



Electro-mechanical studies of MgB_2 wires and cables prepared by different processes



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Motivation:

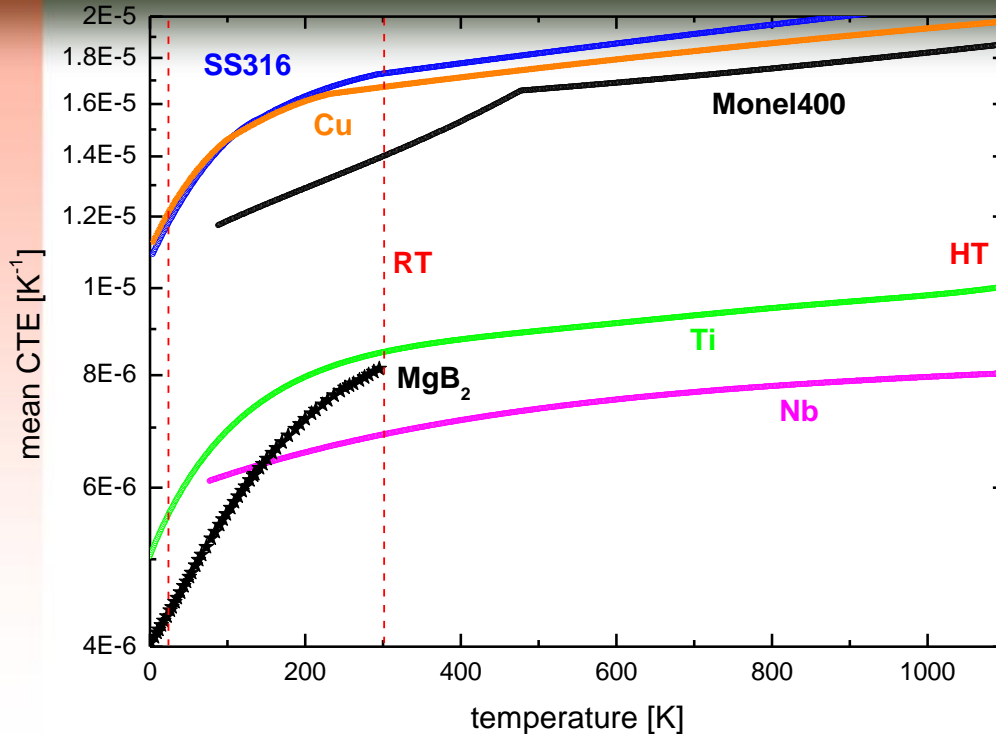
MgB_2 wires are composites consisting of **brittle filaments** inside one or more **ductile metallic sheaths**. Coils winding and magnets performance are not possible without **mechanical stress**.

Final properties of MgB_2 coils are **affected also by mechanical stresses** during the handling and operation.

Therefore, the **effects of mechanical loads** (e.g. **tension, pressure, bending and torsion**) have to be studied.

This presentation: effect of **tension at 4.2K, bending at RT and torsion** after HT applied to MgB_2 conductors of **variable design** and manufacturing **process**

MgB₂ phase in the cooled composite wire



Generally, **MgB₂ filaments** are under **pressure stress** due to higher **CTE** of used **metallic sheaths**.

Additional stress (**tension or bending**) is superposed to residual one, which is **increasing or decreasing** (compensating) the final stress - affecting sup. properties (decreased **T_c** and **J_c**).

Young modulus of the composite can be described by **the rule of mixtures** if metallic matrix and filaments are **elastic**:

$$E = f_f E_f + f_m E_m$$

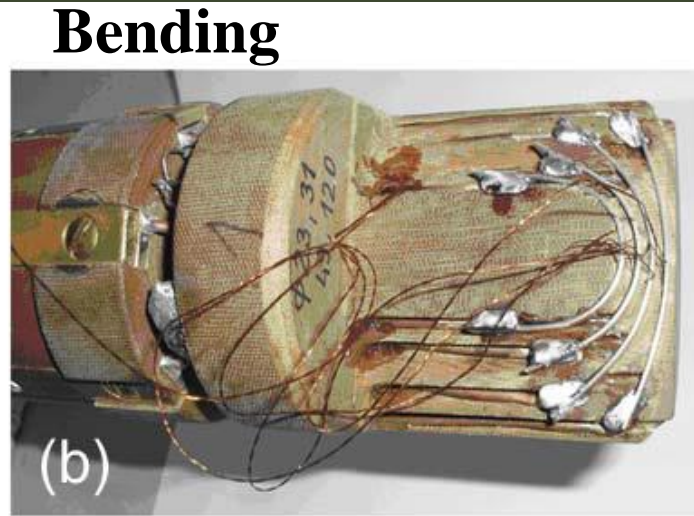
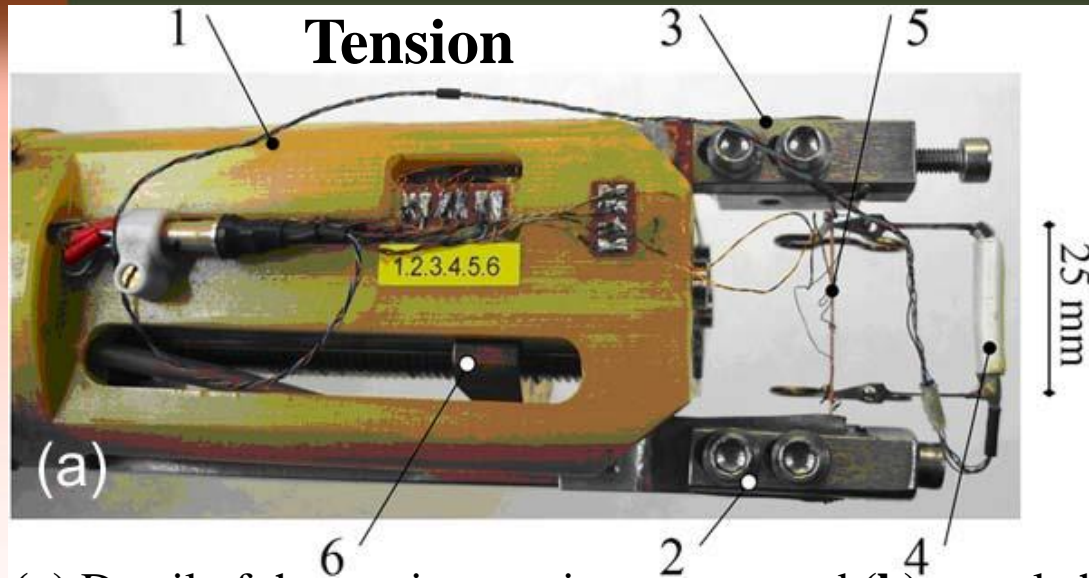
where f_f and f_m are the volume fractions of filaments and matrix.

