



**UNIVERSITÉ
DE GENÈVE**

FACULTÉ DES SCIENCES

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Electromechanical Properties of Technical Superconductors

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Department of Quantum Matter Physics
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Outline

Forces, stress and strain in a magnet

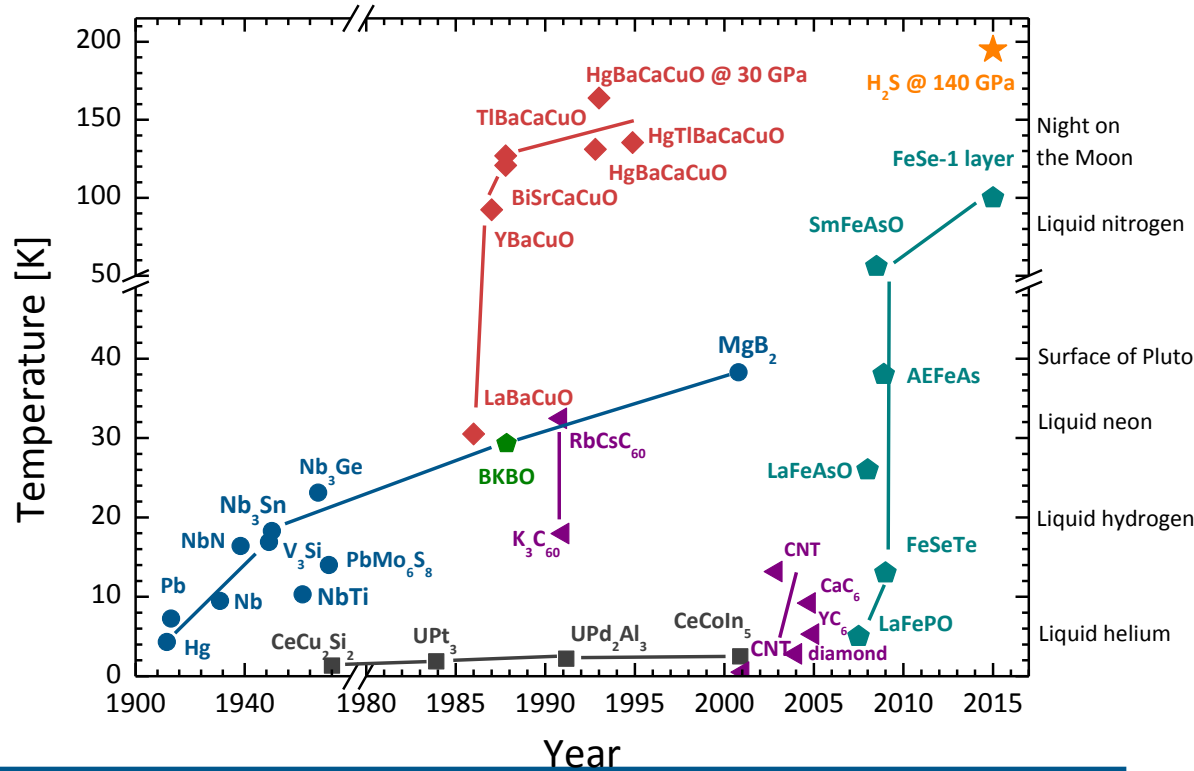
Technical superconductors under mechanical loads

- ***Reversible vs. irreversible effects***
- ***Mechanisms behind the irreversible degradation of the critical current***
- ***Intrinsic vs. extrinsic effects***

Focus on two materials: Nb₃Sn and YBCO

From superconducting materials...

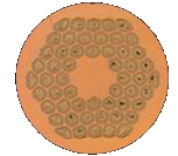
...to technical superconductors



NbTi



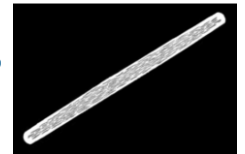
Nb₃Sn



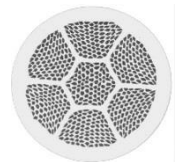
MgB₂



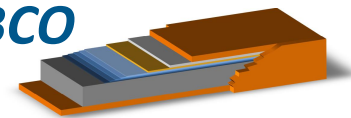
Bi2223



Bi2212



YBCO



- | | |
|------------------------------------|--------|
| 1. Superconducting ? | 10'000 |
| 2. $T_c > 4.2K$ & $B_{c2} > 10T$? | 100 |
| 3. $J_c > 1000 A/mm^2$? | ~10 |

Operate at **high current density** is a necessary condition for applications, but it is **not sufficient**

Other crucial requirements:

- **Have high tolerance to stress** *Magnetic forces*
- **Be safe in case of magnet quench** *Quench detection, NZPV*
- **Have low magnetization** *Applications to NMR, MRI, HEP magnets*
- **Have a persistent joint technology** *Applications to NMR, MRI*

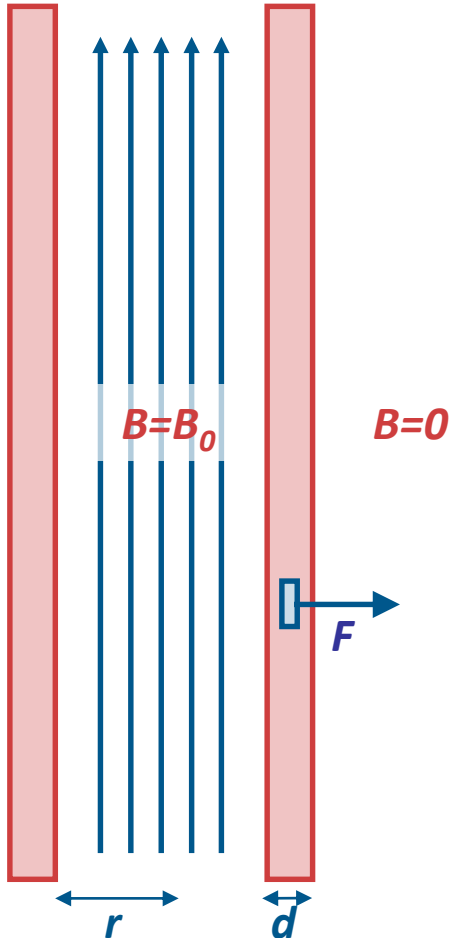
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Introduction to Forces and Stresses in a Magnet

Infinite solenoid
thin wall



The magnetic field is $B_0 = \mu_0 J d$

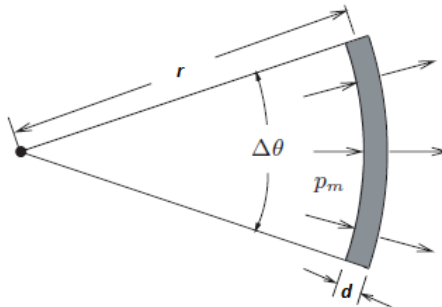
The expression of the magnetic force density (per unit of volume) is

$$f = J \times B$$

The average field in the winding is $\frac{B_0}{2}$

The (radial) force on a winding volume element is

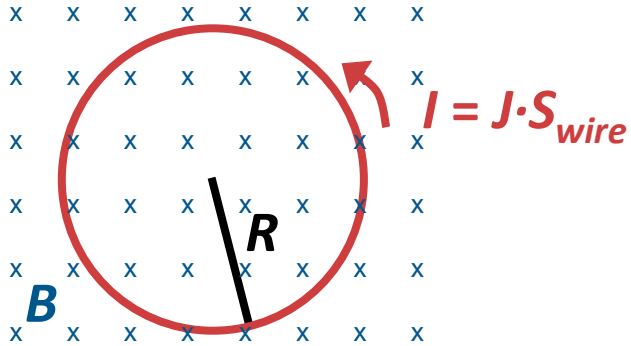
$$\begin{aligned} \Delta F &= J \frac{B_0}{2} \Delta v = J \frac{B_0}{2} r \Delta \theta \Delta z d = \frac{B_0^2}{2\mu_0} r \Delta \theta \Delta z \\ &= \frac{B_0^2}{2\mu_0} \Delta s = p_m \Delta s \end{aligned}$$



$$p_m(B_0 = 10T) = 400 \text{ bar}$$

Hoop stress in a ring

A ring carrying a current I in a field B

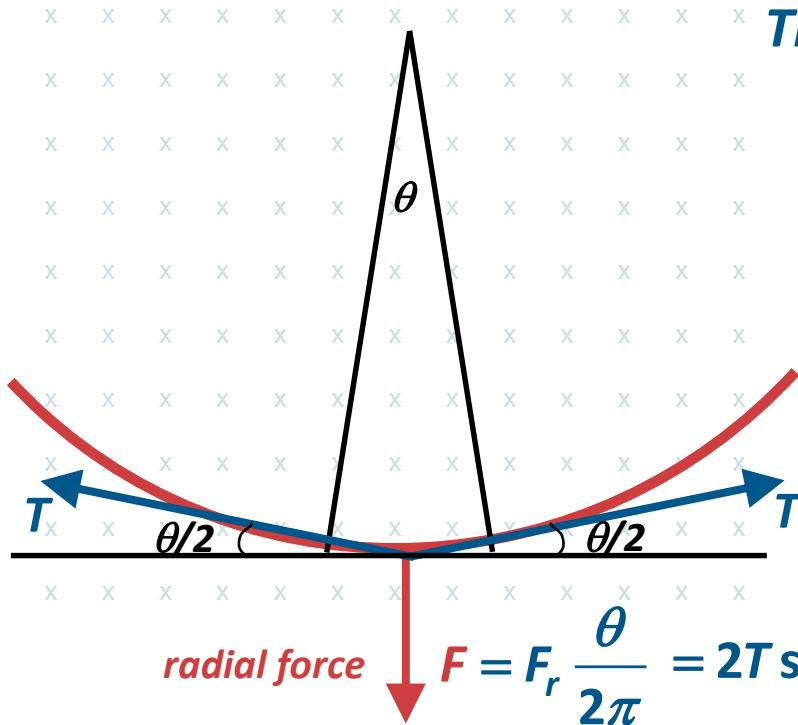


The total radial magnetic force on the ring is

$$\begin{aligned} F_r &= \int_{\text{loop}} dv f = \int dv J \times B \\ &= 2\pi R I B \end{aligned}$$

Hoop stress in a ring

A ring carrying a current I in a field B



The total radial magnetic force on the ring is

$$F_r = \int_{loop} dv f = \int dv J \times B$$

$$= 2\pi R I B$$

A tension is developed within the ring

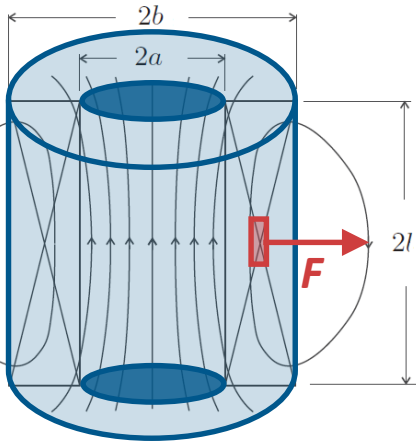
$$F = F_r \frac{\theta}{2\pi} = 2T \sin \frac{\theta}{2} \approx 2T \frac{\theta}{2} = T\theta \Rightarrow T = \frac{F_r}{2\pi} = R I B$$

The so-called "hoop stress" on the wire is $\sigma_\theta = \frac{T}{S_{wire}} = R J B$

Hoop stress levels **above 100 MPa** are common, the NHMFL 32 T magnet operates at **400 MPa**

Electromagnetic stresses in a finite solenoid

In a winding adjacent turns will press on each other and develop a radial stress σ_r which modifies the hoop stress σ_θ



Considering the solenoid as a continuous uniform medium, from the Hooke's law

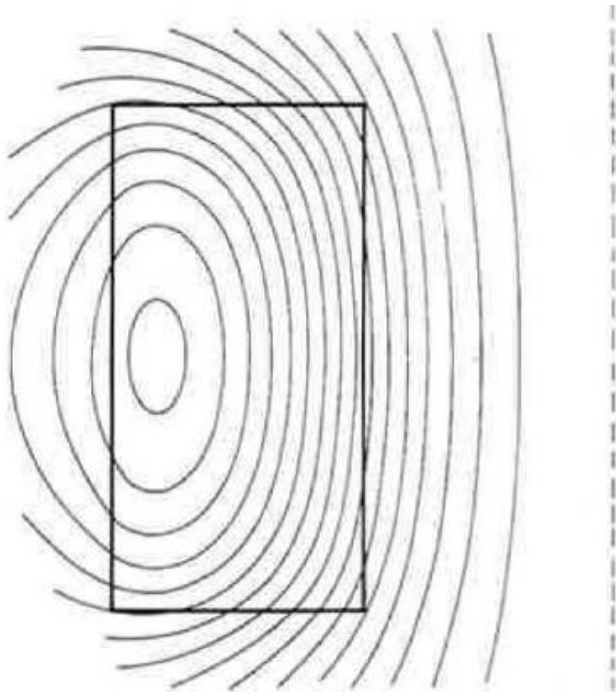
$$\sigma_r = \frac{E}{1-\nu^2} \left(\frac{du}{dr} + \nu \frac{u}{r} \right) \quad \text{and} \quad \sigma_\theta = \frac{E}{1-\nu^2} \left(\frac{u}{r} + \nu \frac{du}{dr} \right)$$

where u is the local displacement in the radial direction, E is the Young's modulus and ν is the Poisson's ratio

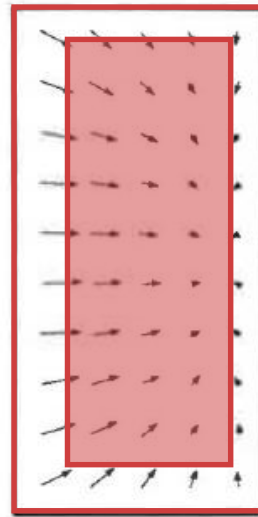
The condition for equilibrium between radial stress σ_r , hoop stress σ_θ and body force BJr is given by the equation

$$\frac{1}{r} \frac{d}{dr} \left(r \frac{du}{dr} \right) - \frac{u}{r^2} = \frac{1-\nu^2}{E} BJ$$

Electromagnetic stresses in a finite solenoid

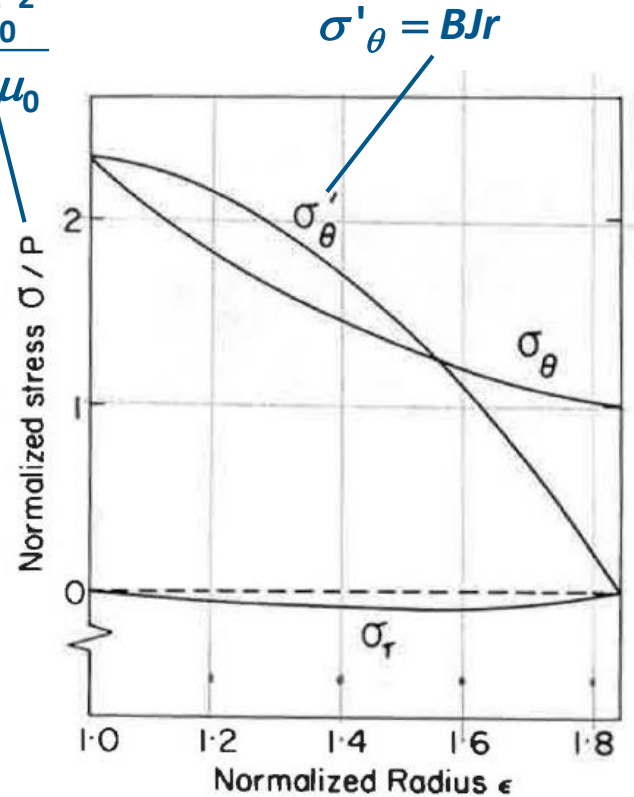


magnetic lines of force



vectors of electromagnetic force per unit volume

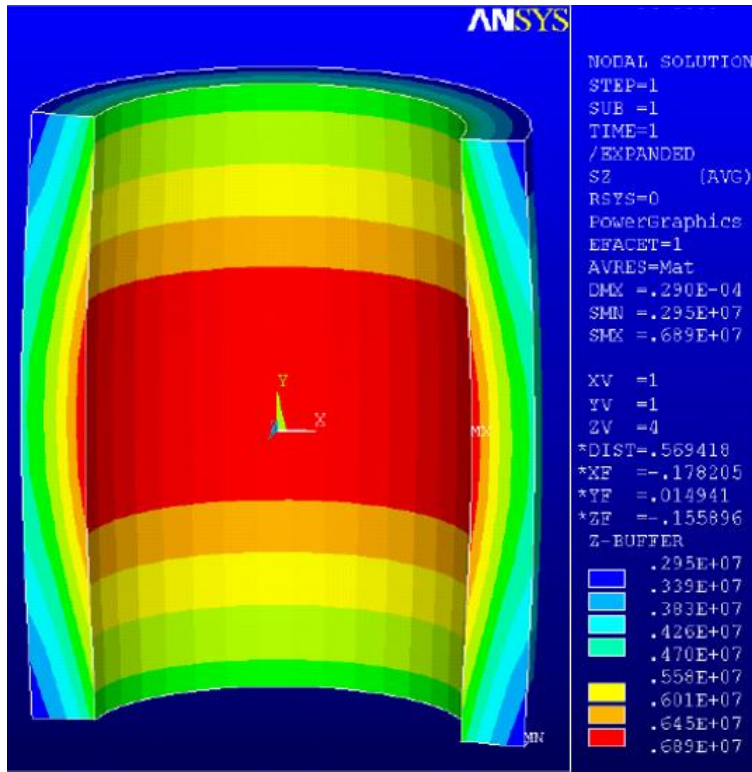
$$p = \frac{B_0^2}{2\mu_0}$$



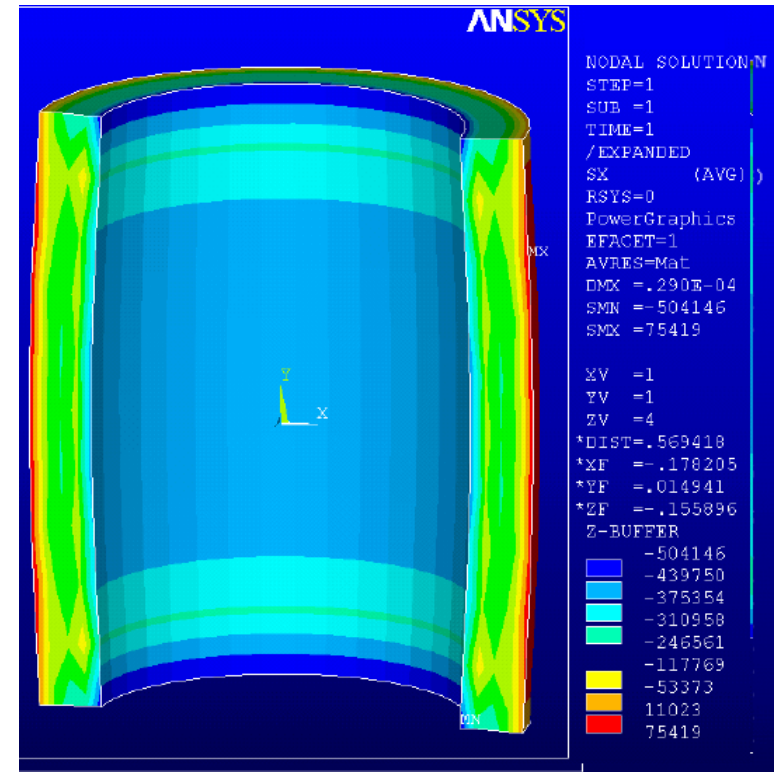
$$\epsilon = \frac{r}{\text{bore radius}}$$

Example: for B_0 of the order of 10 T, on the winding σ_θ of > 200 MPa (> 2000 bar)

