

# A comparative study of new Ge-based additions to MgB<sub>2</sub>

National Institute  
of Materials Physics

D. Batalu<sup>1</sup>, S. Ciuca<sup>1</sup>, R.E. Dumitrescu<sup>1</sup>, G. Aldica<sup>2</sup>, S. Popa<sup>2</sup>, M. Burdusel<sup>2</sup>, P. Badica<sup>2</sup>  
Contact: dan\_batalu@yahoo.com

<sup>1</sup> Metallic Materials Science, Physical Metallurgy, University POLITEHNICA of Bucharest, Romania

<sup>2</sup> Lab. of Magnetism and Superconductivity, National Institute of Materials Physics, Magurele, Romania

## A. Background

Different Ge-based additions [1-4] were shown to improve the superconducting functional characteristics of MgB<sub>2</sub>, such as the critical current density ( $J_c$ ) and the irreversible magnetic field ( $H_{irr}$ ). Additives of Ge, GeO<sub>2</sub>, Ge(OM), GeTe were tested and results are compared. All samples show a critical onset temperature of 37-39 K. For each additive we obtained an optimum starting composition ( $x=0.005$  for Ge and GeO<sub>2</sub> additions, and  $x=0.0014$  for Ge<sub>2</sub>C<sub>6</sub>H<sub>10</sub>O<sub>7</sub> addition).

### REFERENCES:

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[2] Dan Batalu, G. Aldica, M. Burdusel, S. Popa, M. Enculescu, I. Pasuk, D. Miu, P. Badica, *J Supercond Nov Magn* (2015) 28:531–534.

[3] Dan Batalu, G. Aldica, S. Popa, A. Kuneser, V. Mihalache, P. Badica, *Solid State Sciences* (2015) 48:23–30.

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## B. Experimental

Raw powders and compositions were:

(1) MgB<sub>2</sub> (99.5 % purity, 1-2  $\mu$ m, decomposition temperature  $T_{dec}=890$  °C, Alfa Aesar), Ge (99.999 % purity, 152  $\mu$ m, melting point  $T_m=937$  °C, Alfa Aesar): MgB<sub>2</sub>Ge<sub>x</sub>,  $x=0$  (a), 0.005 (b), 0.01 (c), and 0.03 (d) (Fig. 1).