$\sigma(\epsilon)$ curve of MgB₂ wire is **not linear** and the composite modulus must be determined by:

$$E = f_f E_f + (d\sigma_m / d\epsilon_m)_{\epsilon_f} \cdot f_m$$

where $(d\sigma_m / d\epsilon_m)_{\epsilon_f}$ is the slope of the matrix stress-strain curve at filament strain ϵ_f .

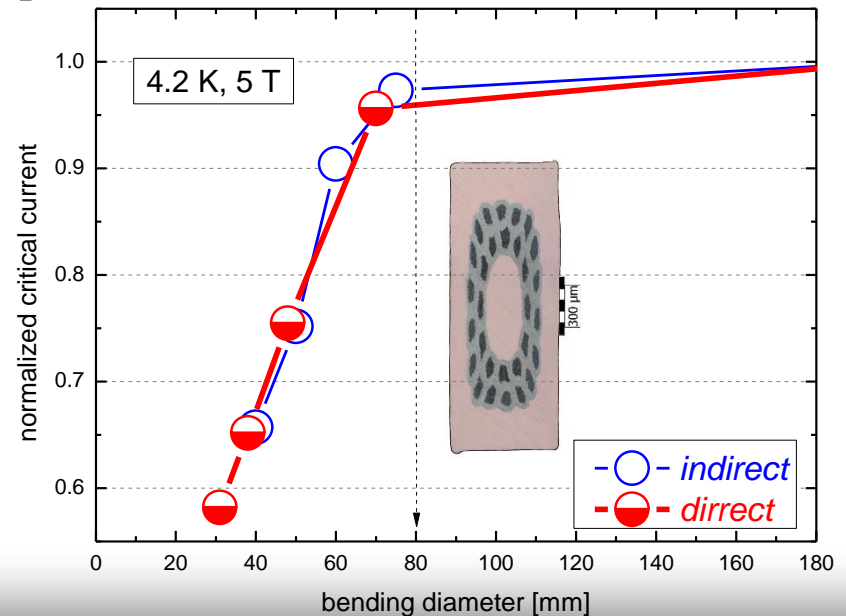
Electro-mechanical characterization of MgB_2 wires



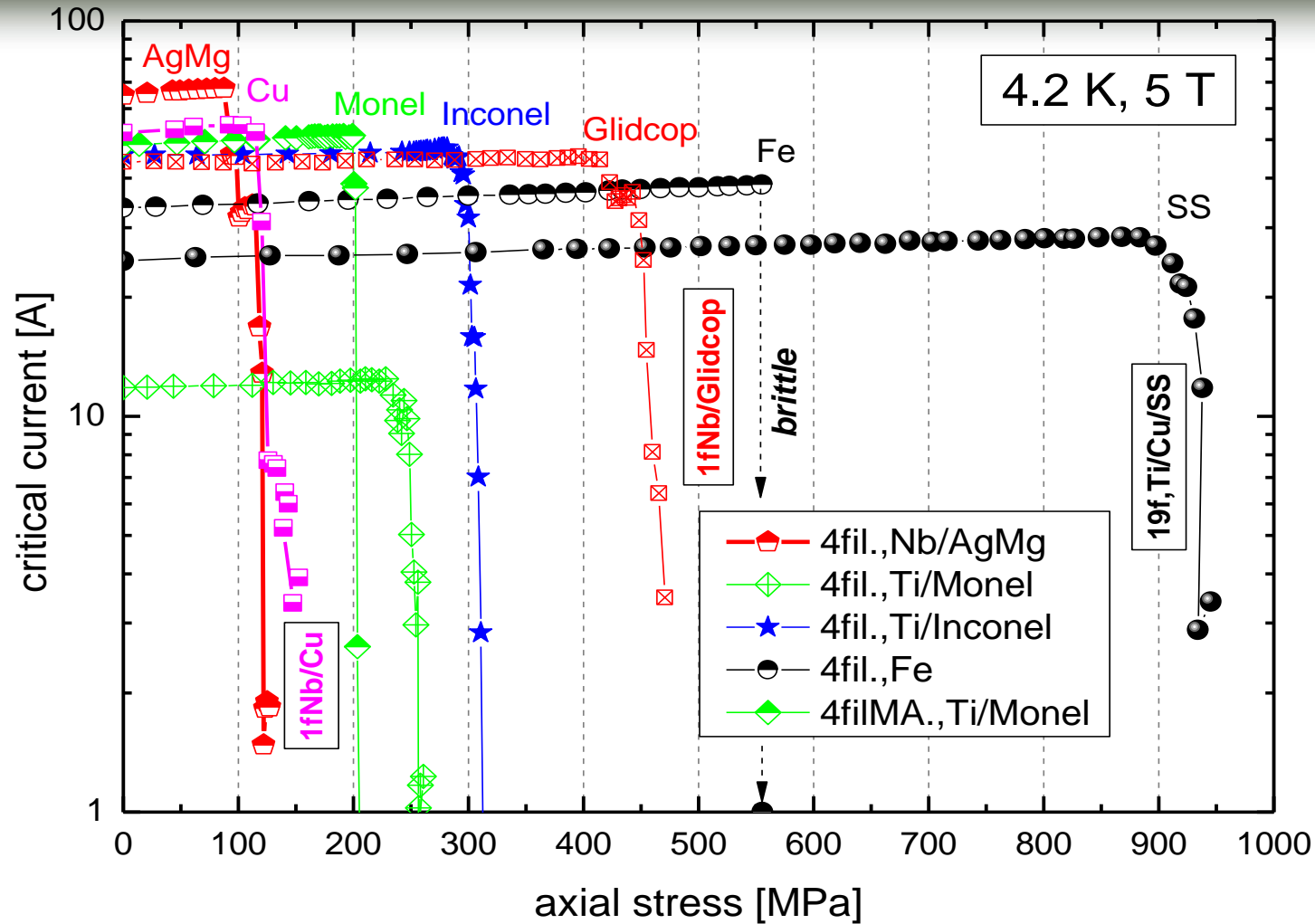
(a) Detail of the tension test instrument and (b) sample holder used for the **direct** measurement of critical current degradation by bending stress.

