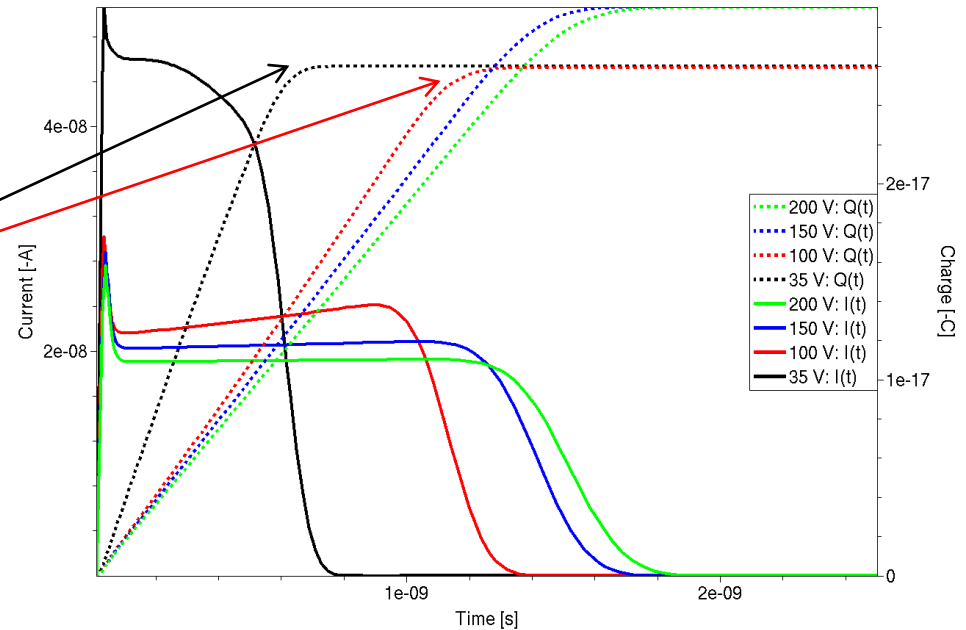
❑ Measured collection time (t_{coll}) rate $t(100 \text{ V})/t(35 \text{ V}) \approx 2$ is reproduced

❑ t_{coll} starts to saturate at high V as in measurement

❑ Absolute t_{coll} & transient current shapes are not reproduced → need to insert deep level traps to the simulation



❑ **Preliminary:** Deep level traps (deep donors from slide 10) are introduced to the GaAs bulk



Conclusions

- ❑ GaAs X-ray detectors for mammographic applications have successfully been processed at Micronova nano-fabrication center
- ❑ GaAs pixel detectors have been flip-chip assembled to Medipix readout

- ❑ Thickness of semi-insulating GaAs layer is $\sim 110 \mu\text{m}$, grown by CVPE
- ❑ $V_{\text{fd}} \approx 25 \text{ V}$ and $J_{\text{leak}} \approx 10 \text{ nA/cm}^2$
- ❑ Electron transient time measured by TCT is $\sim 7\text{-}8 \text{ ns}$, i.e. less than radiative recombination lifetime
- ❑ DLTS results reveal **several defects or lattice distortions**:
 - $N_t(\text{deep e traps}) \sim 5 \times 10^{11} \text{ cm}^{-3}$, $N_t(\text{shallow h traps}) \sim 5 \times 10^{10} \text{ cm}^{-3}$
 - Further investigations needed to determine the nature of defects (related to impurities or lattice deformations?)
- ❑ TCAD simulations agree with experimental data
 - Further TCT tuning by inserting traps

- ❑ **More information:** X. Wu et al., *Radiation Detectors Fabricated on High-purity GaAs Epitaxial Materials*, iWoRID 2014, June 2014, Trieste, Italy