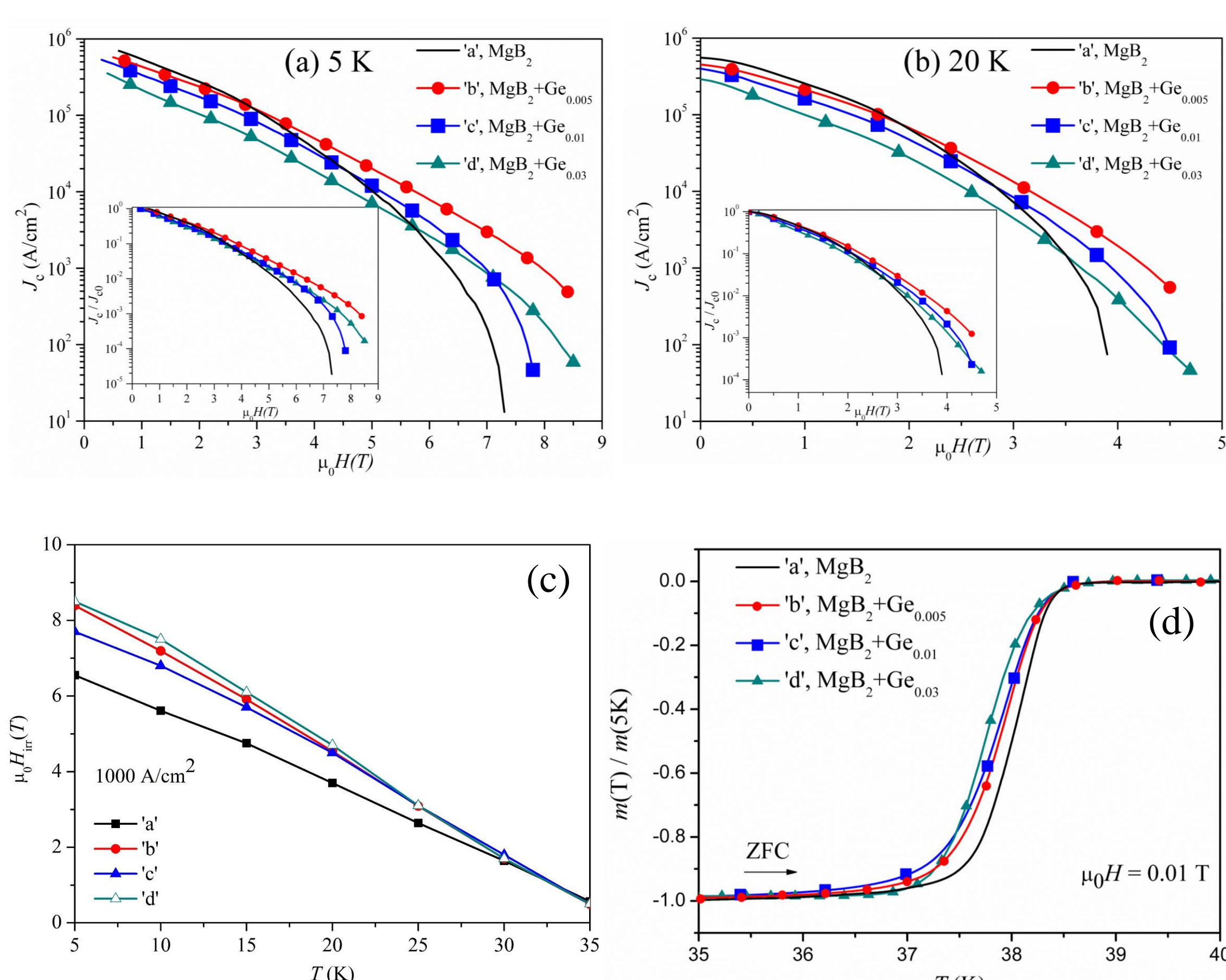
(2) MgB<sub>2</sub> (Alfa Aesar), GeO<sub>2</sub> (99.999%, 14.3  $\mu$ m,  $T_m=1086$  °C, Alfa Aesar): MgB<sub>2</sub>(GeO<sub>2</sub>)<sub>x</sub>,  $x=0$  (A), 0.005 (B), 0.01 (C), and 0.03 (D) (Fig. 2).

(3) MgB<sub>2</sub> (Alfa Aesar), Ge<sub>2</sub>C<sub>6</sub>H<sub>10</sub>O<sub>7</sub> (99.7%,  $T_{dec}=320$  °C, Alfa Aesar): MgB<sub>2</sub>(Ge<sub>2</sub>C<sub>6</sub>H<sub>10</sub>O<sub>7</sub>)<sub>x</sub>,  $x=0$  (MgB<sub>2</sub>), 0.0007 (GEP0.7), 0.0014 (GEP1.4), 0.0025 (GEP2.5), 0.005 (GEP5), 0.015 (GEP1.5) (Fig. 3).