Tension tests: $\sigma(\epsilon)$, $I_c(\epsilon)$ and $I_c(\sigma)$, which allow to estimate: **irreversible strain** (σ_{irr}) and **stress** (ϵ_{irr})

Bending tests (direct or indirect): $I_c(d_b)$, which allow to estimate: **critical diameter**, $I_c(\epsilon_b)$



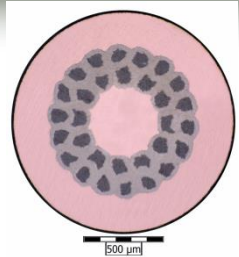
Tolerance to axial tension of in-situ wires at 4.2 K



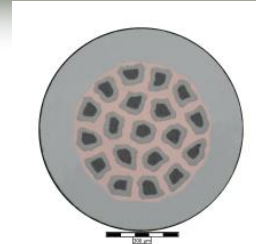
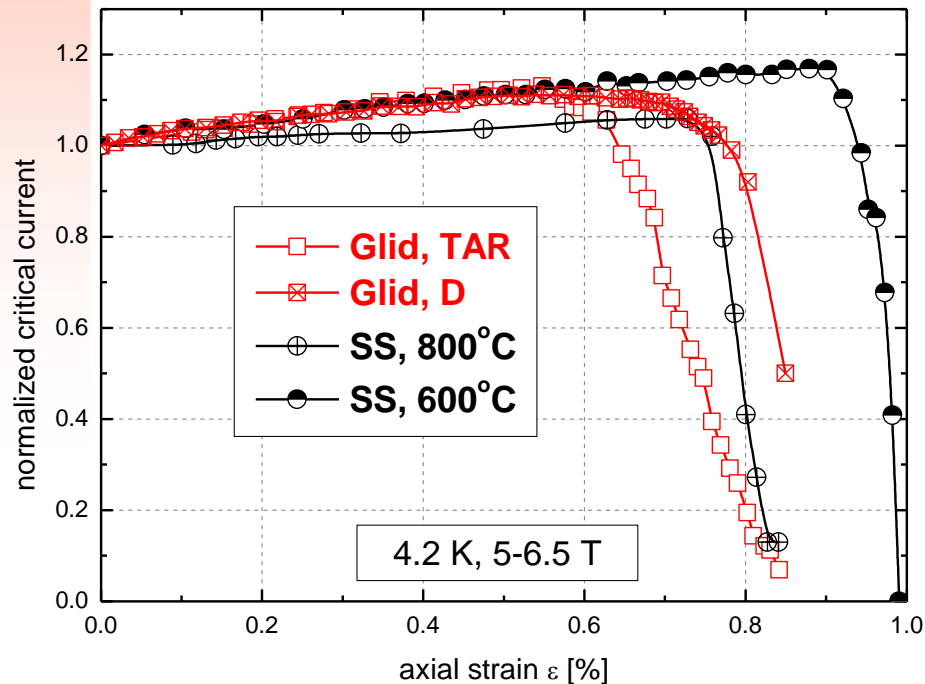
Tolerance to tensile stress is dominantly influenced by **outer sheath**:

- the **lowest** for AgMg and Cu, Monel-GlidCop, Fe - **extremely brittle**, SS - the **highest**

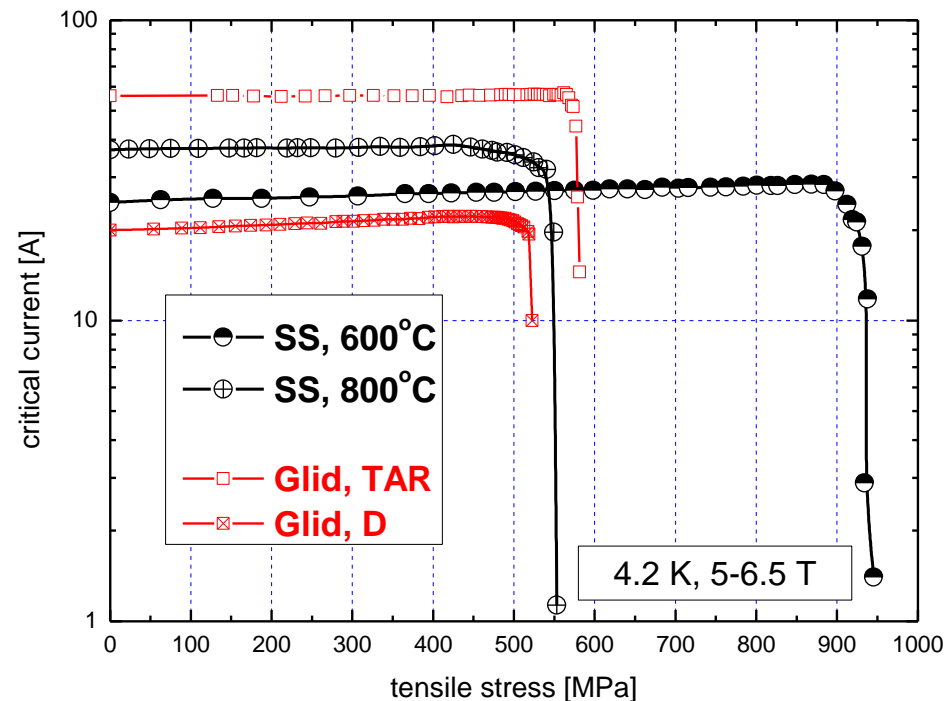
Tolerance of filamentary MgB_2 to tension



MgB_2 /Ti/GlidCop
Drawing (D) and rolling
(TAR)



MgB_2 /Ti/Cu/SS
HT at:
600 and 800°C



Tolerance to tensile strain/stress is affected by the **filaments density** and also by **HT conditions**:

- more dense filaments > higher σ_{irr} /lower ϵ_{irr} , - the strength of SS is reduced by HT at 800°C.

Reversible behaviour...

M. Dhallé, *Sup Sci and Technology* **18** (2005) S253

Dhallé M et al.: Up to the irreversible tensile strain, behaviour at different temperatures and magnetic fields (i.e. $I_c(T, B, \varepsilon)$ for $\varepsilon < \varepsilon_{\text{irr}}$), the **critical current varies linearly** with applied strain:

$$I_c(T, B, \varepsilon) = I_c(T, B, 0) [1 + K_I(T, B)\varepsilon] \quad (1)$$

where $K_I(T, B)$ is the (normalized) **slope** of the reversible strain response. Using the approach presented by eq. (1), the **K_I relation for the constant temperature** (4.2 K) and **magnetic field** (≈ 6 T) will be:

$$K_I = \frac{1}{I_c} \frac{dI_c}{d\varepsilon} = \frac{1}{I_c} \frac{dI_c}{dp} \frac{dp}{d\varepsilon} = \frac{E}{I_c} \frac{dI_c}{dp} \quad (2)$$

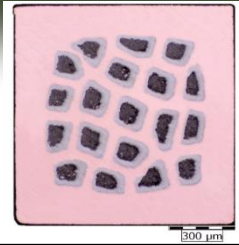
where **p is external pressure acting on the MgB_2 compound.**

Kitaguchi et al. have analyzed **the slope of I_c** versus the reduced field $b = B/B_{\text{irr}}$:
it falls on an universal line independently of temperature, K_I is \uparrow with B close to B_{irr} .

