



A-M Valente-Feliciano



Development of NbTiN based multi-layered structures for SRF applications

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SRF Application beyond Nb: SIS Multilayers

Taking advantage of the high $-T_c$ superconductors with much higher H_c without being penalized by their lower H_{c1} ...

Alex Gurevich, *Appl. Phys. Lett.* 88, 012511 (2006)

Alex Gurevich, *AIP ADVANCES* 5, 017112 (2015)

T. Kubo, *Applied Physics Letters* 104, 032603 (2014)

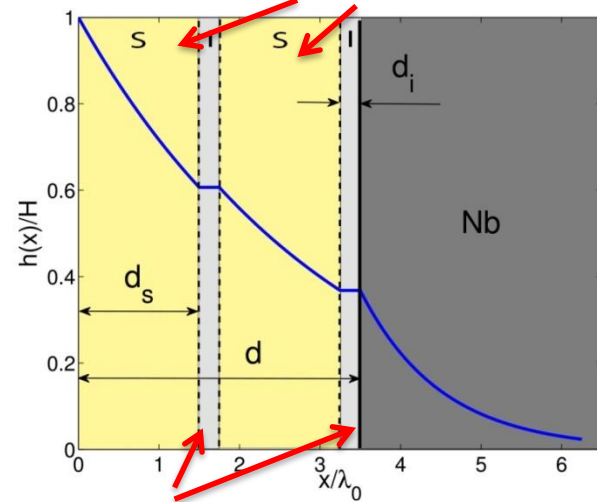
**Multilayer coating of SC cavities:
alternating SC and insulating layers with $d < \lambda$**

Higher T_c thin layers provide magnetic screening of the Nb SC cavity (bulk or thick film) without vortex penetration

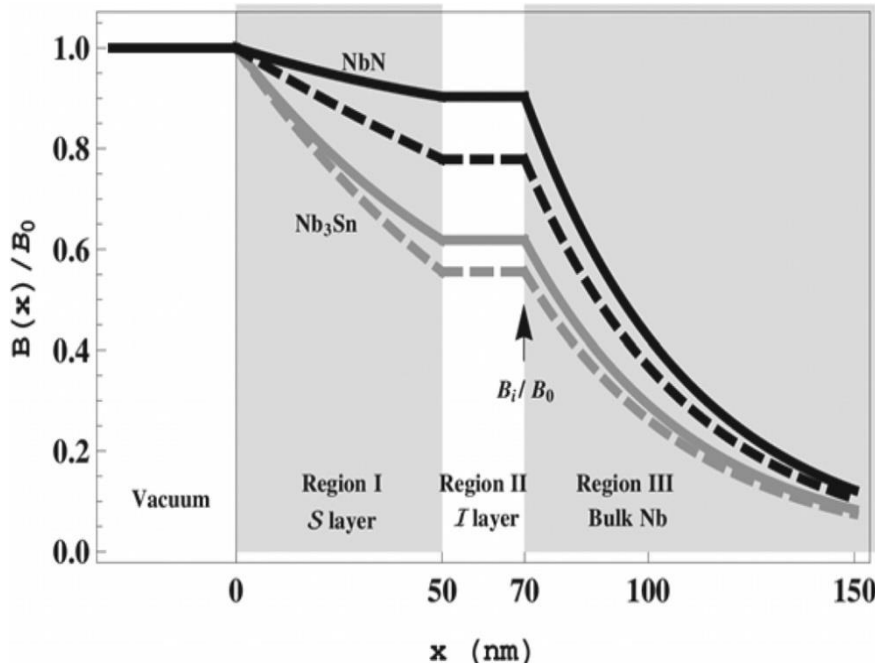
- Strong increase of H_{fp} in films allows using RF fields $> H_c$ of Nb, but lower than those at which flux penetration in grain boundaries may become a problem => no transition, no vortex in the layer
- High H_{fp} , applied field is damped by each layer
- Insulating layer prevents Josephson coupling between layers
- Applied field, i.e. accelerating field can be increased without high field dissipation
- SC layers with higher T_c , Δ (Nb₃Sn, NbN, etc.) => Strong reduction of R_{BCS} (ie high Q_0)

Possibility to move operation from 2K to 4.2K

Higher- T_c SC: NbN, Nb₃Sn, etc



Insulating layers

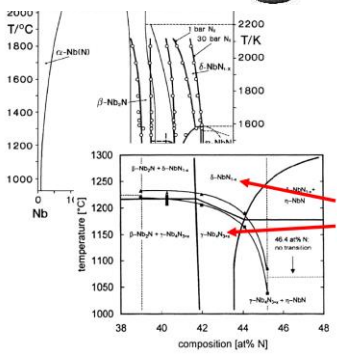
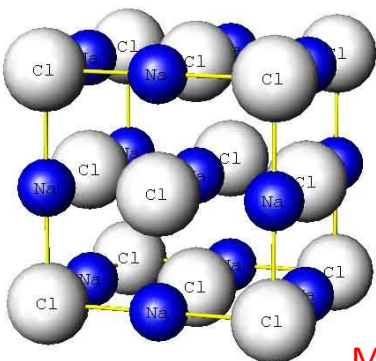


APPROACH

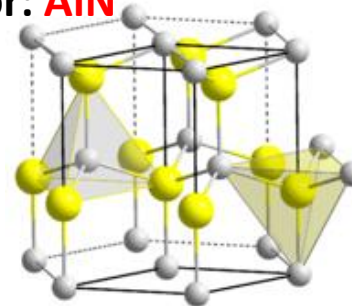
Superconductor

Ternary Nitride $(\text{Nb}_{1-x}\text{Ti}_x)\text{N}$
 $T_c = 17.3 \text{ K}$, $a = 4.341 \text{ \AA}$ for δ -phase
 Presence of Ti found to reduce significantly the resistivity

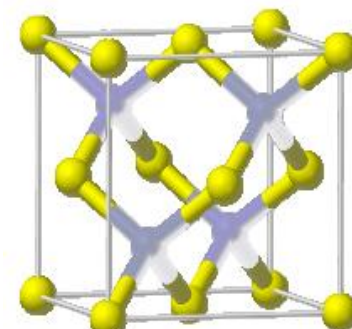
More metallic nature and better surface properties than NbN should result in better RF performance



Insulator: AlN



Wurtzite structure



Sphalerite structure

- Grown with a wurtzite (hcp, $a=3.11\text{\AA}$, $c=4.98\text{\AA}$) or sphalerite (B1 cubic, $a=4.08\text{\AA}$) structure.
- Found to enhance the properties (T_c) for very thin NbN and NbTiN films .
- Large thermal conductivity (3.19W/cm.K at 300K, comparable with Cu, 4.01W/cm.K)