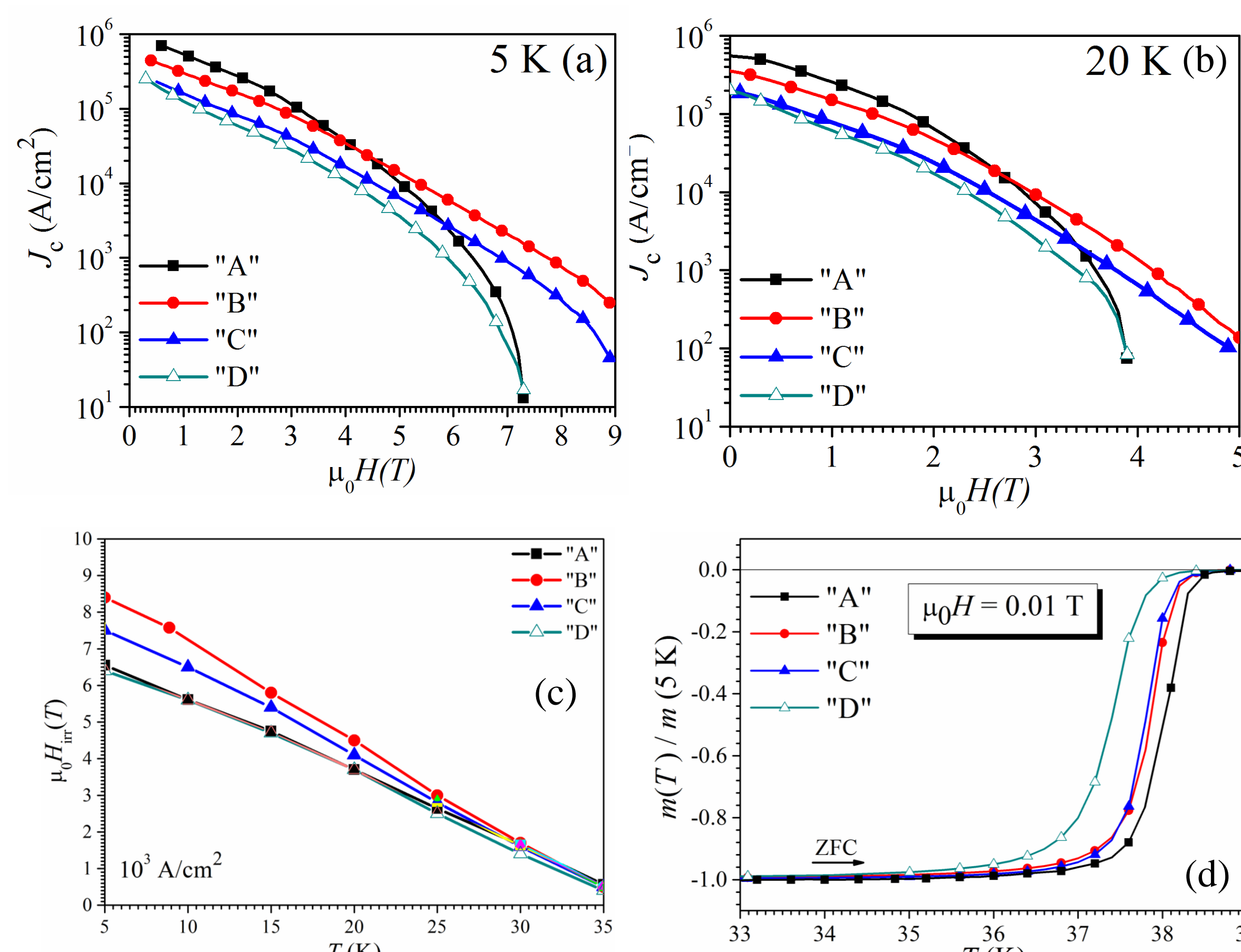
(4) MgB<sub>2</sub> (Pavezyum, Turkey, purity >95%, particle size ~30 nm, product number PVZ-nMgB<sub>2</sub>-008), OM=Ge<sub>2</sub>C<sub>6</sub>H<sub>10</sub>O<sub>7</sub> (RGe\* – Alfa Aesar, 99.7%,  $T_{dec}=320$  °C), RGe\*\* (Asai Germanium Research Institute, Japan), PGe (Asai), SP (Asai): MgB<sub>2</sub>(OM)<sub>0.0028</sub> (Fig. 4).

(5) MgB<sub>2</sub> (Pavezyum), crystalline GeTe, amorphous Ge20Te80 (synthesized at University of Pardubice, Faculty of Chemical Technology, Czech Republic): MgB<sub>2</sub>(Ge<sub>x</sub>Te<sub>1-x</sub>)<sub>0.01</sub>,  $x=0.5$ , and 0.2 for GeTe, and Ge20Te80, respectively (Fig. 5).

