



# Development of *kilometer-grade* $\text{MgB}_2$ superconducting wires at NIN

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**Xifeng Pan**, Guo Yan, Qingyang Wang, Fang Yang, Yong Feng, Pingxiang Zhang

Northwest Institute for Non-Ferrous Metal Research (NIN)  
Western Superconducting Technologies Co., Ltd. (WST)

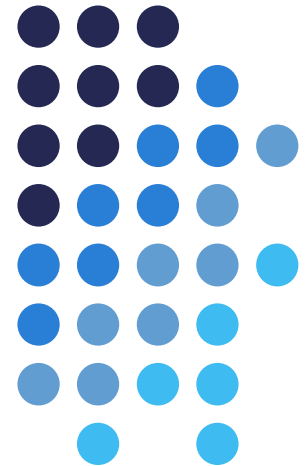
2015, 07/01

# Outline

- Introduction
- Fabrication of *km*-grade  $\text{MgB}_2$  wires by *in-situ* PIT method
  - ◆ Preparation of precursor powders
  - ◆ Conductor structure design
  - ◆ Fabrication of *km*-grade  $\text{MgB}_2$  wires and their superconducting properties
- Conclusions



# Introduction



# Advantages of $\text{MgB}_2$ superconductor

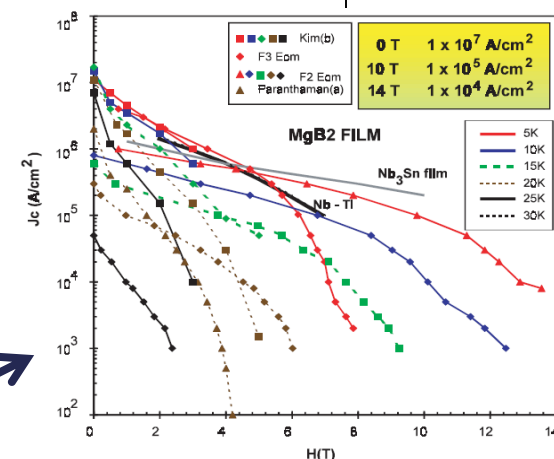
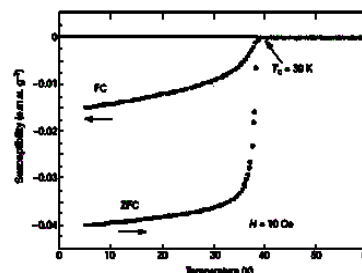
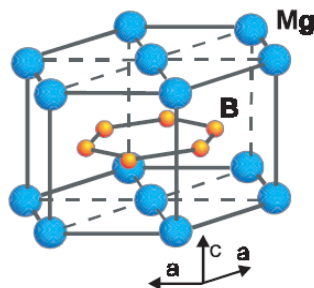
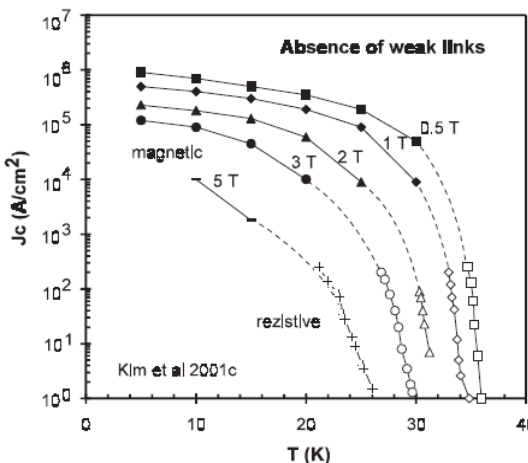


Absence of weak links

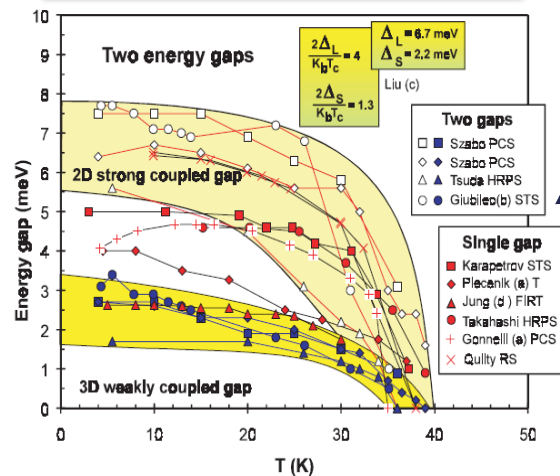
Simple crystal structure

Moderately high  $T_c$

High current densities



Two energy gaps

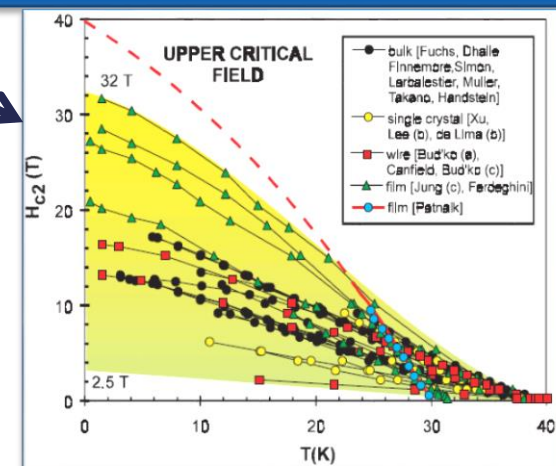


**$\text{MgB}_2$**

Low cost – low weight  
precursors price: 150 €/Kg

Density: 2.5 kg/dm³

Potentially high critical field

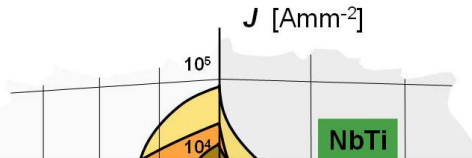


# Disadvantages of $\text{MgB}_2$ superconductor



$J_c$  decrease with increasing field quickly

Fragile  $\rightarrow$  Difficult to making wires and tapes directly



Flux pinning Enhancement:

- ✓ Irradiation
- ✓ Doping; like C or Ti

Sheath & conductor structure design

High conductivity  
Matrix (e.g. Cu)

Superconducting  
Filaments

Porosity  
structure

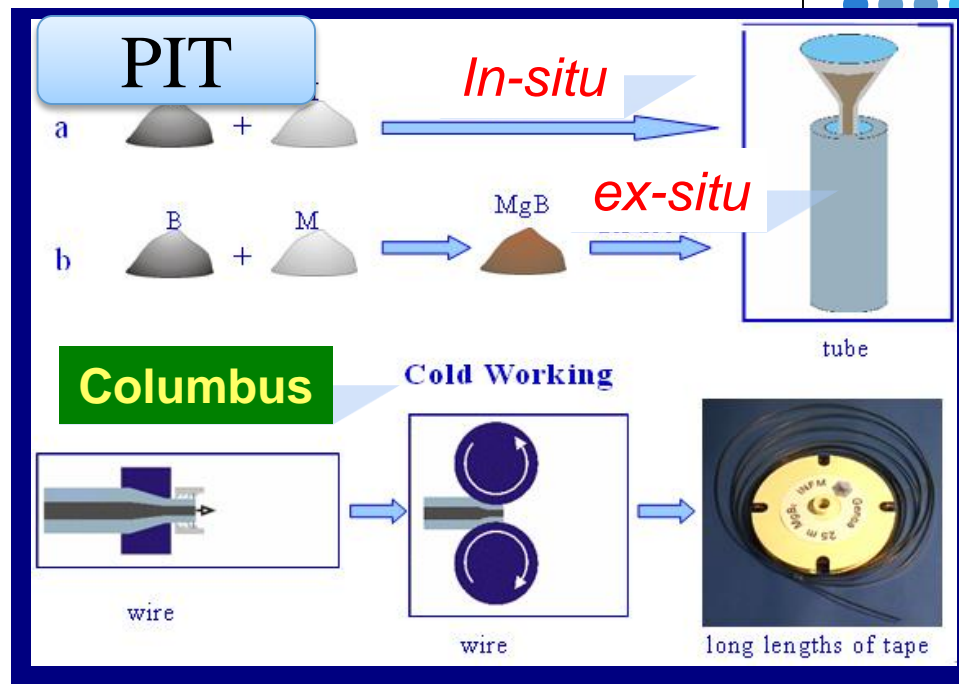
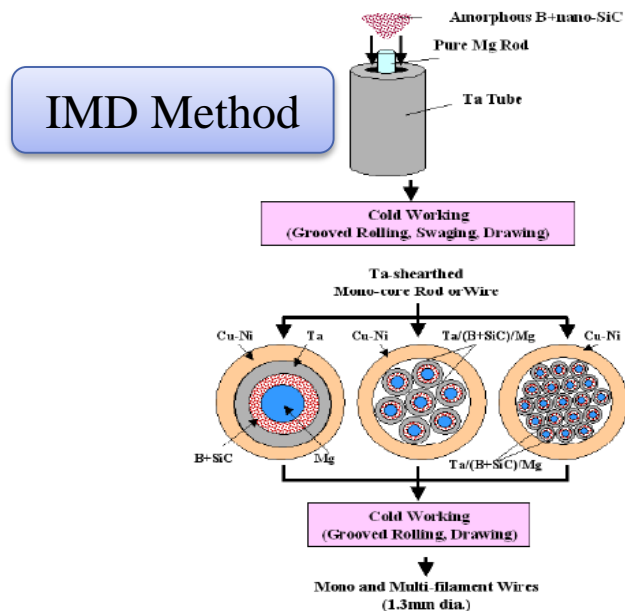
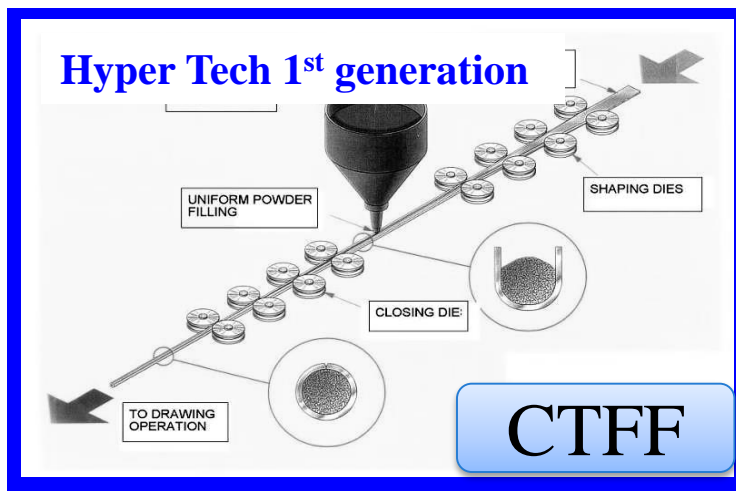
Densification:

- ✓ Hot-press heat-treatment
- ✓ Magnetic Processing
- ✓ High-energy ball milling

Homogeneous  
deformation for  
the powders is  
not easy.