CHALLENGES

- Develop good quality and uniform thin layers
- Sharp interfaces
- Growth of equally performing S/I and I/S layers

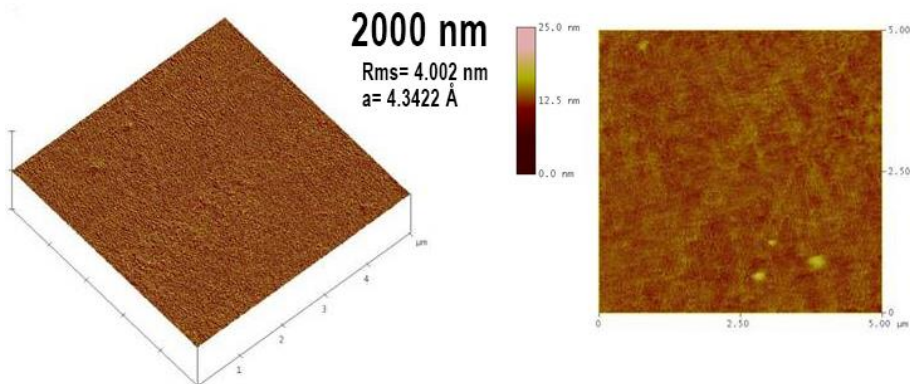
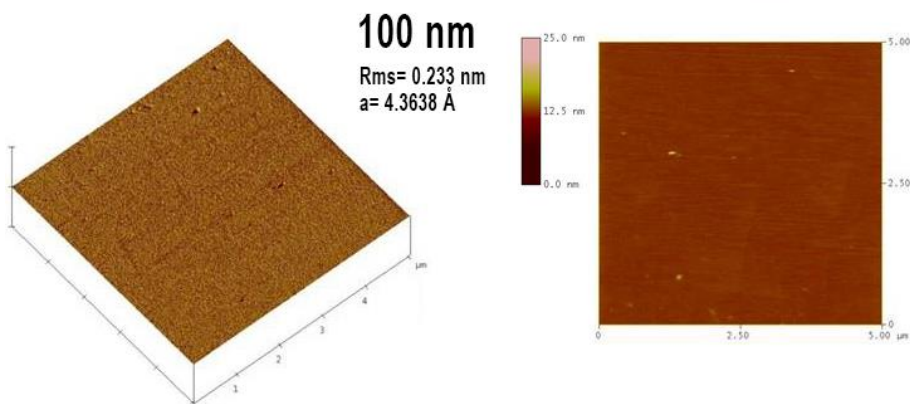
Base pressure range: 10^{-10} Torr

- dc-Magnetron Sputtering (reactive mode)
- HiPIMS (Huettinger 2000 V, 3000 A)
- Central sample stage

Substrates:

MgO (ideal)
 AlN ceramic (worst case)
 Bulk Nb
 ECR Nb films (real)

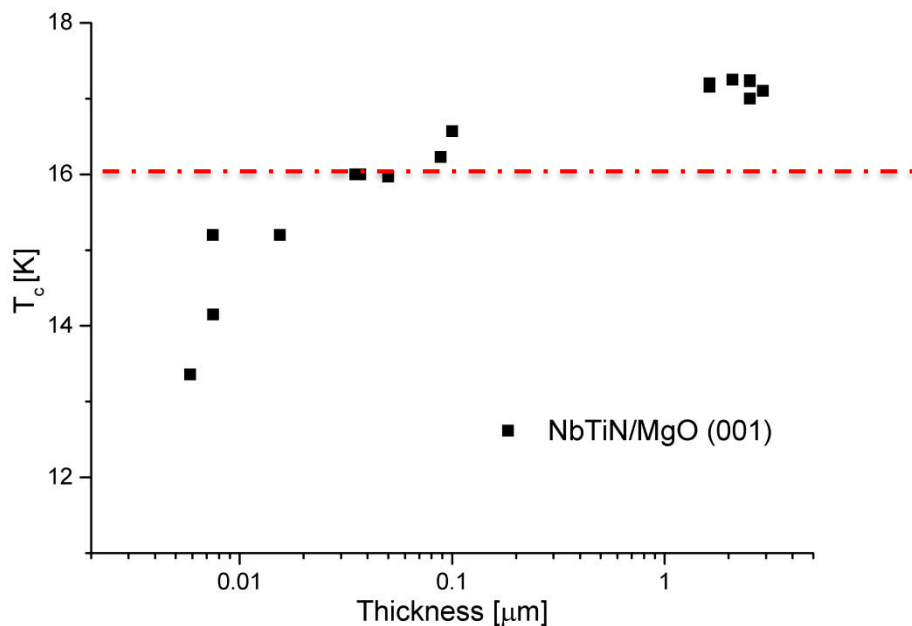
NbTiN Films – Influence of Thickness on T_c