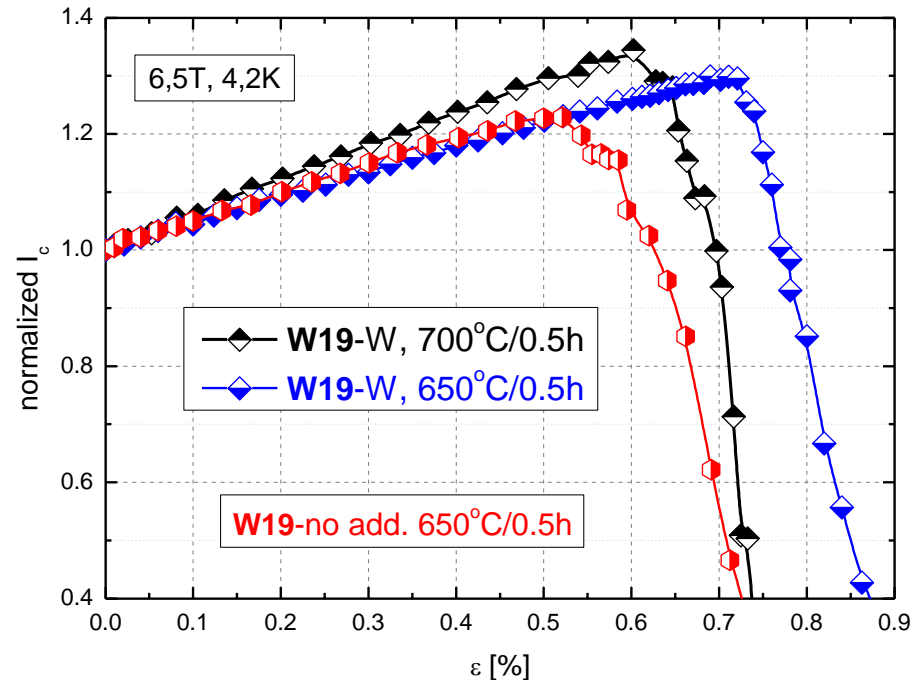
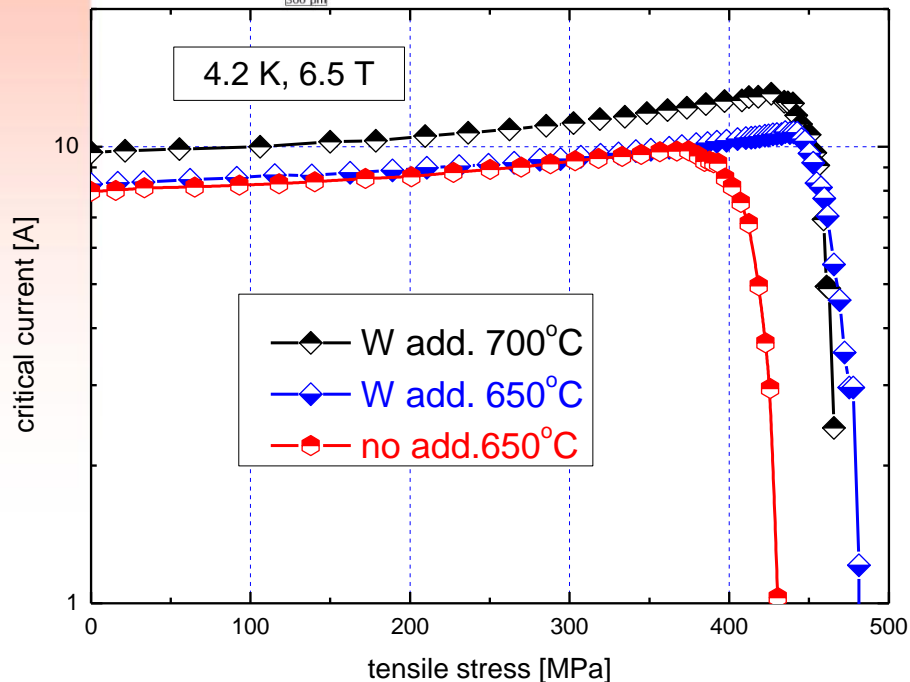
Kitaguchi H and Kumakura H *Supercond Sci Technol.* **18** (2005) S284