# Fabrication Technique for $\text{MgB}_2$ wires and tapes



## Advantages of *in-situ* PIT:

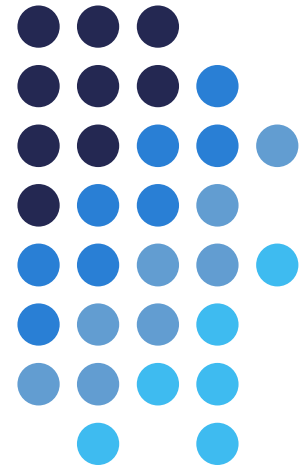
- Simple fabrication process
- Easy to introduce doping

## Disadvantages of *in-situ* PIT:

- Reaction between precursor and sheath

# *In-situ* PIT method for *km*-grade $\text{MgB}_2$ wires

- ✓ Preparation of Precursor powders
- ✓ Conductor structure design
- ✓ Fabrication of *km*-grade  $\text{MgB}_2$  wires and their superconducting properties

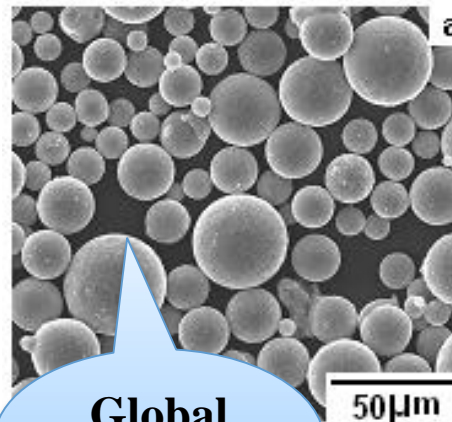
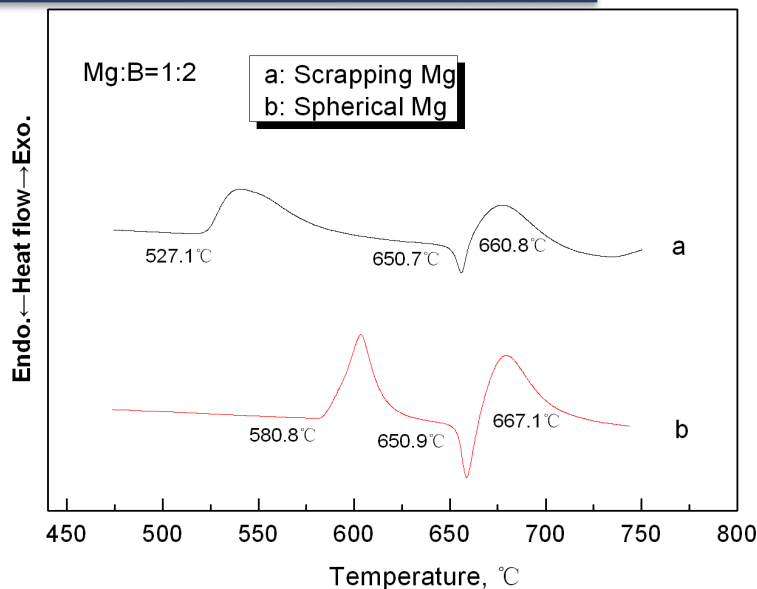




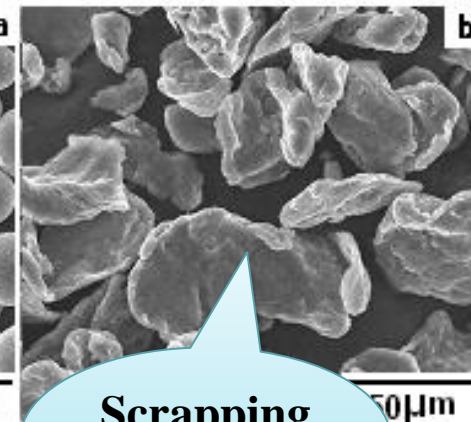
# Preparation of Precursor powders



## Shape of Mg powders



Global  
Mg

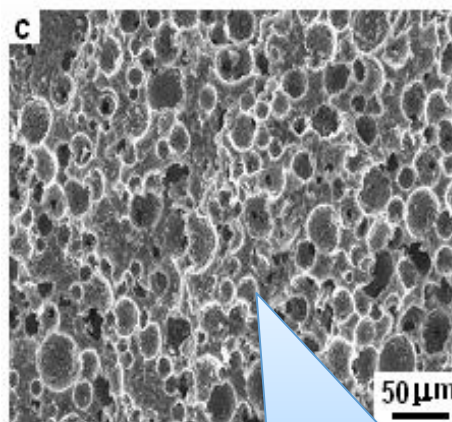


Scrapping  
Mg

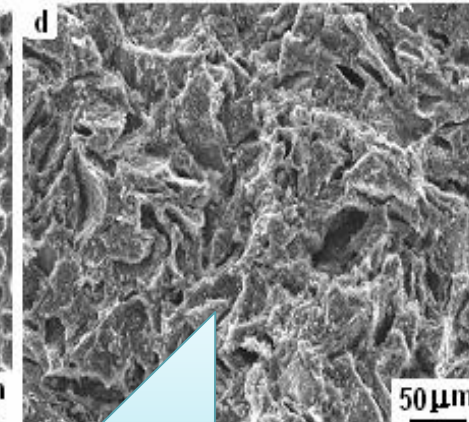
### ● Scrapping Mg:

- ✓ Higher specific surface area;
- ✓ Larger surface energy;
- ✓ Lower solid-solid starting reaction temperature.

● The shape and size of holes depend on the shape and size of raw Mg powder.



(Global Mg+ 2B)  
After sintering



(Scrapping Mg+ 2B)  
After sintering

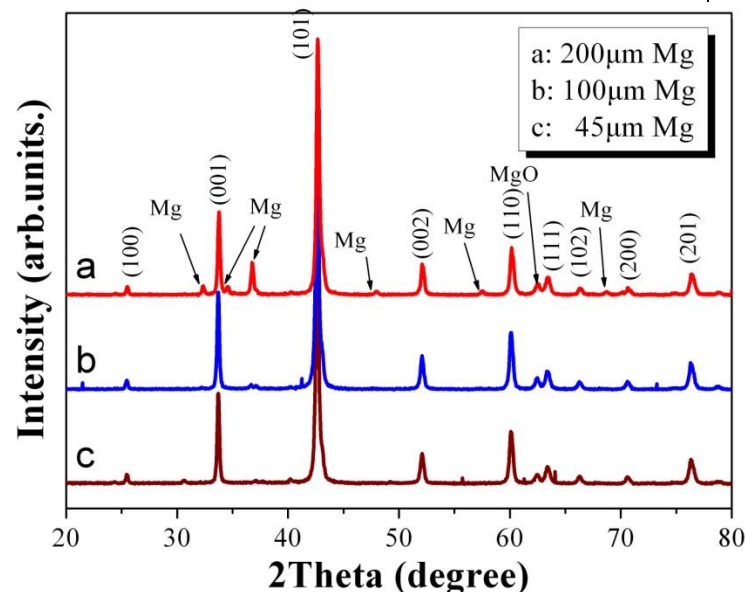


# Preparation of Precursor powders

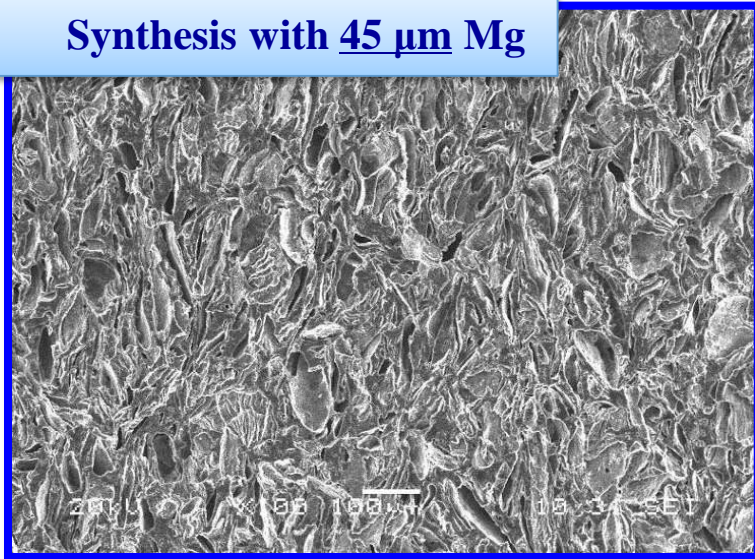


## Size of Mg powders

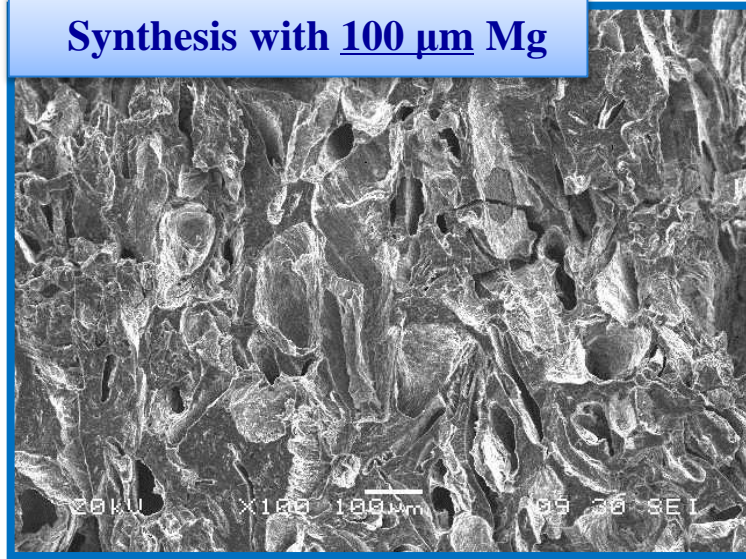
Complete Reaction, higher density and higher phase purity can be acquired using scrapping Mg powder with the size of 45  $\mu\text{m}$  as the precursor powder



## Synthesis with 45 $\mu\text{m}$ Mg



## Synthesis with 100 $\mu\text{m}$ Mg



# Preparation of Precursor powders

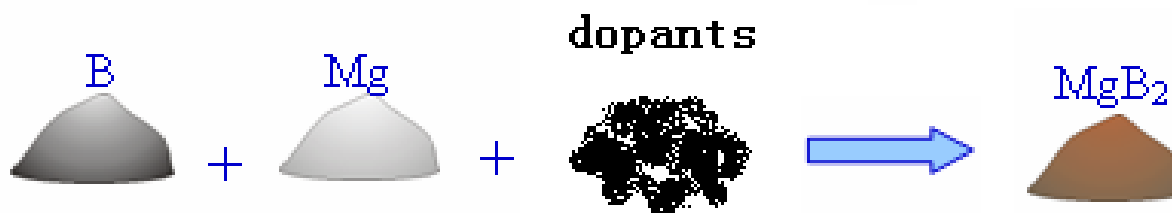


## Two-step reaction method

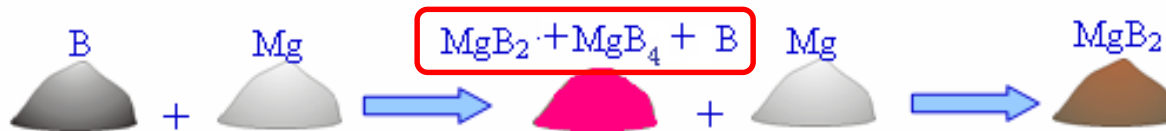
Single-step reaction:  
Mass hole formed!