Process Conditions for NbTiN

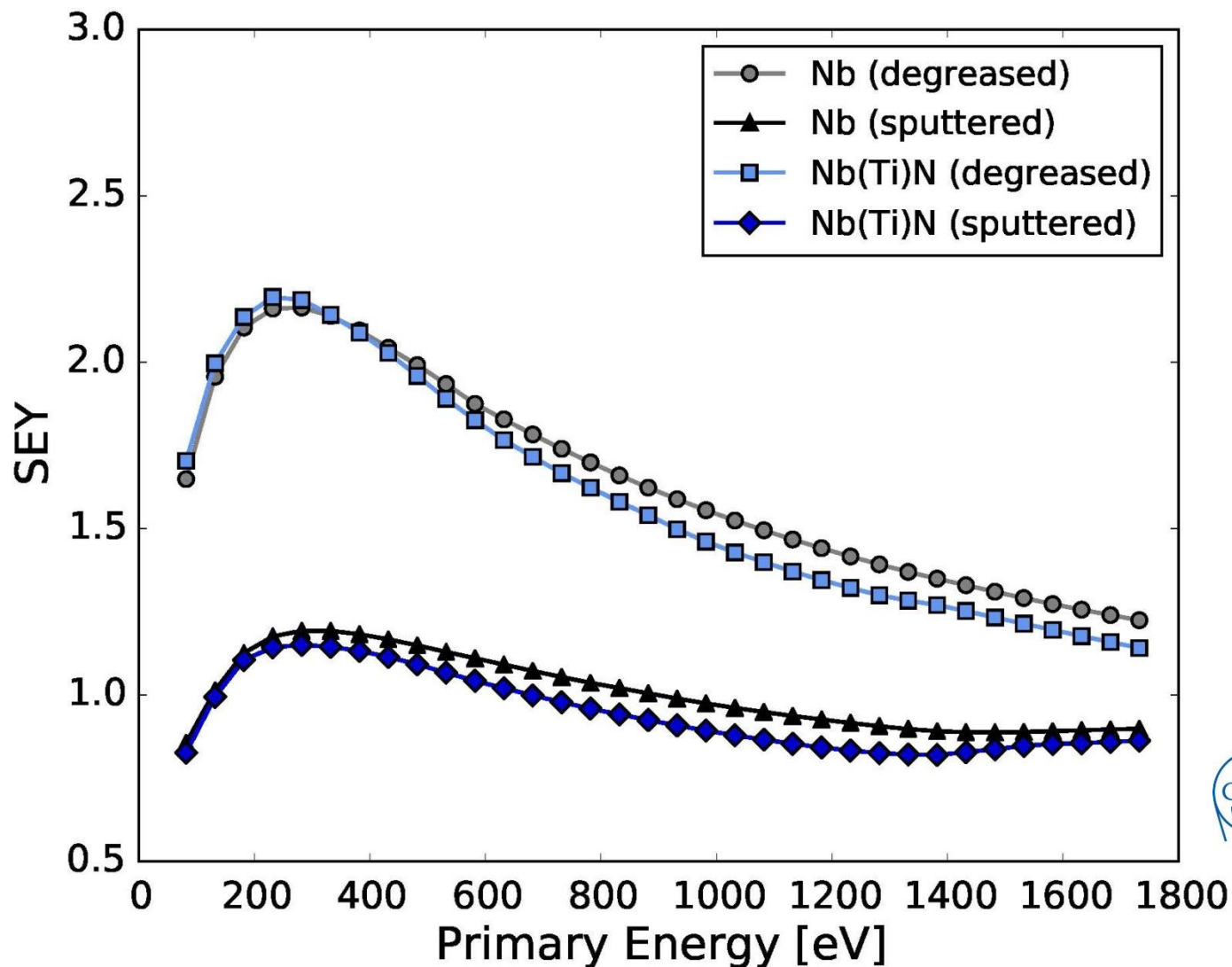
N_2/Ar	0.23
Total pressure	2×10^{-3} Torr
Sputtering Power	300 W
Deposition rate	~ 20 nm/min

Single crystal NbTiN/MgO films (XRD/EBSD)
Very smooth films (\sim substrate)



**$T_c \geq 16$ K for
film thickness > 35 nm**

Secondary Electron Yield of NbTiN Films



Measurements at room temperature

Max. SEY = 2.2 ± 0.1
comparable to EP Nb

After sputtering away
~ 3 nm,
SEY down to 1.15



Process Conditions for AlN

N ₂ /Ar	0.33
Total pressure	2x10 ⁻³ Torr
Sputtering Power	50 -100 W
Deposition rate	~ 4nm/min

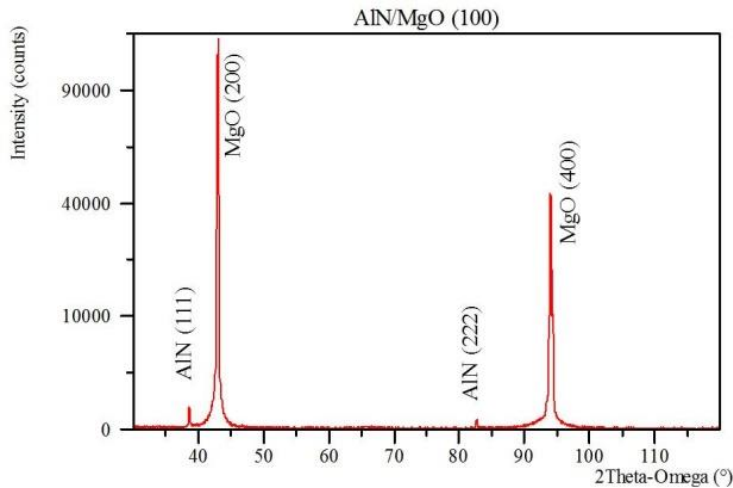
AlN Films

Structure

AlN films were coated by reactive sputtering with different parameters. They were found to become fully transparent for N₂/Ar ratios of ~33%.

Good quality AlN are readily produced at 600 and 450°C by dc-reactive magnetron sputtering.

The films exhibit the cubic structure (single crystal) at 600 °C and the hexagonal structure (polycrystalline) at 450 °C .



Dielectric Behavior



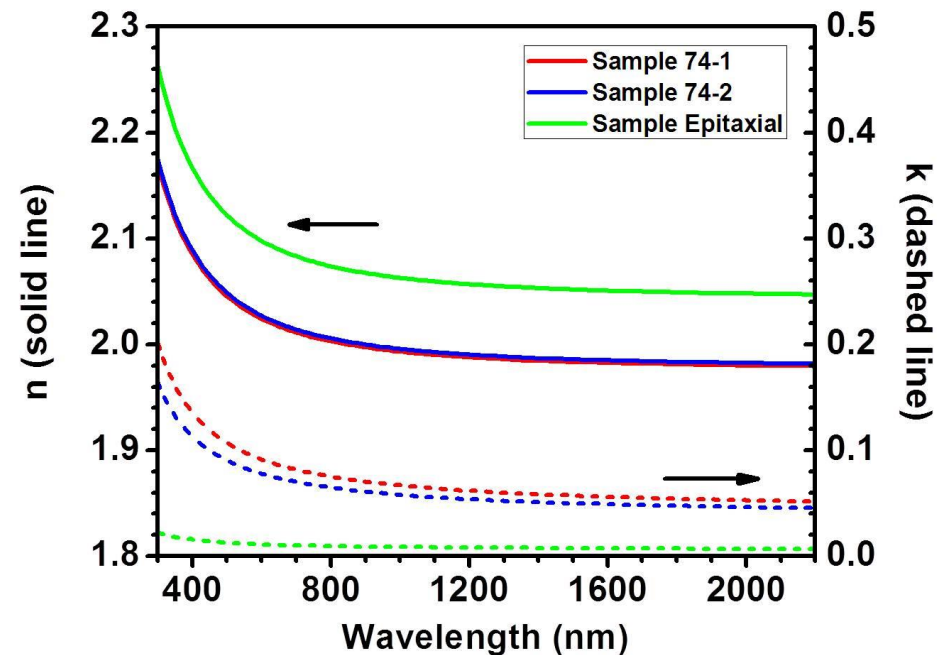
Roughness - EMA with 50% Void (XRR thickness used)