Colour mapping of σ_θ and σ_r in a finite solenoid

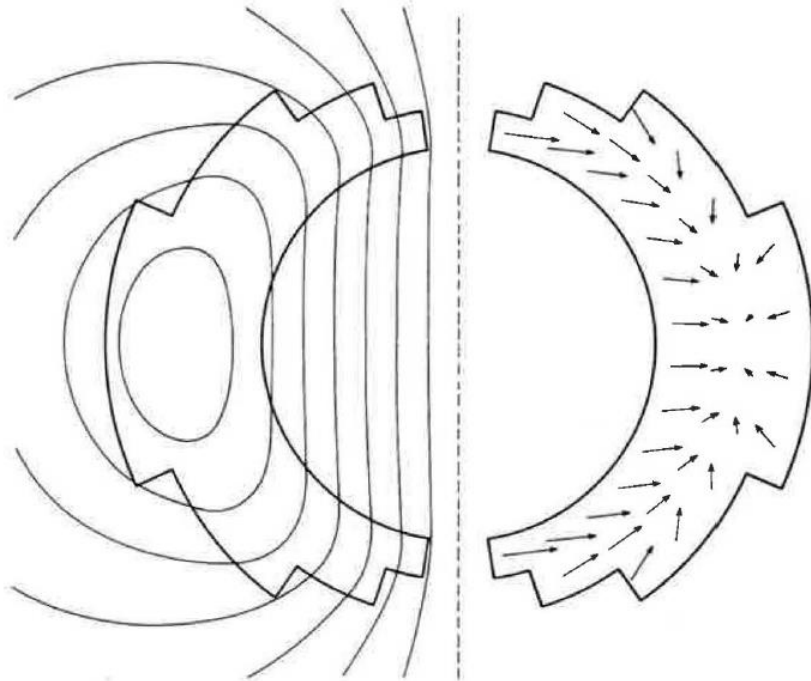


Hoop stress



Radial stress

Electromagnetic stresses in an accelerator dipole

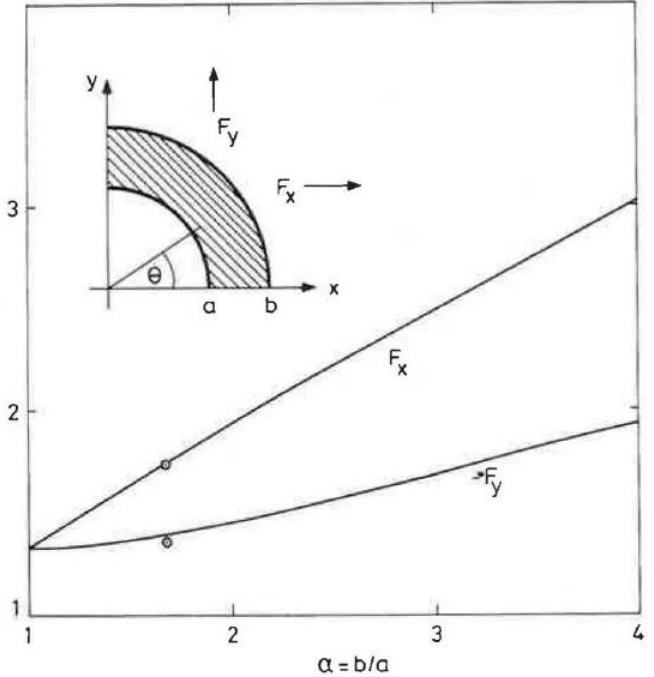


magnetic lines of force

vectors of electromagnetic force per unit volume

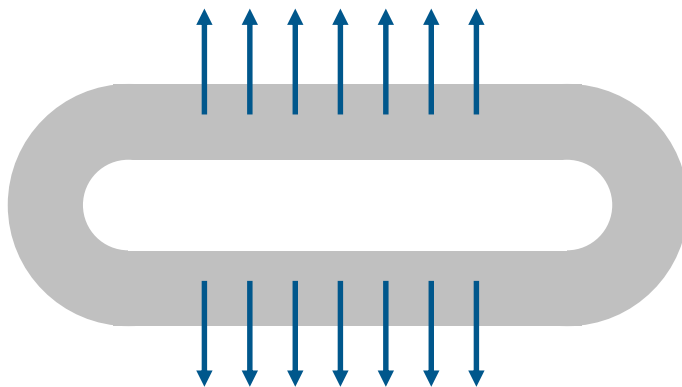
$$p = \frac{B_0^2}{2\mu_0}$$

F/Pa



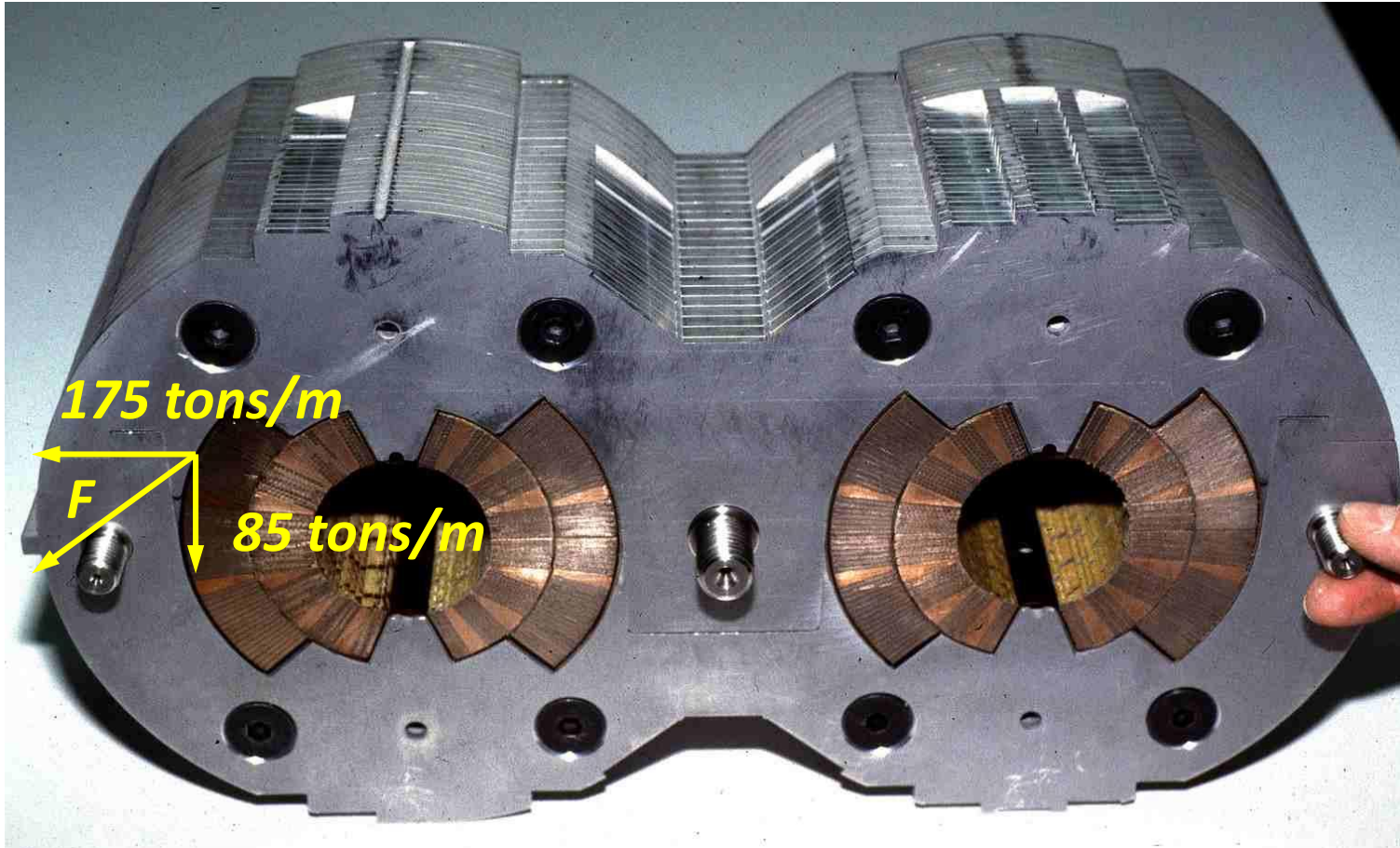
For $B_0=6T$ and $(a+b)/2=100$ mm

$F_x=200$ tons/m



In straight-sided coils such as dipoles and quadrupoles the conductor is unable to support the magnetic forces in tension

Collaring and Pre-stress in accelerator magnets

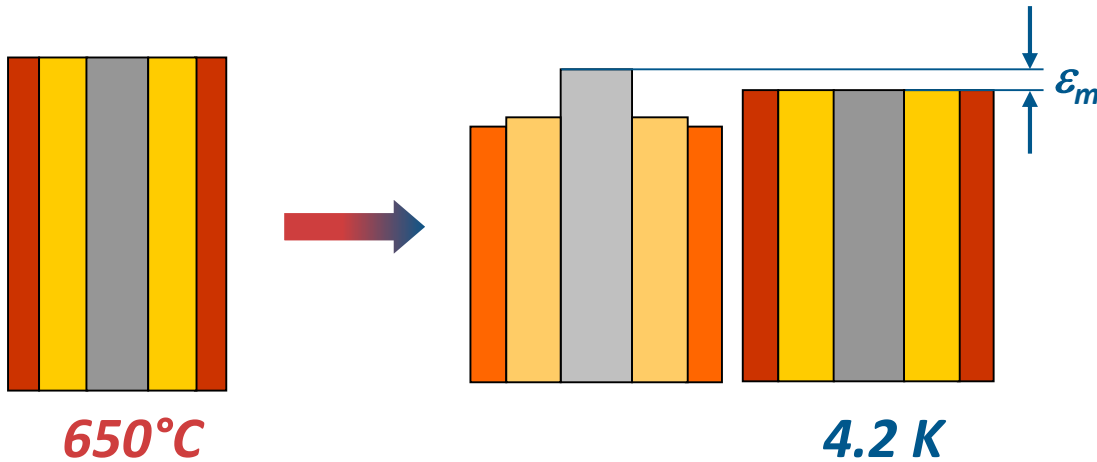


*The windings of accelerator magnets are clamped in a solid collar
Collar provide pre-stressing to prevent the movement of the coil in the
presence of electromagnetic forces*

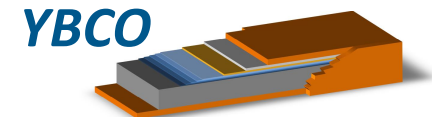
Thermal precompression of the superconductor

All technical superconductors are composite

Mismatch in thermal contraction within the composite induces a precompression in the SC






When the SC is cooled at the operating temperature, its crystal structure deforms and this induces changes on T_c , B_{c2} (and thus on J_c)



In the following, the focus will be on



Nb_3Sn → Today high field superconductor, up to 23.5 T in solenoidal coils, the material for  and  and only candidate material for the 16 T dipoles of 

YBCO → Tomorrow high field superconductor, more than 40 T demonstrated in solenoidal insert coils, the way to get 20 T dipoles and beyond 



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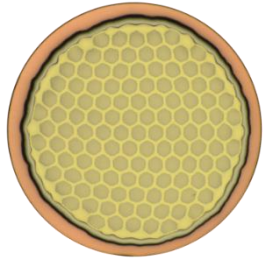


YBCO → Tomorrow high field superconductor, more than 40 T demonstrated in solenoidal insert coils, the way to get 20 T dipoles and beyond

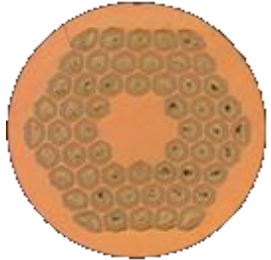


Industrial fabrication of Nb_3Sn wires

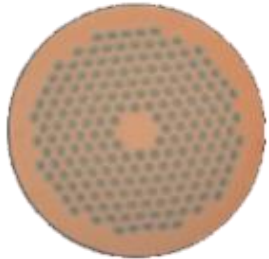
Three technologies have been developed at industrial scale



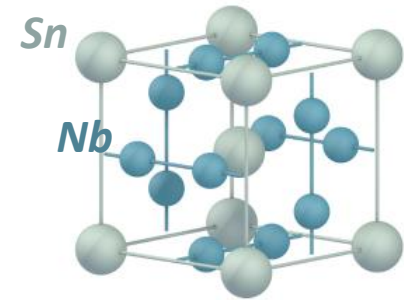
- *Bronze route*



- *Internal Sn diffusion*



- *Powder-In-Tube (PIT)*



The Sn source is the main difference

Presently produced by



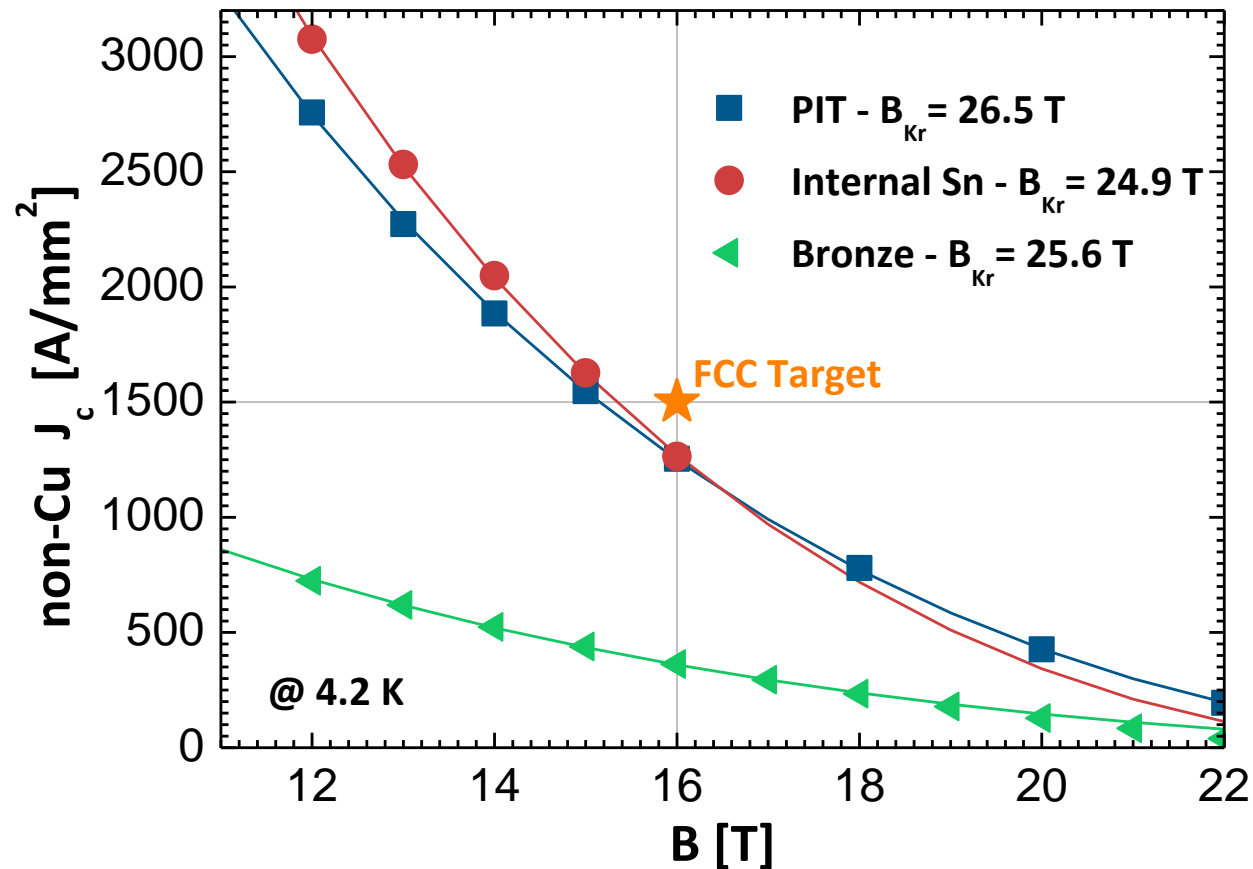
LUVATA

Western Superconducting Technologies Co.,Ltd.



Critical current density vs. magnetic field

Best performance achieved so far in industrial wires



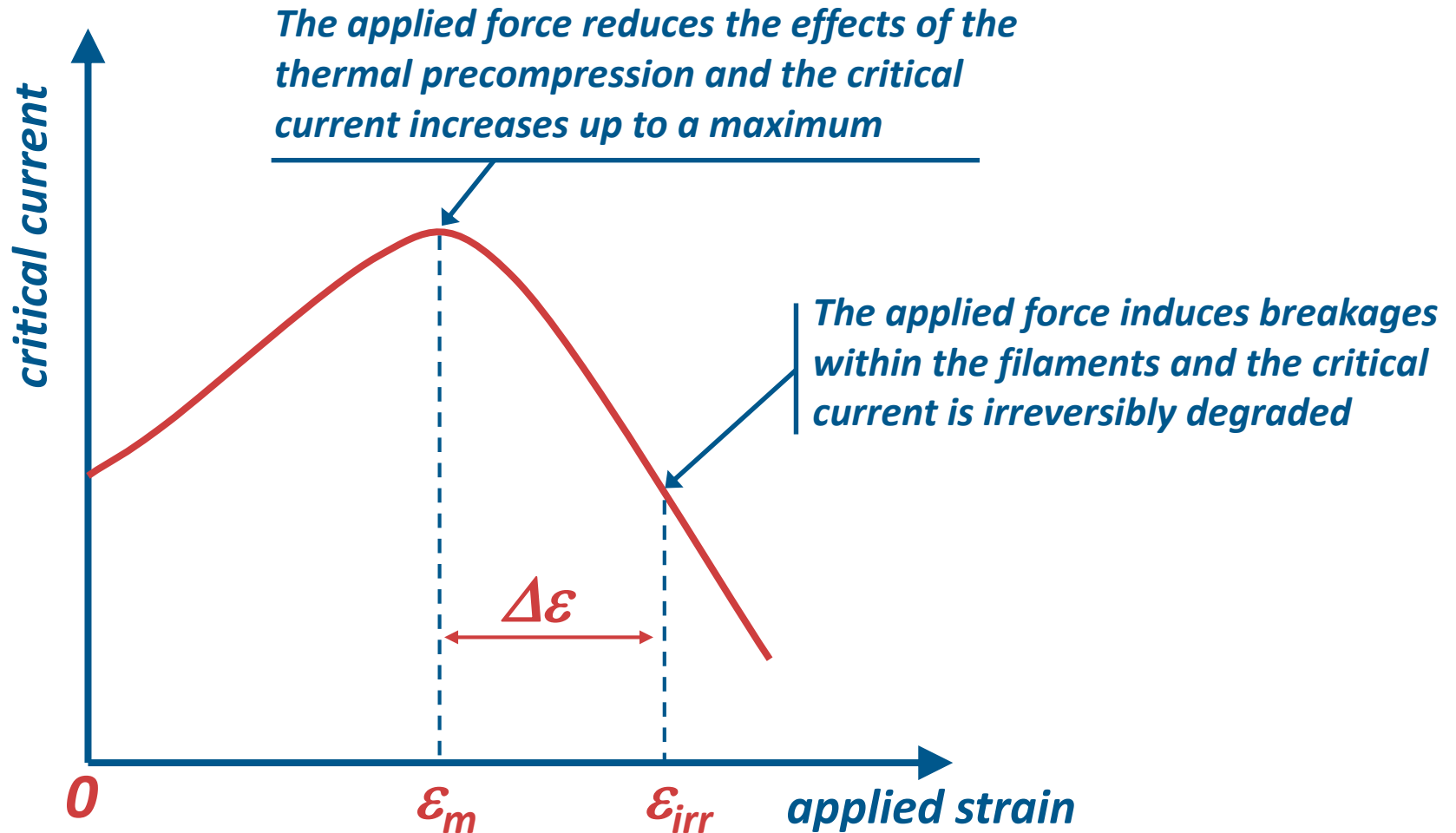
T. Boutboul et al., IEEE TASC 19 (2009) 2564

J. Parrell et al., AIP Conf. Proc. 711 (2004) 369

V. Abächerli et al., IEEE TASC 17 (2007) 2564

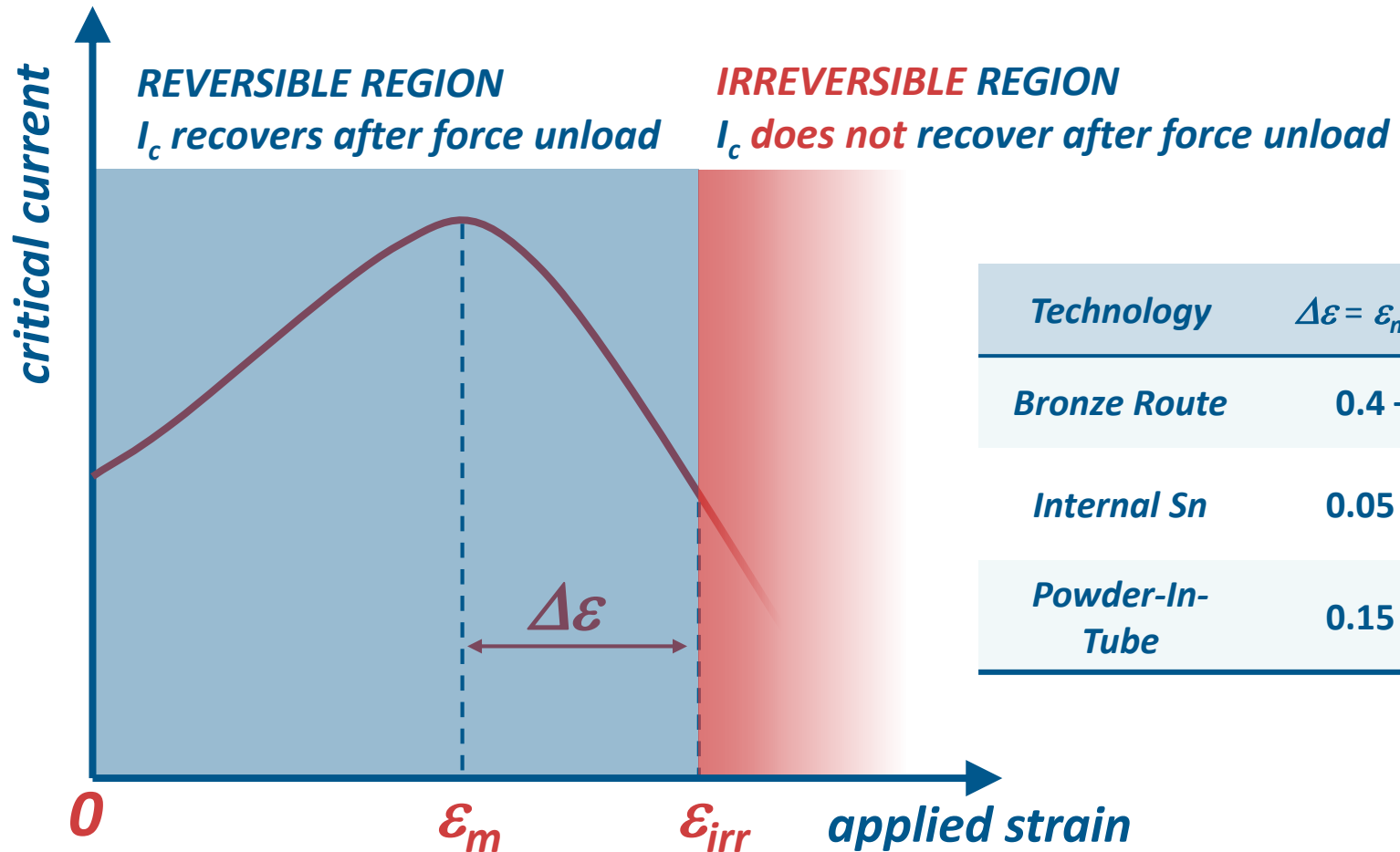
Strain-induced changes in the critical current