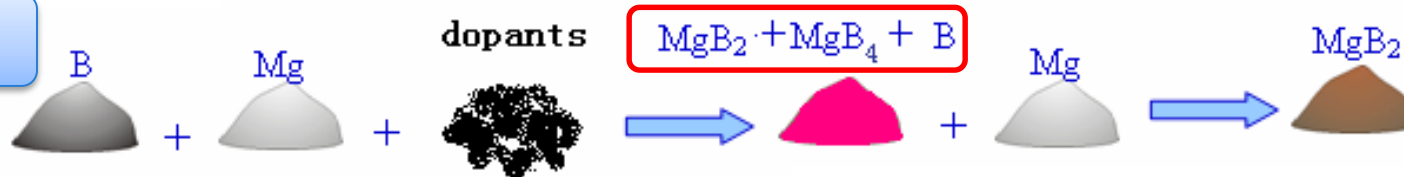
$Mg:B=1:2$



Two-step reaction:  
Improved the density  
and homogeneity



$Mg:B=1:4$



S.C. Yan *et al.* *Rare metal mat. Eng.* 36(2007)1260-1262

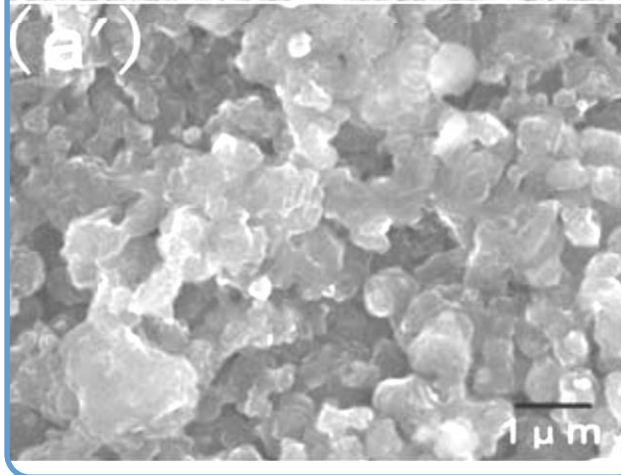
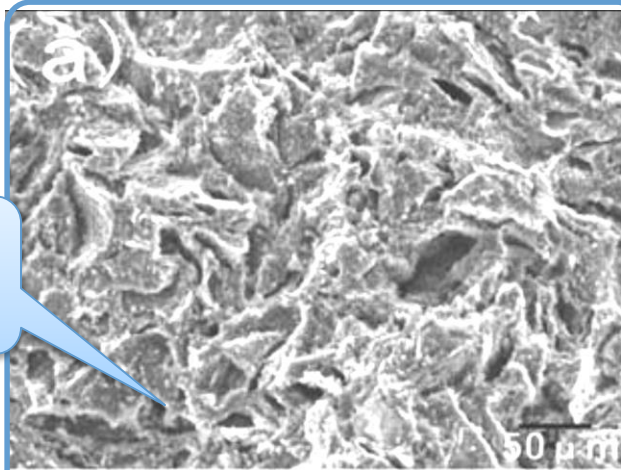
The hole in  $MgB_2$  is depending on the size of magnesium. The two-step method could decrease the initial Mg ratio and thus improve the density.

# Preparation of Precursor powders

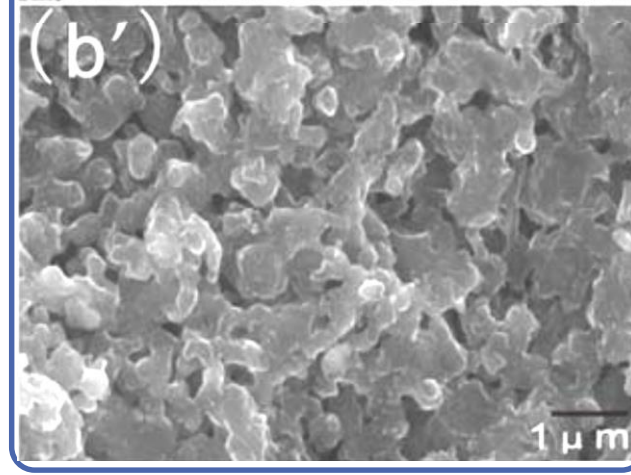
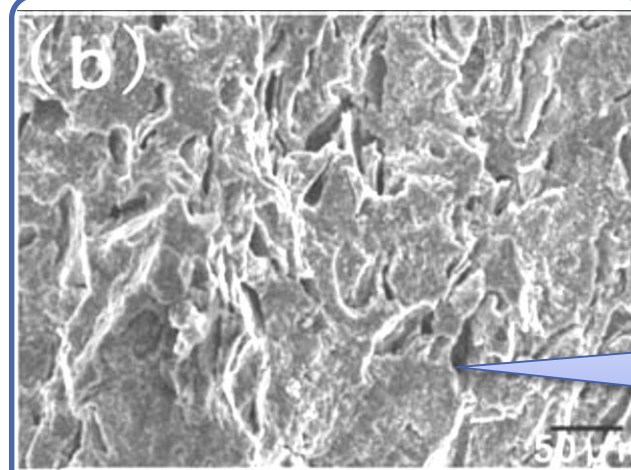


## Two-step reaction method

One-step  
reaction



Two-step  
reaction



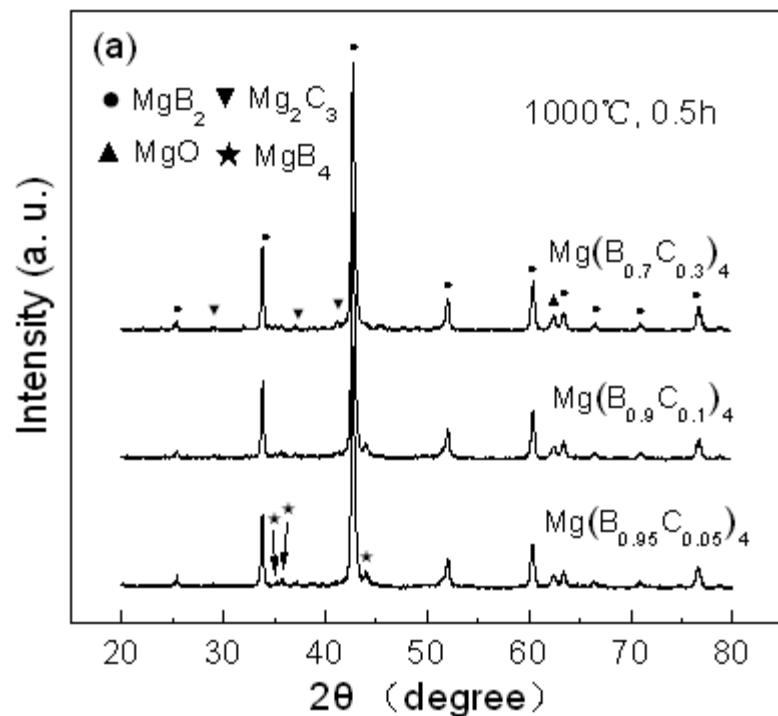
- ✓ Porosity decrease obviously, the density increased from 1.5 to 1.9 g/cm<sup>3</sup>
- ✓ The Grains are still with very small size.

# Preparation of Precursor powders

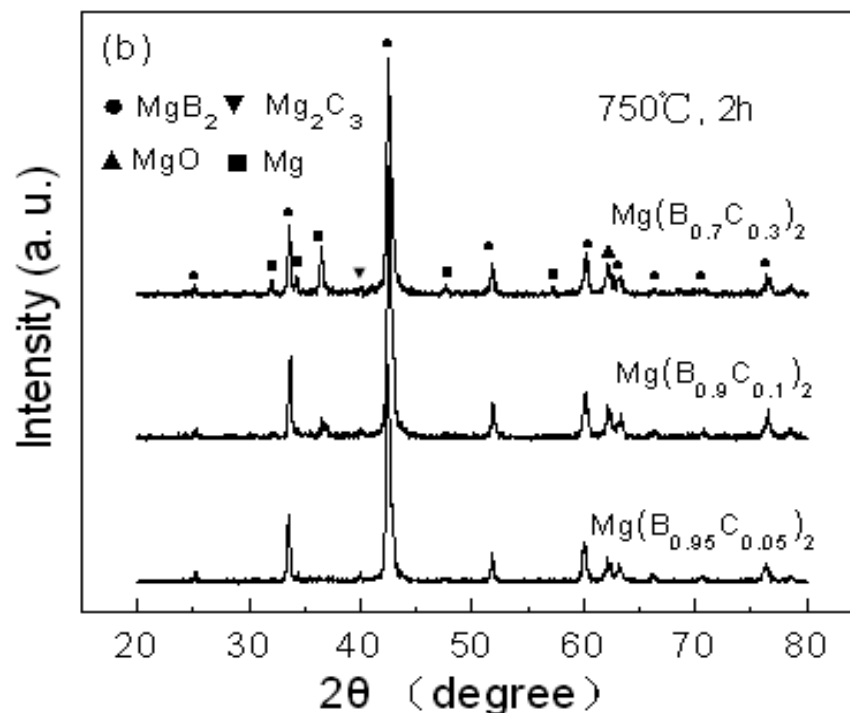


## Two-step reaction method

### First step sintering with C doping



### Second step sintering with C doping



S.C. Yan et al. *J. Alloys Compd.* 459(2008)452-456

Two-step reaction method is beneficial for carbon substitution at B sites.

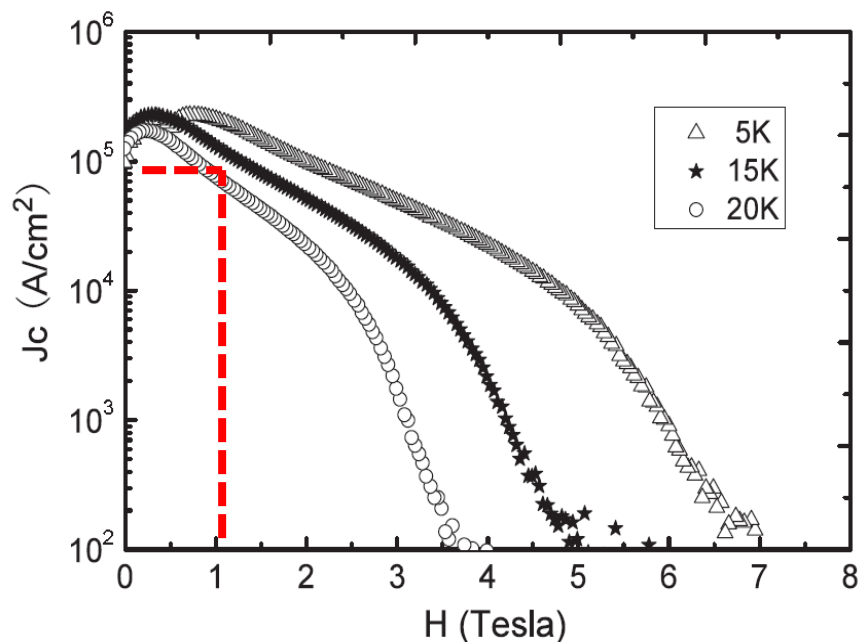


# Preparation of Precursor powders

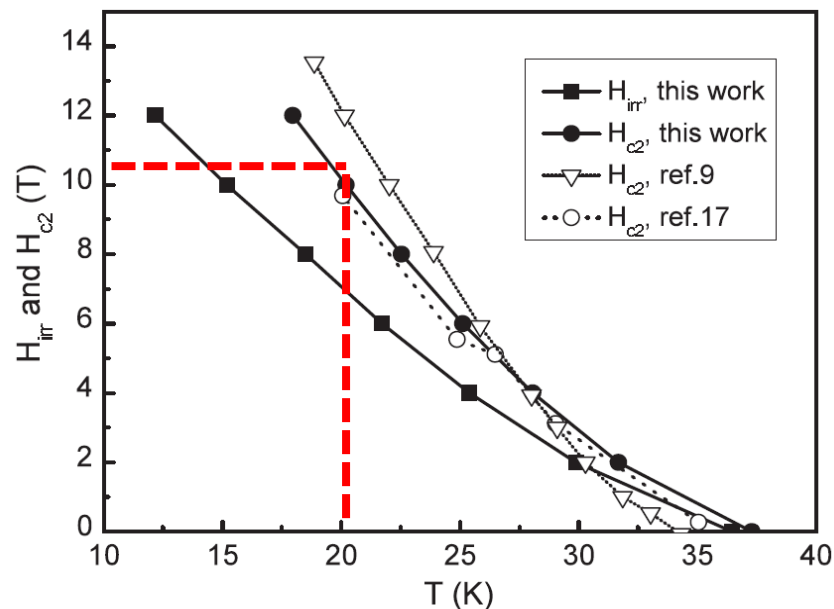


## Two-step reaction method

$J_c$  of  $\text{Mg}(\text{B}_{0.95}\text{C}_{0.05})_2$



$H_c$  and  $H_{ir}$  of  $\text{Mg}(\text{B}_{0.95}\text{C}_{0.05})_2$



S.C. Yan et al. *Supercond. Sci. Technol* 20(2007)377-380

For the sub-micron carbon-doped  $\text{MgB}_2$  by a two-step reaction:

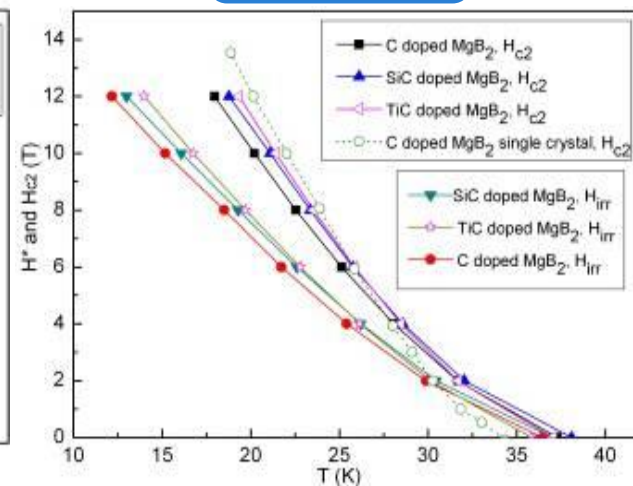
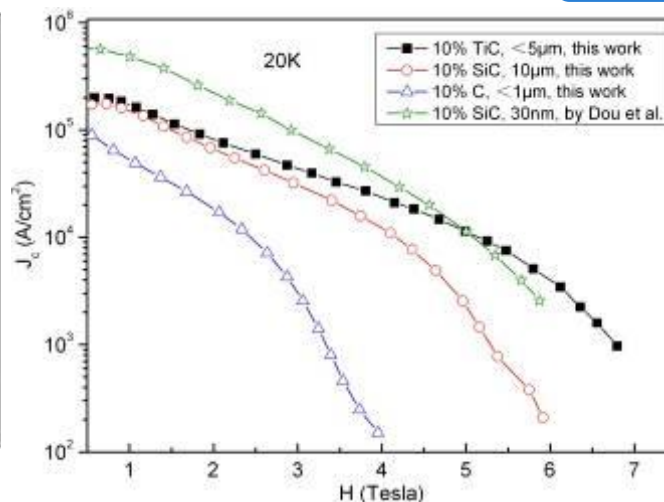
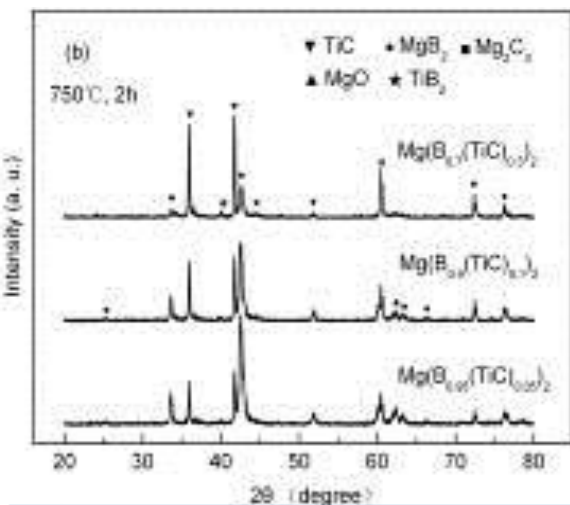
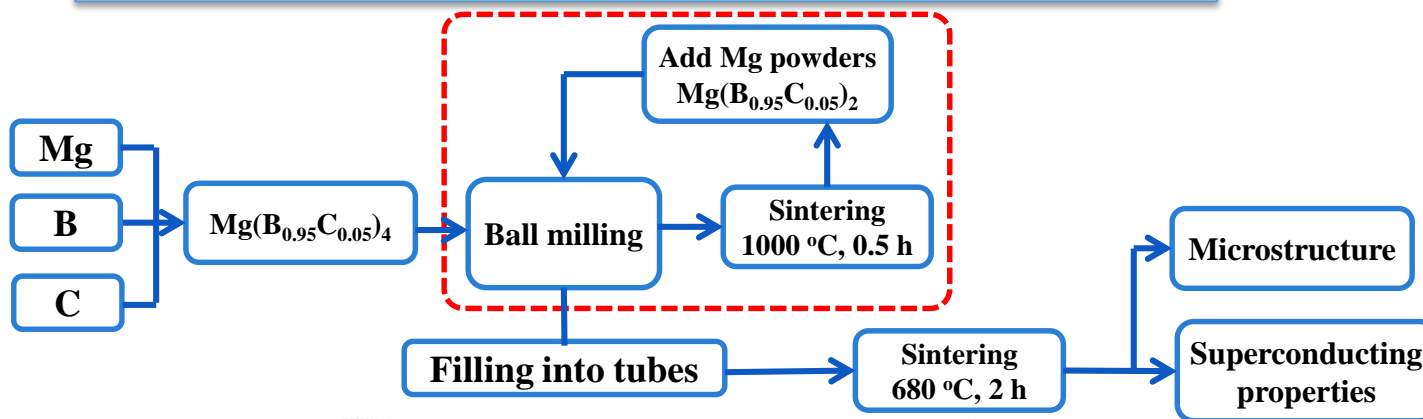
- ✓ The upper critical field  $H_{c2}$  reached 10.5 T at 20 K.
- ✓ The critical current density  $J_c$  is  $7.8 \times 10^4$  A cm<sup>-2</sup> at 20 K and 1 T.

# Preparation of Precursor powders



## Two-step reaction method

### Synthesis flow chart of the two-step reaction method



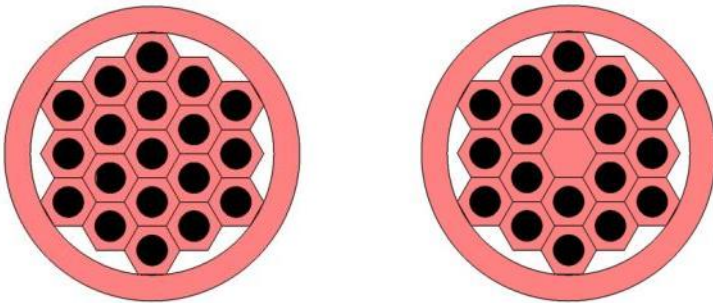
The two-step reaction method is beneficial to the preparation of chemical doped  $\text{MgB}_2$  superconducting materials with high density.

# Conductor structure design



## Conductor structure design

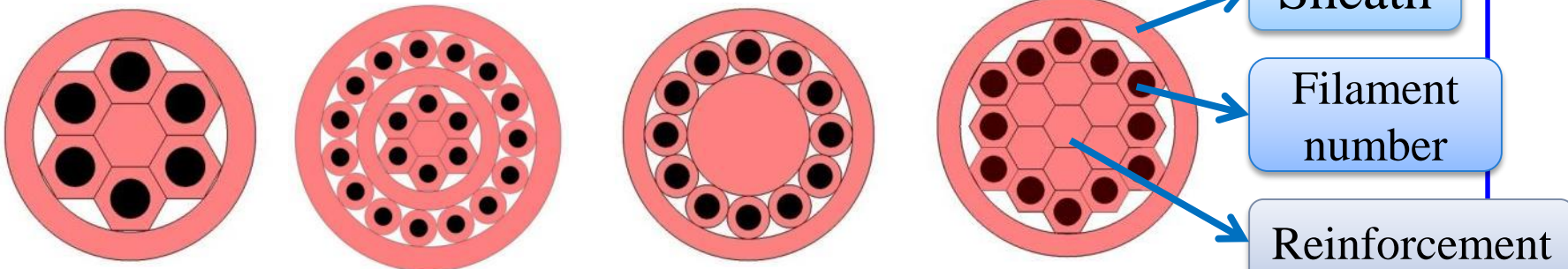
### Low Cu ratio structure



### Principle:

- Stabilization
- Workability
- Critical current density
- Satisfaction for various application required

### High Cu ratio structure



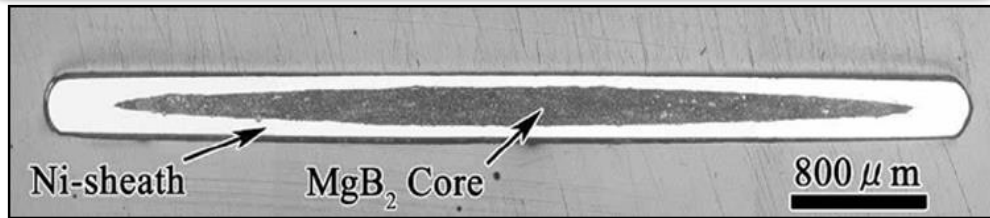


# Conductor structure design

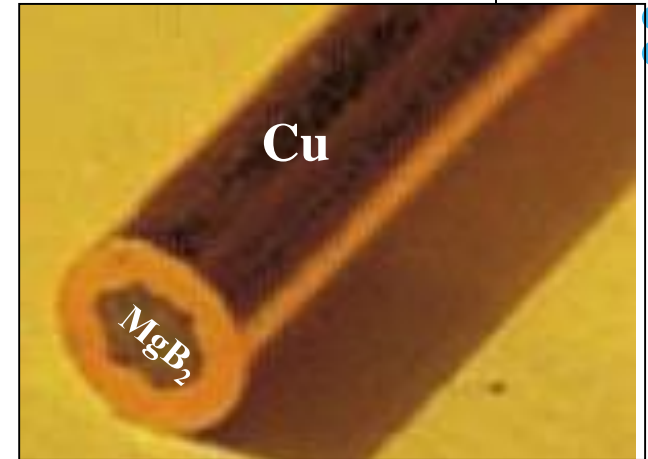
## Sheath materials

Sheath: Ni, Fe, Cu, Ti, Al, SS...