Film - Cauchy w/ Urbach Absorption (XRR thickness used)

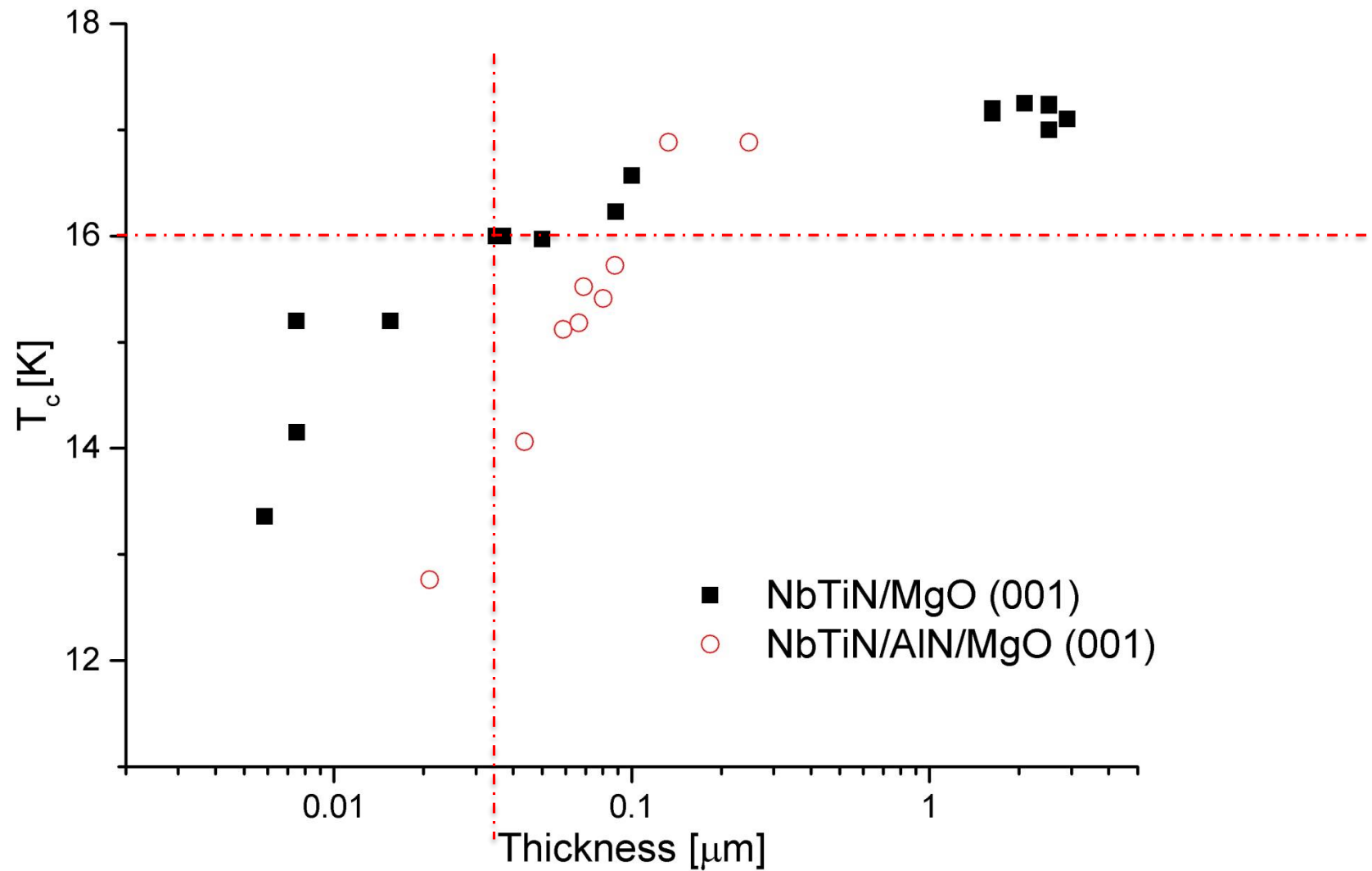
Substrate - Palik bulk optical constants; 0.5 mm

At 450 °C, 30 nm AlN films exhibit dielectric properties of polycrystalline AlN films

