



# Characterization of irreversible degradation of Nb<sub>3</sub>Sn Rutherford cables due to transversal compression stress at room temperature

*Patrick Ebermann<sup>1,2</sup>, Jerome Fleiter<sup>1</sup>, Oleg Kalouguine<sup>1</sup>, Dietmar Meinel<sup>4</sup>, Giuseppe Peiro<sup>1</sup>, Christian Scheuerlein<sup>1</sup>, Daniel Schörling<sup>1</sup>, Davide Tommasini<sup>1</sup>, Felix Wolf<sup>1,3</sup>, Friedrich Lackner<sup>1</sup> and Michael Eisterer<sup>2</sup>*

<sup>1</sup>European Organisation for Nuclear Research (CERN), Geneva, Switzerland

<sup>2</sup>TU Wien, Vienna, Austria

<sup>3</sup>TU Bergakademie Freiberg, Freiberg, Germany

<sup>4</sup>Federal Institute for Materials Research and Testing (BAM), Berlin, Germany



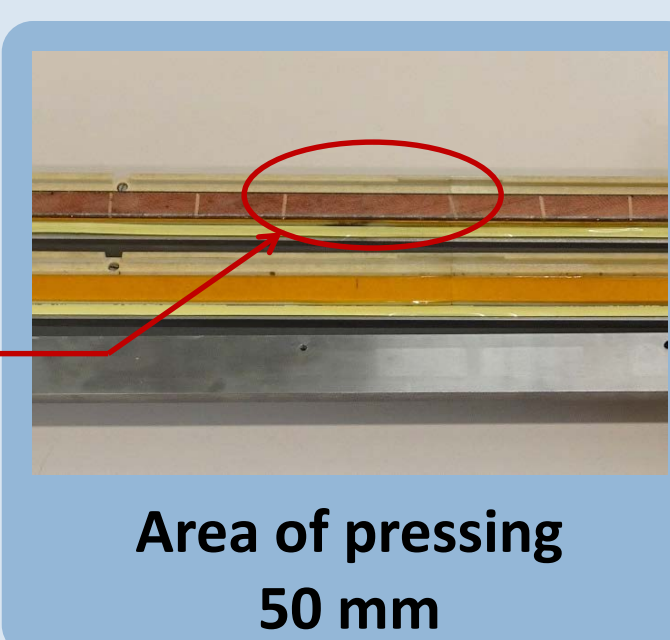
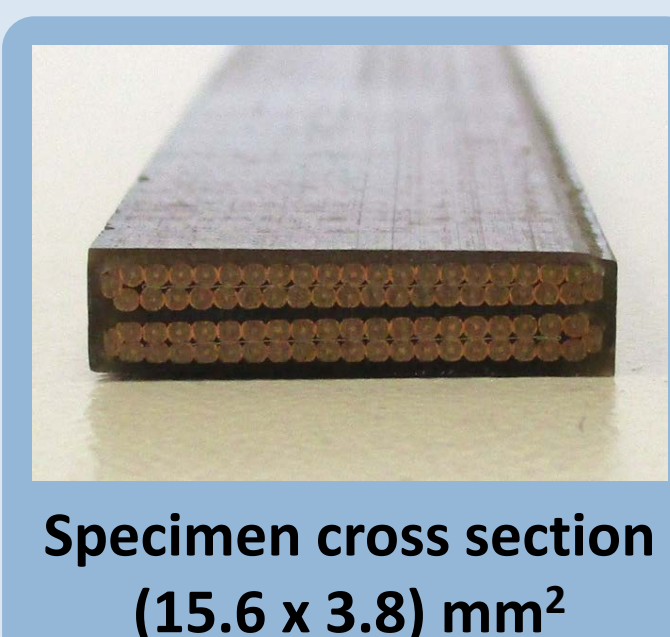
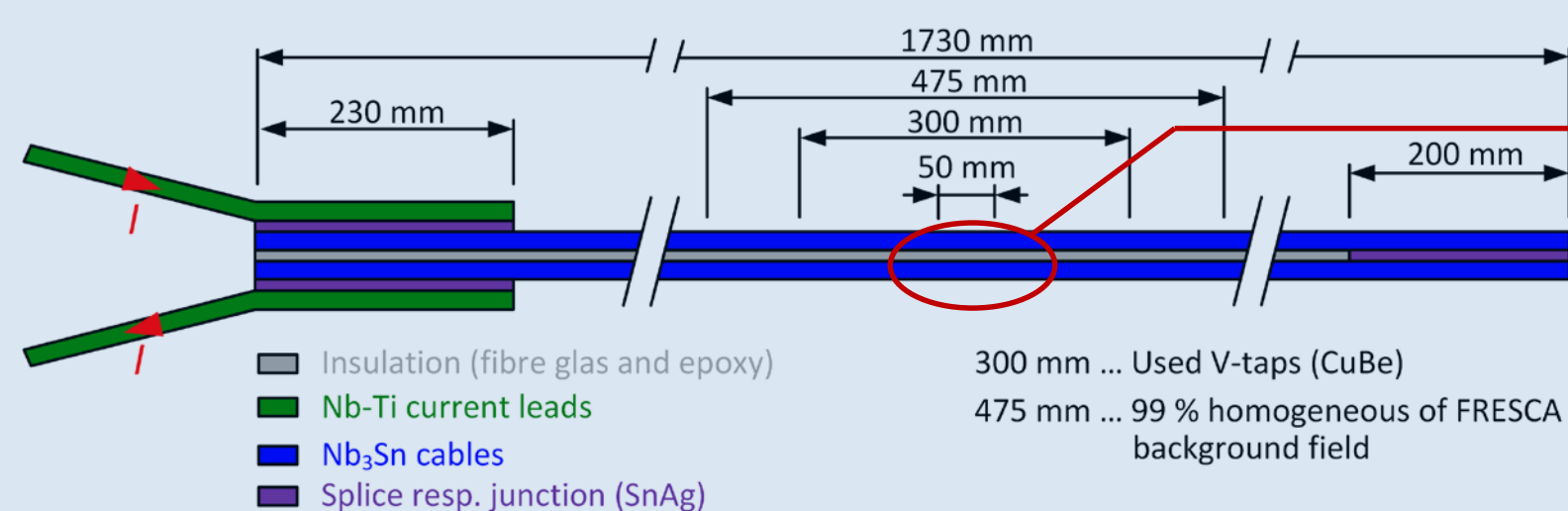
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## 1) Introduction and objectives