Effects of the longitudinal strain



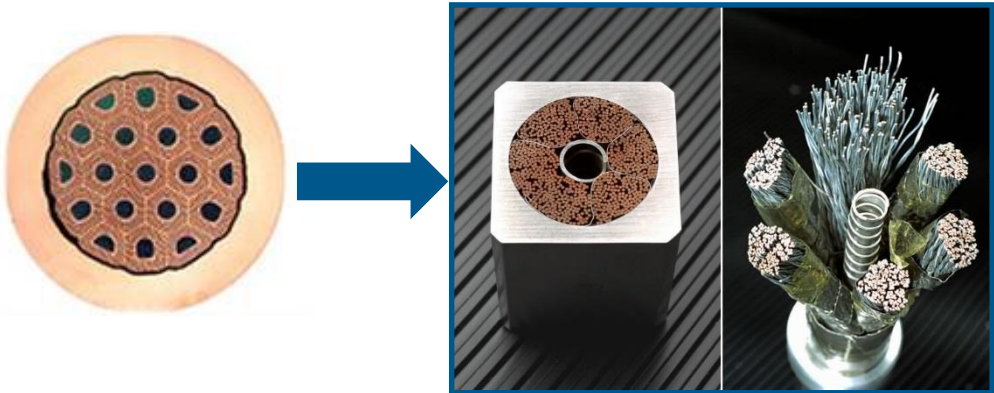
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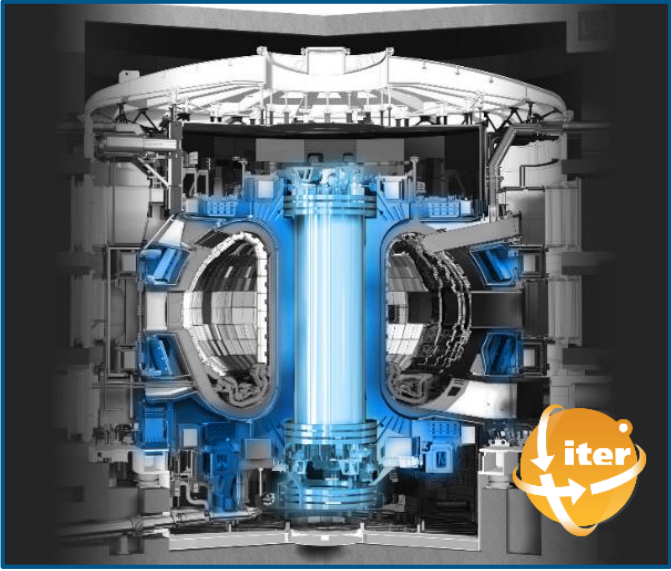


Technology	$\Delta\epsilon = \epsilon_m - \epsilon_{irr}$ [%]
Bronze Route	0.4 – 0.6
Internal Sn	0.05 – 0.2
Powder-In-Tube	0.15 – 0.3

Wire cabling for the ITER coils

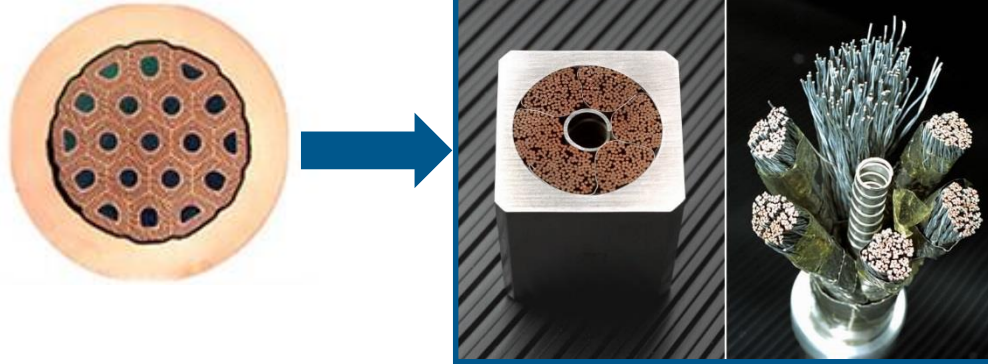


CICC = Cable-In-Conduit Conductor



Wire cabling for the ITER coils

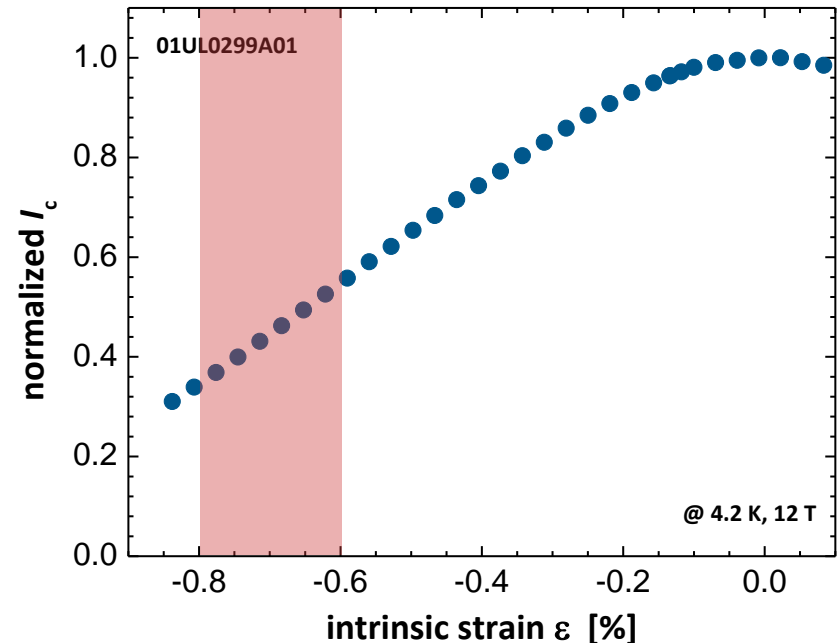
Reversible variation of I_c under strain matters



CICC = Cable-In-Conduit Conductor

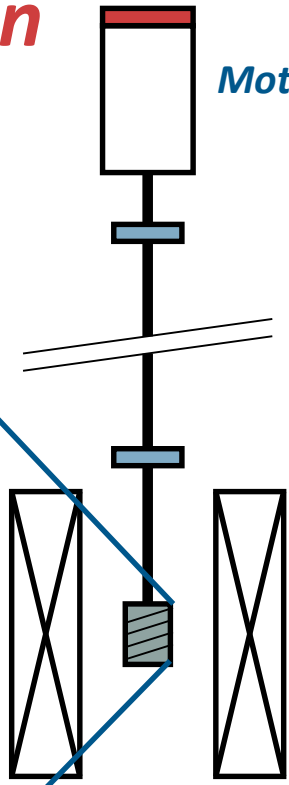
Because of the thermal precompression, wires in a CICC experience an effective axial compressive strain ranging between **-0.6% and -0.8%**

The performance of the superconductor is **limited to ~40%** of the maximum achievable current



How to measure I_c vs. axial strain

The WASP (Walters Spring) probe



Motor @ room temperature

B. Seeber et al., Rev. Sci. Instr. 76 (2005) 093901

Max current	1000 A
Sample length	1.1 meter
Voltage taps distance	126 mm
Electrical field criterion	0.01 $\mu\text{V}/\text{cm}$
Strain (ϵ)	up to $\pm 1.2 \%$

Why do superconducting properties depend on strain ?

1. Reversible effects of strain

Reversible effects of strain

Why superconducting properties depend on strain

Nb_3Sn exhibits strong-coupling superconductivity

$\langle \omega \rangle$ is an average phonon frequency

electron-phonon coupling phonon DOS

$$\lambda = 2 \int_0^{\infty} d\omega \frac{\alpha^2(\omega) F(\omega)}{\omega} \propto \frac{N(E_F)}{\langle \omega^2 \rangle}$$

McMillan parameter

$$T_c = \frac{\hbar \langle \omega \rangle}{1.2 k_B} \exp \left[- \frac{1.04(1 + \lambda)}{\lambda - \mu^* (1 + 0.62\lambda)} \right]$$

μ^* gives the Coulomb repulsion at the Fermi surface

Strain induces change both in the phonon spectrum and the electronic bands

So far the question whether the strain sensitivity is due mainly to electrons or phonons has not been settled

Reversible effects of strain

Why superconducting properties depend on strain

Some recent theoretical studies

D. Markiewicz, *Cryogenics* 44 (2004) 767

This work emphasizes the role of anharmonicity as a source of the strain dependence in Nb₃Sn

D. Taylor and D. Hampshire., *Supercond. Sci. Technol.* 18 (2005) S241

Microscopic theory analysis suggests that the uniaxial strain effects are predominantly due to changes in the average phonon frequencies

G. De Marzi et al., *J. Phys.: Condens. Matter* 25 (2013) 135702

Investigation of the strain sensitivity of Nb₃Sn from first principle calculations based on the density functional theory

M. Mentink, PhD thesis (2014), University of Twente

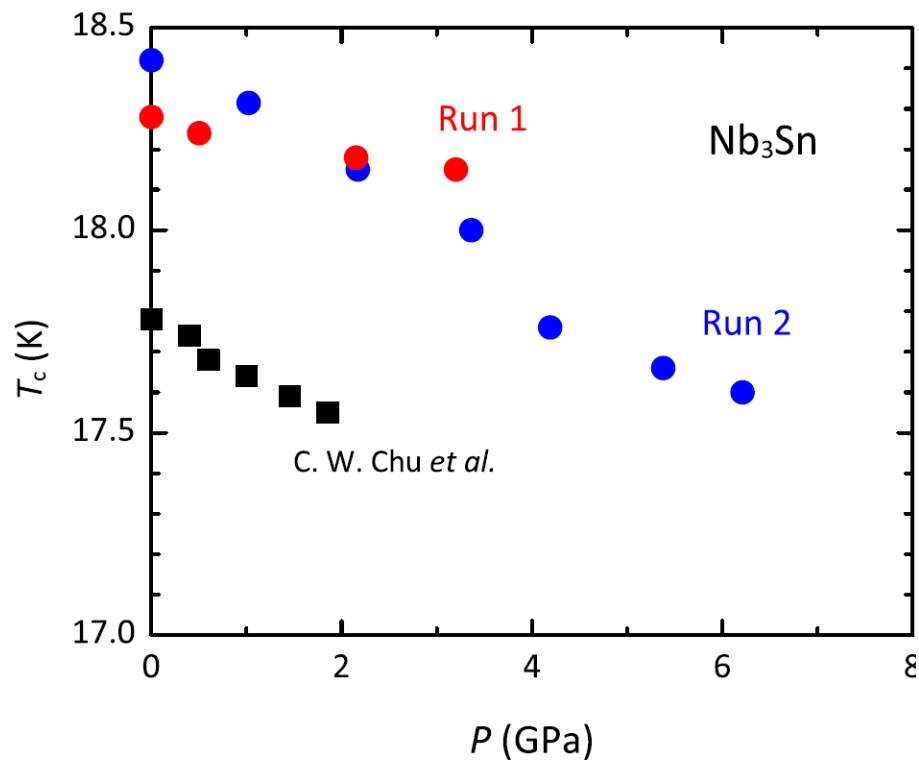
Ab-initio calculations used to evaluate electronic, and vibrational properties, as well as T_c and B_{c2} as a function of disorder, crystal orientation, and strain

D. Valentinis et al., *Supercond. Sci. Technol.* 27 (2014) 025008

A theory of the strain-dependent critical field in Nb₃Sn based on anharmonic phonon generation

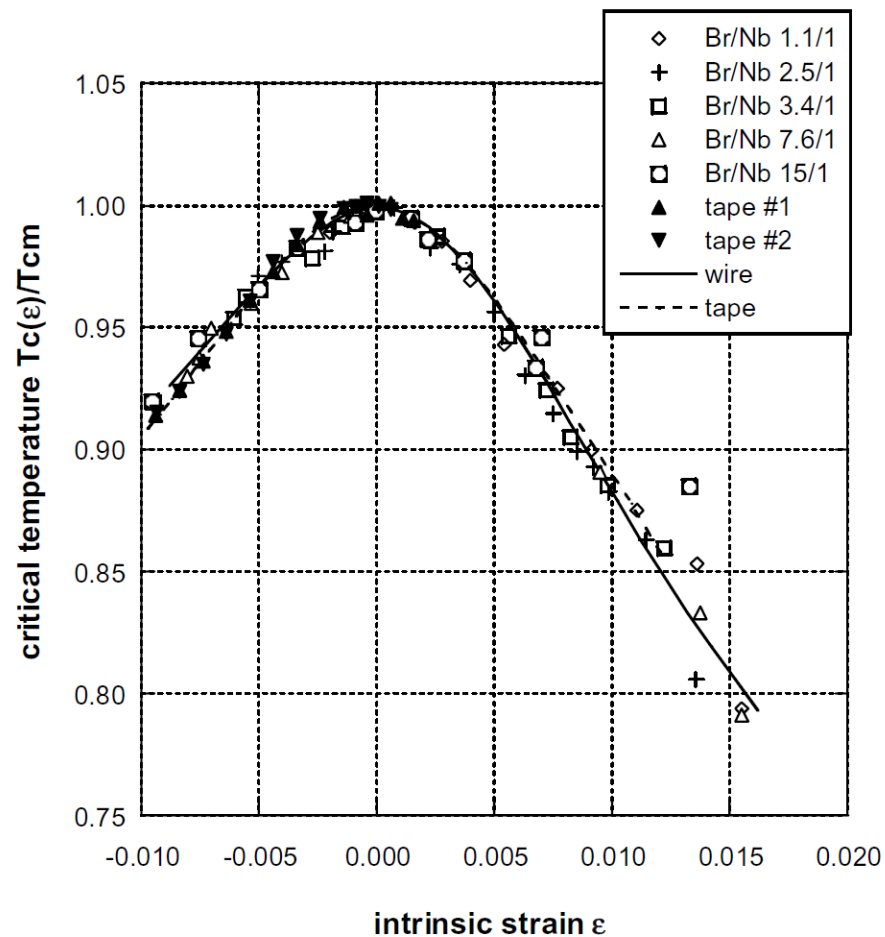
Reversible effects of strain

Hydrostatic vs. uniaxial



Critical temperature vs. hydrostatic compression

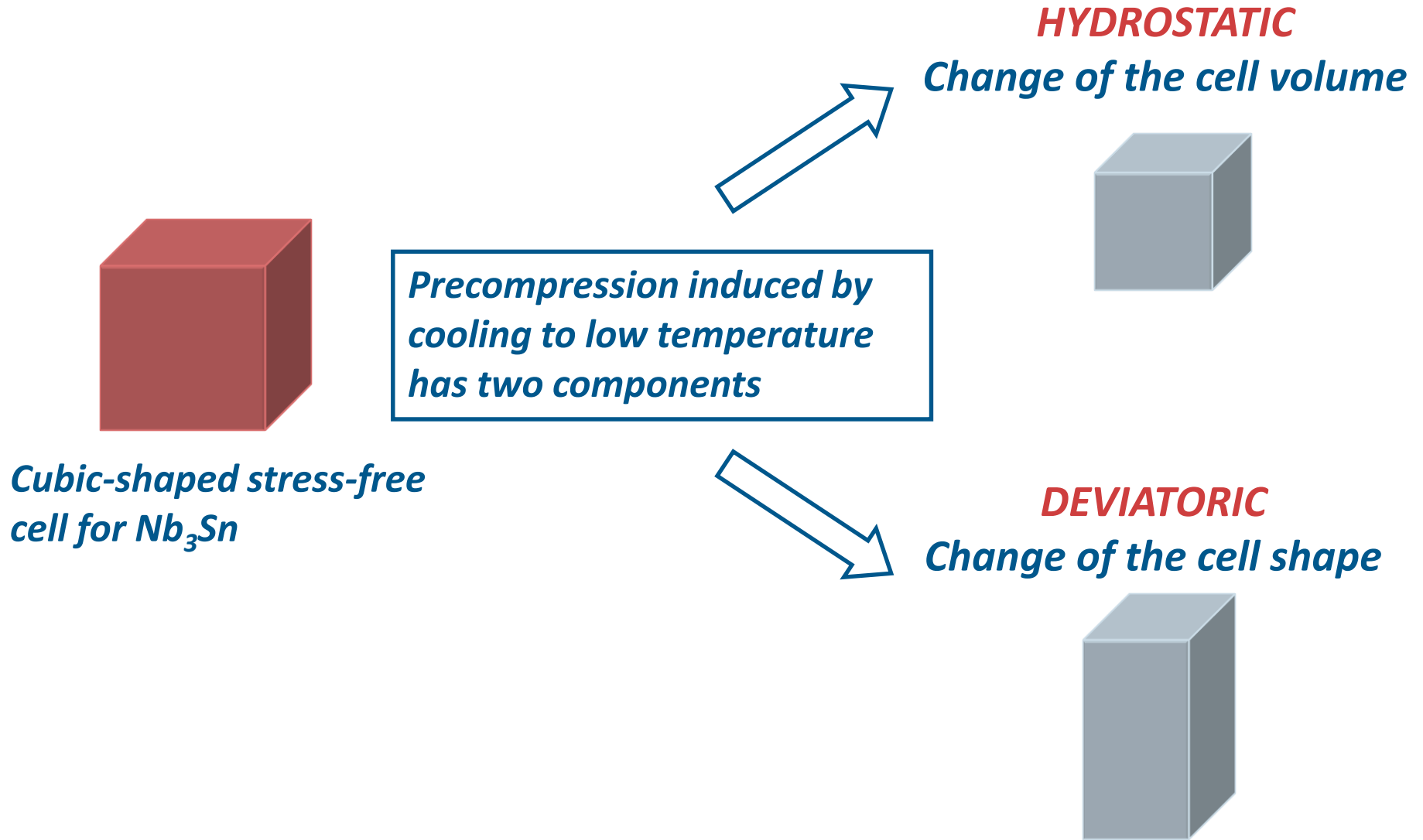
S. Tanaka et al., J. Phys. Soc. Jpn. 81 (2012) SB026



Reduced critical temperature vs. uniaxial strain

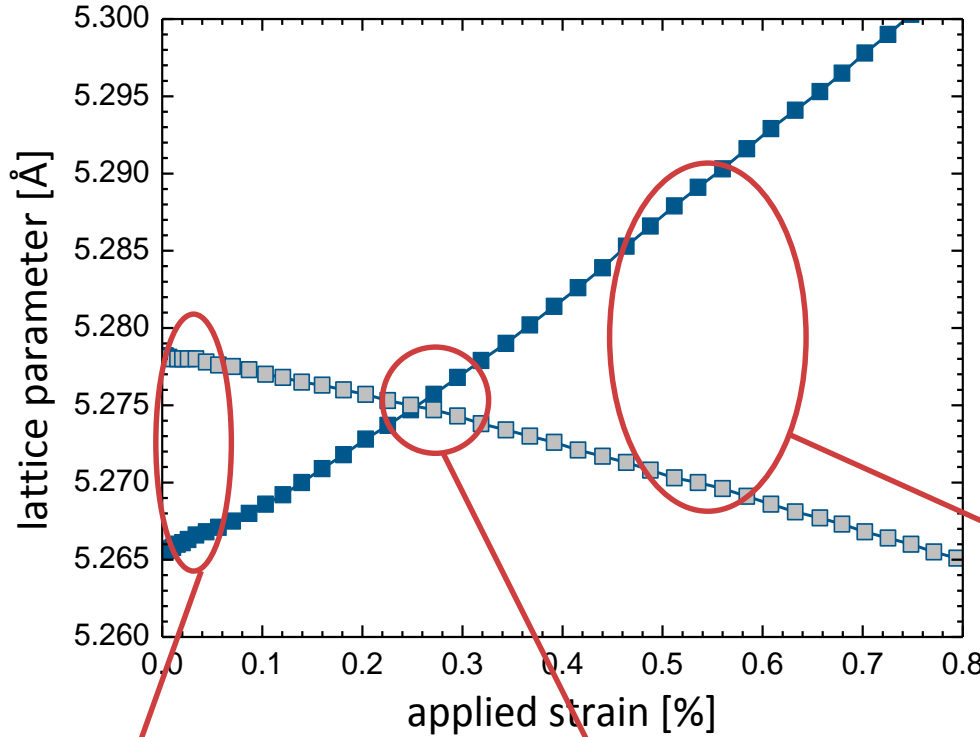
D. Markiewicz, Cryogenics 44 (2004) 767

Reversible strain effects in Nb_3Sn wires



Reversible strain effects in Nb₃Sn wires: lattice parameters

Bronze route wire: Nb₃Sn lattice parameters vs uniaxial strain @ 4.2 K



XRD experiments @ ESRF Grenoble

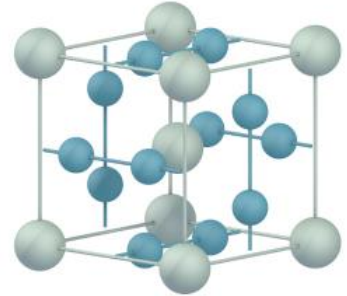
C. Scheuerlein et al., IEEE TAS 19 (2009) 2653

L. Muzzi et al., SUST 25 (2012) 054006

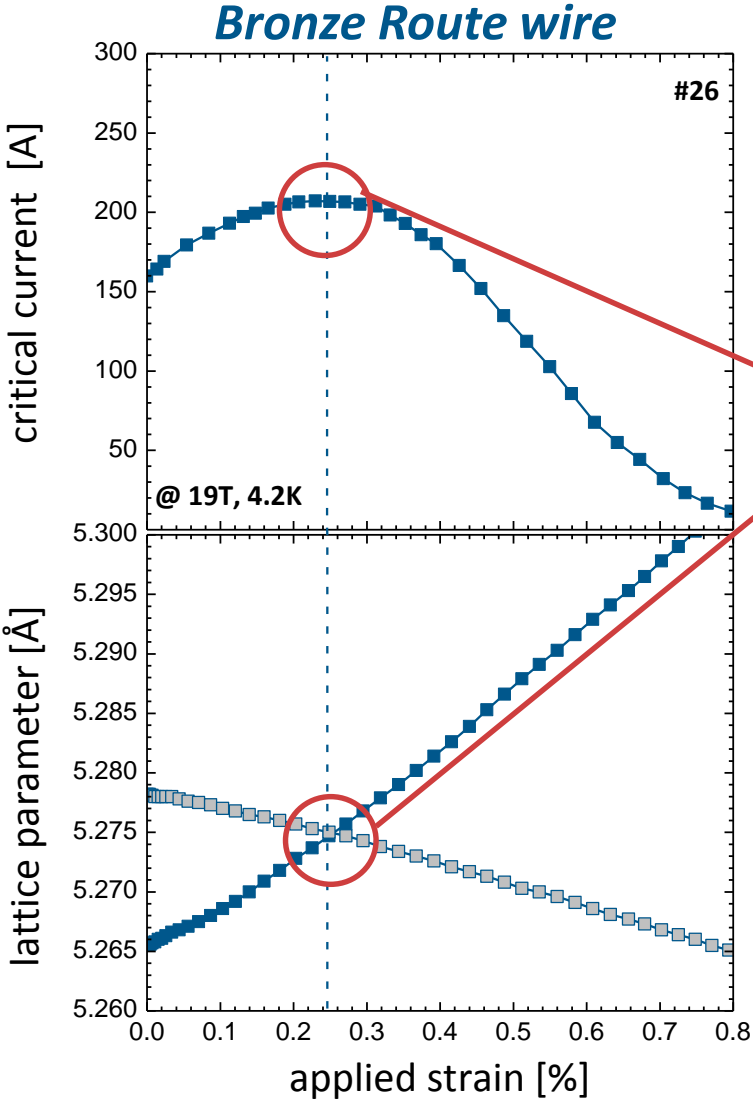
HIGH APPLIED STRAIN
Again Hydrostatic+Deviatoric