n in the range of 1.98- 2.15



NbTiN/AlN bi-layers – Influence of thickness on T_c



SRF Multilayer Structures Based on NbTiN

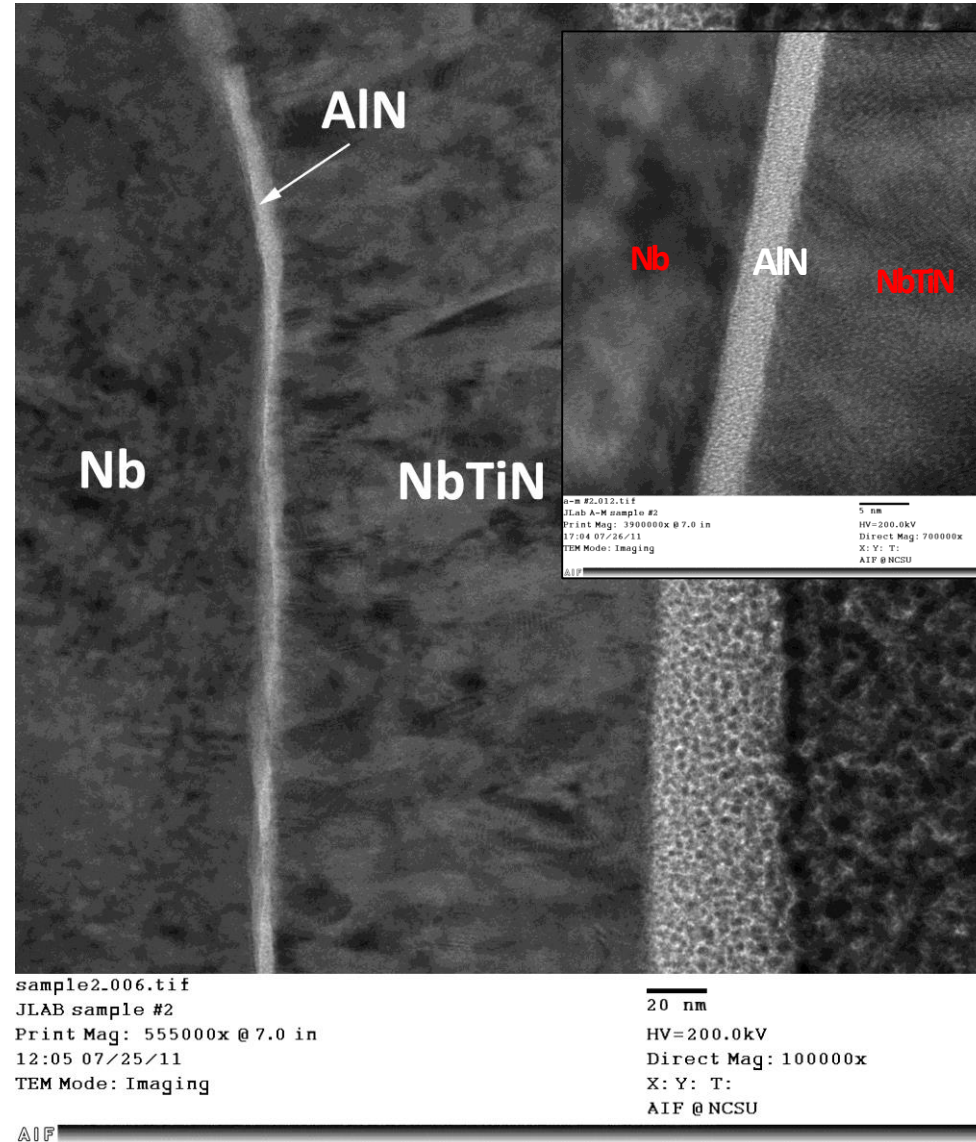
Influence of coating temperature

NbTiN/AlN/Nb film at 600 °C

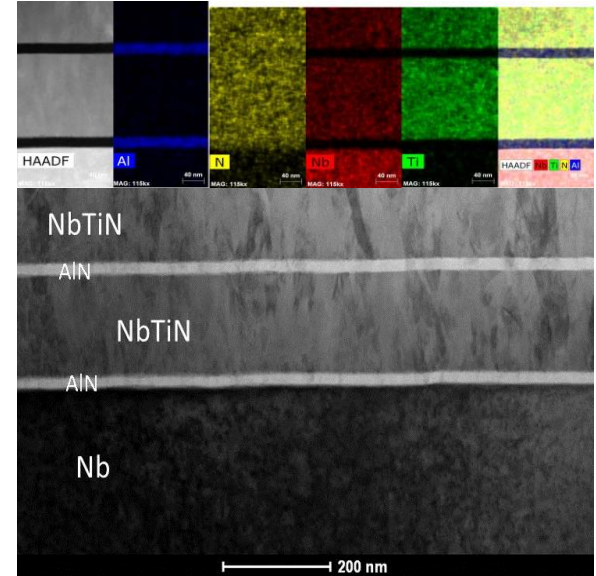
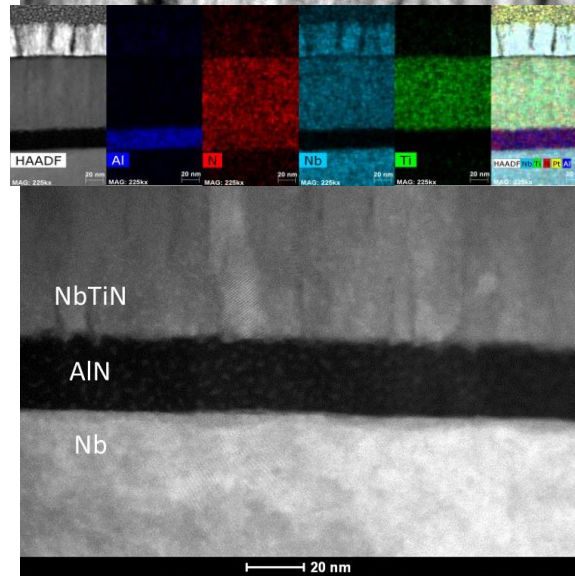
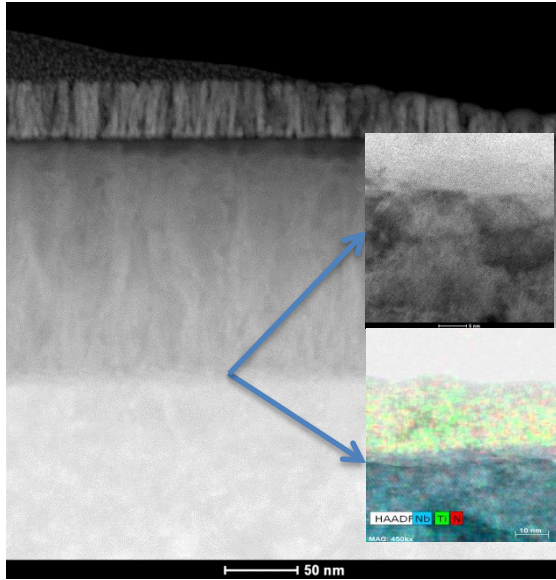
	AlN	NbTiN
N ₂ /Ar	0.33	0.23
Total pressure [Torr]	2x10 ⁻³	2x10 ⁻³
Sputtering Power [W]	100	300
Deposition rate [nm/min]	~ 2.5	~ 18
Thickness [nm]	5	100
T _c [K]	N/A	14

TEM cross-section (FIB cut)
of NbTiN/AlN/Nb/Cu
structure

Miscibility of AlN into Nb and NbTiN
at 600 °C



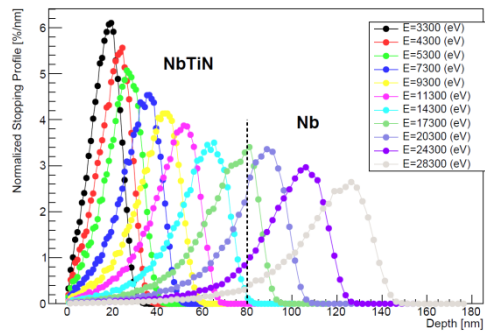
Development of SIS NbTiN/AlN structures on Nb surfaces



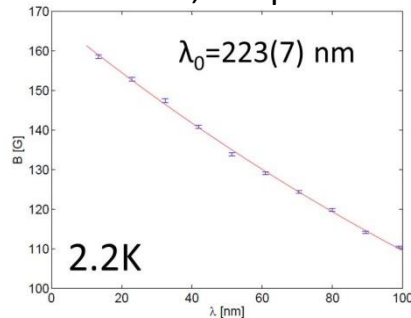
μ SR measurement (T. Junjinger, A. Sutter) on NbTiN/Nb (SS) & NbTiN/AlN/Nb (SIS)

- Measure above T_c of Nb
- Field enters from both sides
- **Significantly different λ_0 and d for SS**