- In the frame of the **Future Circular Collider (FCC)** design study for a 100 TeV circular hadron collider, **16 T superconducting bending magnets** based on Nb<sub>3</sub>Sn technology are being developed. In order to counteract the Lorentz force during operation, **pre-stress is applied on the superconductive coil during the magnet assembly at room temperature** [1]. The electrical properties of the **brittle** Nb<sub>3</sub>Sn superconductor are **strain sensitive** and an excessive pre-stress leads to an **irreversible degradation of the superconductor**.
- The behaviour of Nb<sub>3</sub>Sn, -strands and -cables **during stress at cryogenic temperature** have already been investigated, however the irreversible performance loss of recently developed superconducting cables have not been measured [2]-[5]. The knowledge of the described pre-stress was missing and the **limitation within the community was set to 150 MPa** [6].
- In order to **determine the level of acceptable pre-stress during the magnet assembly process**, reacted and impregnated Nb<sub>3</sub>Sn cables were exposed to **compressive transversal stress up to 200 MPa at room temperature**. After each stress cycle the critical current of the cable specimens was characterized at **4.2 K in the FRESKA test station**.

## 2) FRESKA-compatible specimen

- Two insulated 1.73 m long Rutherford cables
- Cables are **impregnated and keystone-compensated** arranged to reach a planar surface for transversal stress application.
- Top ends** connected to Nb-Ti current leads.
- Bottom ends soldered** together, i.e. the current flows anti-parallel.
- Area of pressing** was chosen in the centre of the specimen between the inner voltage-taps to ensure a homogenous field condition over the measured area.



Strand properties	
Type	RRP 144/169
Cu to Non-Cu volume ratio	1.08
Strand diameter	0.7 mm
Sub-element diameter	41 µm
Sub-element pitch length	14 mm
Virgin RRR	150
Virgin I <sub>c</sub> (4.2 K, 12 T)	438 A
Heat Treatment	48 h / 210 °C
	48 h / 400 °C
	50 h / 650 °C