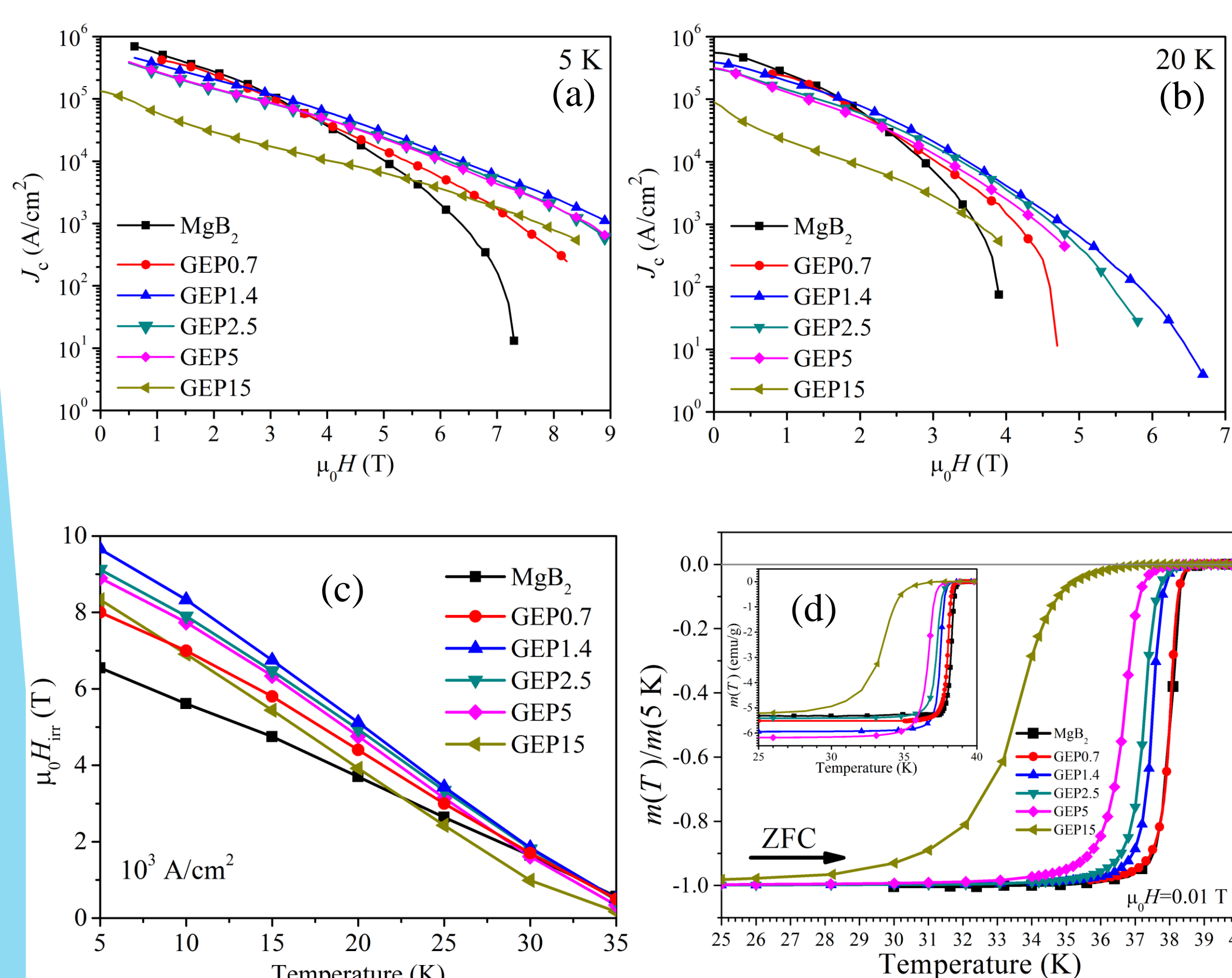
The powders were mixed in an agate mortar and sintered by SPS (FCT Systeme GmbH – HP D 5, Germany) for 3 minutes at 1150 °C, under a uniaxial pressure of 95 MPa. During sintering secondary phases formed: MgO, MgB<sub>4</sub>, Mg<sub>2</sub>Ge, and MgTe.  $J_c$  (Fig. 1-5.a, b) was calculated based on Bean relation, and by using  $m(H)$  loops.  $H_{irr}$  was determined for the  $10^2$  A/cm<sup>2</sup> criterion (Fig. 1-5.c).  $m(T)$  measurement are presented in Figs. 1d-5.d.