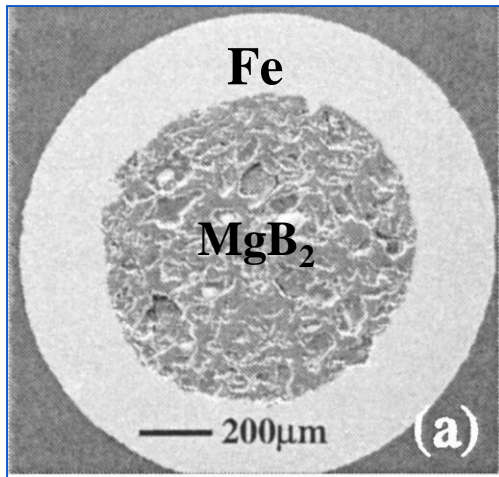
Barrier: Nb, Fe



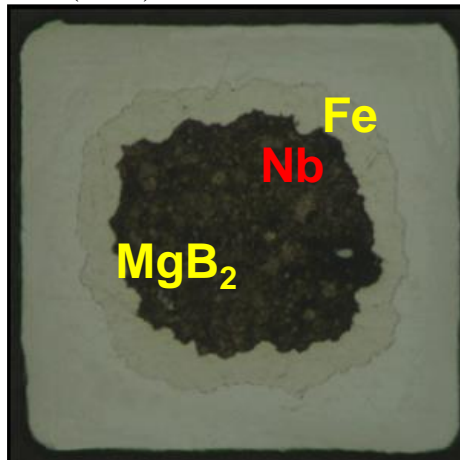
Murase *et al.* *IEEE Trans. Appl. Supercond.*, 16, 1403(2006)



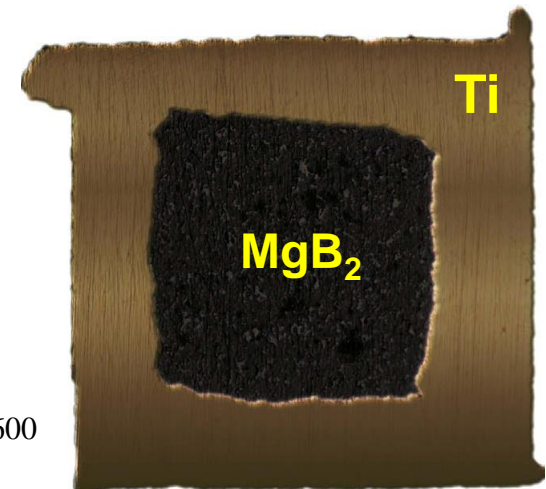
Eisterer *et al.*, *Supercond. Sci. Technol.* 15 (2002) 1088–1091



Pan *et al.*, *J. Appl. Phys.* 96(2004)1150



P Kovac *et al.*, *Supercond. Sci. Technol.* 19 (2006) 600



In our study:  $\text{MgB}_2/\text{Nb}/\text{Cu}$

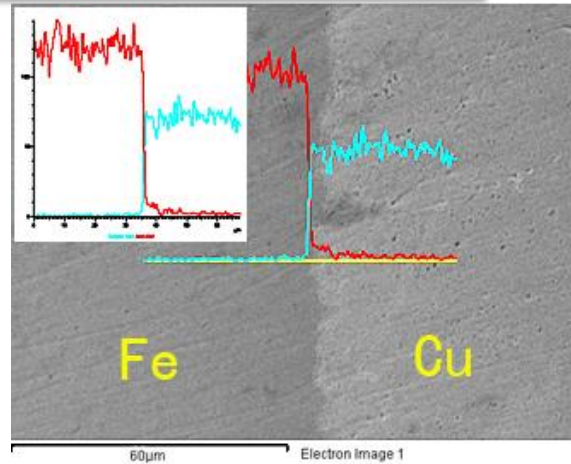
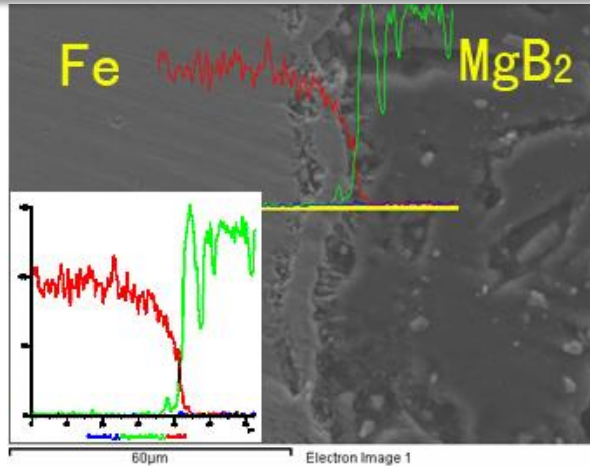
M. Kulich *et al.* *Physica C* 469 (2009) 827–831

# Conductor structure design

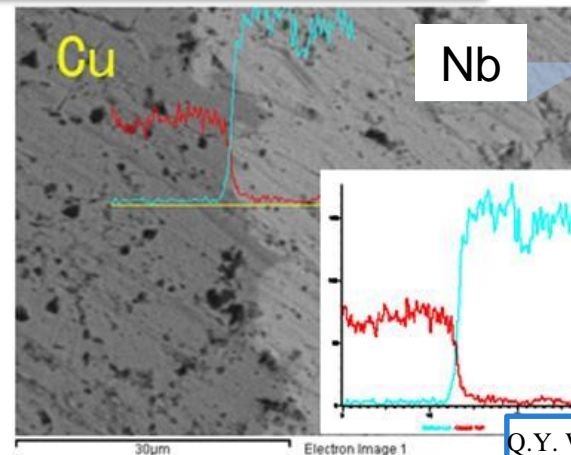
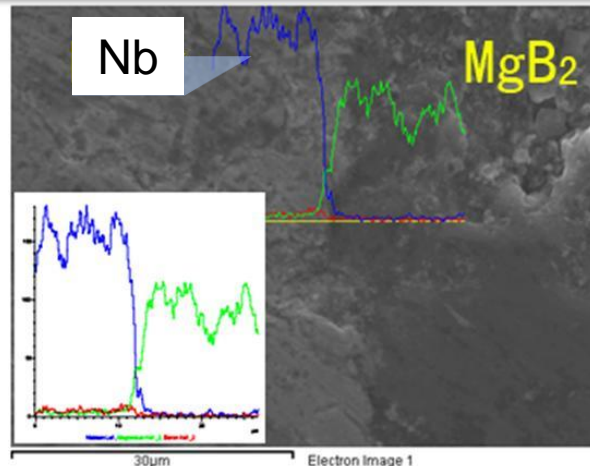


## Choice of sheath/barrier materials

### Interface of $\text{MgB}_2$ /**Fe**/Cu after sintering



### Interface of $\text{MgB}_2$ /**Nb**/Cu after sintering

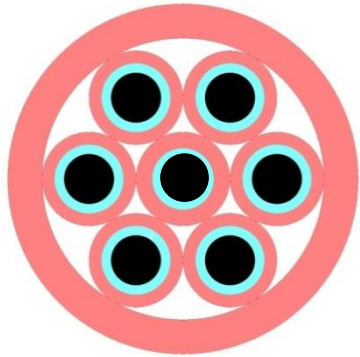


- Diffusion layer appeared between sheath and SC core and obstructed the current transport
- Thinner diffusion layer was obtained, with **Nb** sheath.