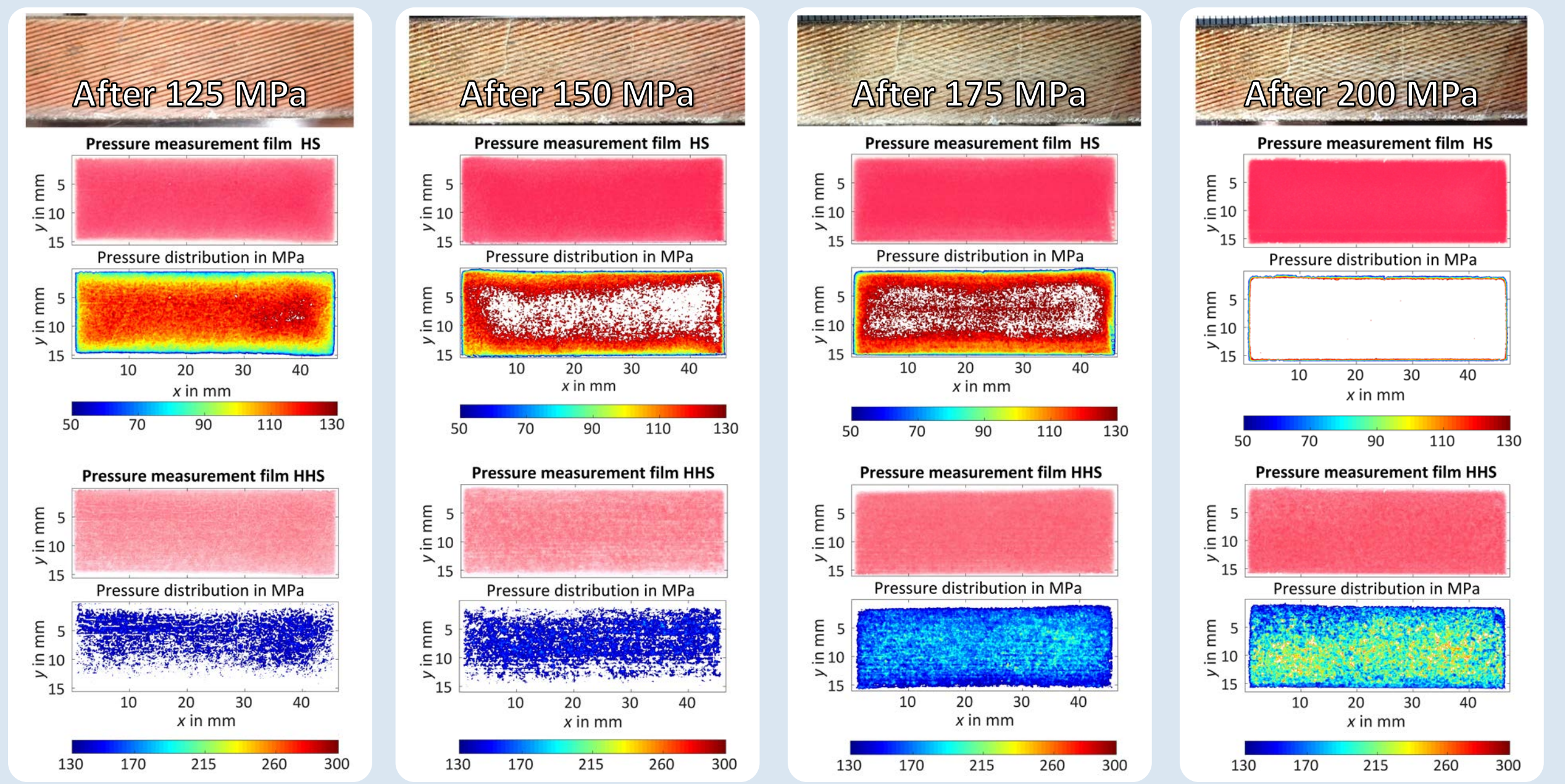
Cable properties	
Cable ID	HT15OC0190
No. of strands	40
Transposition pitch	100 mm
Keystone angle	0.79°
Mid-thickness	1.25 mm
Width	14.7 mm
Insulation	S-2 glass braided
Core	316L
Packing factor	87.3%
Epoxy	CTD-101K

## 3) Application of transversal compressive stress

- Iterative compression **at room temperature** before every I<sub>c</sub> measurement.
- Applying stress by using a hydraulic press and evaluation of force with load cells [7].
- Evaluation of **pressure distribution and optimization of the alignment** parameters by using FUJIFILM Prescale© [8].
- Self-written **image processing software** for evaluation of the Prescale films.
- Multilayer of Prescale films of different sensitivity was used to **cover a higher range**.



Stress application results			
Nominal	$\sum_{i=1}^4 F_{LoadCell,i}$	$\bar{\sigma}_{LoadCells}$	
kN	MPa	kN	MPa
68.64	100	68.93	100.42
85.80	125	85.24	124.18
102.96	150	103.00	150.06
120.12	175	118.00	171.91
137.28	200	137.90	200.90