ZERO APPLIED STRAIN
Nb₃Sn is precompressed
Hydrostatic+Deviatoric

CROSSING POINT
Cubic cell recovered
Hydrostatic component still present



Lattice parameters and I_c under axial strain

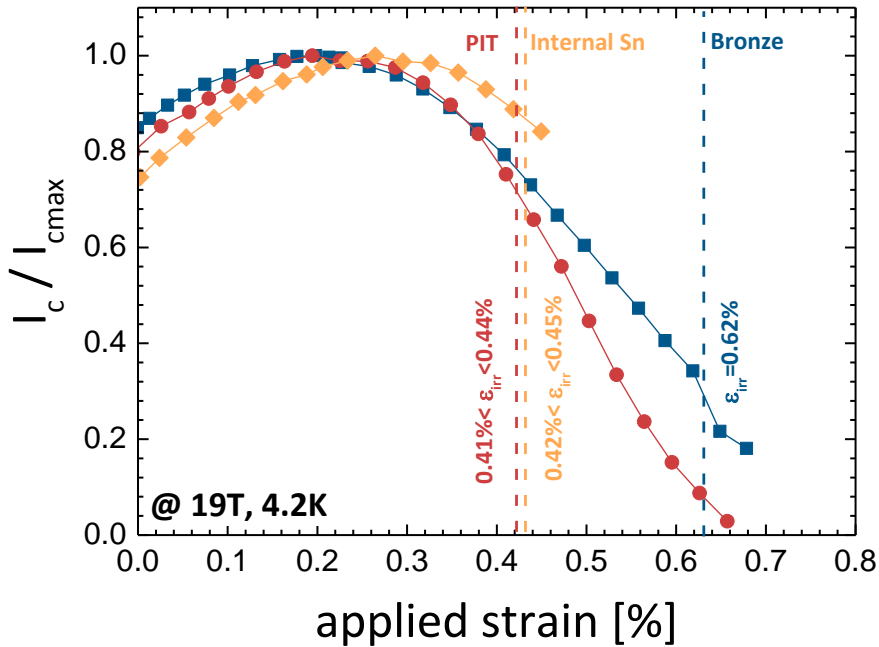


MAXIMUM OF I_c
The maximum of the critical current occurs when the cubic cell is restored

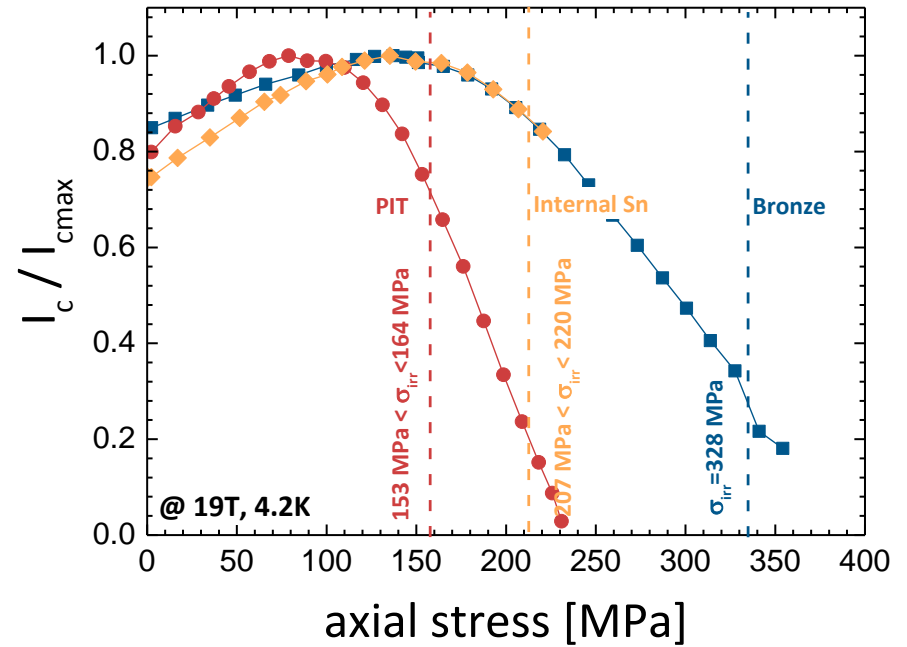
Bronze Route, Internal Sn and PIT

Reversible behaviour and irreversible limit

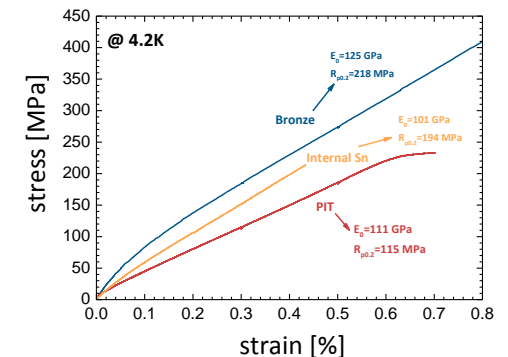
I_c vs. axial strain



I_c vs. axial stress



Technology	σ_{irr}
Bronze Route	330 MPa
Internal Sn	210 MPa
Powder-In-Tube	150 MPa



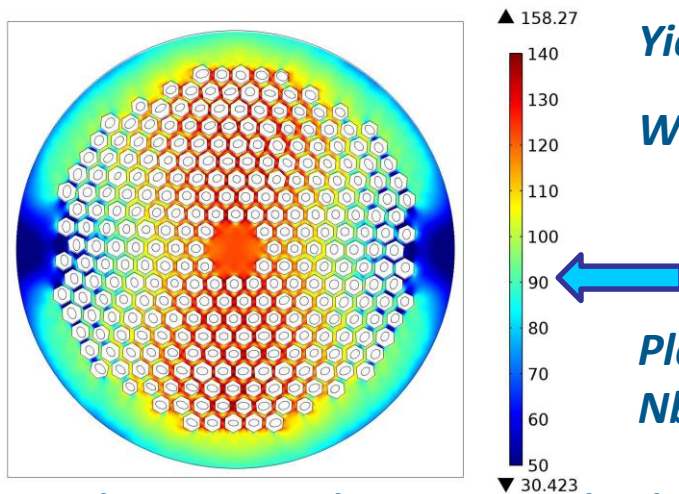
Why do superconducting properties depend on strain ?

2. Irreversible effects of strain

Irreversible degradation phenomena

Two irreversible phenomena play together

- **Plastic deformation of the Cu matrix**



*PIT Nb₃Sn wire under transverse load
Stress map of the Cu matrix*

Yield strength of Cu $\sigma_y \sim 90$ MPa

What is the residual stress on Cu after unload ?

$$\sigma_{res}^{Cu} = \max(\sigma_{Mises}^{Cu} - \sigma_y, 0)$$

Plastically deformed Cu imposes a stress on Nb₃Sn after force unload (= lattice deformation)

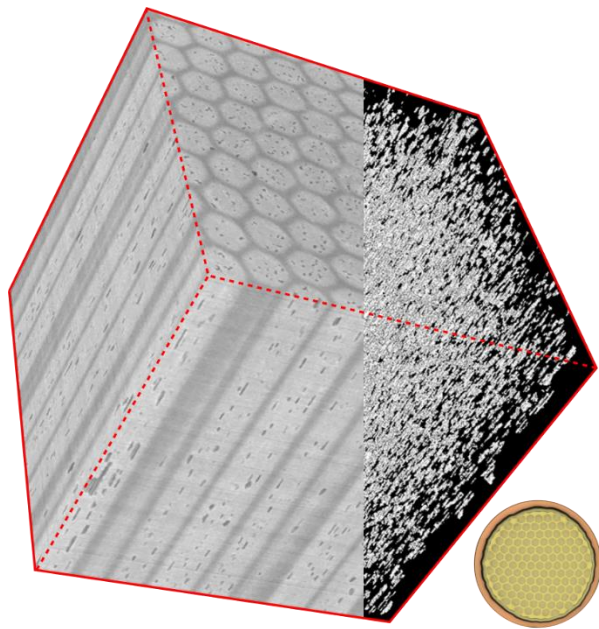
- **Crack formation in Nb₃Sn**

Nb₃Sn is a brittle material and is characterized by a strong propensity to fracture

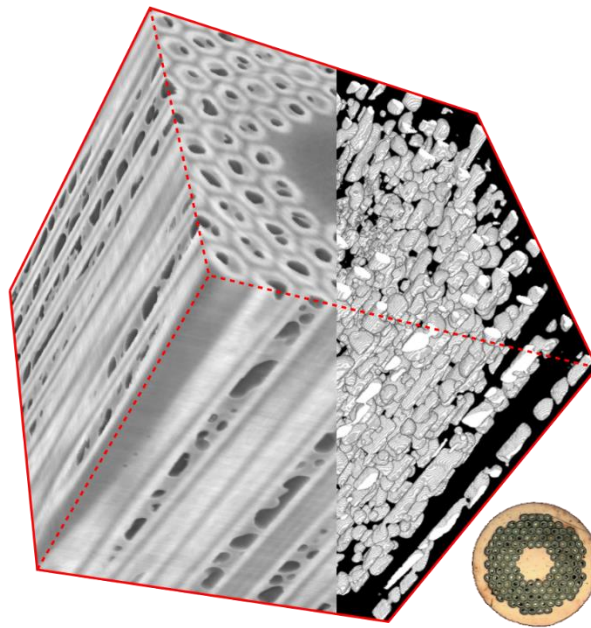
Voids formed during the reaction cause localized stress concentrations where cracks nucleate

Voids in Nb₃Sn wires

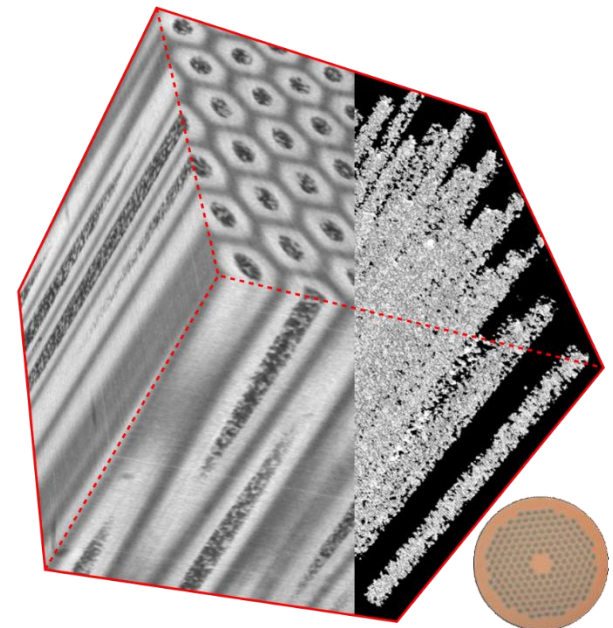
XRD microtomography reconstruction



Bronze Route
121 x 121 filaments



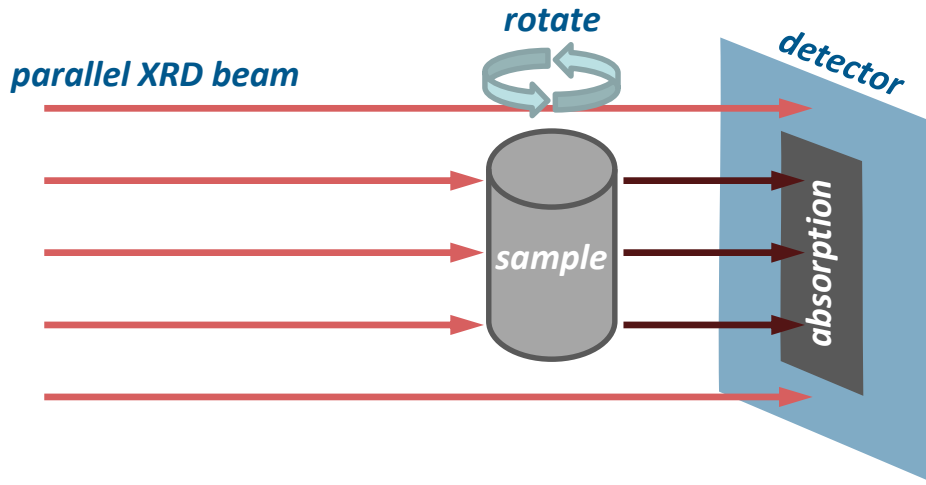
Internal Sn
132/169 subelements



PIT
192 filaments

Can we quantify the impact of voids on the electromechanical limits?

XRD microtomography at ESRF Grenoble

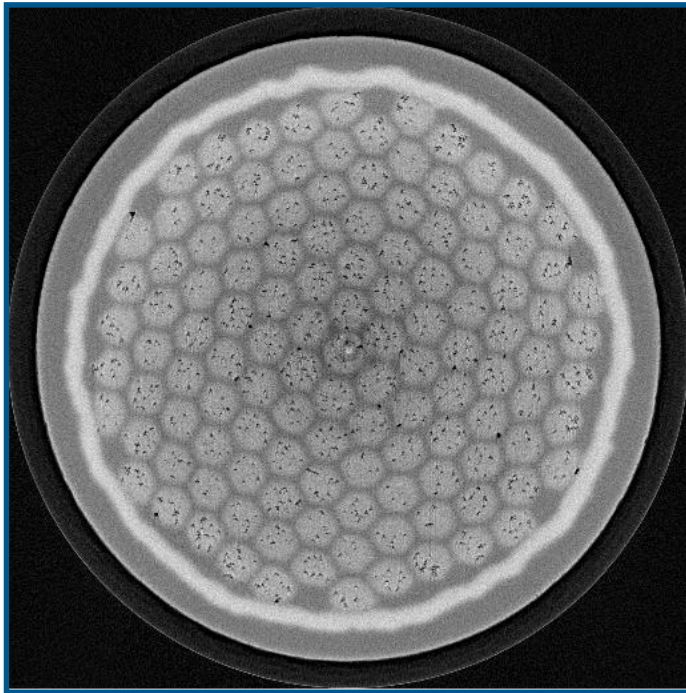


- XRD photon energy = 89 keV
- 360° rotation of the sample
- 30'000 projections
- 2560 x 2160 pixels
- 0.57 $\mu\text{m}/\text{pixel}$ resolution

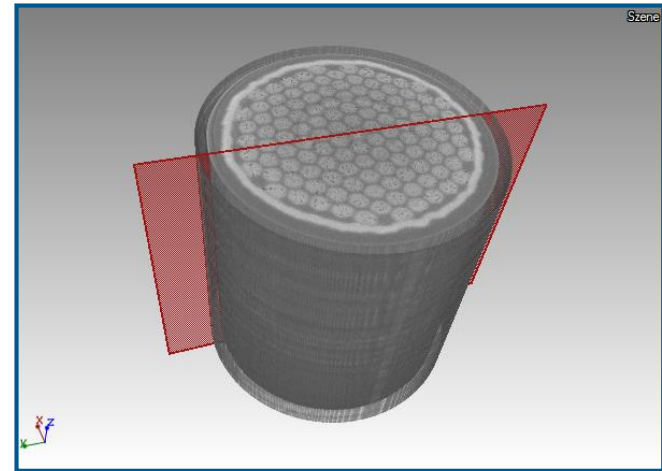


*Non-destructive
3D volume reconstruction
with separation of internal features*

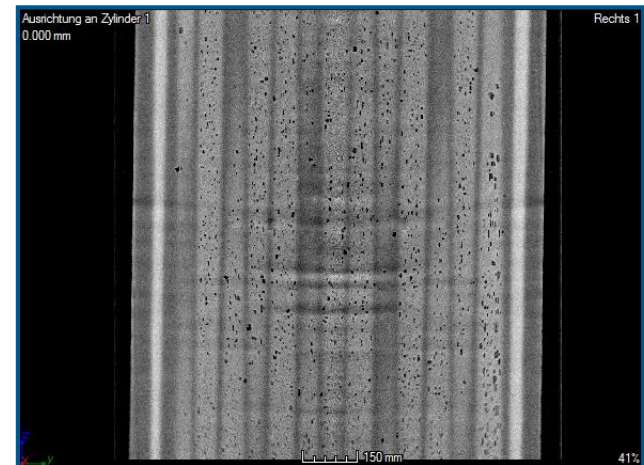
XRD microtomography at ESRF Grenoble



Bronze Route wire



Virtual longitudinal cut



A case study on Bronze Route Nb₃Sn wires

Why Bronze Route wires ?

Because by Hot Isostatic Pressing (HIP) treatment we can reduce the void fraction without degrading the critical current

Possibility to investigate the same wire type with and without voids

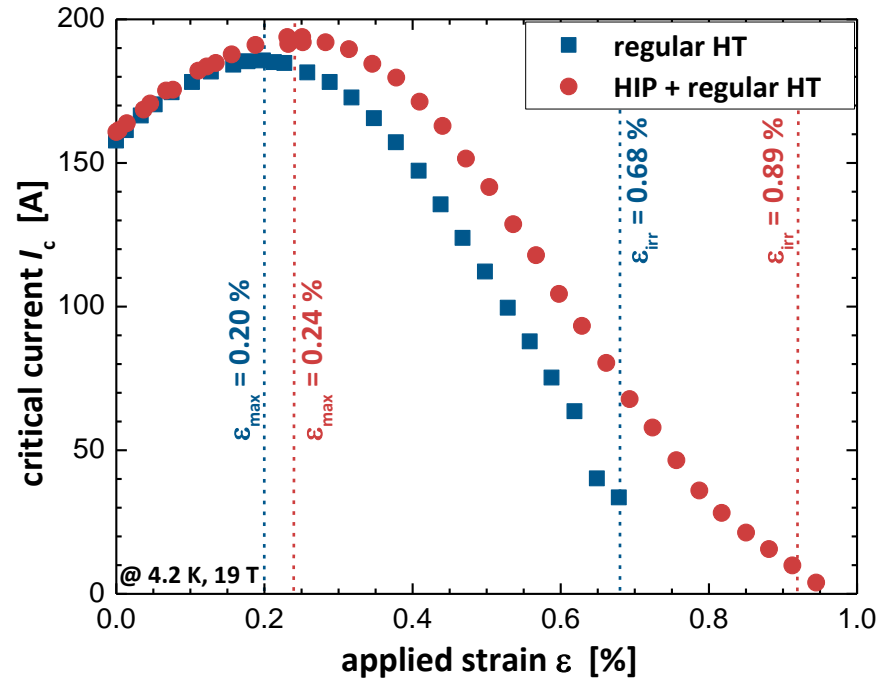
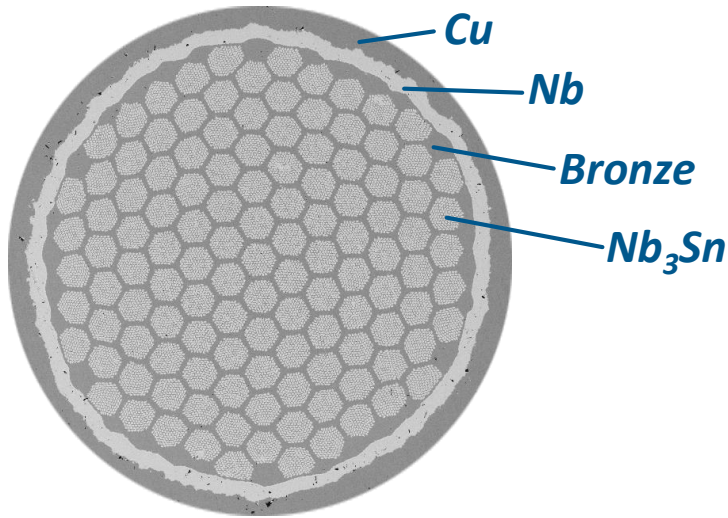
What did we do ?