Effect of W additions into MgB_2 filaments

19 fil. $\text{MgB}_2/\text{Nb}/\text{GlidCop}$



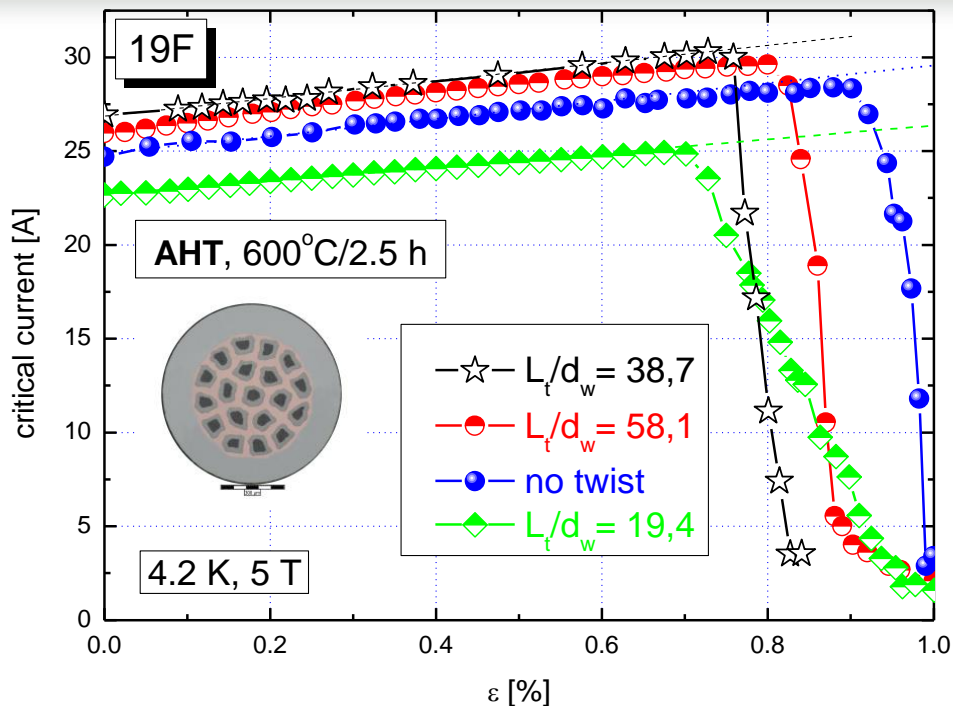
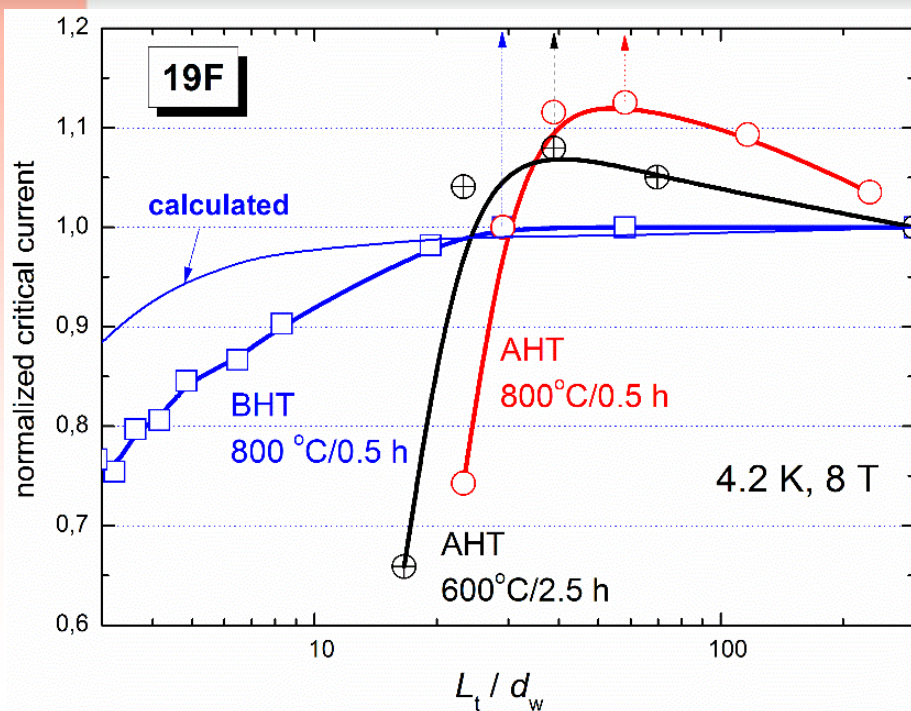
In-situ wire, Mg + B powder mixture and with 15 wt% of W addition of particle size $< 20 \mu\text{m}$



- 15 wt% additions of W particles into filaments **improved** the stress and strain **tolerance**
- W addition allows to **increase** I_c by tension up to 30–35% compared to not stressed one
- Presented improvements are attributed to **mechanical reinforcement** of MgB_2 filaments by distributed and *elongated W particles* (grain connectivity).

Strain tolerances of twisted MgB_2 wires

Twisting of wire before heat treatment (**BHT**) and after heat treatment (**AHT**)

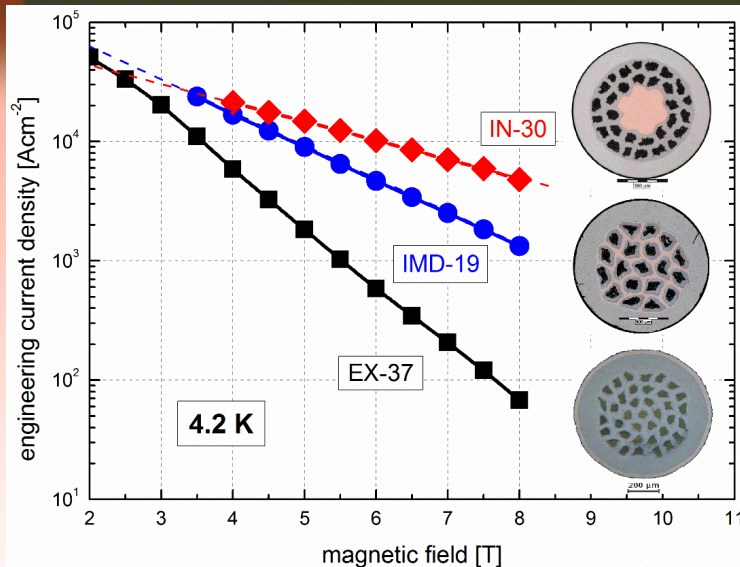


No current improvement has been measured for standard twisting applied before heat treatment (**BHT**).

Twisting applied after heat treatment increases the I_c of filamentary wire by 8–20%.

Improvement of I_c by twisting **AHT** is explained by a partial **compensation of residual stress** through the applied torsion.

Monel sheathed wires under loading and unloading

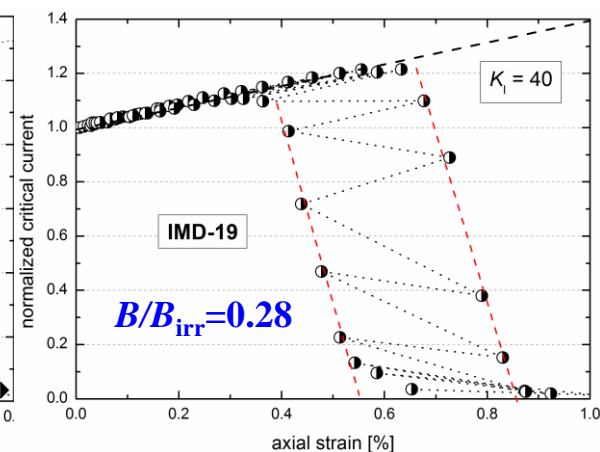
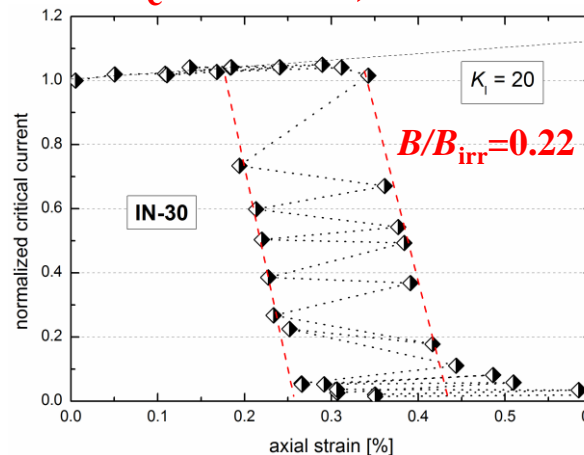
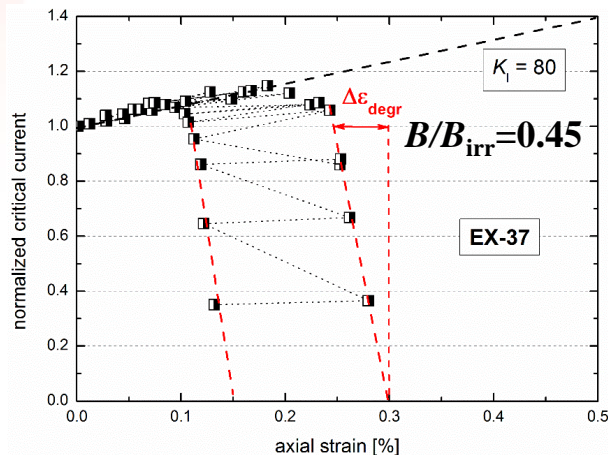
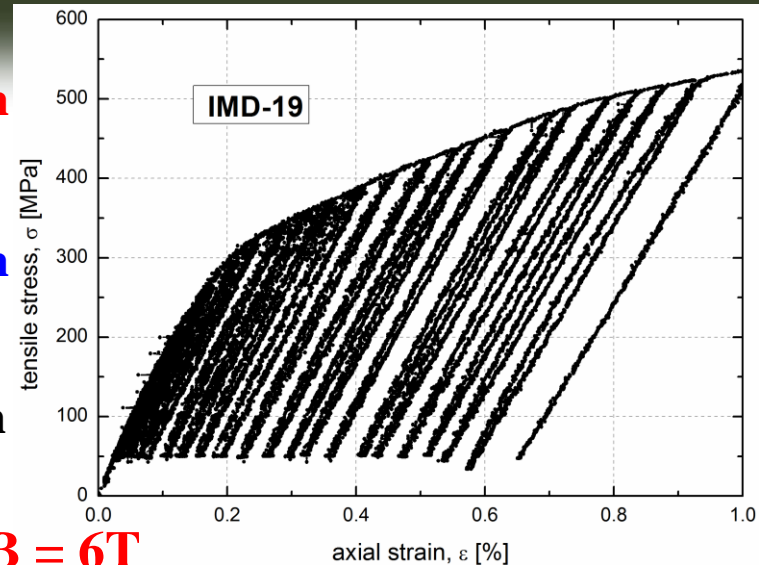