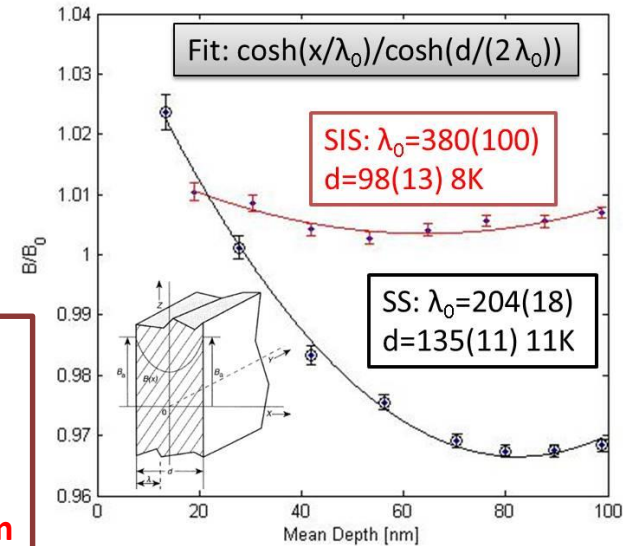
Suggest potential proximity effect for the SS structure, not present for SIS structure



Single exponential decay with $\lambda_0=223(7)$ nm
Proximity effect? Dirty Nb due to diffusion?



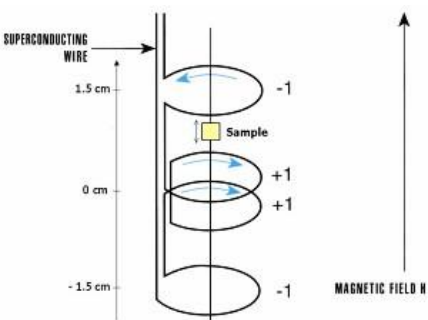
To benefit from the counter current flow an insulating layer is essential at least for the NbTiN/Nb system



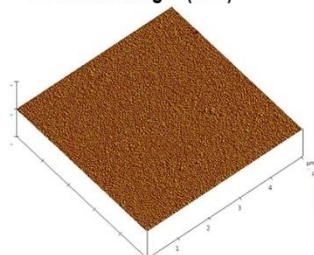
NbTiN/AlN Films (SI) – Flux penetration

SQUID Magnetometry

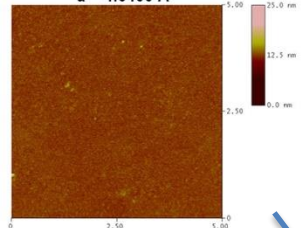
(Prof. A. Lukaszew group, College William & Mary)



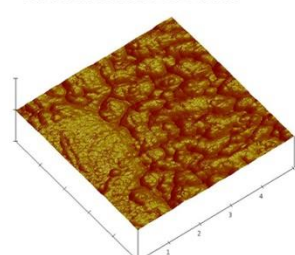
NbTiN/AlN/MgO (100)



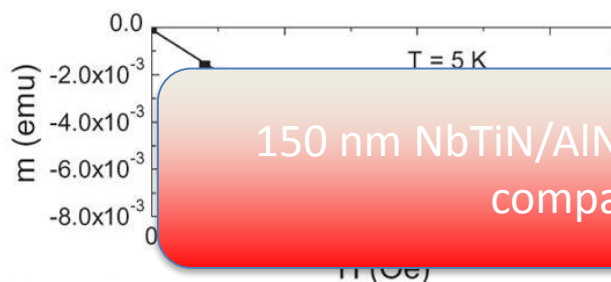
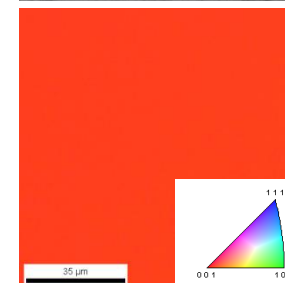
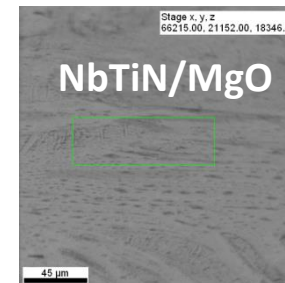
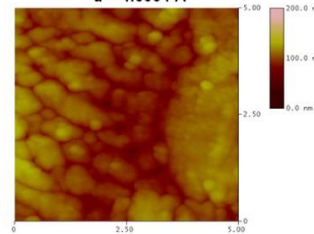
Rms=0.396 nm
a = 4.3455 Å



NbTiN/AlN/AlN ceramic

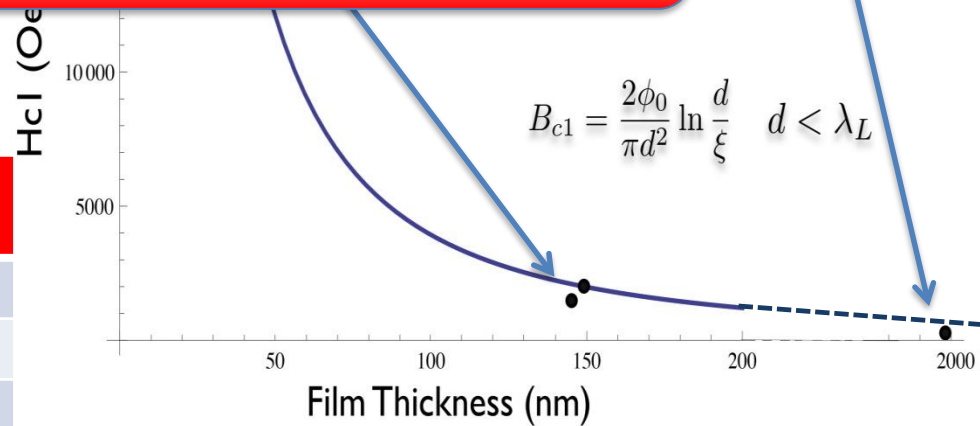


Rms=13.434 nm
a = 4.3584 Å



150 nm NbTiN/AlN SI structures exhibit **H_{fp} enhancement** compared to bulk-like NbTiN film

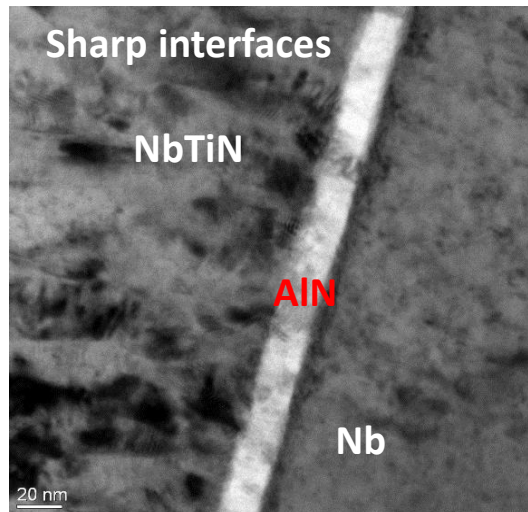
	Thickness [nm]	H _{c1} [mT]	T _c [K]
NbTiN/MgO	2000	30	17.3
NbTiN/AlN/AlN ceramic	145	135	14.8
NbTiN/AlN/MgO	148	200	16.7