Experimental determination of the electromechanical limits

Statistical analysis of the voids size, shape and distribution

Finite Element Modelling (FEM) to quantify the impact of voids

A case study on Bronze Route Nb_3Sn wires




Regular HT: 600°C/100h + 670°/150h

$$\epsilon_c = \epsilon_{irr} - \epsilon_{max} = 0.48\%$$

HIP 550°C/1h/200MPa + Regular HT

$$\epsilon_c = \epsilon_{irr} - \epsilon_{max} = 0.65\%$$

With HIP treatment ϵ_c increases by +0.17 %

manufacturer	 UNIVERSITÉ DE GENÈVE FACULTÉ DES SCIENCES
wire diameter	1.25 mm
# of filaments	121 x 121
filament size	4.5 μ m

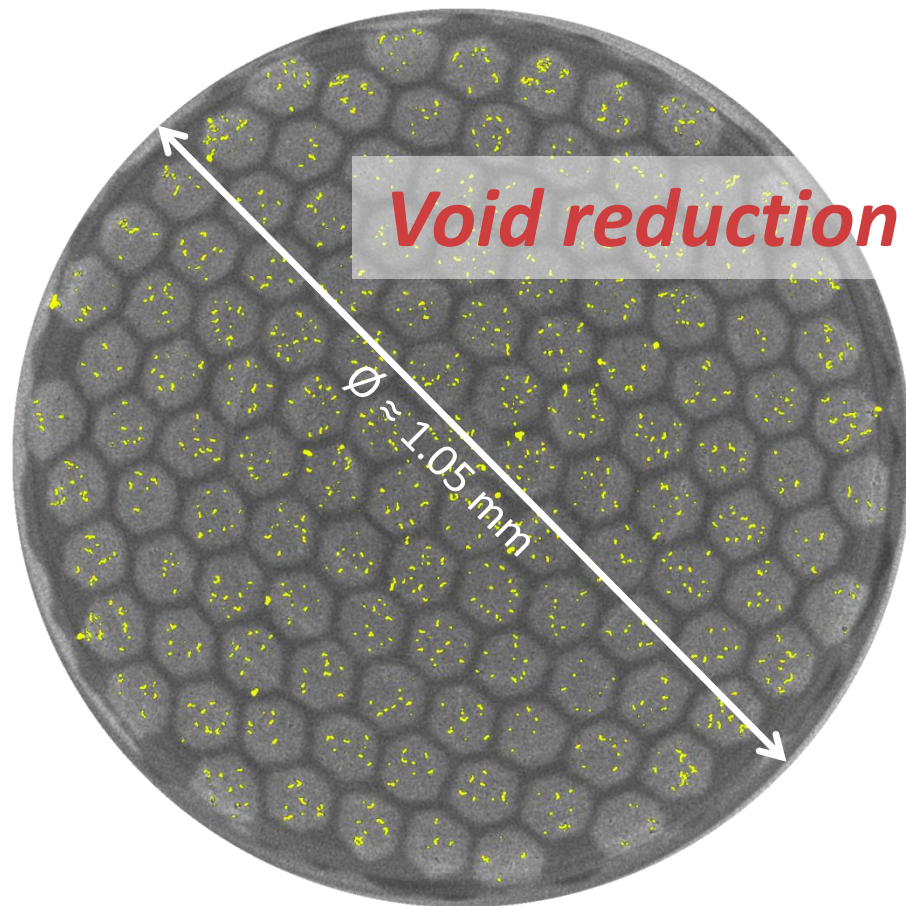
Bronze route wire: Void detection

Without HIP treatment

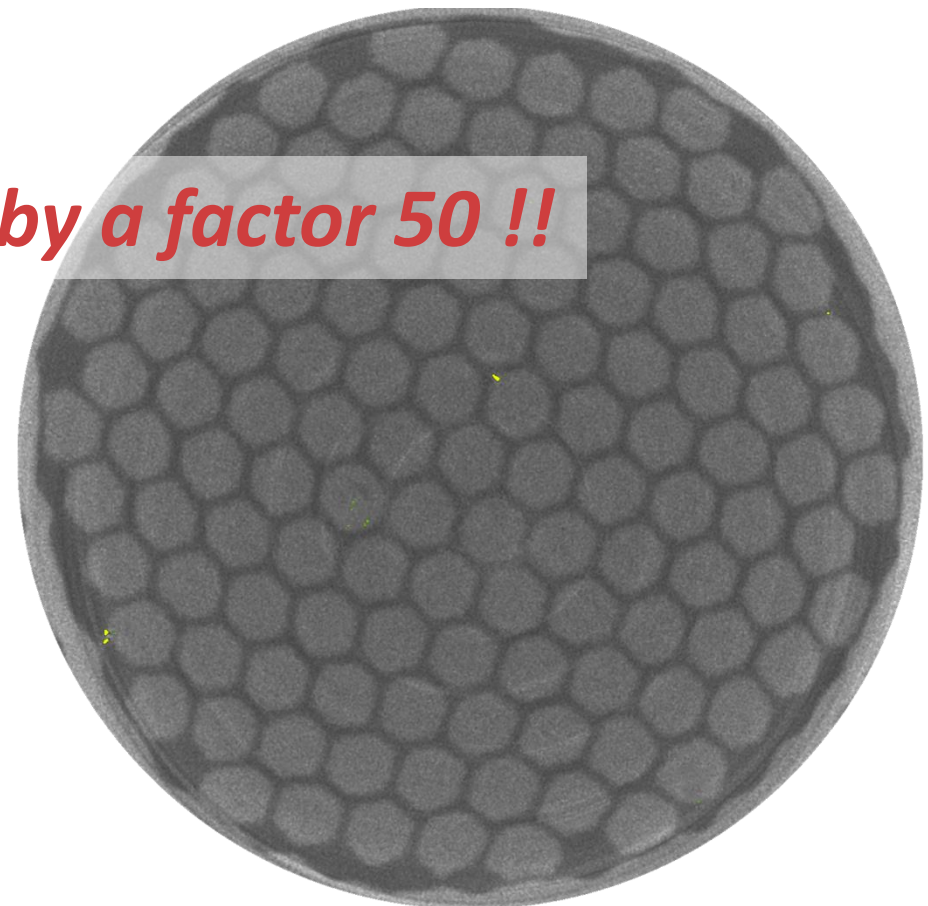
Void fraction = 2.1 %

With HIP treatment

Void fraction = 0.05 %

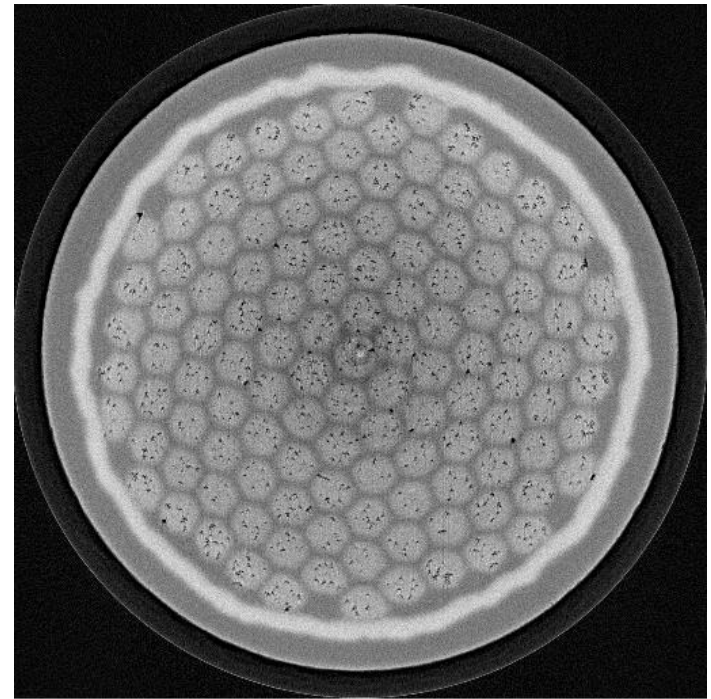
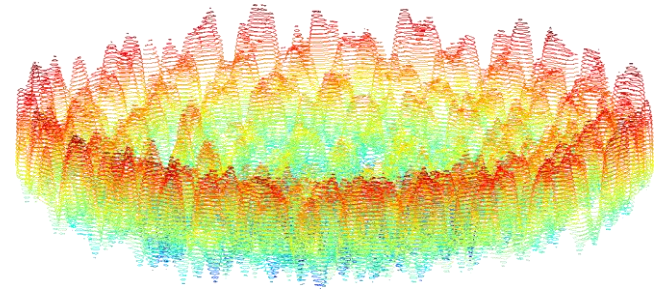


Void reduction by a factor 50 !!



Void detection and analysis

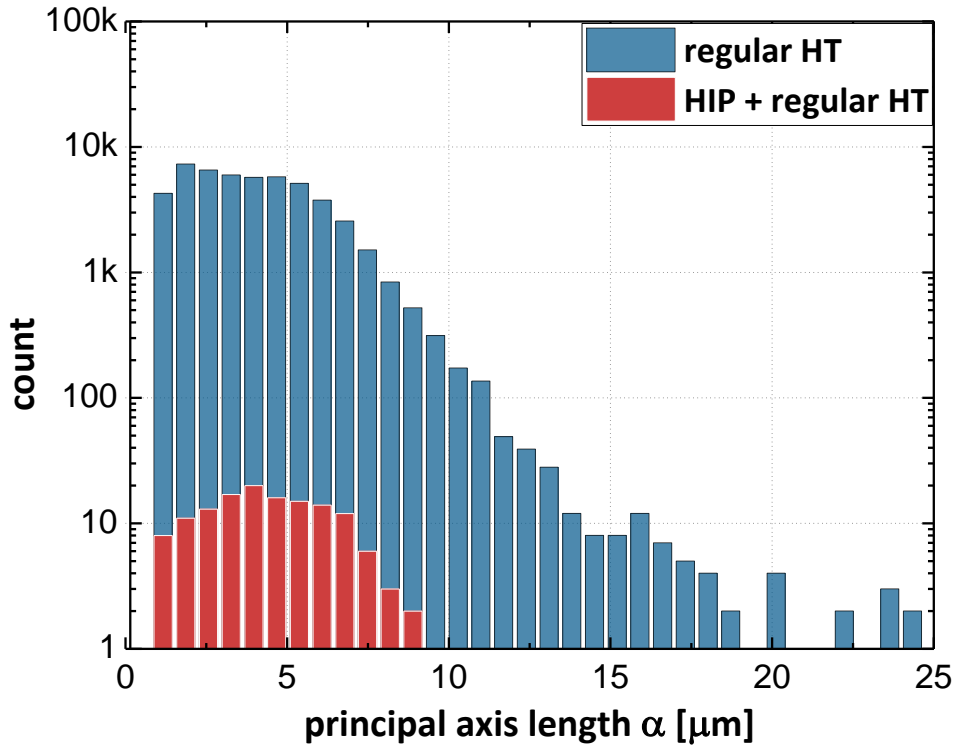
- ***A brightness distribution analysis is used to detect voids***
- ***Position is recorded, including distance to the nearest neighbors and location (matrix, filament bundle, interface)***
- ***Voids are approximated with ellipsoids***
- ***Size and orientation are determined***
 - ***Major axis length***
 - ***Major axis angles φ and θ***
 - ***Eccentricity (sphericity)***



These data become input for FEM

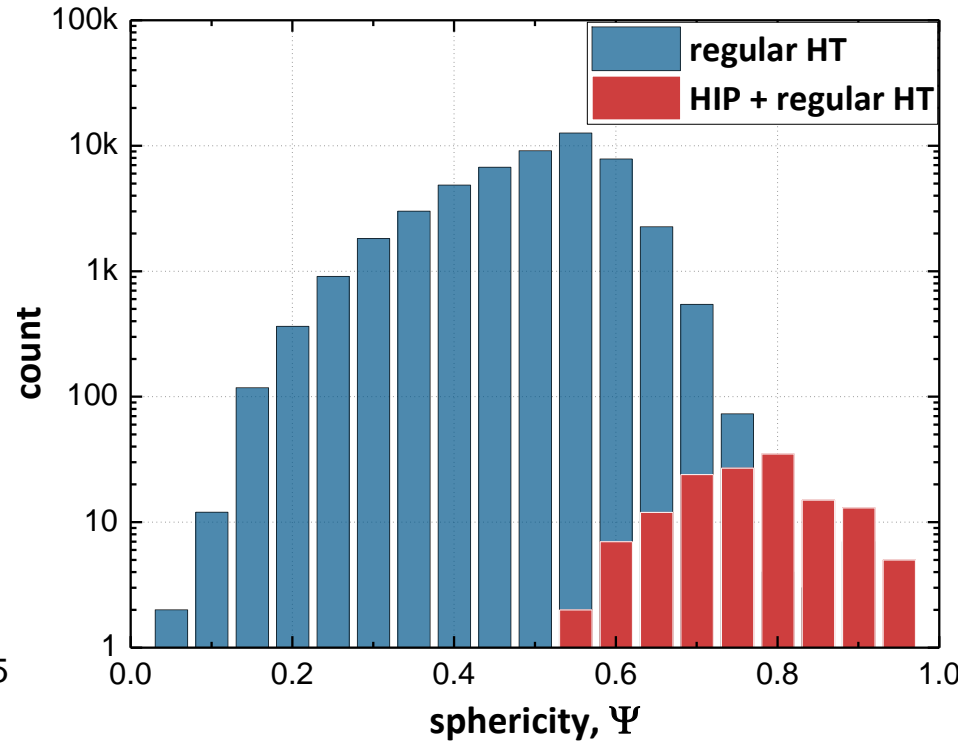
Bronze route wire: Void analysis

Void size



HIP \Rightarrow void size is reduced

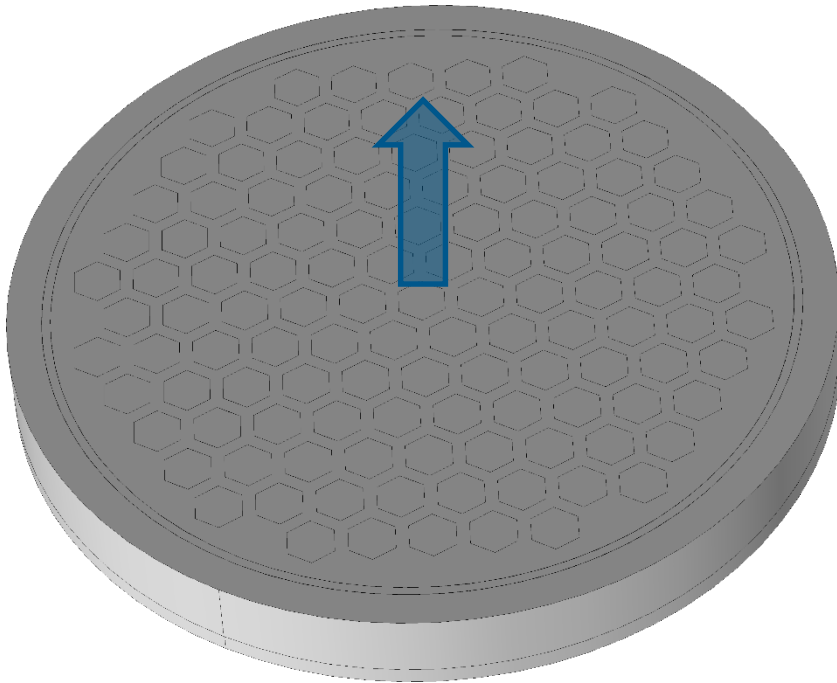
Void shape



HIP \Rightarrow voids are more spherical

Statistical FEM analysis

- *3D model, real dimensions, 0.1 mm long wire*
- *Bottom side is fixed, top side is displaced to apply a mechanical strain*



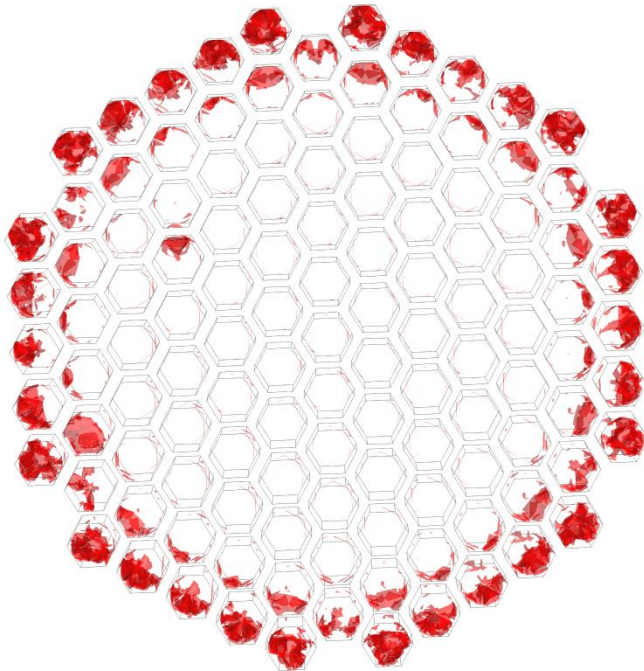
*The **output** of the model is the **von Mises stress distribution** in the **Nb₃Sn** volume at a given strain*

Statistical FEM analysis: 1) no voids

- The experimental ε_c corresponds to an irreversible reduction of I_c by 5%
- **Working hypothesis: 5% of I_c degradation \equiv damage in 5% of the filaments**

Wire with HIP treatment

SIMULATION done at $\varepsilon_c = 0.65\%$



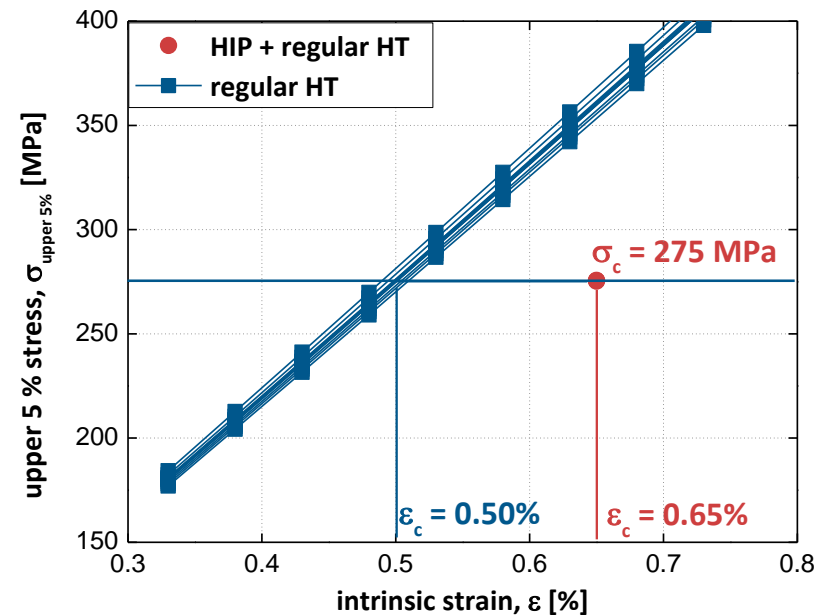
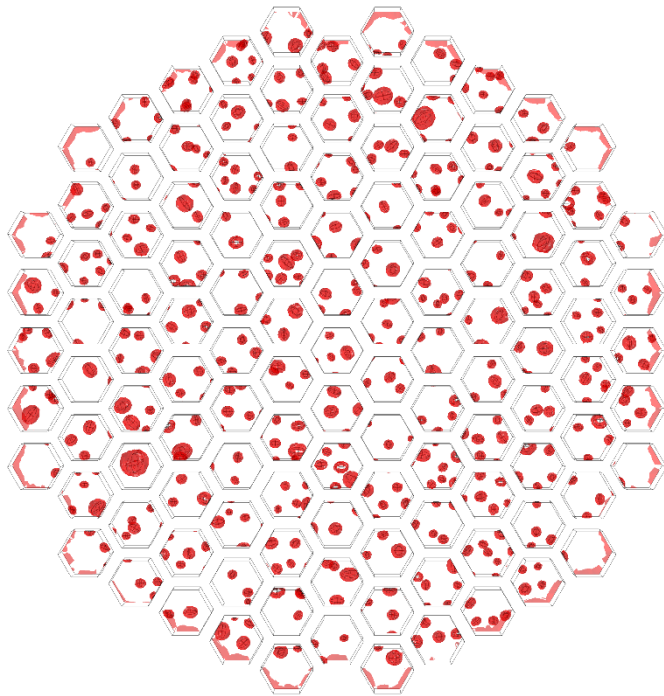
The 5% of the Nb_3Sn volume at the highest stress is highlighted in red

Here the red regions are at $\sigma \geq 275$ MPa

The critical stress is thus $\sigma_c = 275$ MPa

Statistical FEM analysis: 2) with voids

- Voids are generated from the statistical analysis until the void fraction of the wire is reached
- Simulation runs at increasing ε until $\sigma \geq \sigma_c$ in 5% of the filaments
- The goal is to predict the reduction of ε_c induced by the voids

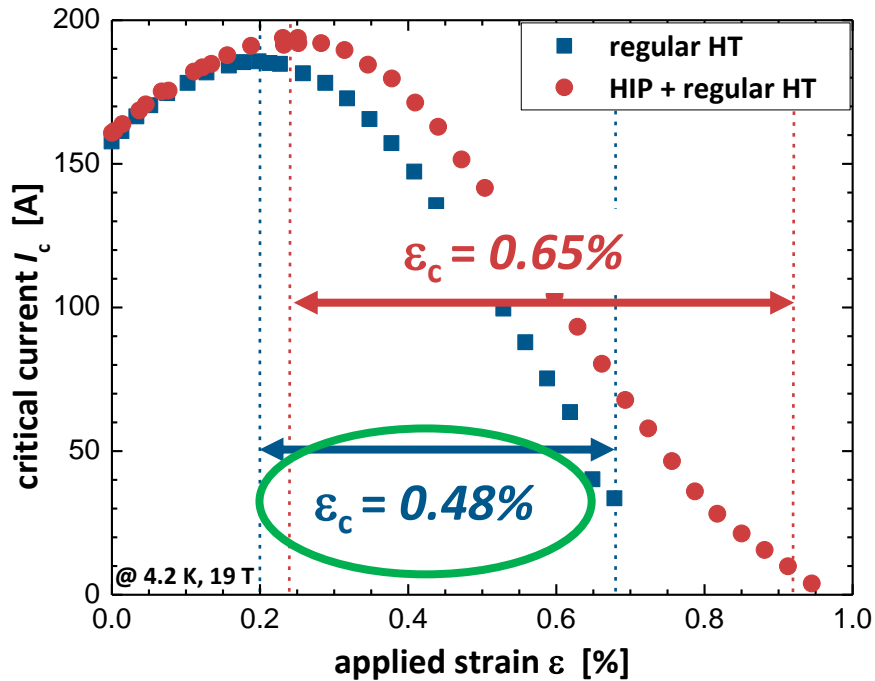


In the presence of voids the model predicts $\varepsilon_c = 0.50\%$

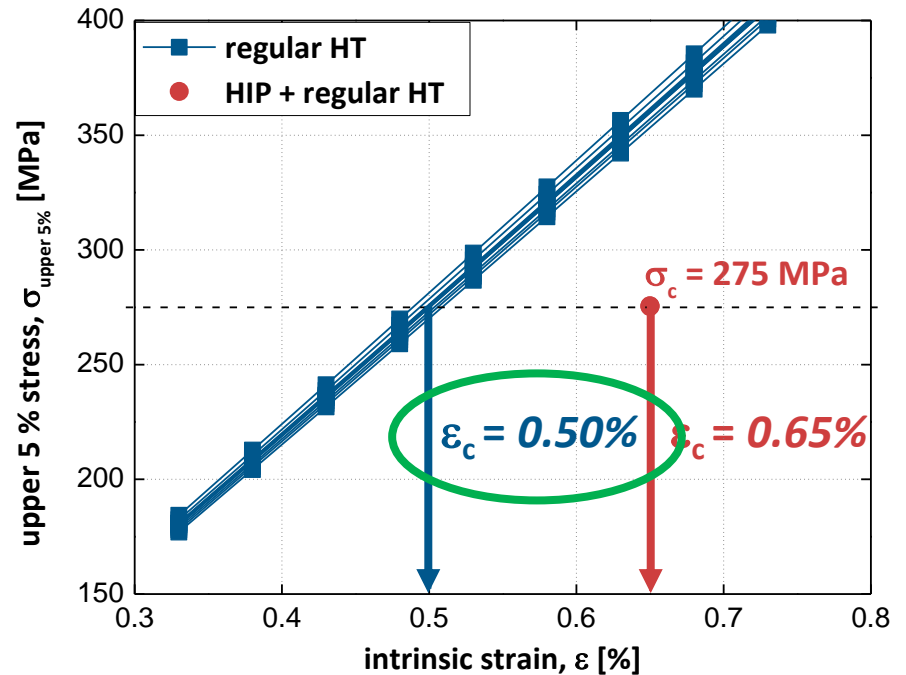
Irreversible limit in the presence of voids

Experiment vs Prediction

I_c vs strain measurement



FEM simulations



The simulations predict the correct value of ϵ_c when voids are introduced

What did we learn ?

