



Joint Laboratory for Characterisation of Defect Centres in Semi-Insulating Materials

Annealing induced evolution of defect centers in epitaxial silicon irradiated with high proton fluences

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Outline

- Samples pad detectors based on epitaxial silicon irradiated with high proton fluence
- Details of HRPITS measurements
- HRPITS images of spectral fringes for radiation defects in standard (ST) and oxygenated (DO) epitaxial layers before and after 1-h isochronal annealing
- Comparison of defect levels in ST and DO epitaxial silicon
- Comparison of defects concentrations in ST and DO epitaxial silicon
- Conclusions



- •Epitaxial n-Si pad-detectors on Cz-substrate produced by ITME/CiS
- -Resistivity : 500 Ωcm, Thickness: ~ 150 µm, Size: 5 x 5 mm²
- -Standard (ST) and oxygen enriched (DO, diffusion for 24 h at 1100°C) material
- -24 GeV/c-proton-irradiation (CERN PS), $\Phi_{eq} = 1.71 \times 10^{16} \text{ cm}^{-2}$

Material	d [µm]	Wafer	Orientation	$\frac{N_{\rm eff,0} [P]}{[10^{12} \rm cm^{-3}]}$	[O] $[10^{16} \text{ cm}^{-3}]$
EPI-ST 150	147	261636-13	<100>	8.8	4.5
EPI-DO 150	152	261636-09	<100>	8	14.0



From:

Charge collection and trapping effects in 75 μ m, 100 μ m and 150 μ m thick n-type epitaxial silicon diodes after proton irradiation

Julian Becker, Eckhart Fretwurst, Jörn Lange, Gunnar Lindström, Hamburg University ,13th RD50 Workshop, CERN, November 2008

Imaging of defect levels for as-irradiated samples and after following annealing stages:

HT1: 1h, 80 °C + 24 h, RT

HT2: (1h, 80 °C + 24 h, RT) + 1h, 160 °C + 24 h, RT

HT3: (1h, 80 °C + 24 h, RT + 1h, 160 °C + 24 h, RT) + 1h, 240 °C + 24 h, RT

Details of HRPITS measurements

- Temperature range: 30 300 K, $\Delta T = 2$ K
- Excitation source: 1 mW, 650 nm laser diode (hv = 1.908 eV)
- Excitation pulse parameters: Period 250 ms, Width 50 ms
- Photon flux: 1.3x10¹⁷ cm⁻²s⁻¹
- BIAS: 20 V
- Gain: 1x10⁶ 1x10⁸ V/A
- AVG: 250 waveforms
- Analysis of photocurrent relaxation waveforms:
 - 2D correlation procedure (multi-window approach) → images of correlation spectral fringes for radiation defect centres
 - 2D inverse Laplace transformation algorithm → images of Laplace fringes for radiation defect centres

Effect of annealing on mobility lifetime product fluence 1.7x10¹⁶ p/cm²



oxygenated epilayer

standard epilayer

HRPITS images – fluence 1.7x10¹⁶ p/cm² – as-irradiated



standard epilayer

HRPITS images – fluence 1.7x10¹⁶ p/cm² – 80 °C



standard epilayer

HRPITS images – fluence 1.7x10¹⁶ p/cm² – 160 °C



standard epilayer

HRPITS images – fluence 1.7x10¹⁶ p/cm² – 240 °C



standard epilayer

Comparison of defect levels in standard and oxygenated epitaxial silicon; 1.7x10¹⁶ p/cm²; as-irradiated

$$e_{Tm} = A_m T^2 \exp(-E_{am}/k_B T); A_{mn} = \gamma_n \sigma_{mn}; A_{mp} = \gamma_p \sigma_{mp}$$

TS8, TS7, TC1 : shallow donors-STD(H); TS9, TS10, TC2: I aggregates; T7: $V_2(+/0)$; TA4: VO(-/0); TA6: $V_2(-/0)$; TA7: V_4V_5 ; TA8: complex of O with V aggregates (V_4, V_5); TS5: IO₂(?)



Comparison of defect levels in standard and oxygenated epitaxial silicon; 1.7x10¹⁶ p/cm²; annealing at 80 °C

TA1: shallow donors; TA2: I aggregates; TA4: VO(-/0); TS5: IO₂(?)