RF characterization of NbTiN/AlN/Nb structures

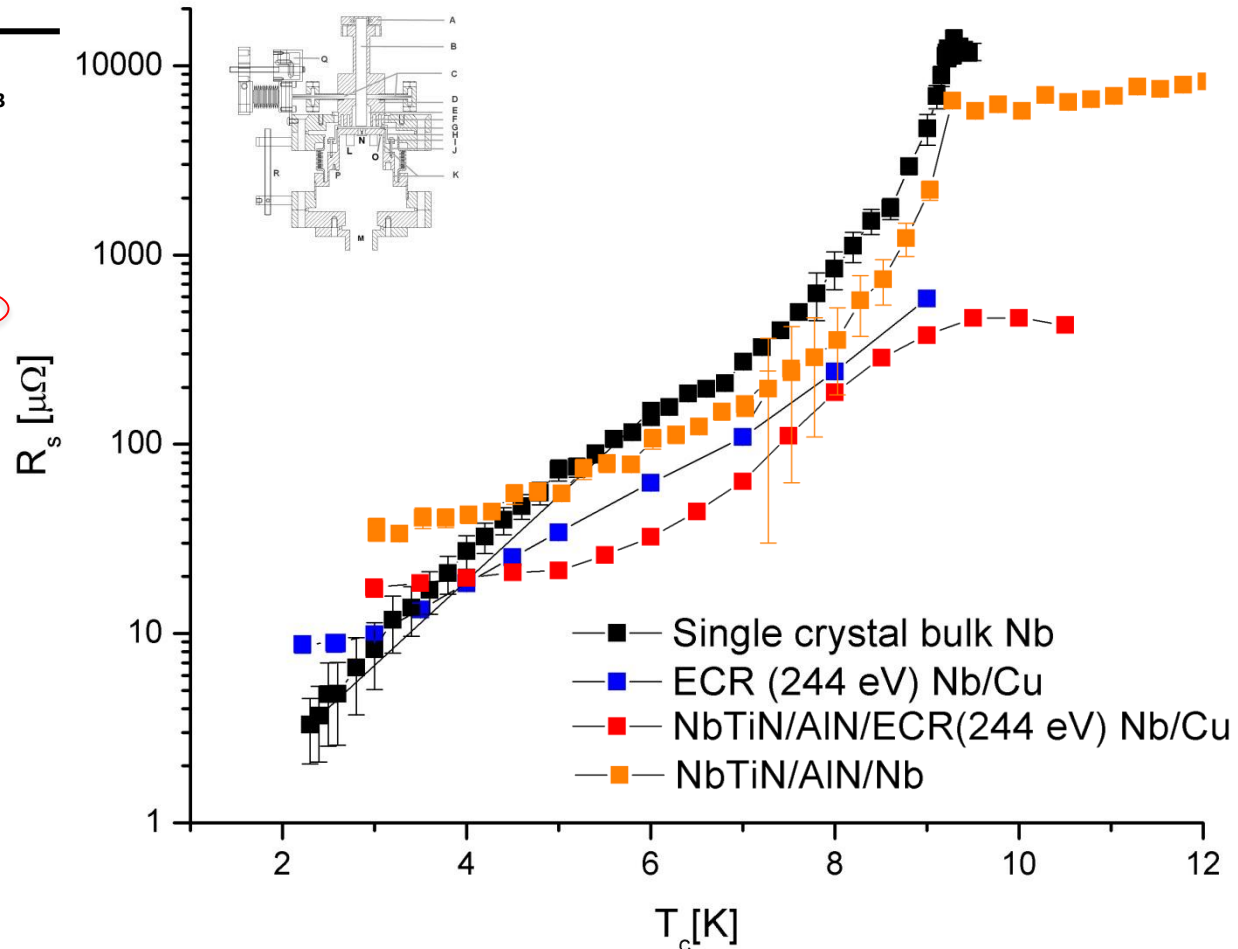
SIS structures coated on ECR Nb/Cu film: 24h-bake, coating and annealing for 4 h at 450°C.

	AlN	NbTiN
N ₂ /Ar	0.33	0.23
Total pressure [Torr]	2x10 ⁻³	2x10 ⁻³
Sputtering Power [W]	100	300
Deposition rate [nm/min]	~ 2.5	~ 18
Thickness [nm]	20	150
T _c [K]	N/A	16.9



TEM cross-section (FIB cut) of NbTiN/AlN/Nb/Cu structure

RF Measurement in 7.5 GHz sapphire-loaded TE₀₁₁ cavity



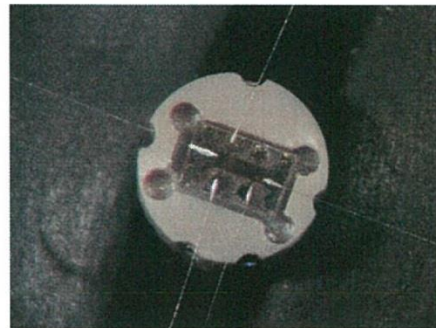
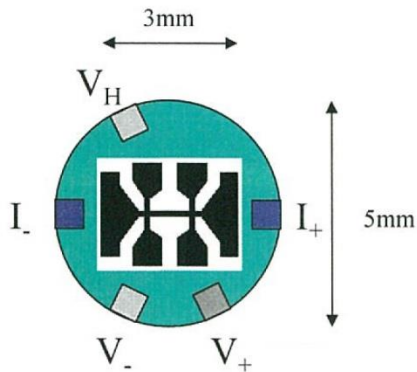
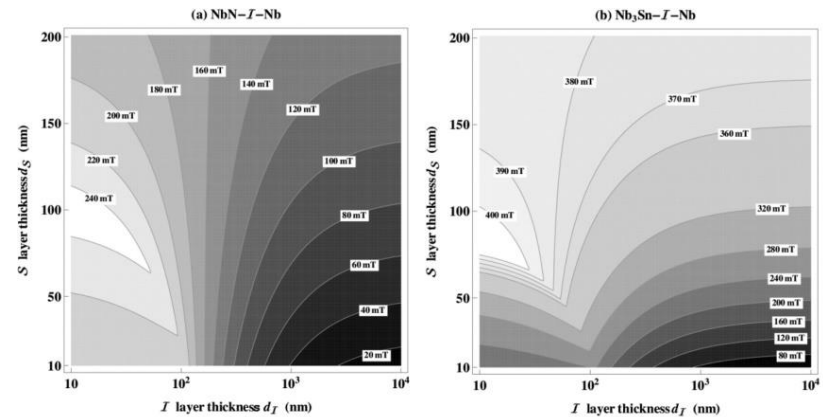
Lower BCS resistance beyond 4 K for SIS coated surfaces compared to standalone ECR film & bulk SC Nb.