## 4) I<sub>c</sub> measurement with FRESKA

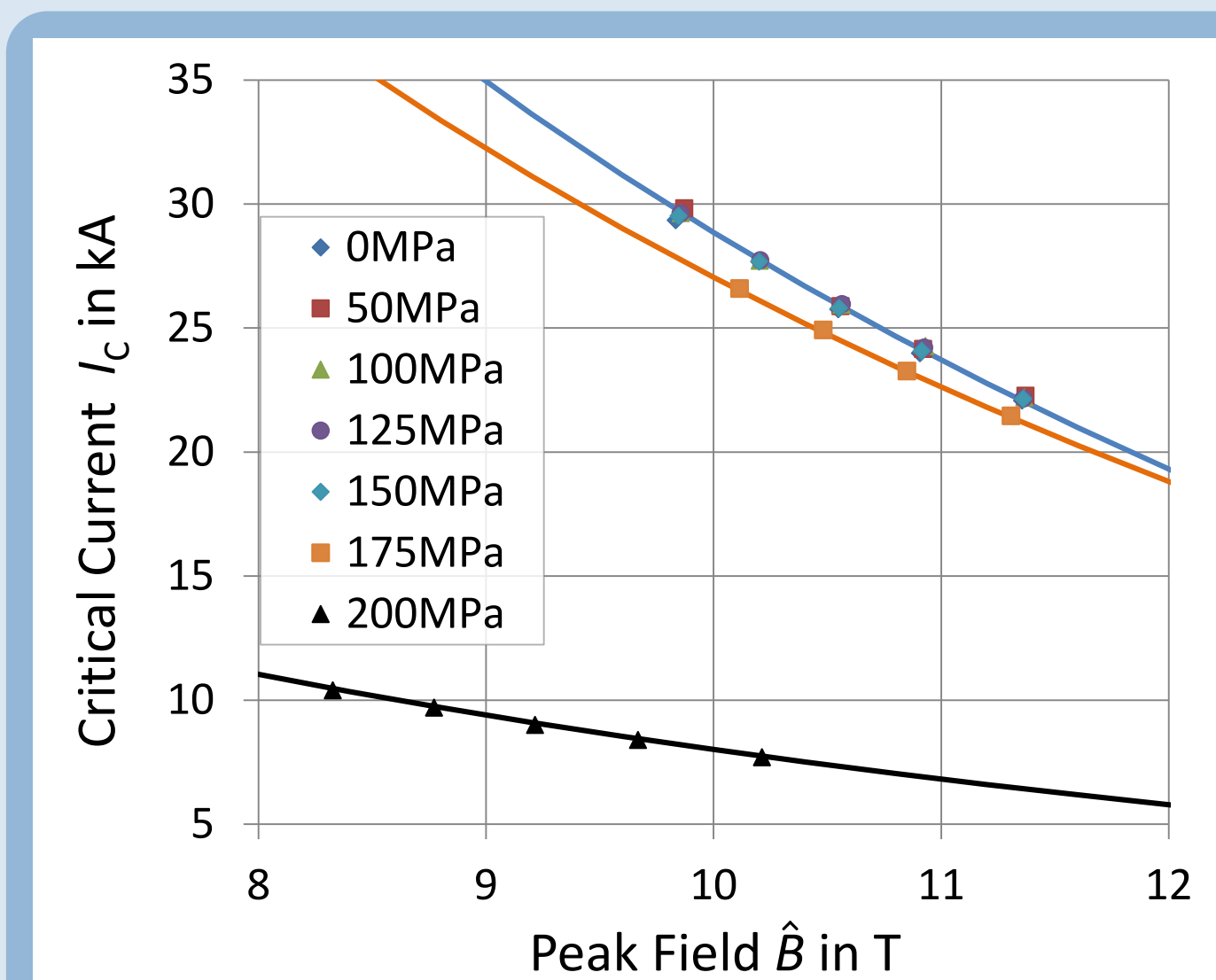
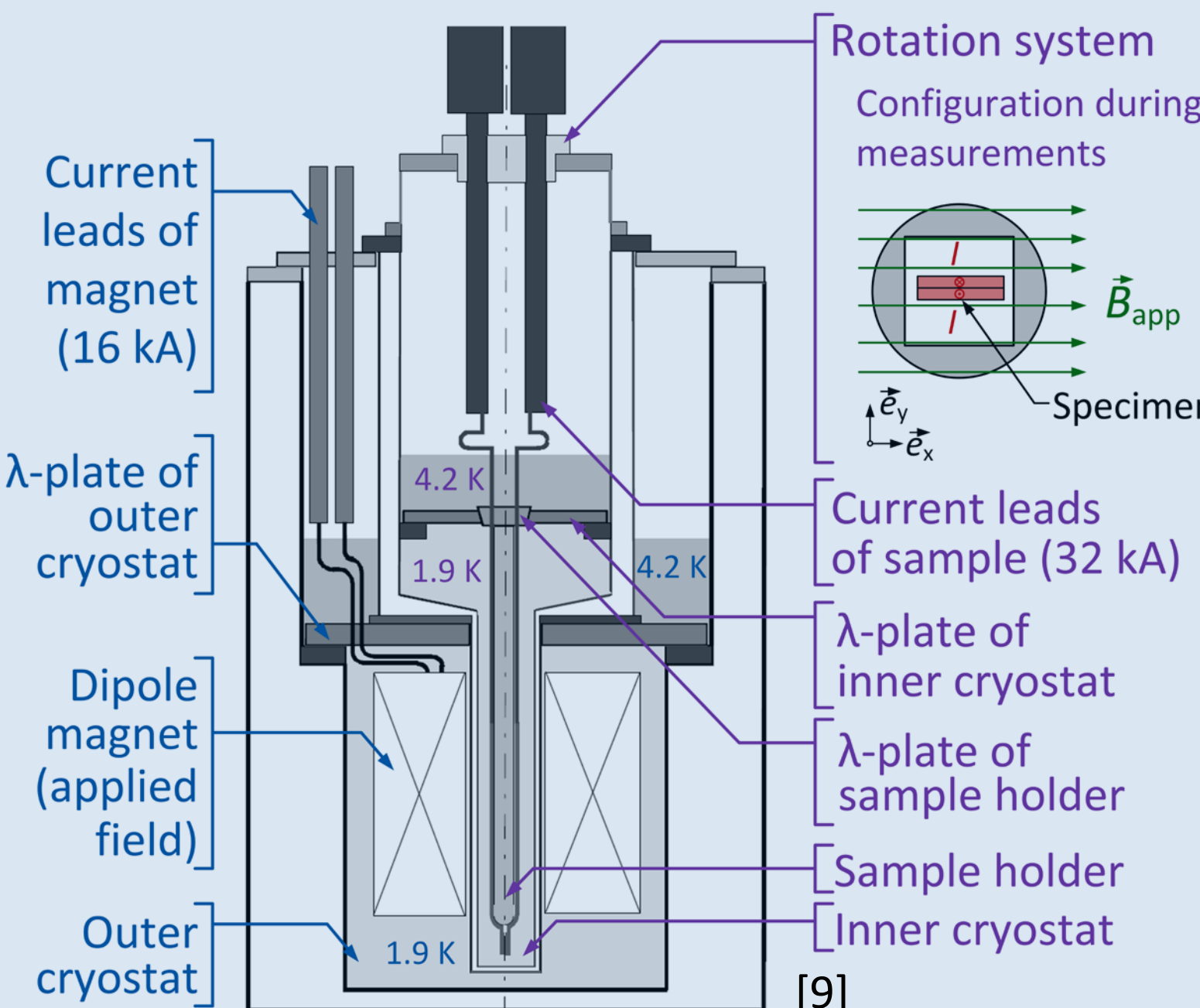
Subsequent to each stress application, the sample was transferred to the FRESKA test station for I<sub>c</sub> measurement.