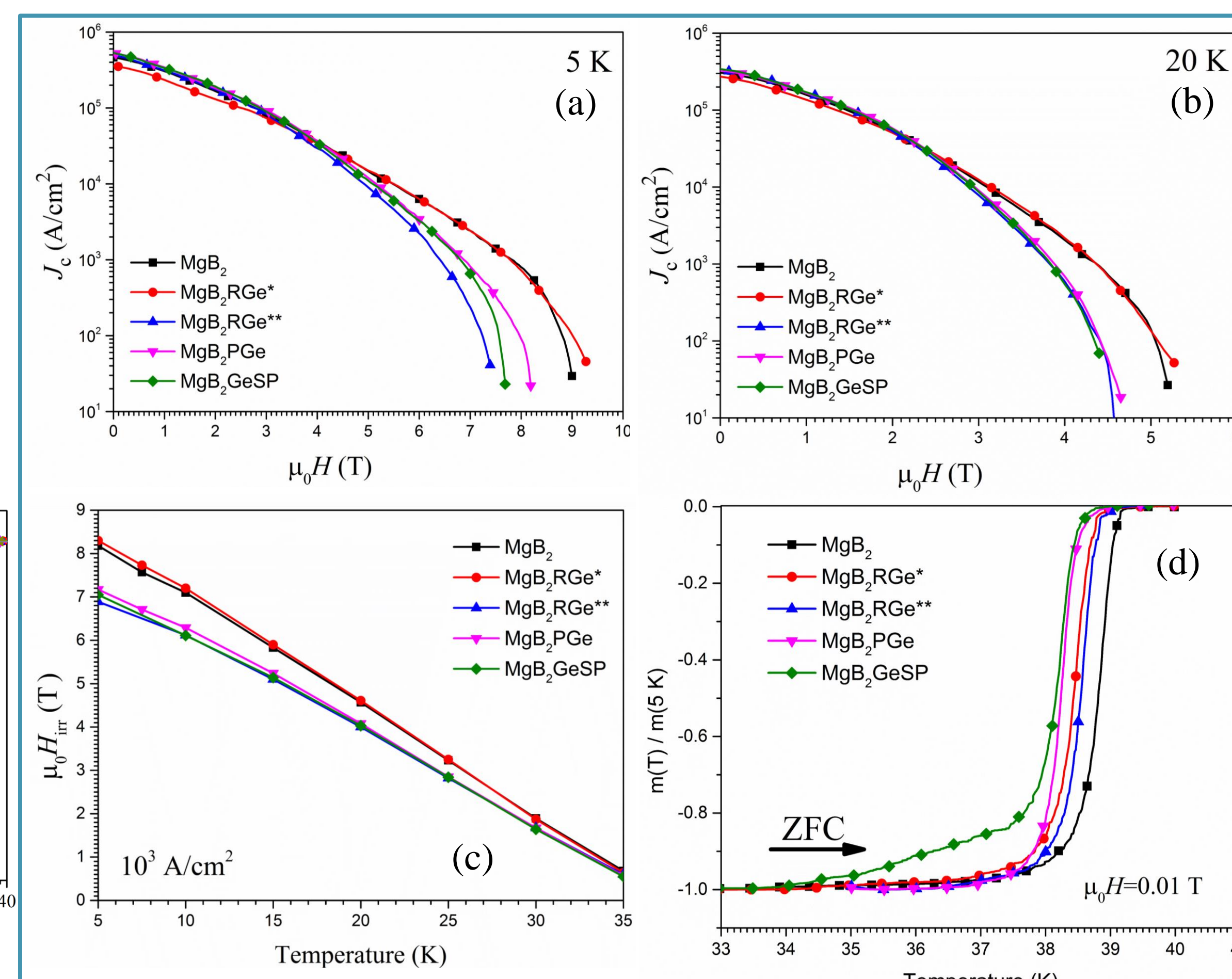
**Fig. 1.** Critical current density vs. magnetic field of SPSed samples at 5 K (a) and 20 K (b),  $H_{irr}$  (c), and ZFC  $m(T)/m(5K)$  (d) for Ge added MgB<sub>2</sub> (Alfa Aesar) [1, 2].