Changes in the voids correlate quantitatively with the changes in the electromechanical limits

More details in SCIENTIFIC REPORTS 

OPEN

Quantitative correlation between the void morphology of niobium-tin wires and their irreversible critical current degradation upon mechanical loading

C. Barth¹, B. Seeber², A. Rack³, C. Calzolaio¹, Y. Zhai⁴, D. Matera¹ & C. Senatore¹

¹Department of Quantum Matter Physics (DQMP), University of Geneva, Geneva, Switzerland. ²Department of Applied Physics (GAP), University of Geneva, Geneva, Switzerland. ³European Synchrotron Radiation Facility (ESRF), Grenoble, France. ⁴Princeton Plasma Physics Laboratory (PPPL), Princeton University, Princeton, NJ, USA. Correspondence and requests for materials should be addressed to C.B. (email: christian.barth@unige.ch)

SCIENTIFIC REPORTS | (2018) 8:6589 | DOI:10.1038/s41598-018-24966-z

Case study on Bronze Route, what about Internal Sn and PIT ?

The same approach may lead to the prediction of how much ϵ_c can be increased by reducing the void fraction

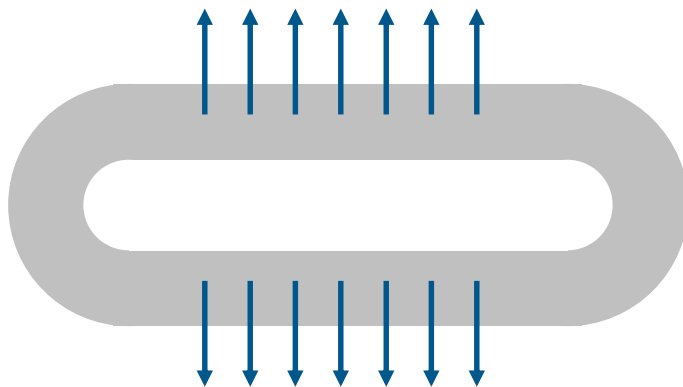
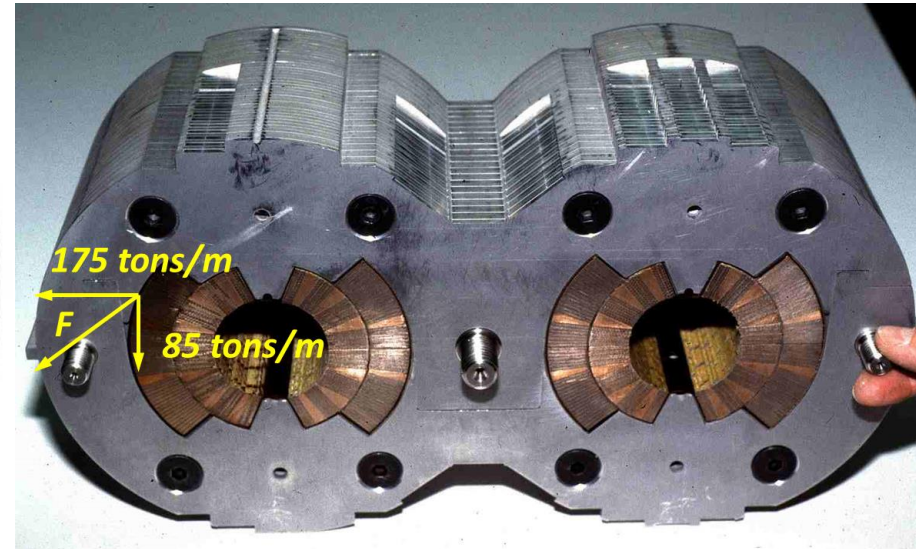
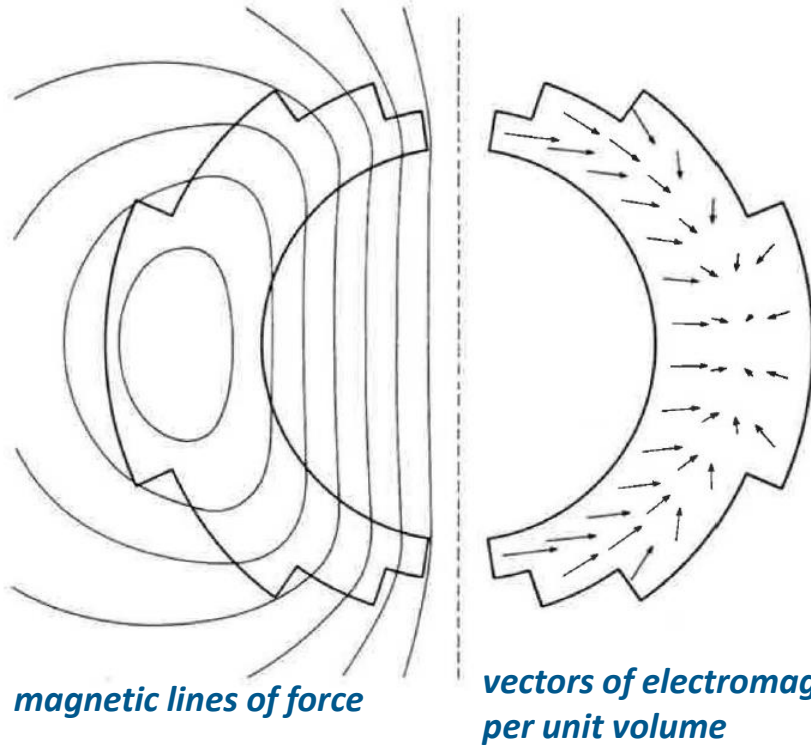
Strain-induced changes in Nb₃Sn wires

Reversible effects of strain ✓

Intrinsic mechanisms behind the irreversible degradation of the critical current ✓

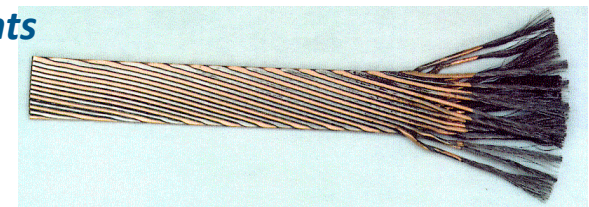
Influence on the stress tolerance of the way the mechanical stresses are exerted

Electromagnetic stresses in an accelerator dipole

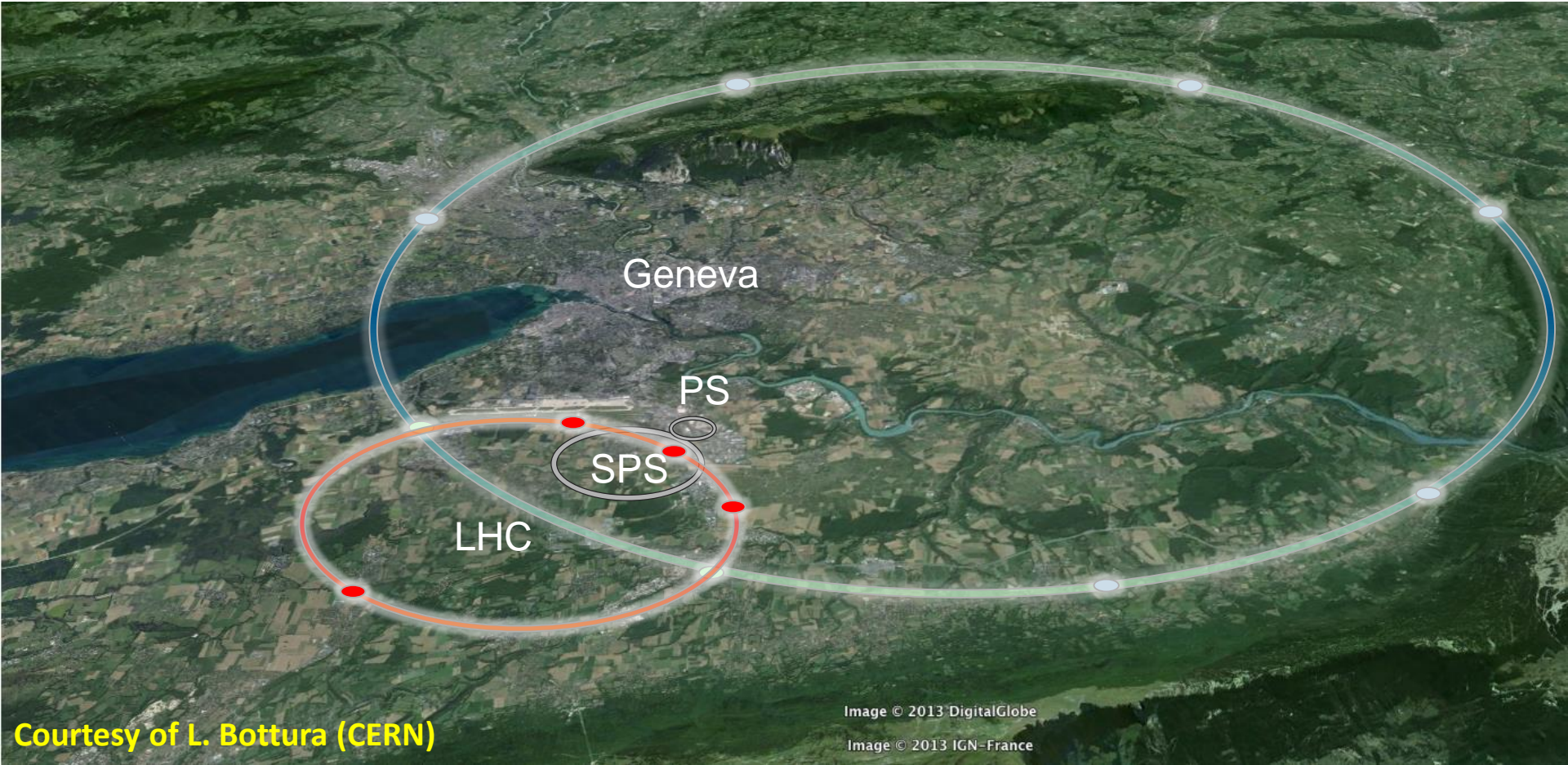


Accelerator magnets operate at ~10 kA to keep inductance low and ease magnet protection

Rutherford cables are used to get these large operating currents



The Future Circular Collider Study



Courtesy of L. Bottura (CERN)

Image © 2013 DigitalGlobe
Image © 2013 IGN - France

LHC	HE-LHC	FCC-hh
27 km, 8.33 T	27 km, 16 T	100 km, 16 T
14 TeV (c.o.m.)	27 TeV (c.o.m.)	100 TeV (c.o.m.)
1300 tons NbTi	~3000 tons Nb ₃ Sn	~9000 tons Nb ₃ Sn

Degradation upon transverse loads

*The 16 T FCC dipoles are being designed with a **peak stress of 200 MPa** at operation*

Are the Nb_3Sn wires in the cable able to withstand such a high stress level? Which degradation is tolerable?



Nb_3Sn Rutherford cable for HL-LHC, 40 strands

- *Nb_3Sn wires are deformed during cabling*
- *Cables are braided with glass fiber*
- *The winding is impregnated with resin*

The irreversible limit under transverse stress is influenced by parameters extrinsic to the wire

- *the type of impregnation (the elastic modulus of the resin)*

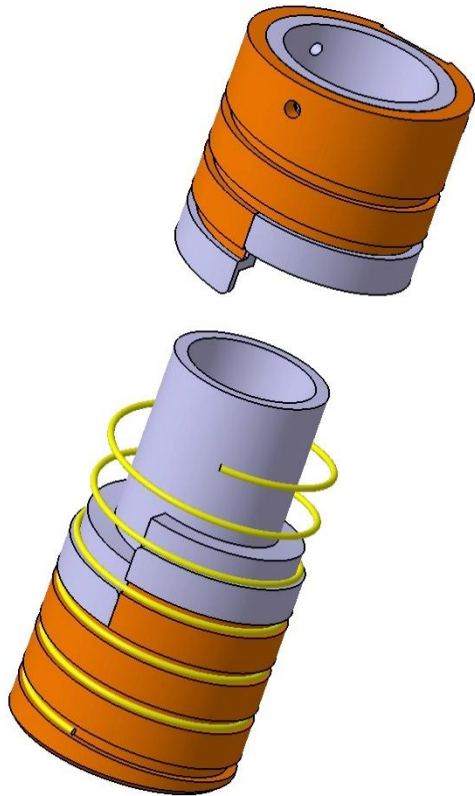
Tested Epoxy L, Glass fiber + Epoxy L, Stycast

- *the redistribution of the applied stress on the wire*

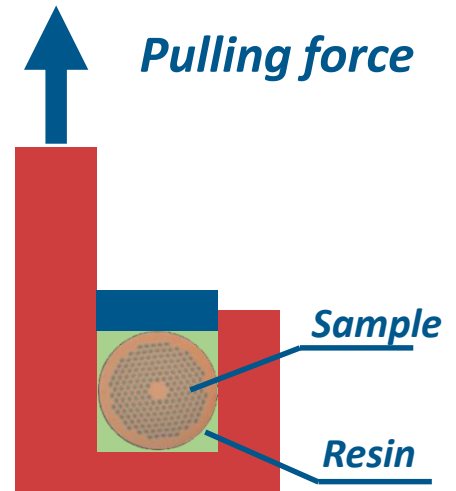
i.e. the deformation of the wire during cabling



The WASP concept for I_c vs. transverse stress

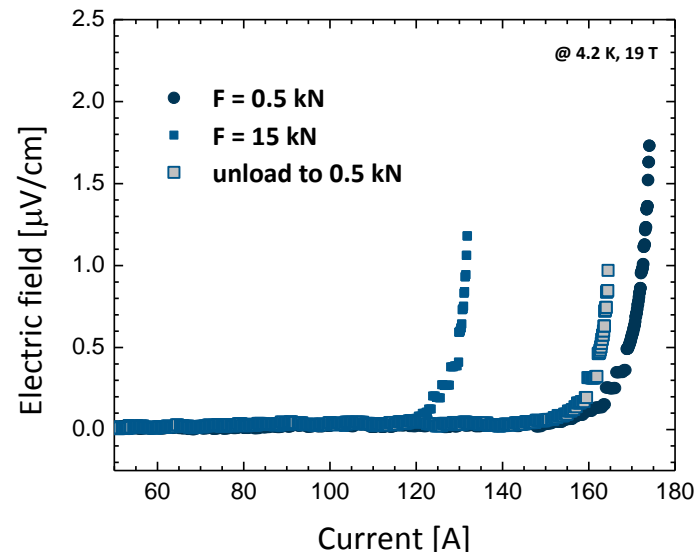
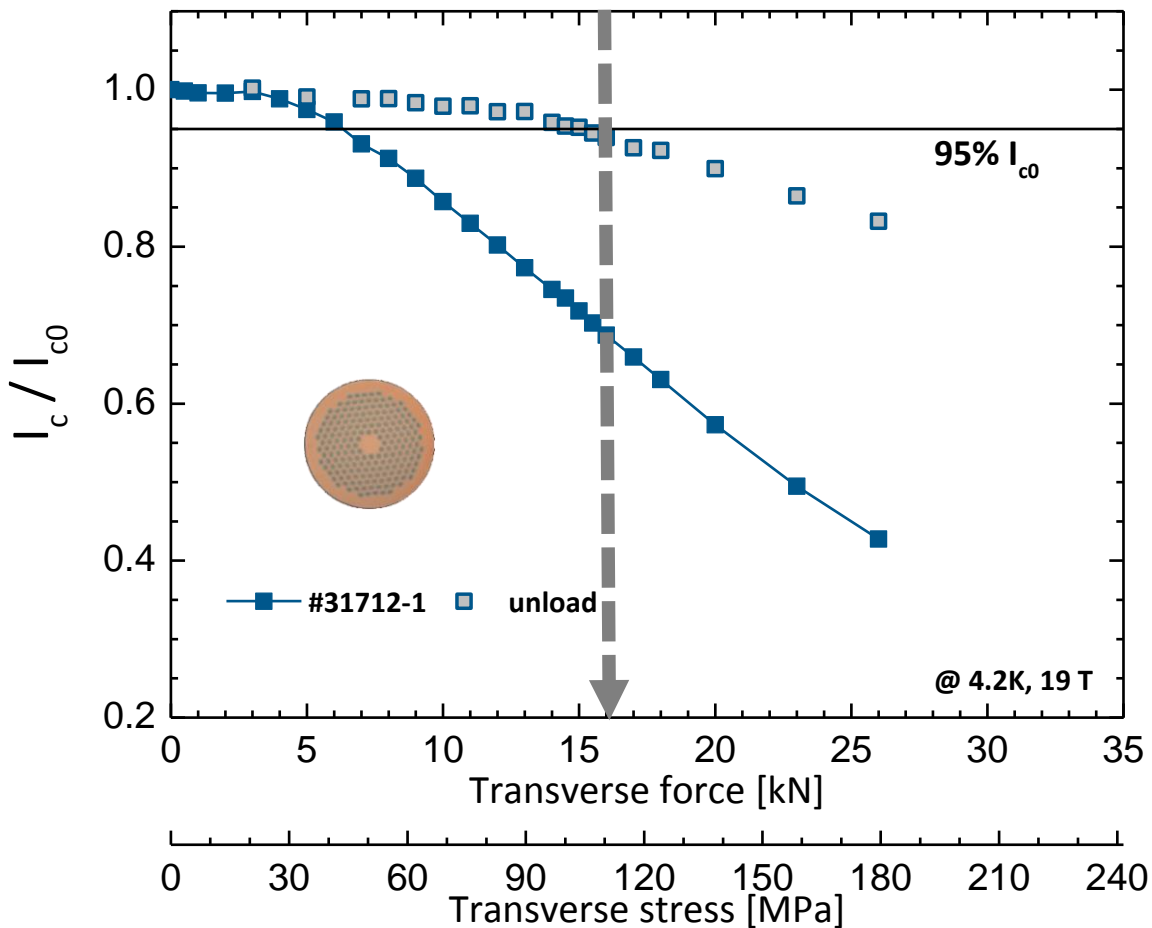


4-WALL + impregnation



I_c vs. transverse stress

PIT 192 + Epoxy L



The irreversible limit is defined at the force level leading to a 95% recovery of the initial I_c after unload