- FRESKA test station properties [9]
  - Operation current I ≤ 32 kA (70 kA)
  - Applied dipole field B<sub>app</sub> ≤ 9.6 T** with self-field up to 12 T
  - Homogeneity B<sub>app</sub>/max(B<sub>app</sub>) ≥ 99 % over a length of 475 mm
- Measurement condition** of specimen
  - Applied field **B<sub>app</sub> = 7 ... 9.6 T**
  - Temperature **T = 4.2 K**
  - B<sub>app</sub> parallel to large surface of specimen and perpendicular to the current.**
  - Both cables were measured and show the same degradation behaviour.

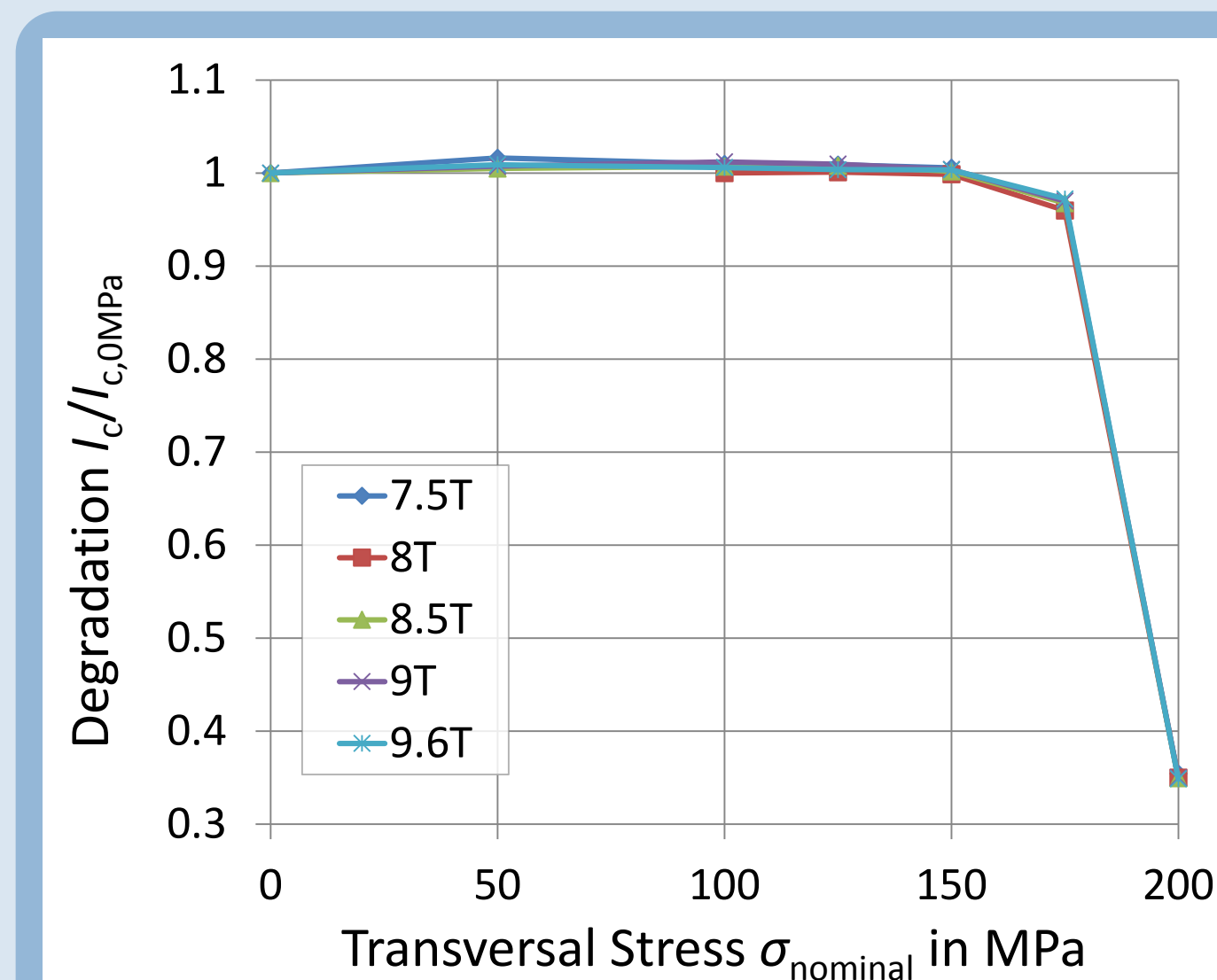
I <sub>c</sub> measurement results at max(B <sub>app</sub> ) = 9.6 T and T = 4.2 K			
$\sigma_{nominal}$ MPa	I <sub>c</sub> * kA	I <sub>c</sub> /I <sub>c,0MPa</sub>	n**
0	22.1	1.00	60
50	22.2	1.00	60
100	22.2	1.00	60
125	22.2	1.00	60
150	22.2	1.00	60
175	21.45	0.97	60
200	7.9	0.36	<10

\* I<sub>c</sub> := I(E = E<sub>c</sub> := 3 µV m<sup>-1</sup>)