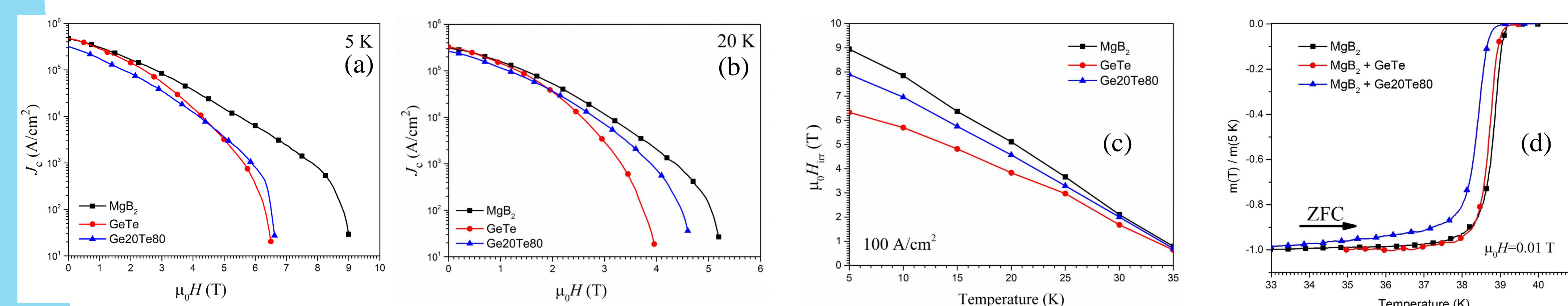
**Fig. 2.** Critical current density vs. magnetic field of SPSed samples at 5 K (a) and 20 K (b),  $H_{irr}$  (c), and ZFC  $m(T)/m(5K)$  (d) for GeO<sub>2</sub> added MgB<sub>2</sub> (Alfa Aesar) [1, 3].



**Fig. 3.** Critical current density vs. magnetic field of SPSed samples at 5 K (a) and 20 K (b),  $H_{irr}$  (c), and ZFC  $m(T)/m(5K)$  (d) for Ge<sub>2</sub>C<sub>6</sub>H<sub>10</sub>O<sub>7</sub> added MgB<sub>2</sub> (Alfa Aesar) [1, 4].



**Fig. 4.** Critical current density vs. magnetic field of SPSed samples at 5 K (a) and 20 K (b),  $H_{irr}$  (c), and ZFC  $m(T)/m(5K)$  (d) for OM added MgB<sub>2</sub> (Pavezyum).



**Fig. 5.** Critical current density vs. magnetic field of SPSed samples at 5 K (a) and 20 K (b),  $H_{irr}$  (c), and ZFC  $m(T)/m(5K)$  (d) for GeTe added MgB<sub>2</sub> (Pavezyum).

## C. Conclusions

1. Dense samples of MgB<sub>2</sub>, with relative density higher than 91 % were obtained by *ex-situ* SPS using different Ge based additions.
2. The best  $J_c$  was obtained for GEP1.4 (Fig. 3. a, b), followed by GEP2.5, MgB<sub>2</sub>Ge<sub>0.005</sub> (Fig. 1. a, b), and MgB<sub>2</sub>(GeO<sub>2</sub>)<sub>0.005</sub> (Fig. 2. a, b).
3. The quality of pristine MgB<sub>2</sub> raw powder strongly influences the addition effect. When using two types of MgB<sub>2</sub> powders (produced by different companies), the efficiency of the addition in enhancing  $J_c$  can strongly change. In fact the effect of the additive vanishes for a raw MgB<sub>2</sub> powder rich in carbon. Samples made from the C-rich MgB<sub>2</sub> raw powder show enhanced  $J_c$  (Fig. 4, 5. a, b).
4. Considering the observations from 3, on the the influence of GeTe, Ge20Te80, and OM (Fig. 4, 5) we plan fabrication of added samples using a C-free raw powder.

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