675°C/60min
HyperTech

650°C/60min

920°C/30min
Columbus

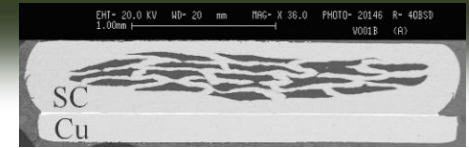
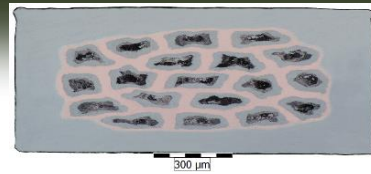
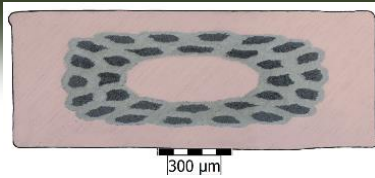
I_c at 4.2 K, $B = 6\text{ T}$



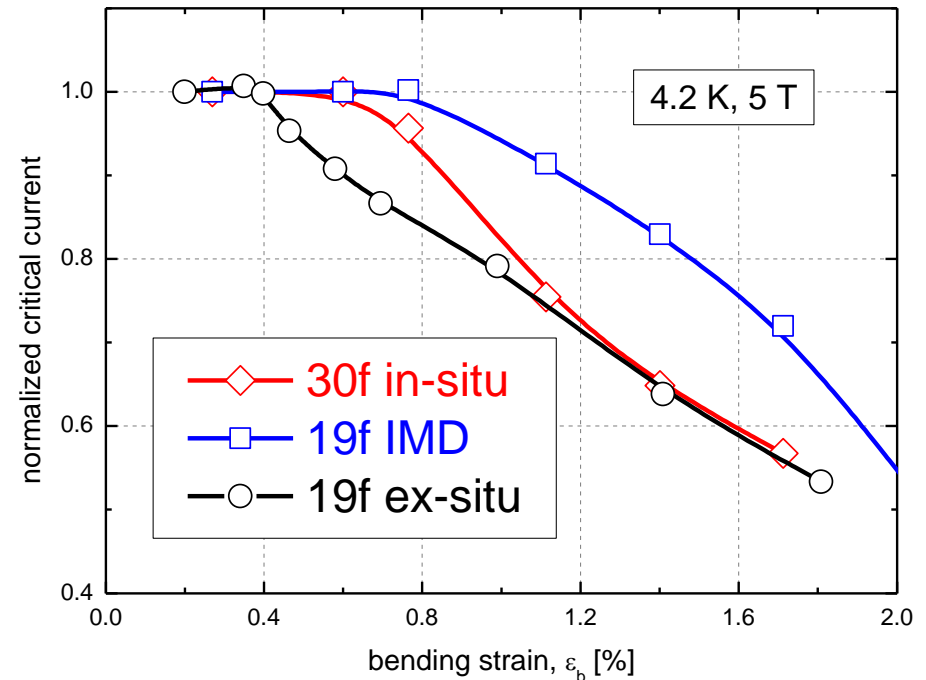
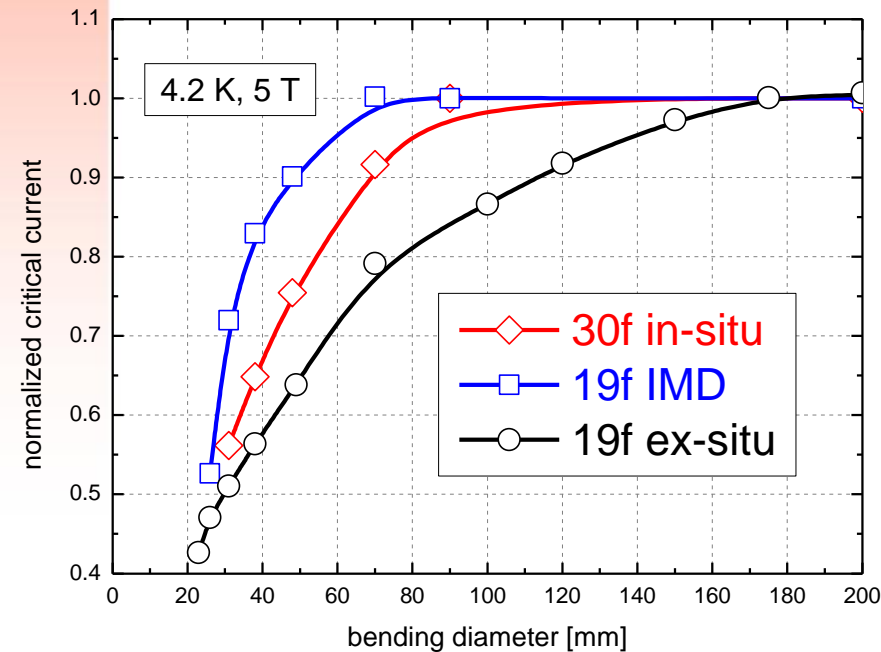
The lowest strain tolerance is measured for EX-37 wire $\epsilon_{\text{irr}} = 0.2\%$, $\epsilon_{\text{irr}} = 0.31\%$ for IN-30 wire, and the best one for IMD-19 wire having $\epsilon_{\text{irr}} = 0.55\%$.