TA6: $V_2(-/0)$; T7: $V_2(+/0)$; TA7: V_4V_5 ; TA8: complex of O with V aggregates (V_4, V_5);



Comparison of defect levels in standard and oxygenated epitaxial silicon;1.7x10¹⁶ p/cm²; annealing at 160 °C

TA1: shallow donors; TA2, TS3: I aggregates; TA4: VO(-/0);

TA6:V₂(-/0); T7: V₂(+/0); TA7: V₄V₅; TA8: complex of O with V aggregates (V₄,V₅)



Comparison of defect levels in standard and oxygenated epitaxial silicon; 1.7x10¹⁶ p/cm²; annealing at 240 °C

TA1: shallow donor; TS1, TA2, TS2, T1, T2, T3, T4: I aggregates; T7: V₂(+/0); TA4: VO(-/0); TA6: V₂(-/0); T8: V₂O (2-/-); T13: I center (V₃); T14: V aggregate (V₅)



Comparison of defects concentrations in standard and oxygenated epitaxial silicon; 1.7x10¹⁶ p/cm²; as-irradiated

TS7, TS8, TC1 : shallow donors; TS9, TS10, TC2: I aggregates; T7: $V_2(+/0)$; TA4: VO(-/0); TA6: $V_2(-/0)$; TA7: V_4V_5 ; TA8: complex of O with V aggregates (V_4, V_5); TS5: IO₂ (?)



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Comparison of defects concentrations in standard and oxygenated epitaxial silicon; 1.7x10¹⁶ p/cm²; annealing at 80 °C

TA1: shallow donors; TA2: I aggregates; TA4: VO(-/0); TS5: IO₂(?)

TA6: $V_2(-/0)$; TA7: V_4V_5 ; T7: $V_2(+/0)$; TA8: complex of O with V aggregates (V_4, V_5);



standard

oxygenated

Comparison of defects concentrations in standard and oxygenated epitaxial silicon; 1.7x10¹⁶ p/cm²; annealing at 160 °C

TA1: shallow donors; TA2, TS3: I aggregates; TA4: VO(-/0); TA6:V₂(-/0);

T7: $V_2(+/0)$; TA7: V_4V_5 ; TA8: complex of O with V aggregates (V_4 , V_5)



Comparison of defects concentrations in standard and oxygenated epitaxial silicon; 1.7x10¹⁶ p/cm²; annealing at 240 °C

TA1: shallow donor; TS1, TA2, TS2, T1, T2, T3, T4: I aggregates; T7: V₂(+/0);

TA4: VO(-/0); TA6: V₂(-/0); T8: V₂O (2-/-); T13: I center (V₃); T14: V aggregate (V₅)



oxygenated

Conclusions

For the first time we have shown the results of both qualitative and quantitative analysis of defect levels for epitaxial silicon after 24 GeV/*c* protons irradiation with a fluence of 1.7×1016 cm-3. The radiation defect levels have been scanned by High-Resolution Photoinduced Transient Spectroscopy with implementation of the imaging procedure.

➤The effect of the oxygen concentration in the epitaxial layers on the traps properties and concentrations was studied. It was found that the higher oxygen concentration affects mainly the properties of shallow traps related to interstitial aggregates. After annealing at 240 °C, the high concentrations of midgap traps with activation energies of 565 and 575 meV are observed independently on the oxygen concentration.

➤After annealing at 80 and 160 °C, the greatest number of the same traps was observed both for standard and oxygenated epitaxial layers.

Further studies aimed at verifying the values of traps concentrations are in progress.

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Model of Photocurrent Relaxation Waveforms

 $i_m(t, T) = I_m(\lambda, T) \exp(-e_{Tm}t);$ $I_m(\lambda, T) = n_{tom}(T) e_{Tm}(T) \mu_T(T) \tau_T(T) C (\lambda, T) qE$ $i(t, T) = \sum_{m=1}^M I_k(\lambda, T) \exp(-e_{Tm}t);$

$$e_{Tm} = A_m T^2 \exp(-E_{am} / k_B T)$$
$$A_{mn} = \gamma_n \sigma_{mn} ; A_{mp} = \gamma_p \sigma_{mp}$$

For Si: $\gamma_n = 1.07 \text{ x } 10^{21} \text{ cm}^{-2} \text{K}^{-2} \text{s}^{-1}$; $\gamma_p = 2.64 \text{ x } 10^{21} \text{ cm}^{-2} \text{K}^{-2} \text{s}^{-1}$