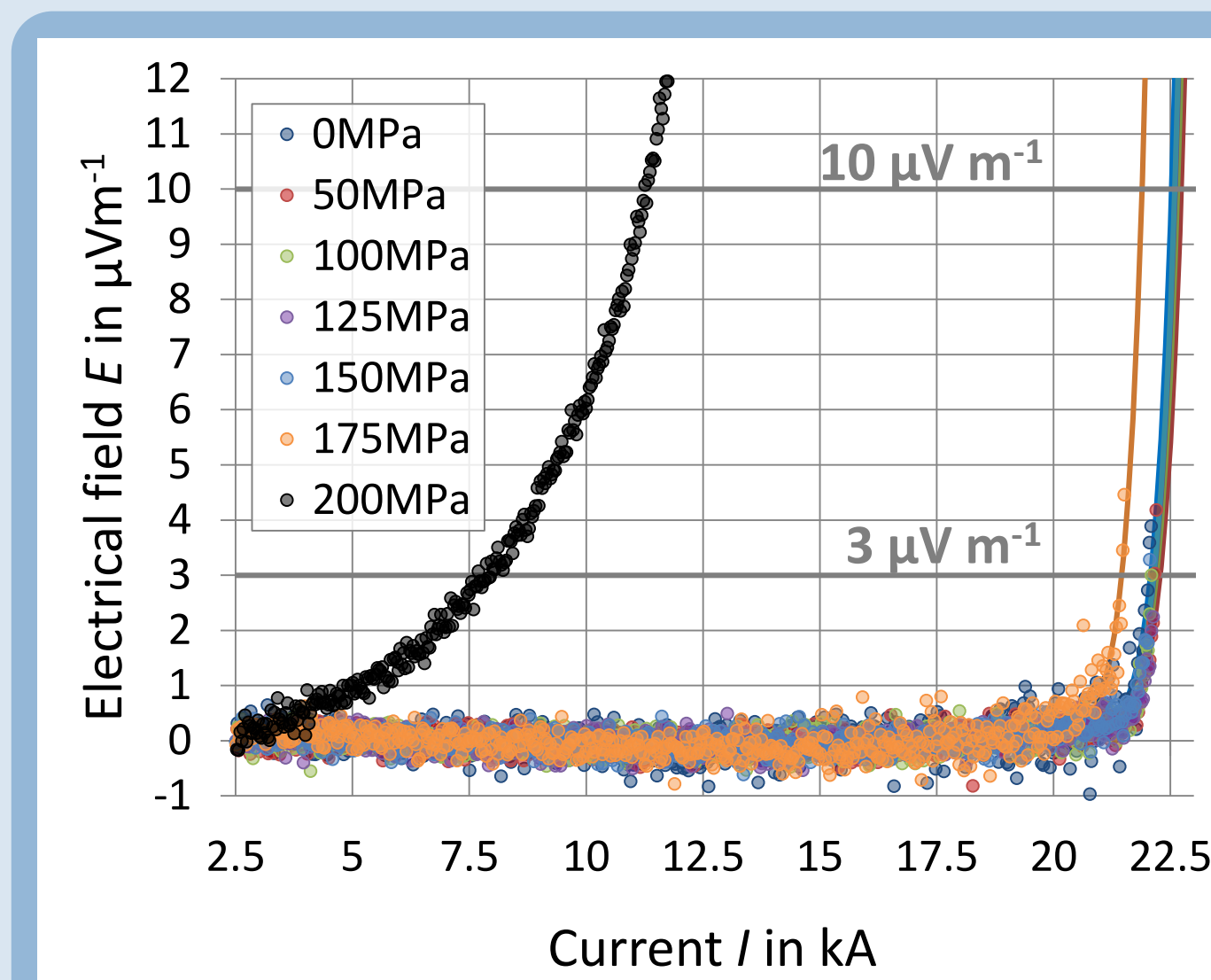
\*\* E/E<sub>c</sub> := (I/I<sub>c</sub>)<sup>n</sup>, [10]



Results of I<sub>c</sub> measurement at T = 4.3 K



Degradation of I<sub>c</sub> at T = 4.2 K (E<sub>c</sub> = 3 µV m<sup>-1</sup>)