It is due to different HT conditions (sheath softening) and grain connectivity – best for IMD.

Bending of MgB_2 wires made by EX, IN and IMD



30f **in-situ**, Ti/GlidCop (0.54 x 1.3 mm²) 19f **IMD**, Nb/Monel (0.5+0.2 x 3 mm²) 19f **ex-situ**, Ni/Cu

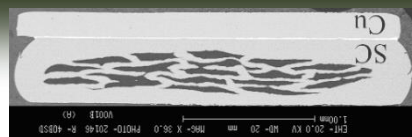


The observed differences in the **strain tolerances** among the **in-situ**, **ex-situ** and **IMD** wires are attributed to different **grain connectivity** and dominantly to mechanical **strengths of sheath materials** affected by the final heat treatment ($\sim 640^\circ\text{C}$ – **IMD**, $\sim 700^\circ\text{C}$ – **in-situ** and above 900°C – **ex-situ**).

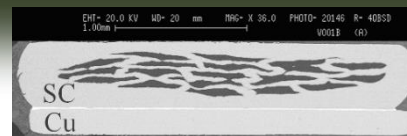
Bending of unsymmetrical MgB_2 conductor

In two directions:

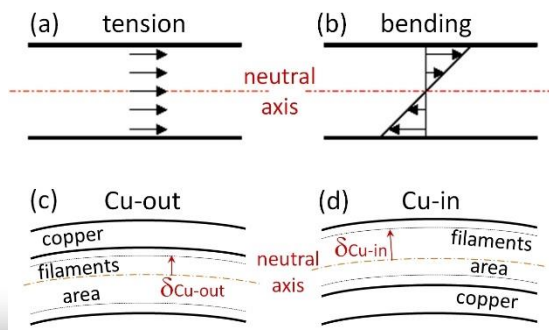
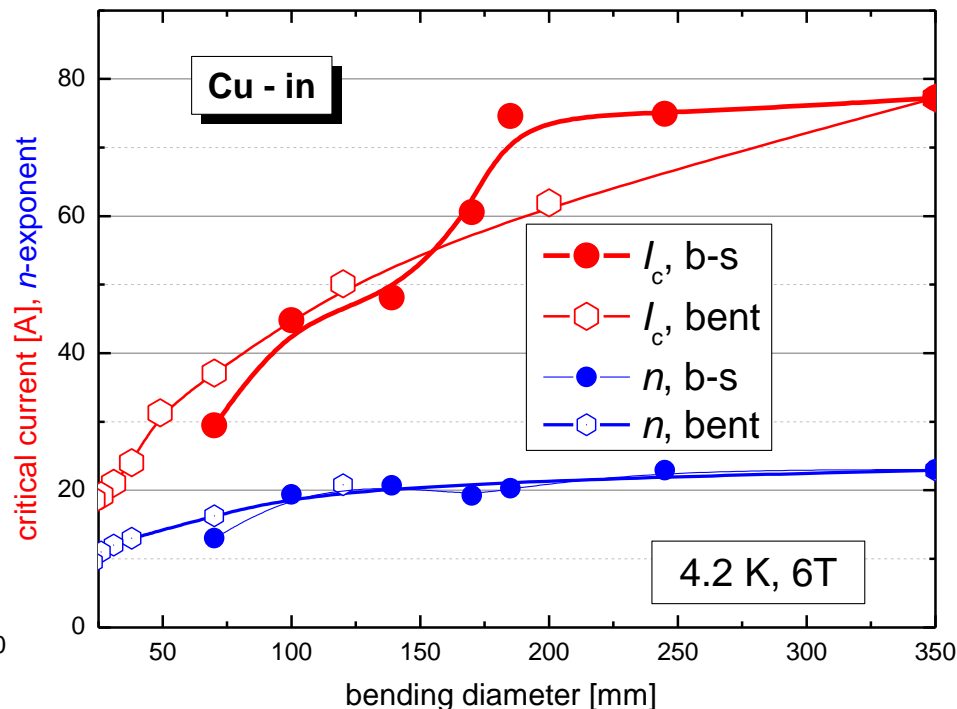
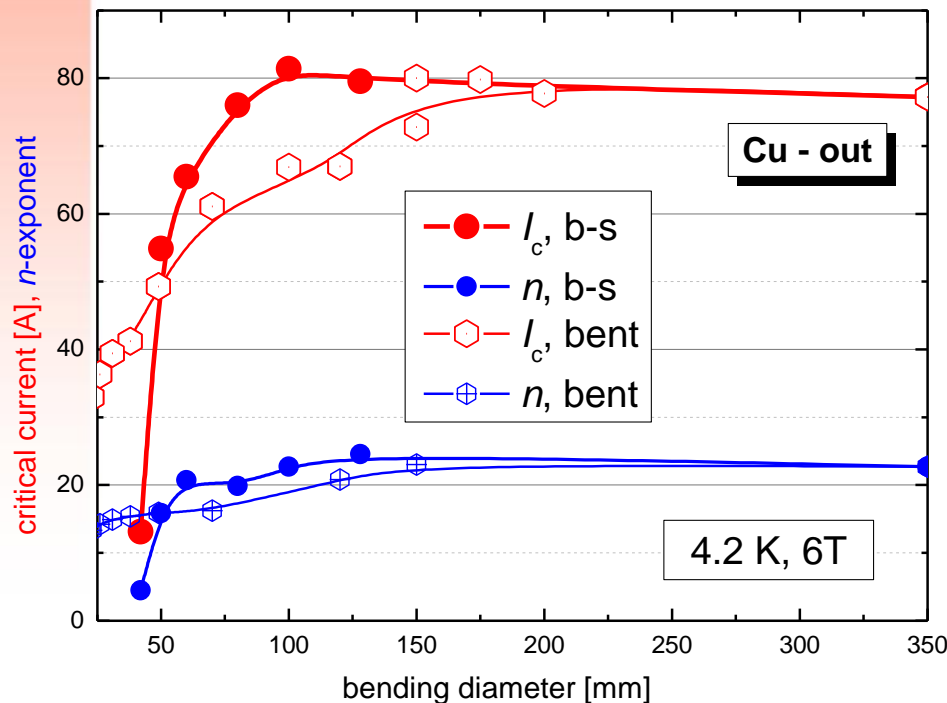
- direct bending – I_c
- straightening – I_c



out



in



I_c degradation is much lower for **Cu-out** (~175 mm) **Cu-in** (> 350 mm) .