Here

$$F_{irr} = 16 \text{ kN}$$

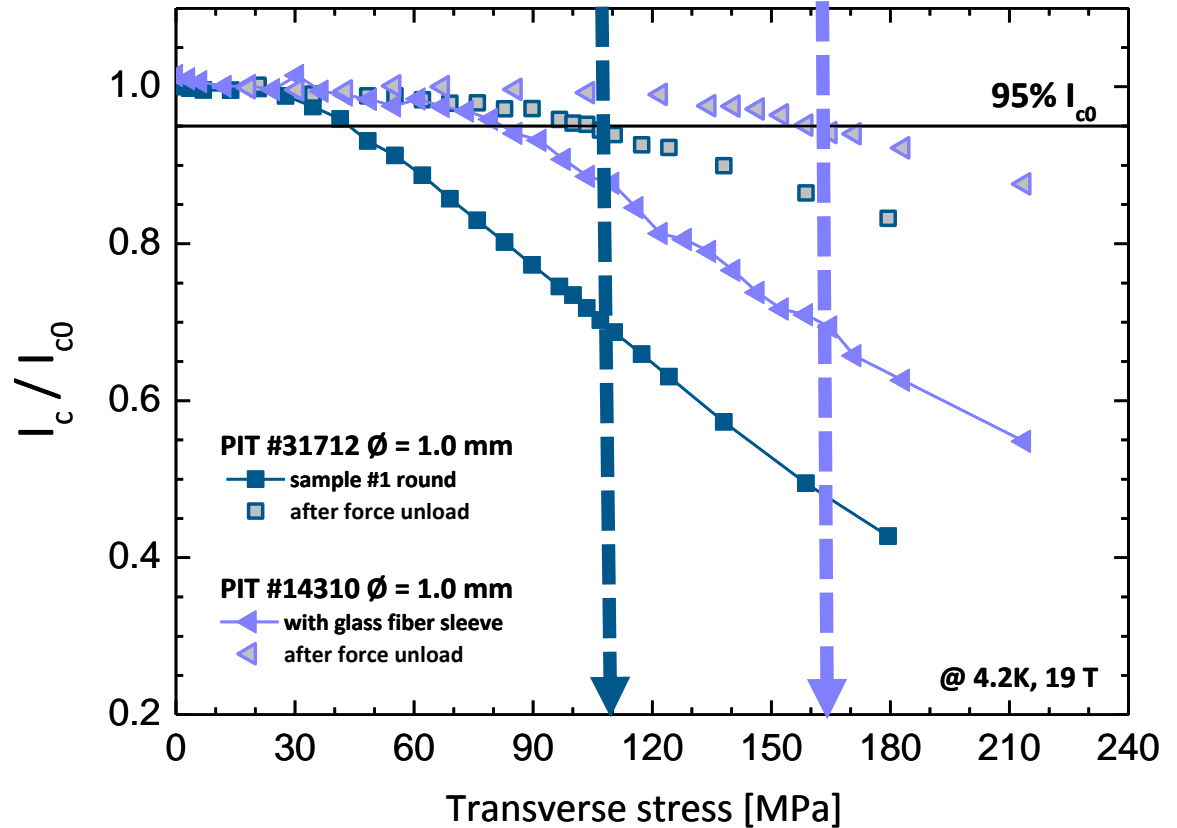
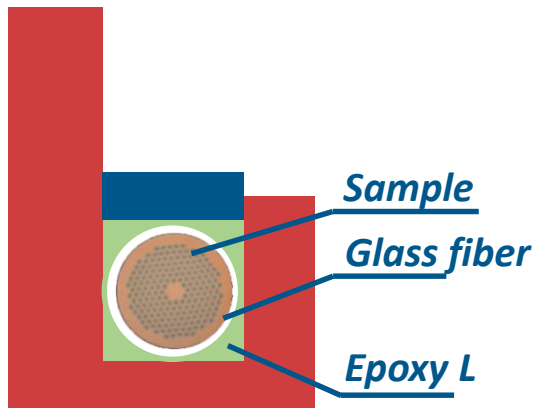
The corresponding irreversible stress limit is

$$\sigma_{irr} = 110 \text{ MPa}$$

where

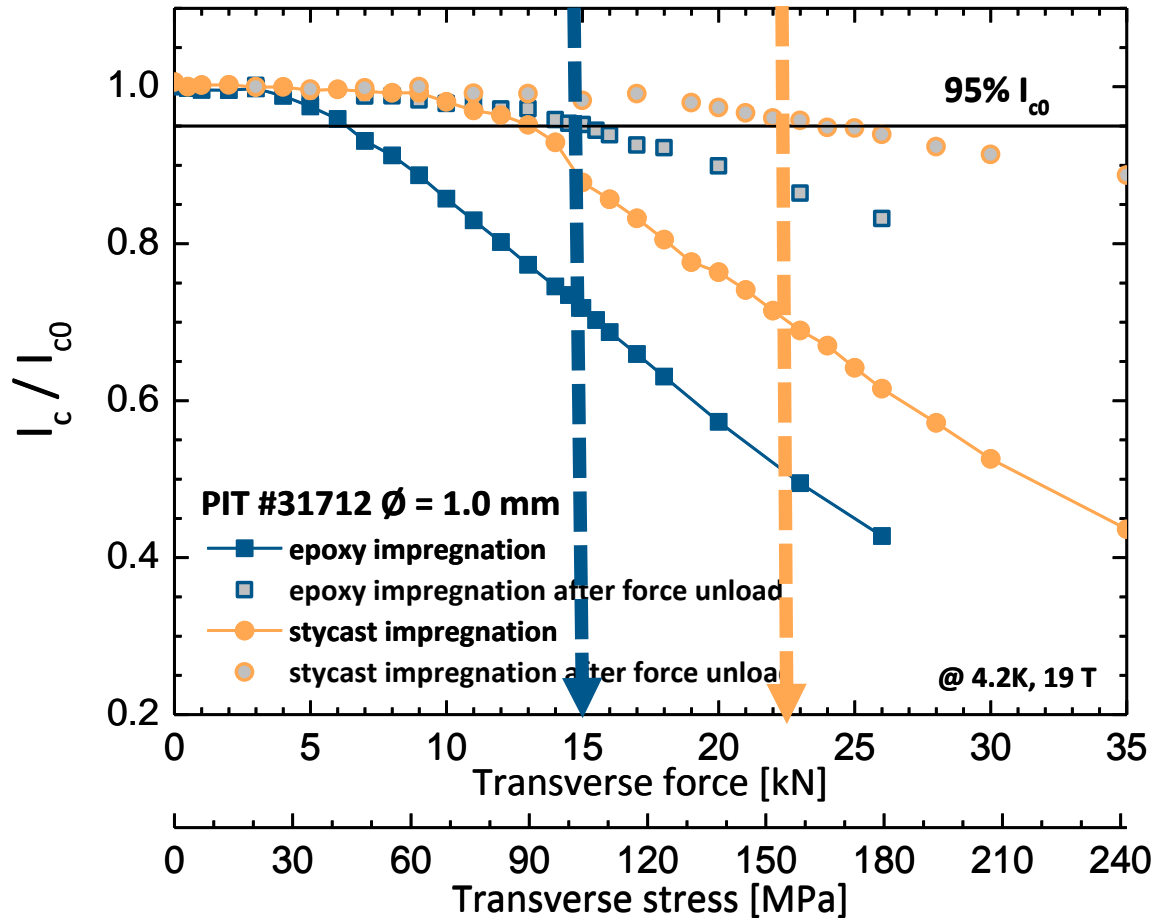
$$\text{Stress} = \frac{\text{Force}}{\text{groove length} \times \text{groove width}}$$

I_c vs. transverse stress: wire in a glass fiber sleeve



Glass fiber adds rigidity to the impregnation \Rightarrow Shift of σ_{irr} by > 50 MPa

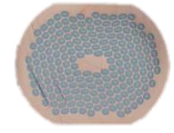
I_c vs. transverse stress: Epoxy L vs. Stycast



The Young modulus of Stycast is 3 to 4 times higher compared to the value of Epoxy L

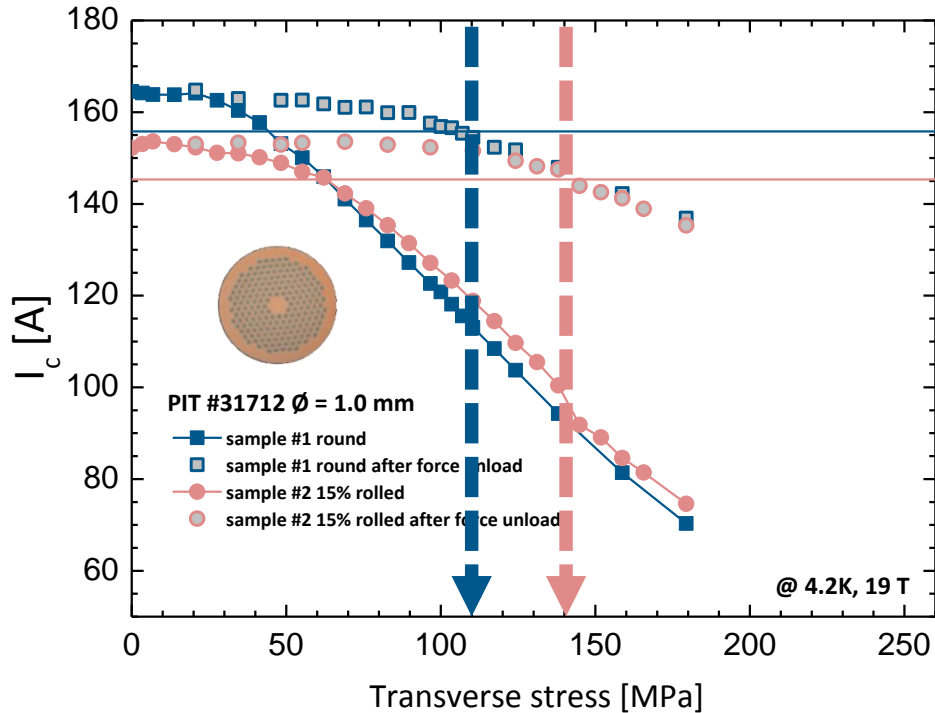
The change of resin, from epoxy to stycast, leads to an increase of σ_{irr} by > 50 MPa
The result is comparable to the value found with epoxy + glass fiber sleeve

I_c vs. transverse stress on flattened wires



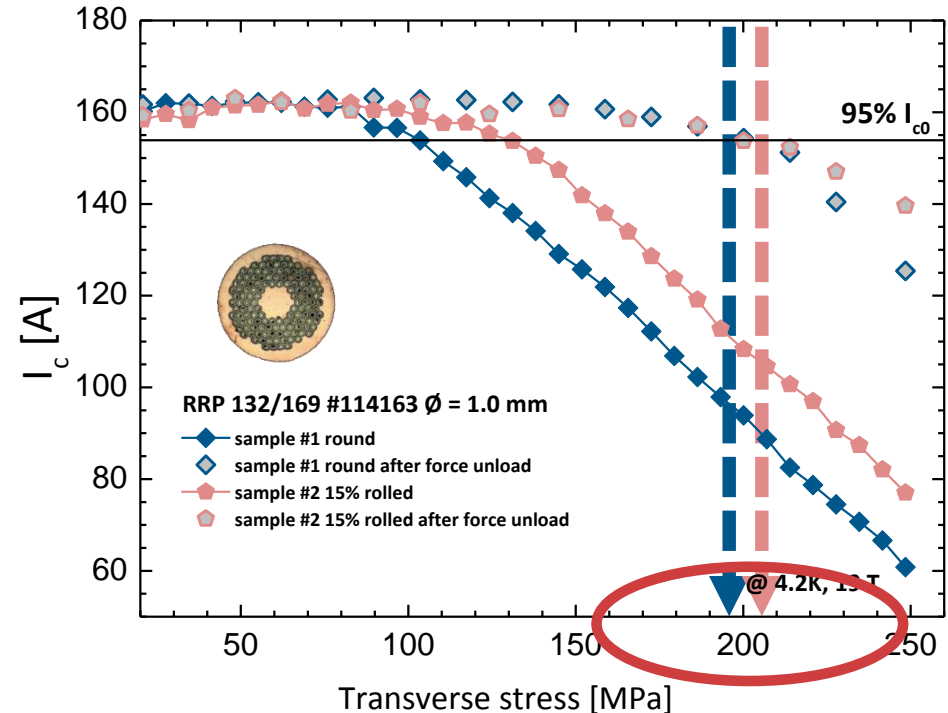
Effects of the stress redistribution

PIT 192



~7.5% I_c reduction by rolling
Shift of σ_{irr} by ~ 40 MPa

Internal Sn 132/169



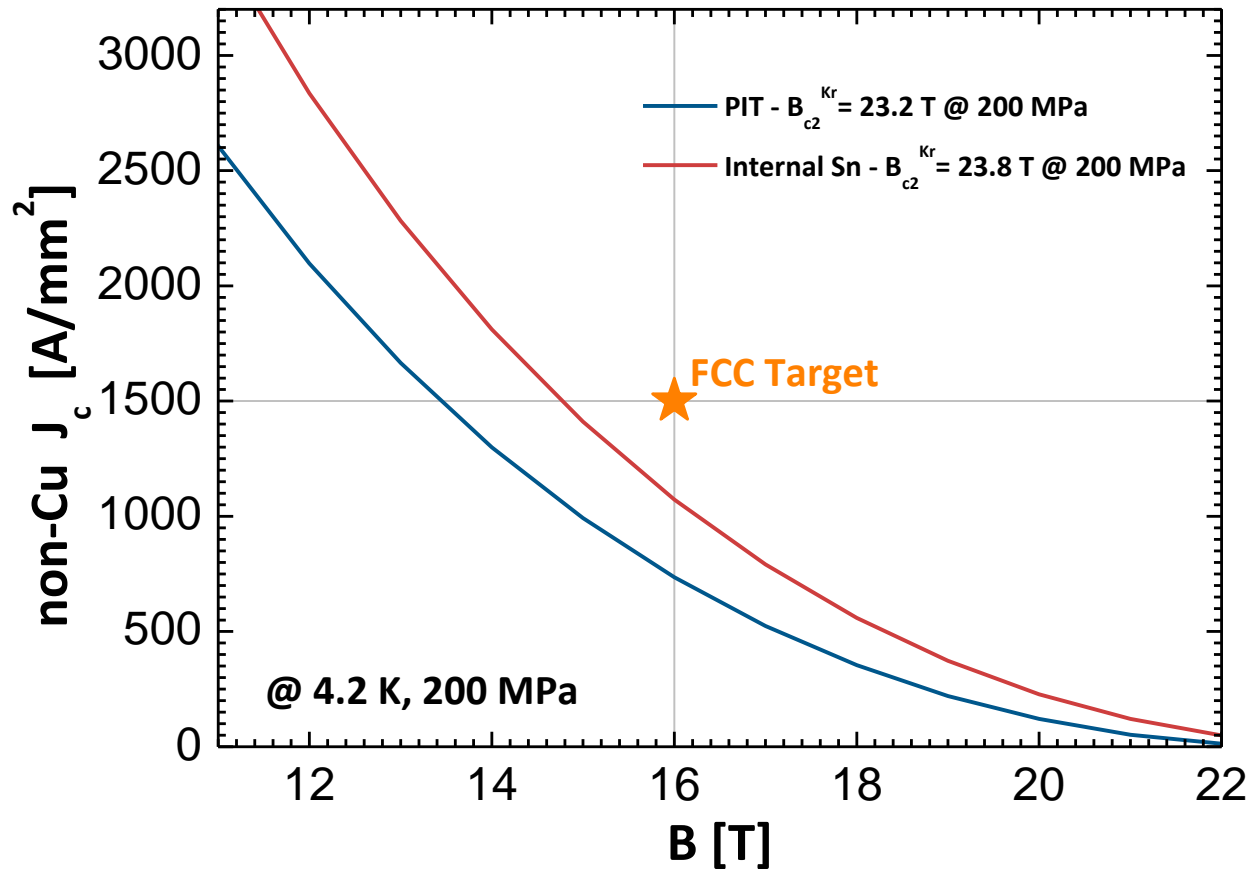
NO I_c reduction by rolling
Shift of σ_{irr} by ~ 15 MPa

This wire can withstand the peak stress of the FCC dipoles !!

Don't forget the reversible effects of stress !!

Critical current density vs. magnetic field @ 200 MPa

Best performance achieved so far in industrial wires



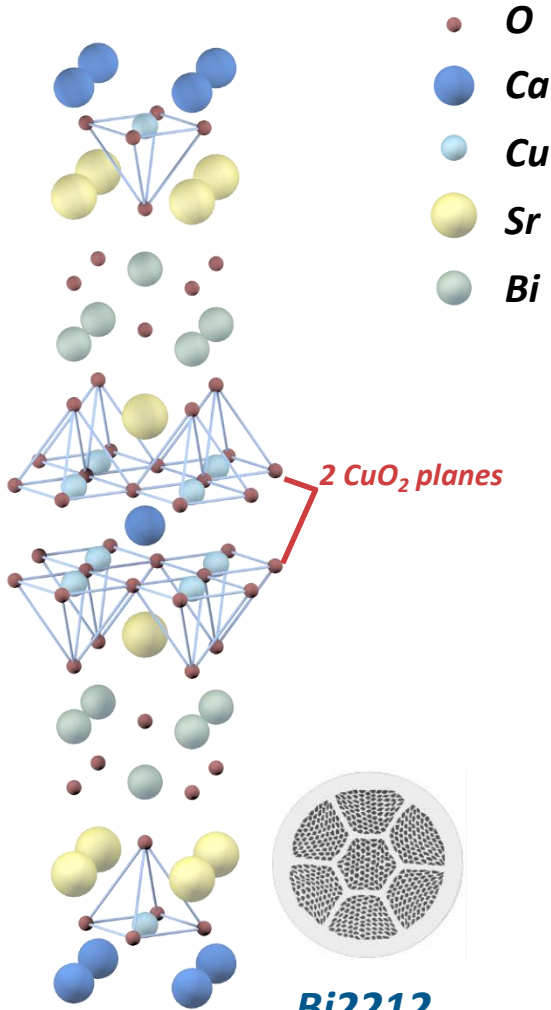
J. Parrell et al., AIP Conf. Proc. 711 (2004) 369

T. Boutboul et al., IEEE TASC 19 (2009) 2564

*High Temperature Superconductors are
different animals ...*

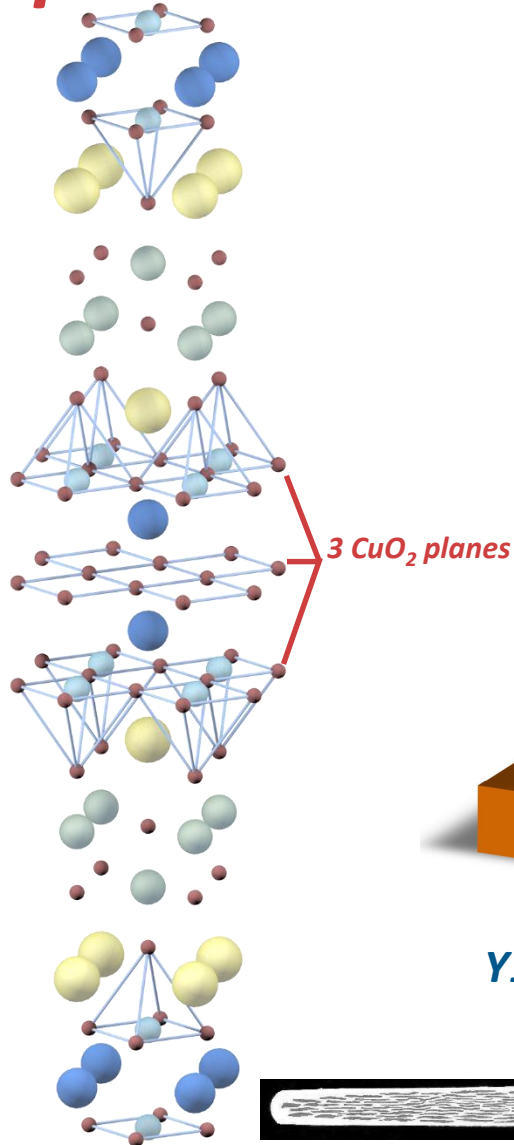


HTS materials for applications

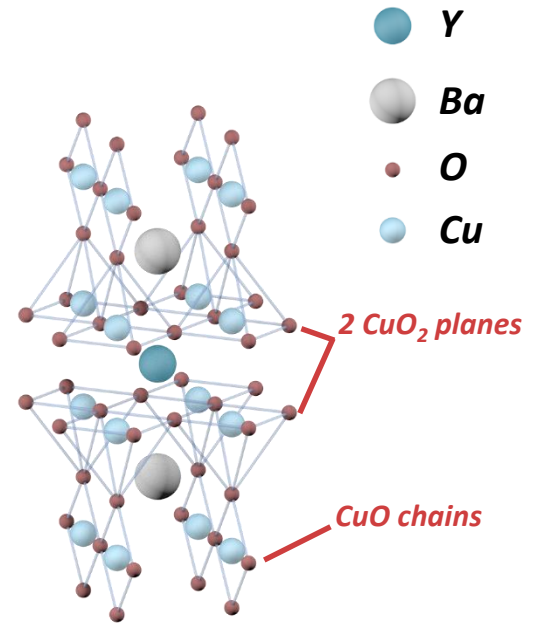


Powder-In-Tube wire

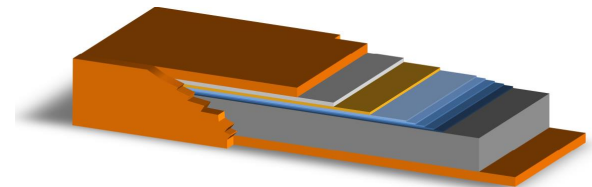
Bi2212



Bi2223 Powder-In-Tube tape

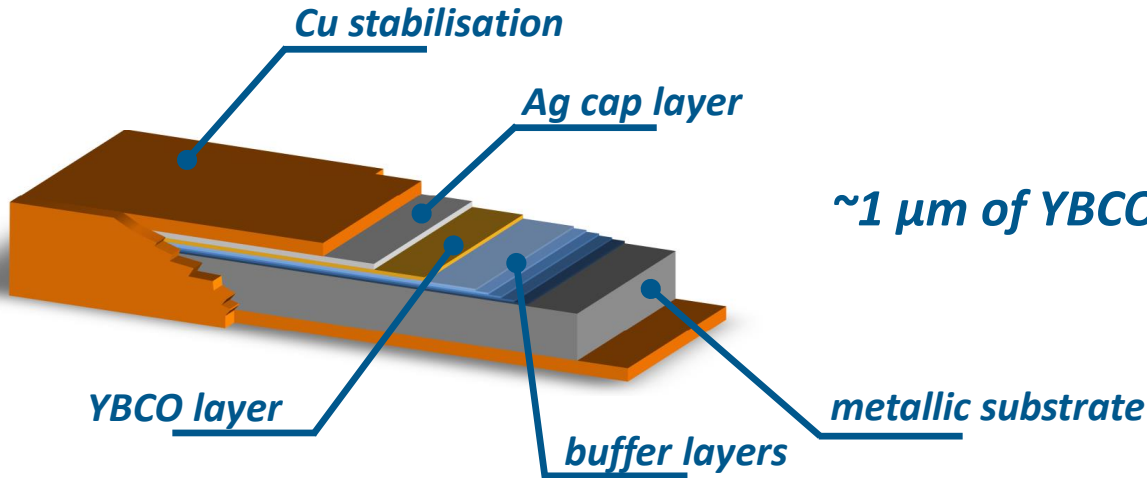
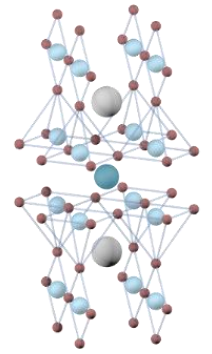