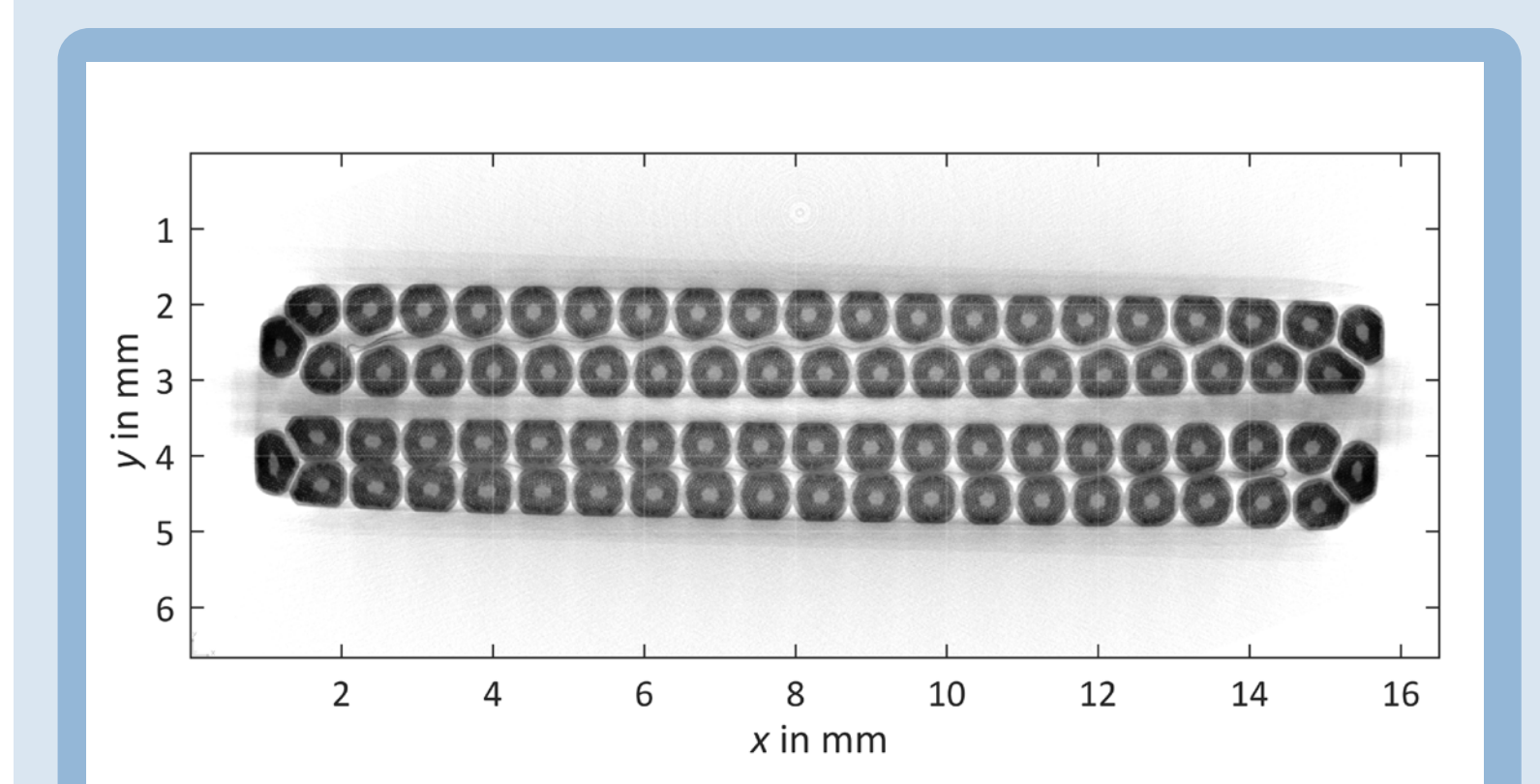


E(I)-comparison at max(B<sub>app</sub>) = 9.6 T

## 5) Post analysis

Investigation carried out after the measurement campaign

- Visual inspection** of the pressing area shows on both sides several cracks of the epoxy resin matrix. An average **displacement in height of 0.02 mm** due to transversal stress was measured.
- Non-destructive 3D X-ray tomography** by the Federal Institute for Materials Research and Testing (BAM) was carried out to identify damages caused by compressive stress.
  - Plastic cable deformations caused by the applied stress are not revealed by the results. The spatial resolution is insufficient to visualise cracks in the expected Nb<sub>3</sub>Sn filaments.
  - Deformations on the edges of the cables and displacements of the core due to specimen manufacturing could be detected independent of the pressing. This leads to a degradation less than 5 % compared to the strand performance [11].



X-ray Tomography Snap Shot

## 6) Conclusion

In order to **determine the limits of pre-stress on the coil assembly for future high field accelerator magnets**, a FRESKA-compatible Rutherford cable specimen was **exposed to transversal compressive stress at room temperature** and subsequently the **critical current at 4.2 K was quantified** in several iterations up to 200 MPa.

A hydraulic press was used to perform transversal compressive stress and the **pressure distribution on the specimen was optimized**. The critical current was measured in the FRESKA test station.

The results show a **small I<sub>c</sub> degradation after an applied stress of 175 MPa** and the specimen was **heavily damaged after the 200 MPa cycle**. The critical current I<sub>c</sub> and the resistive transition index n of the cable are reduced, which is caused by higher current sharing. This measurement campaign also confirms that the **common pre-stress of 150 MPa does not cause any degradation** of the coil based on this study.

First visual inspection has shown **cracks and deformation on the surface of the epoxy resin**. Plastic deformation on the strand level and cracks in the impregnation layer or superconductive sub-elements could not be identified by the X-ray tomography.

## 7) Outlook

Additional to the non-destructive tomography a **microscopy** of the specimen is ongoing to understand the impact of transversal stress during room temperature on the superconductive Rutherford cable, especially the Nb<sub>3</sub>Sn sub-elements of the strands.

**Further campaigns** with cables from the 11 T- and the MQXF- programme have been organised to confirm the observed results.

## 8) Acknowledgements

The main author would like to thank **his working group** for the great support as well as the colleagues from the SCD section, especially the **FRESKA team** for providing the specimen and performing the I<sub>c</sub> measurements.

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