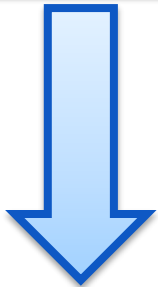
# Conductor structure design



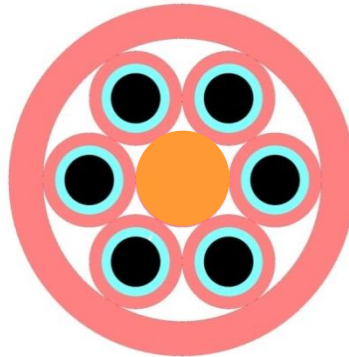
## Central reinforcement Materials



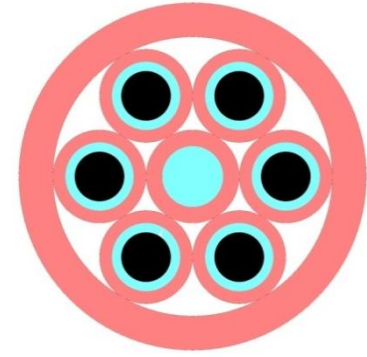
7-MgB<sub>2</sub> filaments



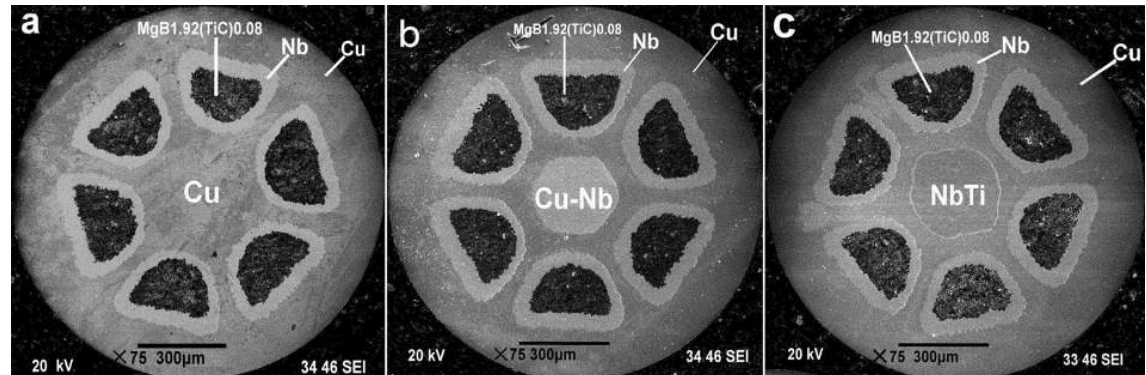
Central filament  
easily be broken



6-MgB<sub>2</sub> filaments  
+ **Cu** reinforcement



6-MgB<sub>2</sub> filaments  
+ **Nb** or **NbTi**  
reinforcement



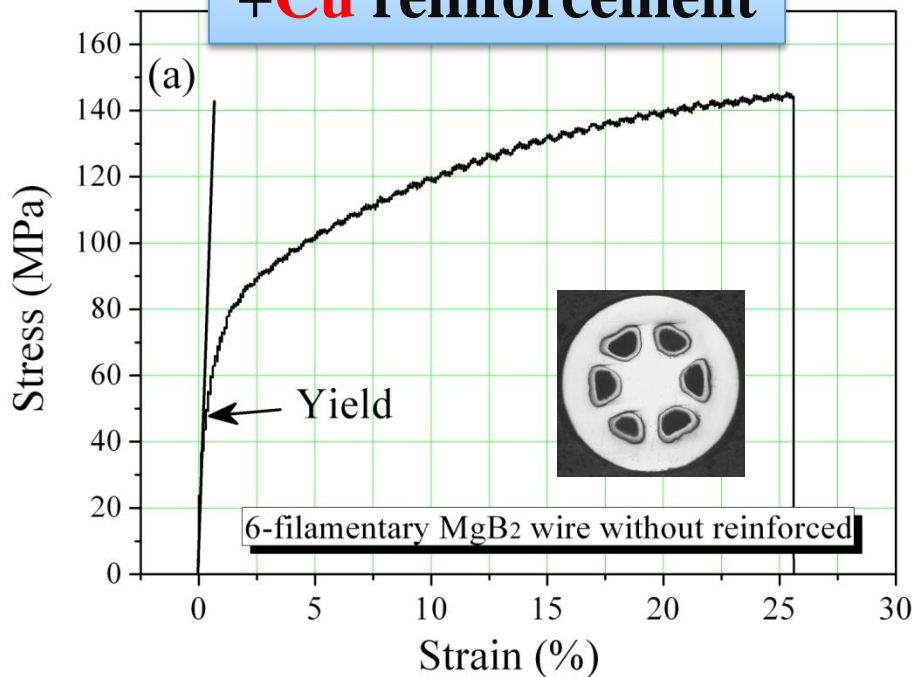


# Conductor structure design

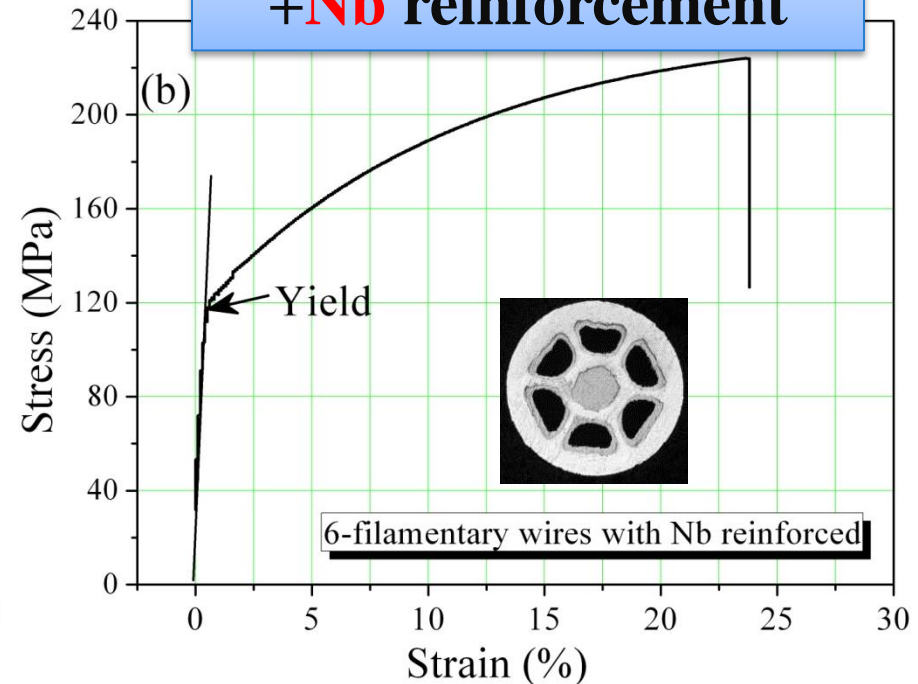


## Mechanical Properties

### 6-MgB<sub>2</sub> filaments + **Cu** reinforcement



### 6-MgB<sub>2</sub> filaments + **Nb** reinforcement



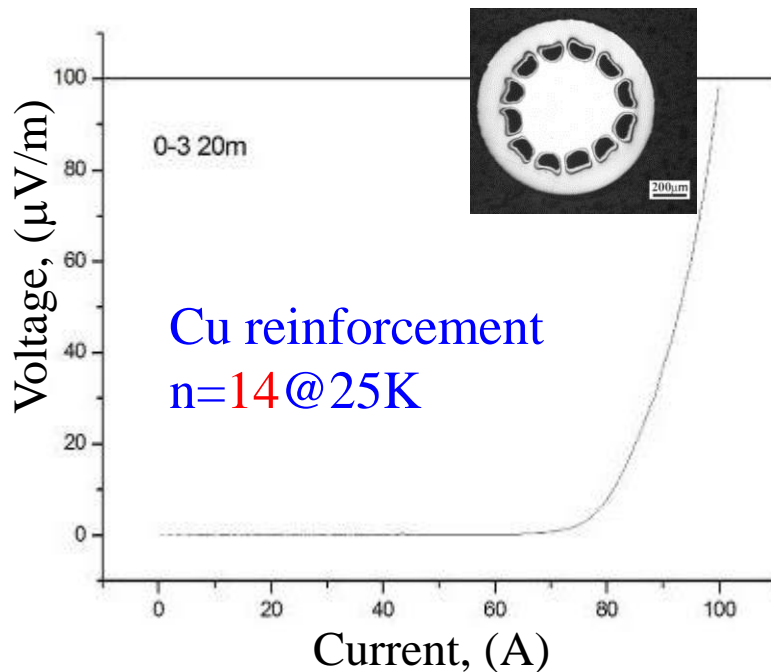
- Nb reinforced wire exhibits higher mechanical properties. The yield strength of the Nb reinforced wire is about **120 MPa** and only **50 MPa** for the wires with Cu reinforced.
- The Nb reinforcement could remarkably enhance the mechanical property of the 6 filaments MgB<sub>2</sub> wires.

# Conductor structure design

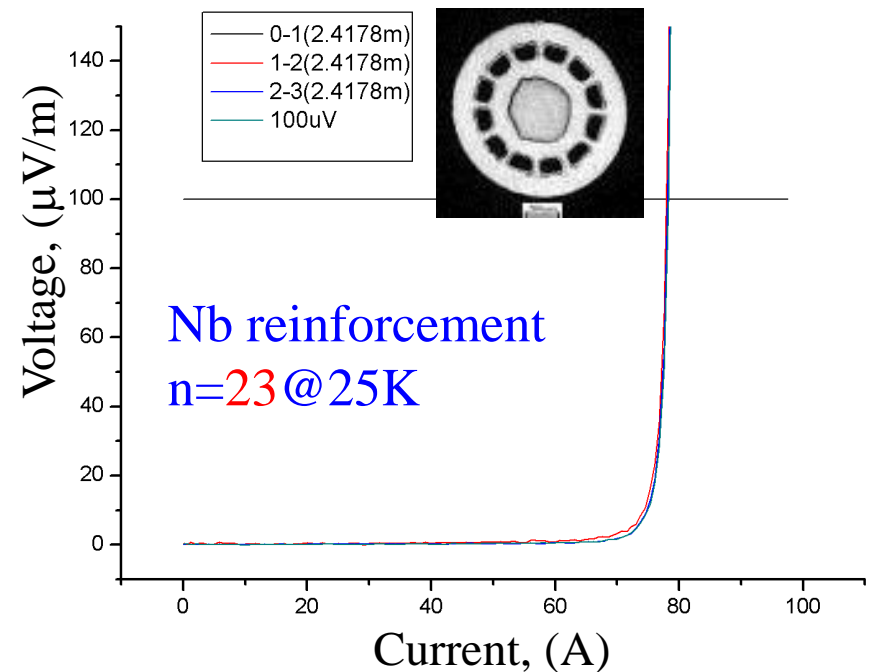


## Thermal stability

12-core  $\text{MgB}_2$   
+ **Cu** reinforcement



12-core  $\text{MgB}_2$   
+ **Nb** reinforcement

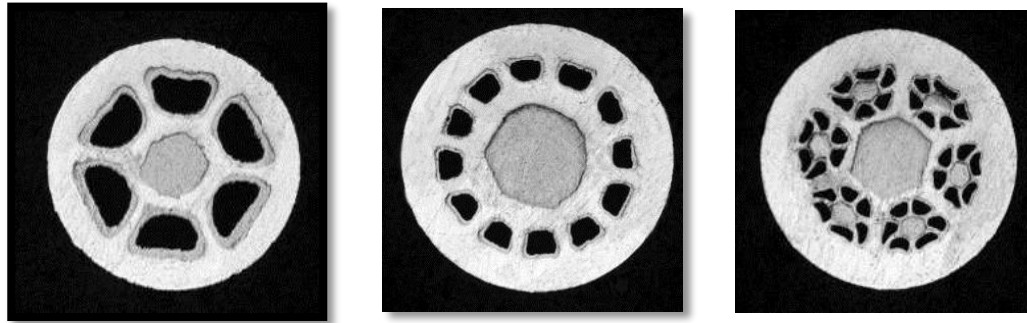


Nb reinforcement is also prone to improve the  $n$  value  
and homogeneity of superconducting wires.

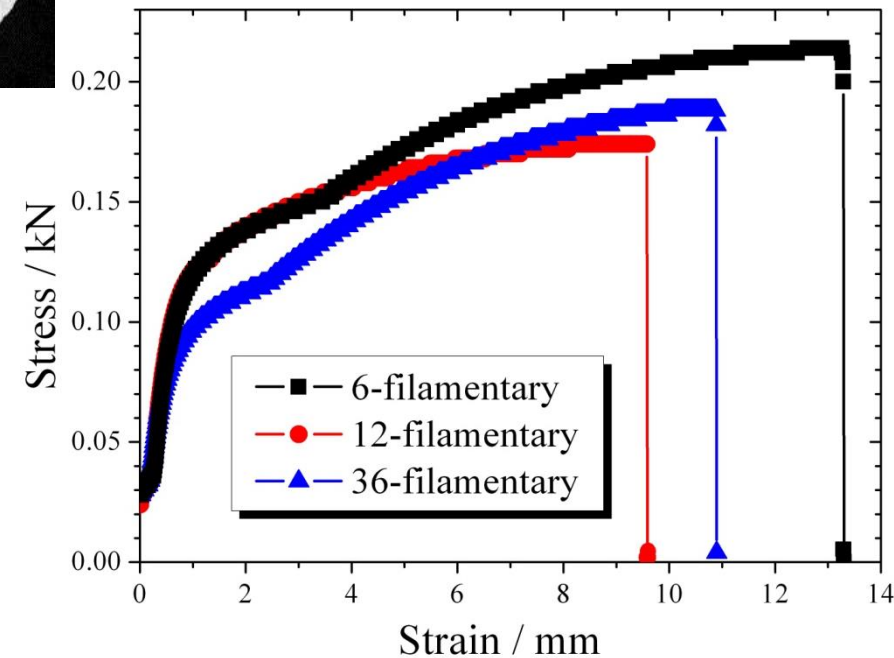
# Conductor structure design



## Different Filament Numbers



Filament	6	12	36
MgB <sub>2</sub>	21.89	14.14	13.95
Nb	23.61	21.88	20.97
Cu	54.50	63.98	65.08