NbTiN based SIS Optimization

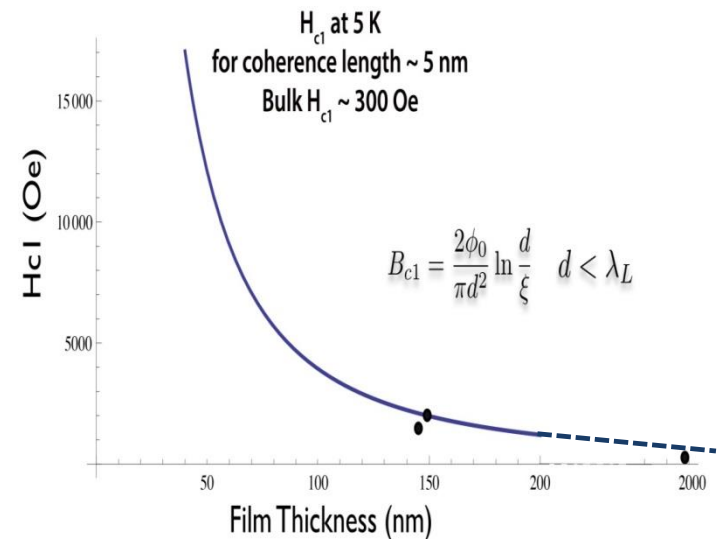
T. Kubo, SRF 2015

❑ Thickness series to **determine/verify optimum layer thicknesses** with H_{fp} measurements

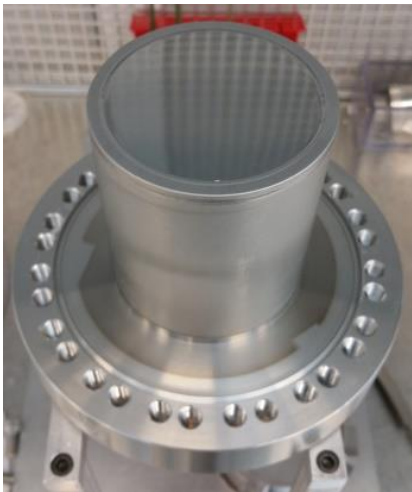
❑ **Implementing energetic condensation via HiPIMS** (High power impulse magnetron sputtering) to lower the coating temperature while maintaining a good quality δ -phase for NbTiN.



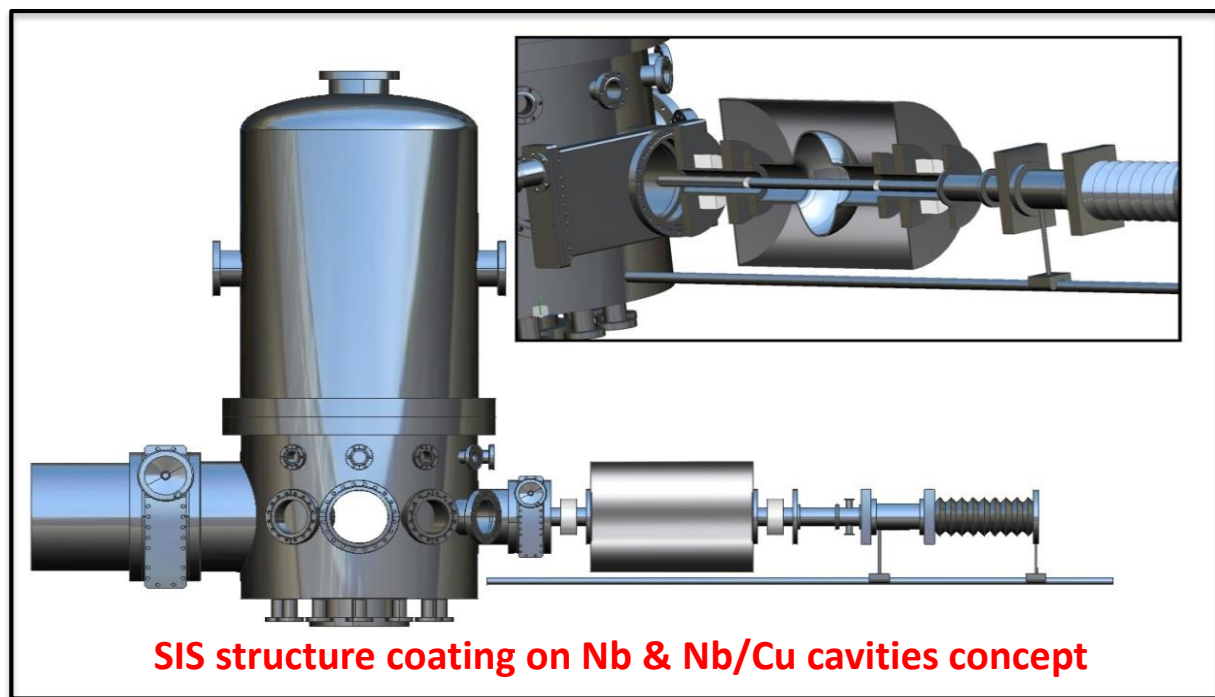
H_{C2} measurements via constrains MR in pulsed field - KU Leuven



NbTiN based SIS Optimization



- **RF measurement** for SIS NbTiN/AlN structures on previously characterized bulk Nb **QPR samples**.



Conclusions

- ✓ Good quality standalone NbTiN & AlN layers
- ✓ SIS NbTiN/AlN layers with a $T_{c, \text{NbTiN}}$ between 16.6 and 16.9 K.
 - **Growth conditions for SIS structures** need to be a compromise between optimum conditions for standalone films and minimizing interaction between layers .
- ✓ **H_{fp} enhancement** (SQUID magnetometry) observed for 150 nm NbTiN films.
- ✓ Completion of **H_{fp} enhancement versus** thickness under way (Kubo curves)
- ✓ **RF characterization of NbTiN/AlN structures coated on Nb surfaces** reveal a promise of delaying flux penetration and lower RF losses for SIS coated Nb surfaces, both bulk and thick film.