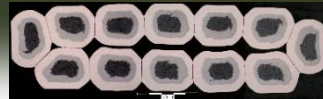
Tape straightening - attributed to partial stress release.

$$\varepsilon_b = h_f / (R_b + h_f)$$

Tolerances of MgB_2 cables to bending stress



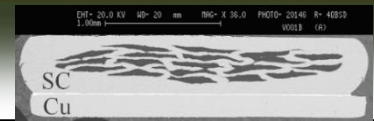
R1



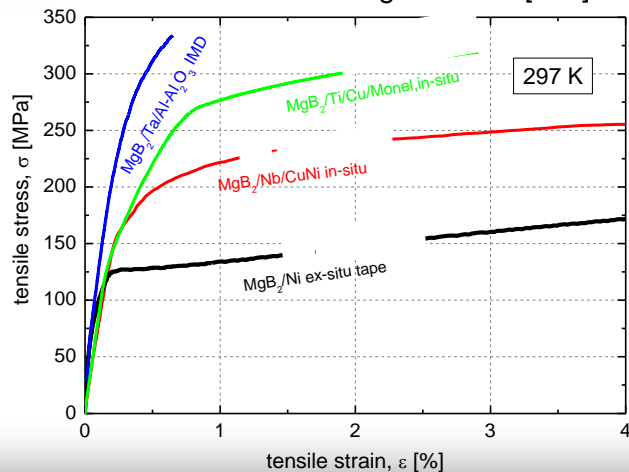
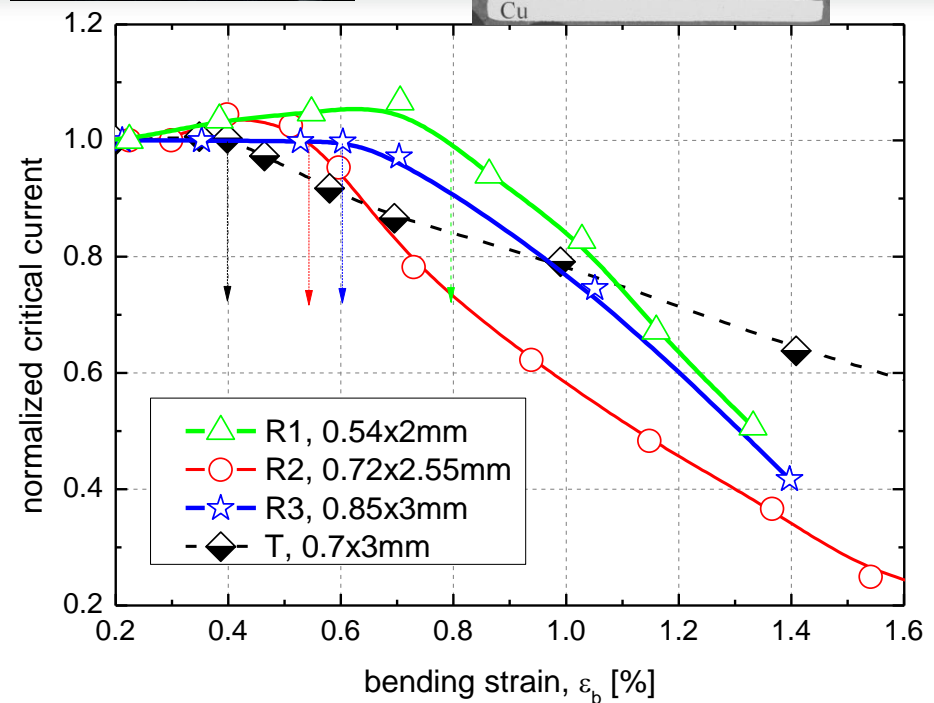
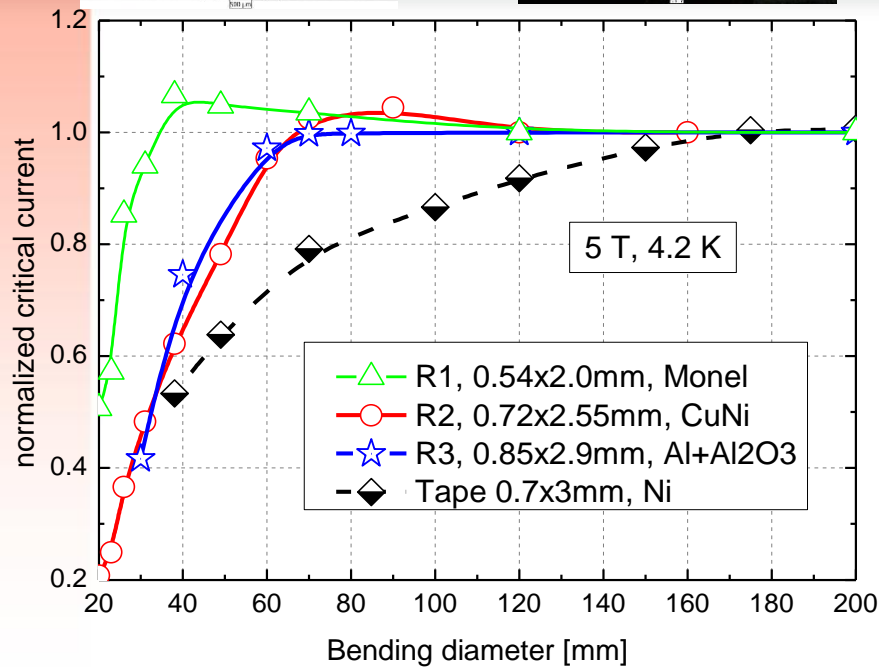
R2



R3



T



- **low I_c degradation** of Rutherford cable by bending **< 35 – 70 mm**. It correlates well with the strands diameter and $\sigma(\epsilon)$ of strands at room temperature
- **important for multi-pole generators – low diameter racetrack coil**

Conclusions

- Electro-mechanical characterization by **tension and bending** has been performed for MgB_2 conductors of different architecture and sheaths mat.
- It was found that **outer sheath has a dominant effect** on the stress tolerance of MgB_2 composite wire.
- The stress tolerance can be **improved by filament structure (W-addition, densified filaments** – rolling or swaging and by **IMD process**).
- MgB_2 conductors made by **IMD shows the best strain tolerance** (by **tension and bending**).
- **Twisting** of filaments is changing the **residual stress** inside the wire and consequently the **strain tolerance is decreased**, but I_c values are **increased**.
- Rutherford **MgB_2 cables** offers the **lowest bending diameters** after HT, which can be interesting for winding of multi-pole motors/generators.