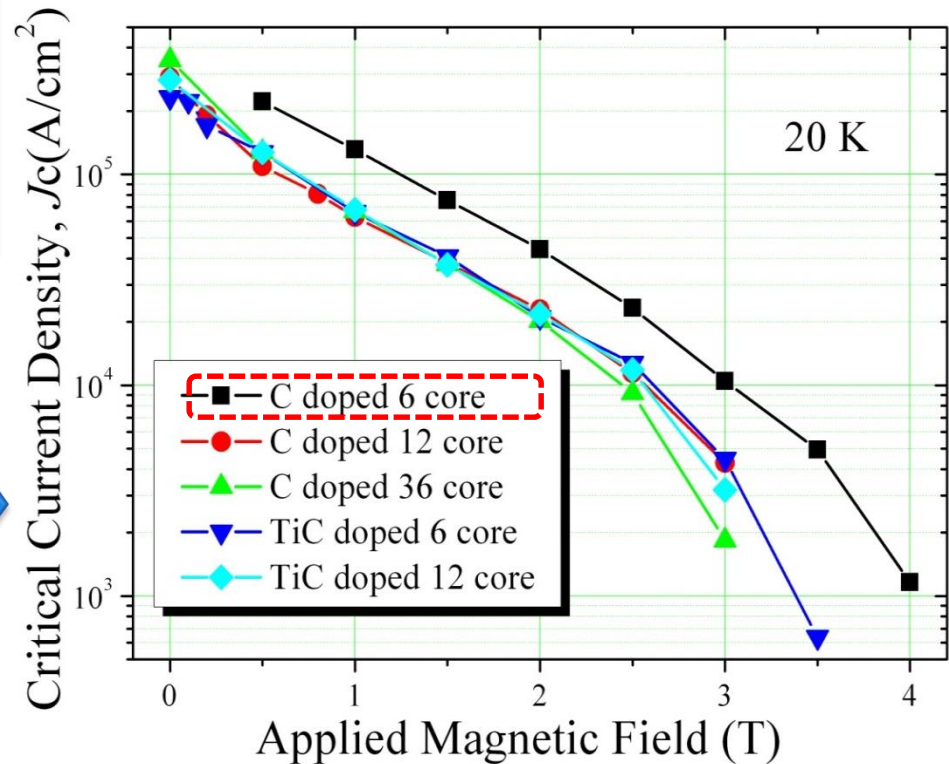
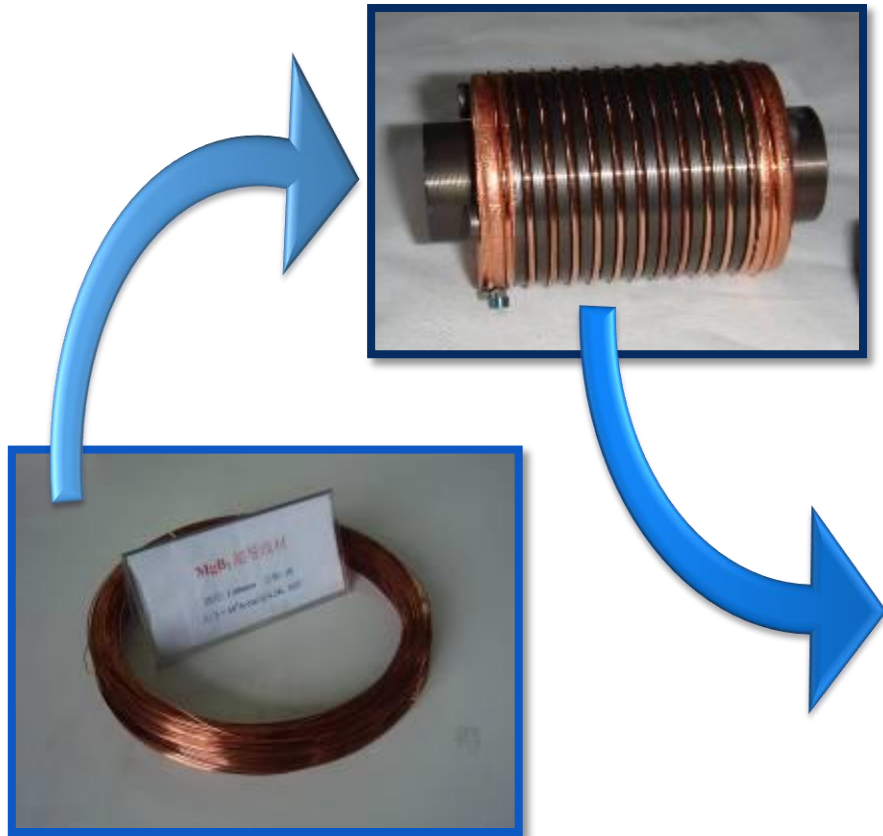


The intensity of 6 filaments wire is higher than that of other wires due to much higher Nb content.

# Conductor structure design



## Transport properties



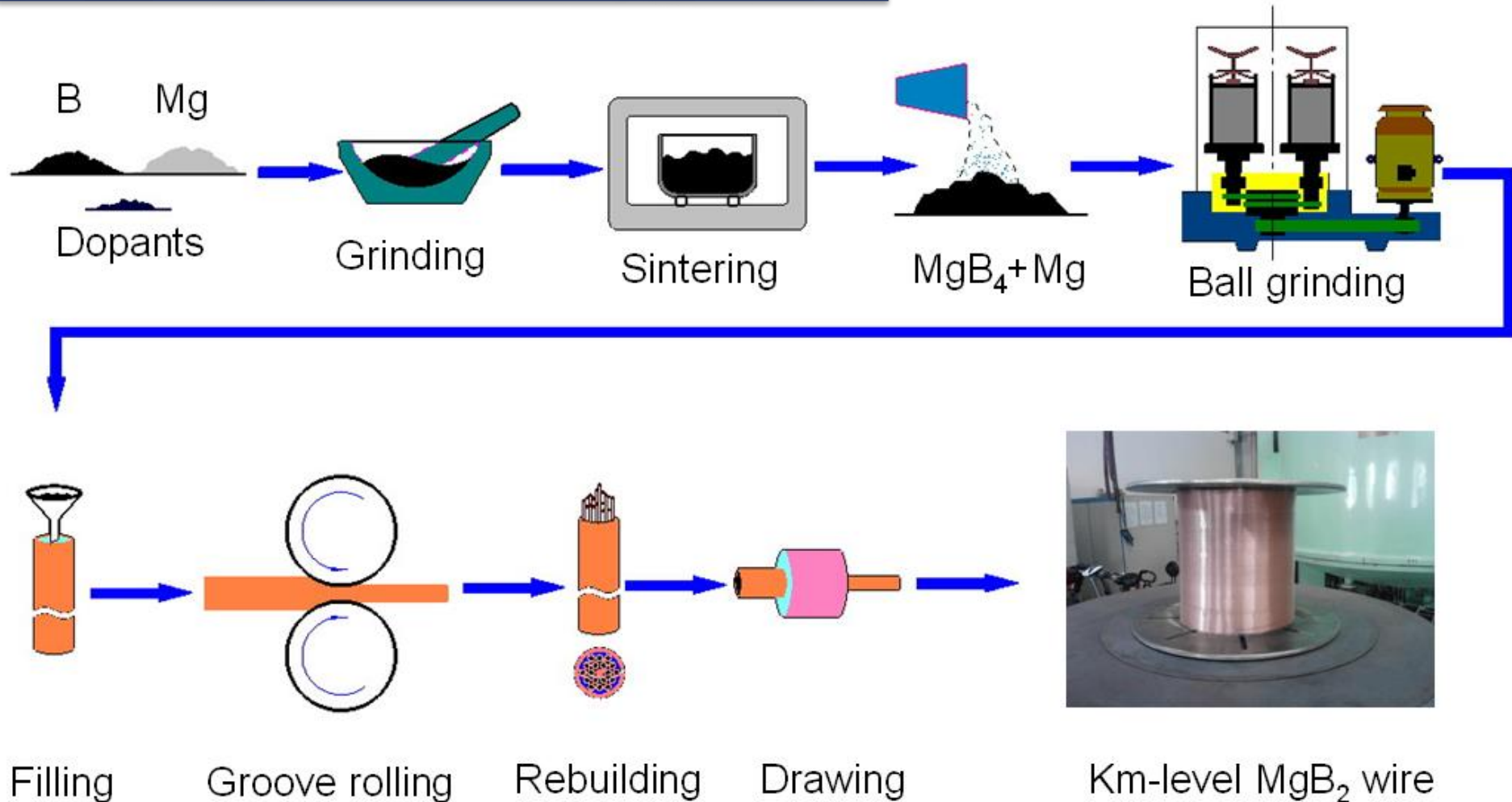
- ✓  $J_c$  is influenced by filament number;
- ✓ The optimal filament number is **6-filament**.



# Fabrication of *km*-grade $\text{MgB}_2$ wires



## In-situ PIT process for $\text{MgB}_2$ long wires



# Fabrication of *km*-grade $\text{MgB}_2$ wires

Production of **1500 m**  $\text{MgB}_2$  wires/tapes

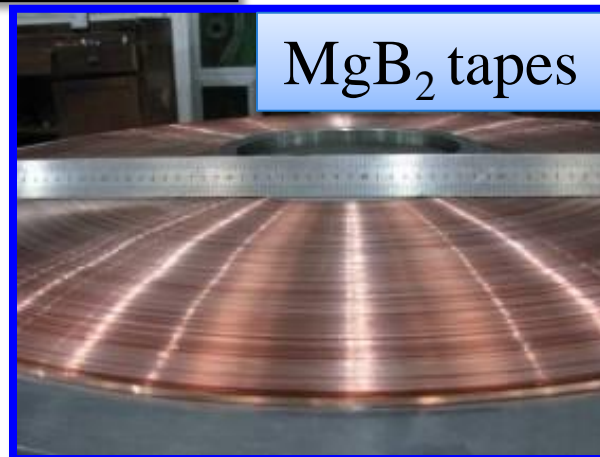
$\text{MgB}_2$  wires



$\text{MgB}_2$  wires



$\text{MgB}_2$  tapes

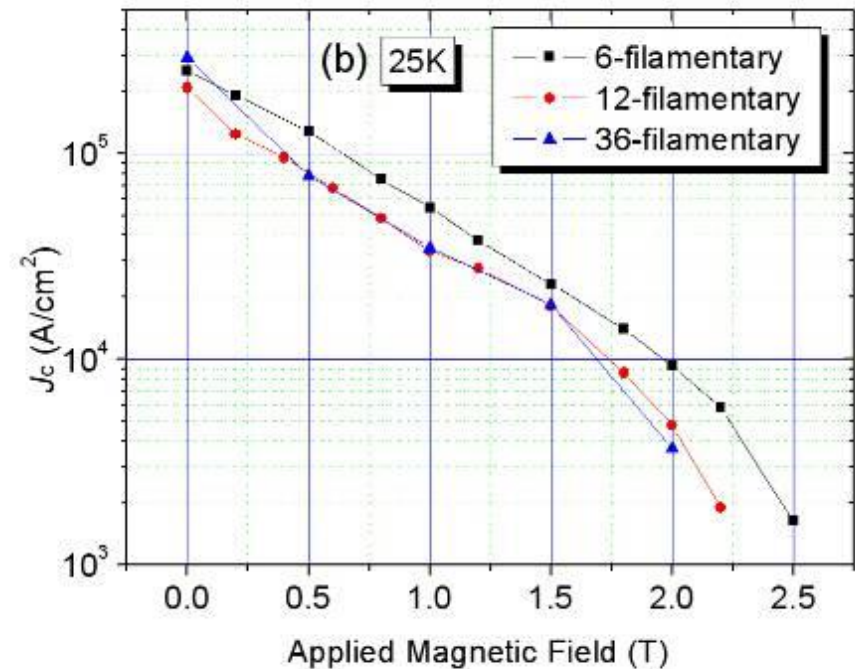
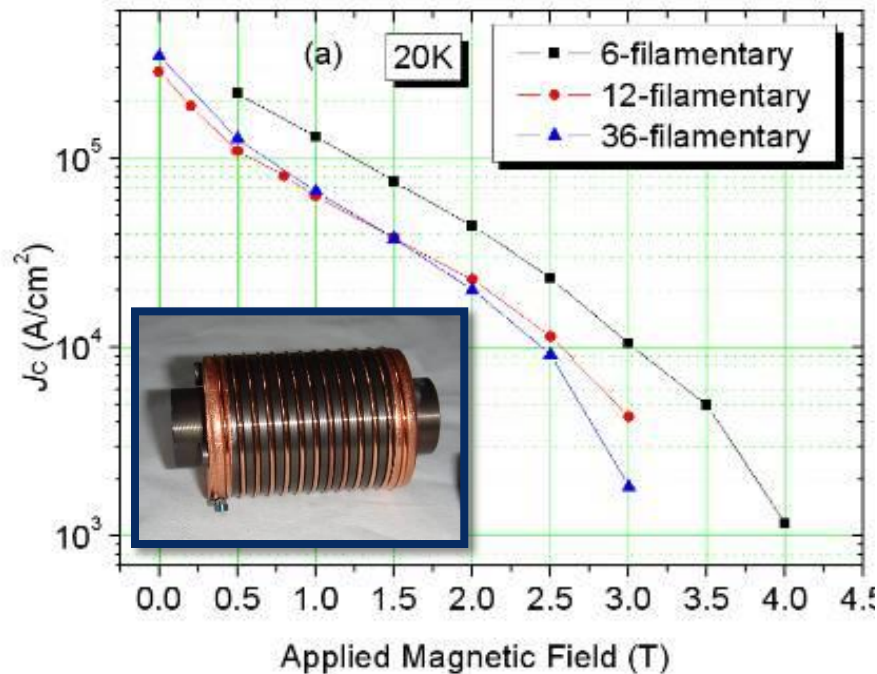


The preparation technology of kilometer  $\text{MgB}_2$  wire is stable, we have prepared **20 kilometers**  $\text{MgB}_2$  superconducting wires.

# Fabrication of *km*-grade MgB<sub>2</sub> wires



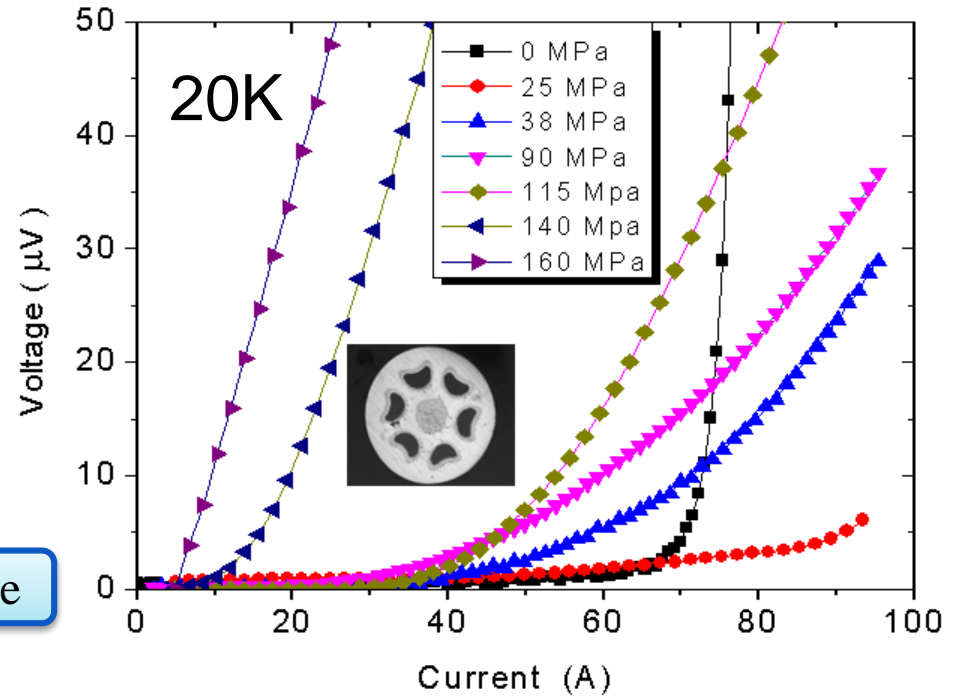
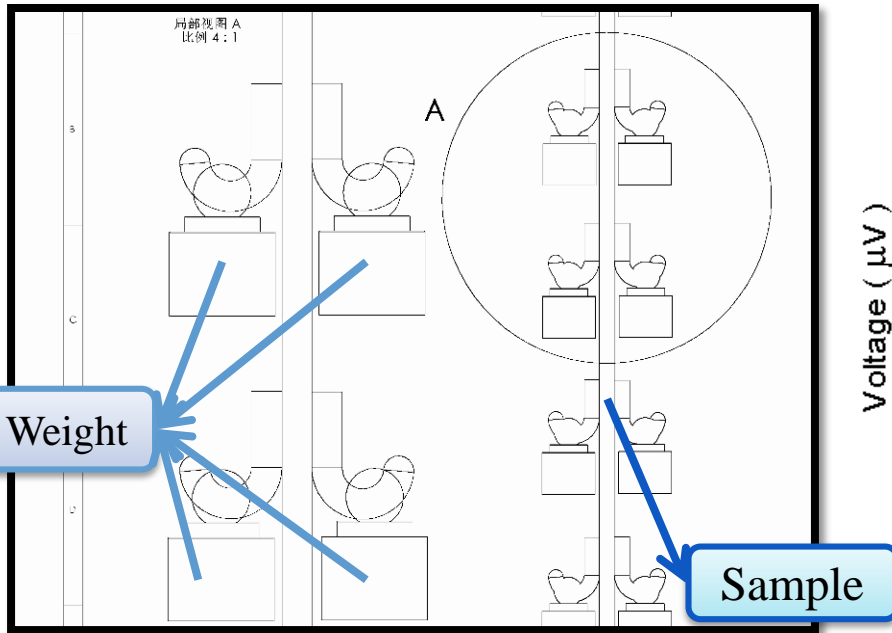
## Superconducting Properties of *km*-grade wires



Now we can produce 1500 meter MgB<sub>2</sub> superconducting wires.  
At 20 K、 2 T,  $J_c = 43,000$  A/cm<sup>2</sup>.

# Fabrication of *km*-grade MgB<sub>2</sub> wires

## Stress dependence of $J_c$ ---Tensile Stress

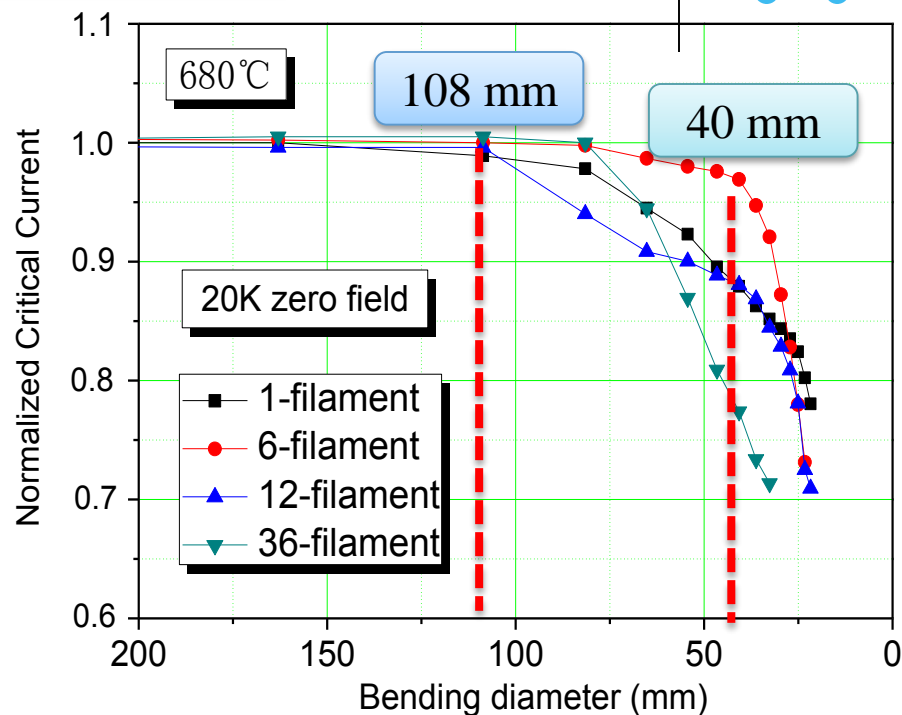
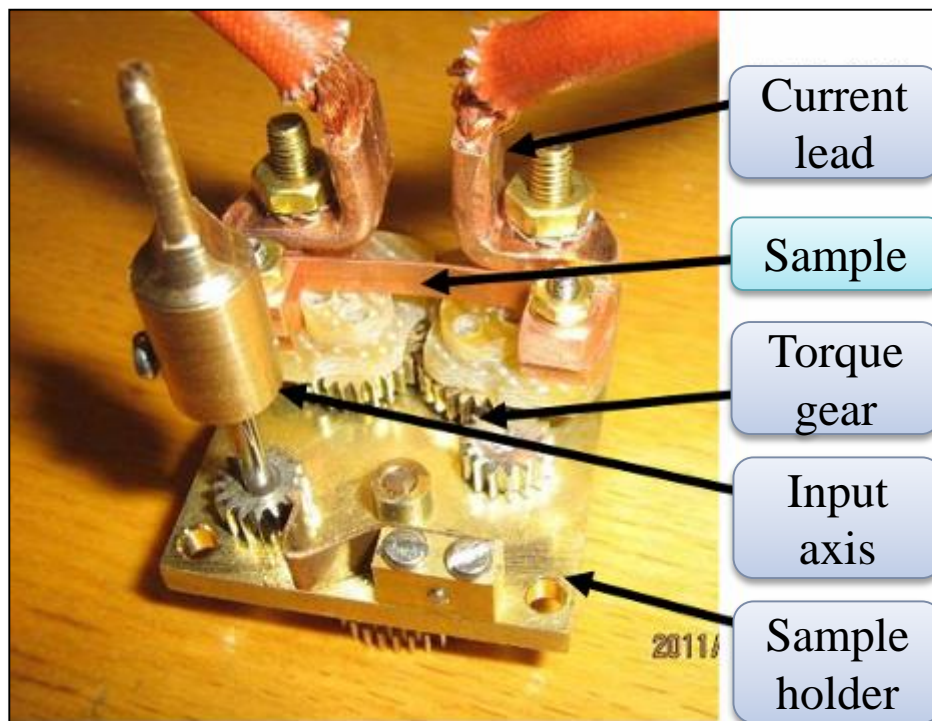


- ✓ In the elastic range, the effect of deformation on  $I_c$  is smaller,  $I_c$  decreased slowly; In the plastic range,  $I_c$  decreases sharply.
- ✓ When the strength of 167 MPa, the superconductivity losses completely.



# Fabrication of $km$ -grade $MgB_2$ wires

## Stress dependence of $J_c$ --Bending Stress

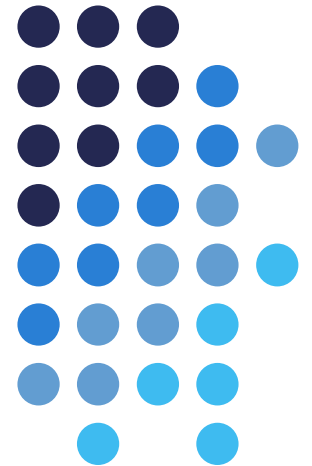


Q.Y. Wang *et al.* *Physica C* 484 (2013) 163–166

- ✓ For all the wires, the critical current density has no loss when the radius is larger than **108 mm**.
- ✓ Bending has smaller effect on the 6-filament wires when bending radius is larger than **40 mm**.



# Conclusions



# Conclusions



- Both the size and shape of Mg affects the density of  $\text{MgB}_2$ , as well as the superconducting properties.
- Nb reinforcement is good to improve the superconducting properties and strength of  $\text{MgB}_2$  wires.
- 20 kilometers  $\text{MgB}_2$  wires have been successfully produced by scale fabrication technique at NIN, which are with a  $J_c$  of  $43000 \text{ A/cm}^2$  ( $\pm 5\%$ ) at 20 K and 2 T.
- For all the wires, the  $J_c$  has no loss when the radius is larger than 108 mm; and 6-filament Nb-reinforcement wires have better bending properties, it has the smallest bending radius, up to 40 mm.





Thank you for your attention