Y123

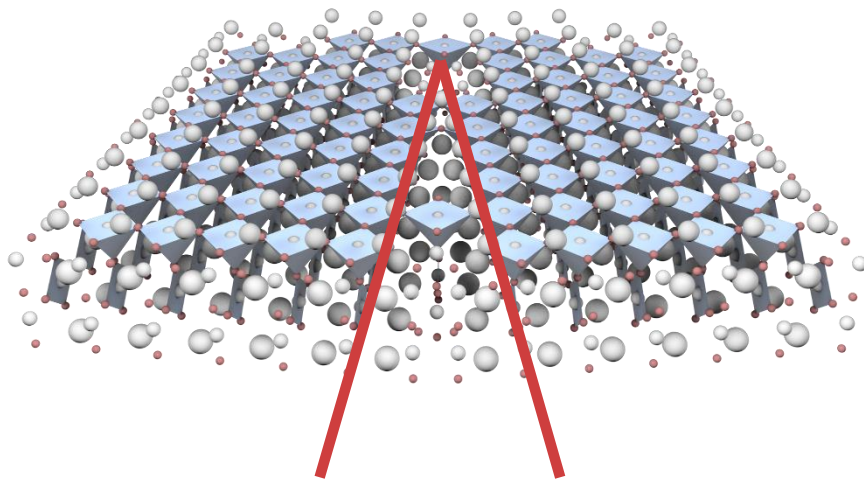


Y123 Coated Conductor

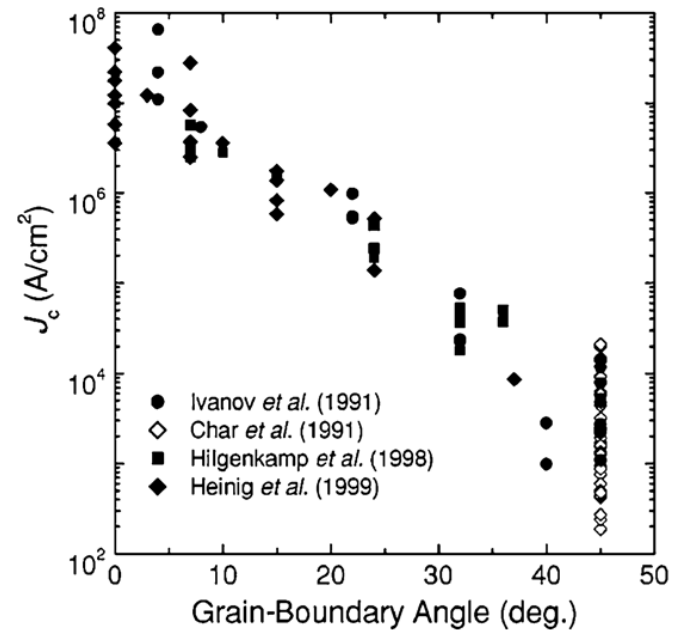
$YBa_2Cu_3O_{7-x}$ (YBCO) coated conductors



$\sim 1 \mu\text{m}$ of YBCO in a $\sim 100 \mu\text{m}$ thick tape



[001] tilt grain boundary

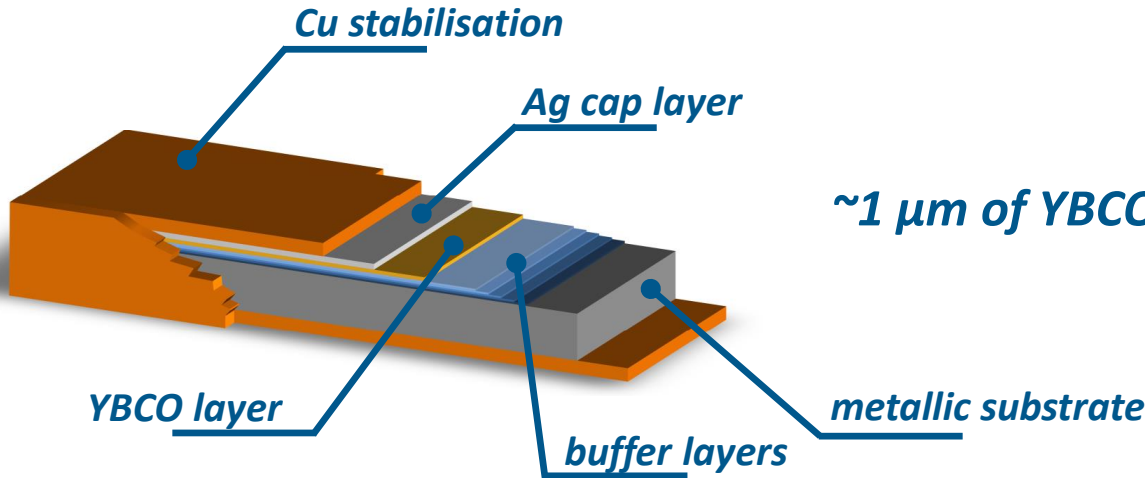


REBCO, RE = Rare Earth

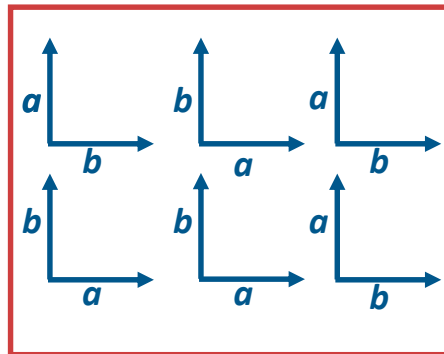
$YBa_2Cu_3O_{7-x}$ (YBCO) coated conductors

Iijima et al., APL 60 (1992) 769

Goyal et al., APL 69 (1996) 1795



$\sim 1 \mu\text{m}$ of YBCO in a $\sim 100 \mu\text{m}$ thick tape



Top view

The template is a metallic substrate coated with a multifunctional oxide barrier

Biaxial texturing – within $< 3^\circ$ – is obtained

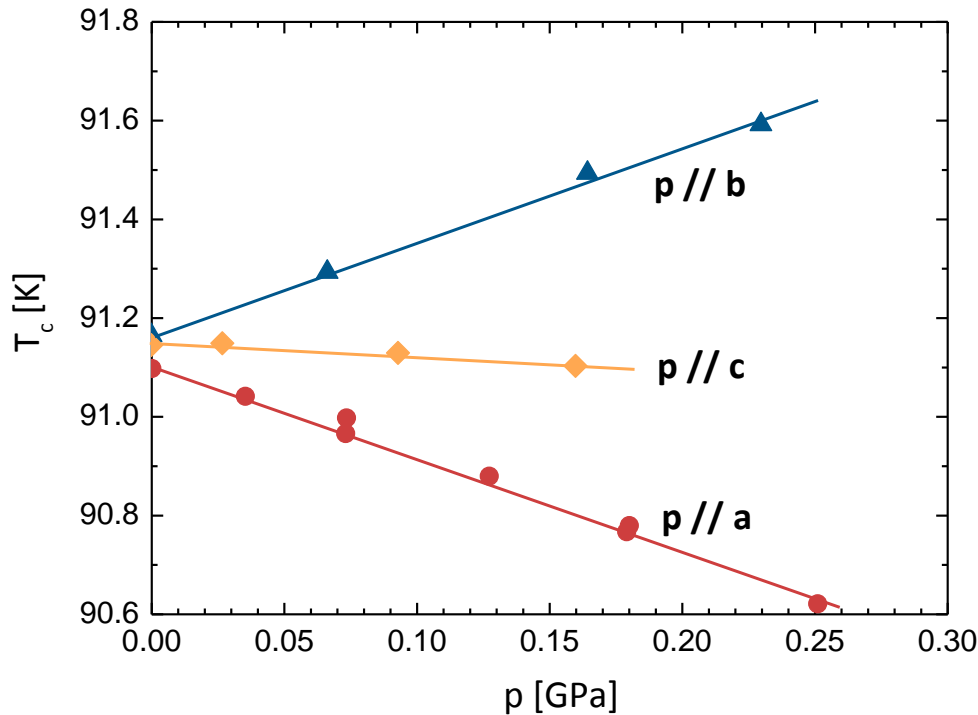
but with some also drawbacks:

- pronounced anisotropic behaviour
- complex and expensive manufacturing process

Presently produced by

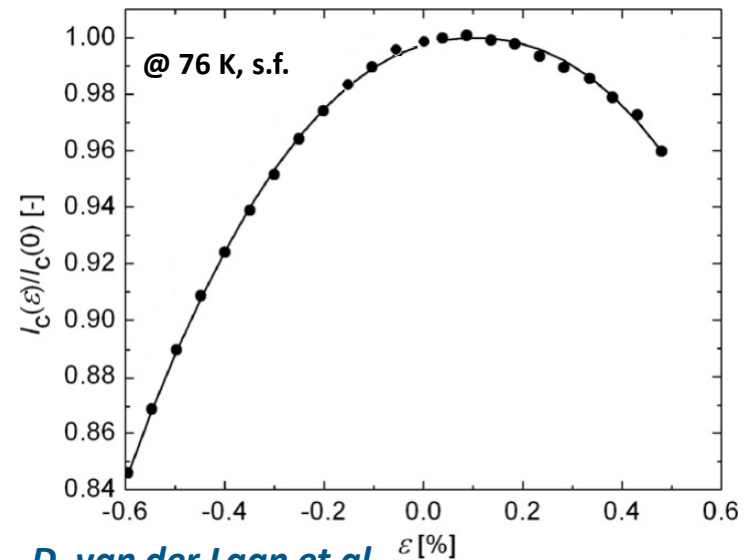


Reversible effects of strain on YBCO



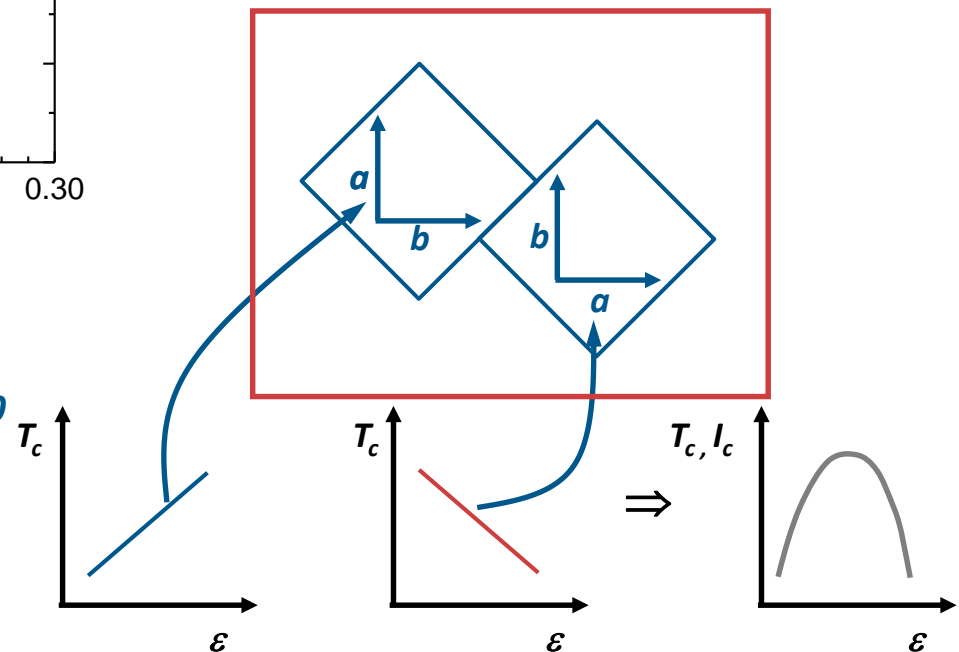
Critical temperature vs. uniaxial strain in single crystals

U. Welp et al., Phys. Rev. Lett. 69 (1992) 2130



D. van der Laan et al., Supercond. Sci. Technol. 24 (2011) 115010

Critical current vs. uniaxial strain in coated conductors



In the following, the focus will be on

Nb_3Sn → *Today high field superconductor, up to 23.5 T in*

solenoidal coils, the material for



and



and only candidate material for the 16 T dipoles of



YBCO → *Tomorrow high field superconductor, more than 40 T demonstrated in solenoidal insert coils*, the way to get 20 T dipoles and beyond*























**tested in a resistive outsert*



All superconducting magnets beyond LTS

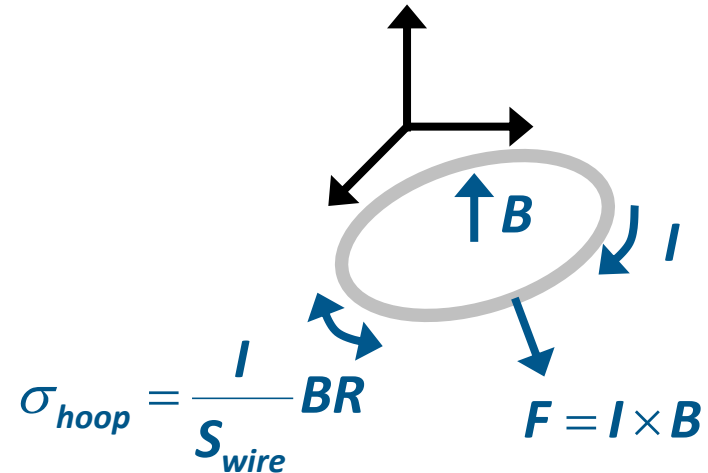
The World cup of high magnetic fields



		Maximum field [T]	HTS insert	HTS coil field [T]	Winding technology
		32	REBCO	17	DP
		27.6	Bi2223 / REBCO	4.5 (Bi2223) 6 (REBCO)	LW
		27.2	REBCO	12.2	DP
 	 	26.4	REBCO	26.4	DP
		25.7	REBCO	10	DP
 		25	REBCO	4	LW
 		24.6	Bi2223	10.6	DP
		24.2	Bi2223	3.66	LW

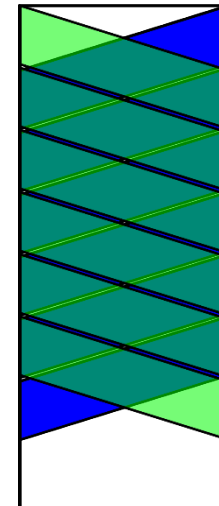
Stress management in Ultra High Field magnets

Hoop stress levels **above 100 MPa** are common, the NHMFL 32 T magnet operates at **400 MPa**



Other constraints to the winding come from the tape geometry of the conductor

In the case of layer winding the direction changes going from one layer to the next impose **in-plane hard bending** of the tape

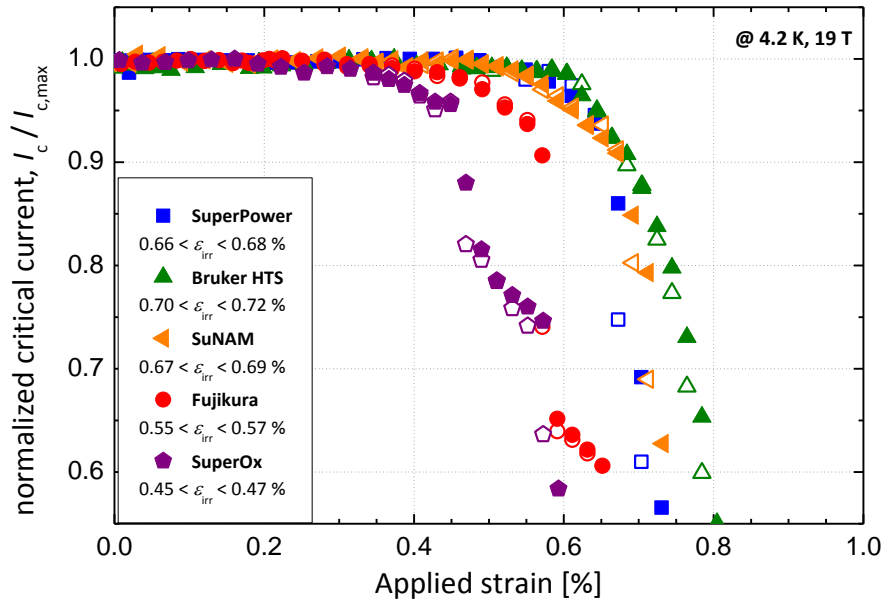


REBCO CCs: Dependence of I_c on axial loads

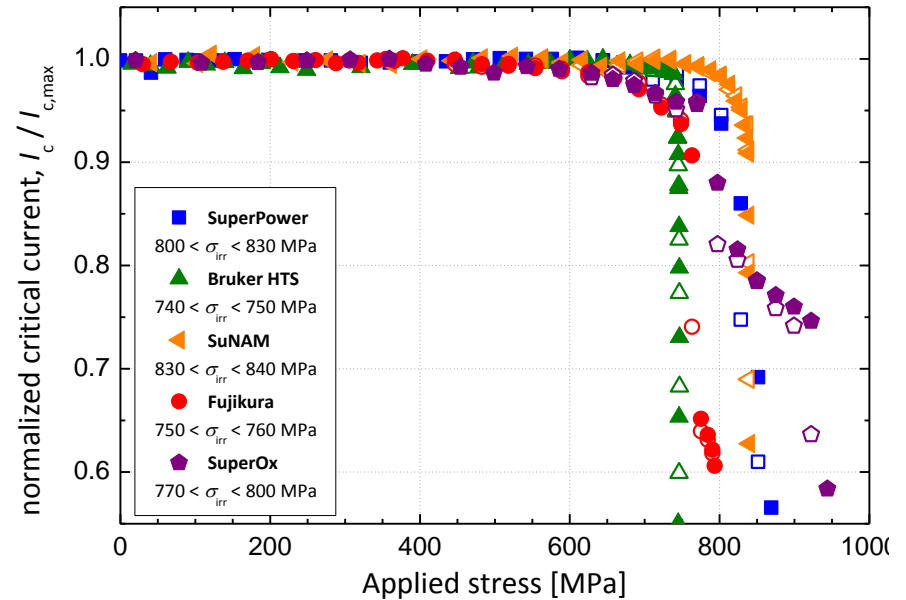
Reversible behaviour and irreversible limit



I_c vs. axial strain



I_c vs. axial stress

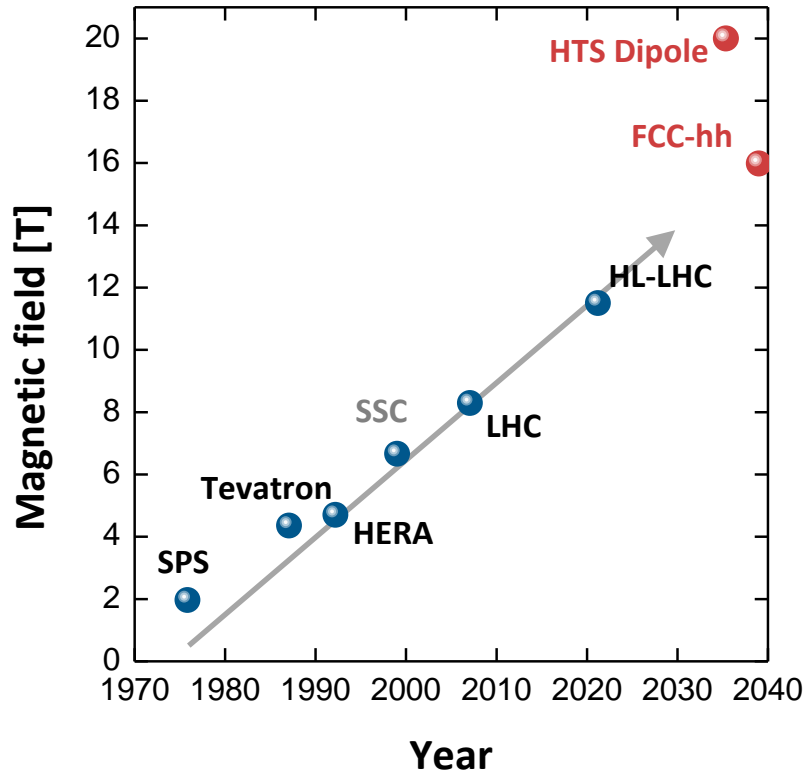


- REBCO CCs are inherently strong, ~50% is a high strength alloy
- Very low stress effect \rightarrow curves are flat in rev. region
- Irreversible stress limits above 500 MPa
- The only weakness is delamination...



HTS for accelerator magnets

The goal of 20 T in an accelerator quality dipole calls for HTS

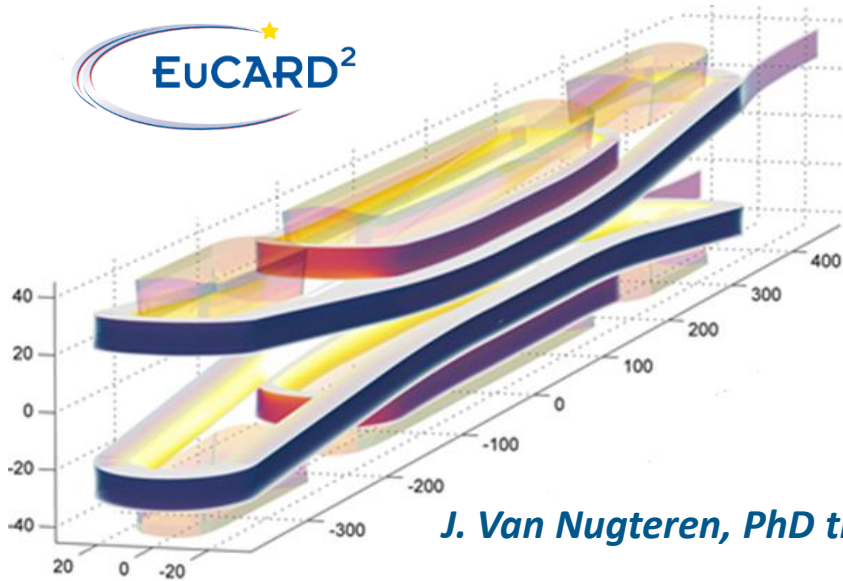


Accelerator magnets operate at ~10 kA to keep inductance low and ease magnet protection

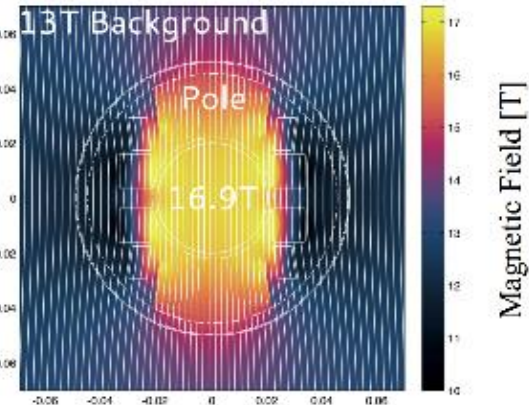
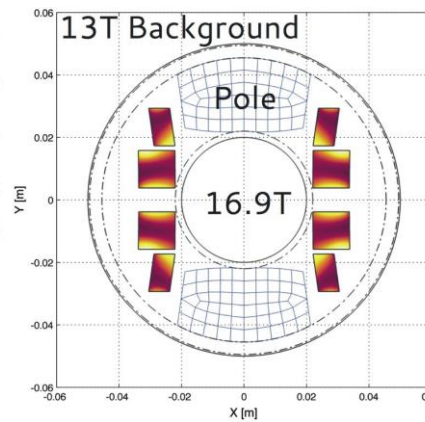
The most promising HTS CONDUCTOR for accelerator magnets is the Roebel Cable (10 kA-class cable)



A coil of Roebel cable: a Short Dipole Demonstrator

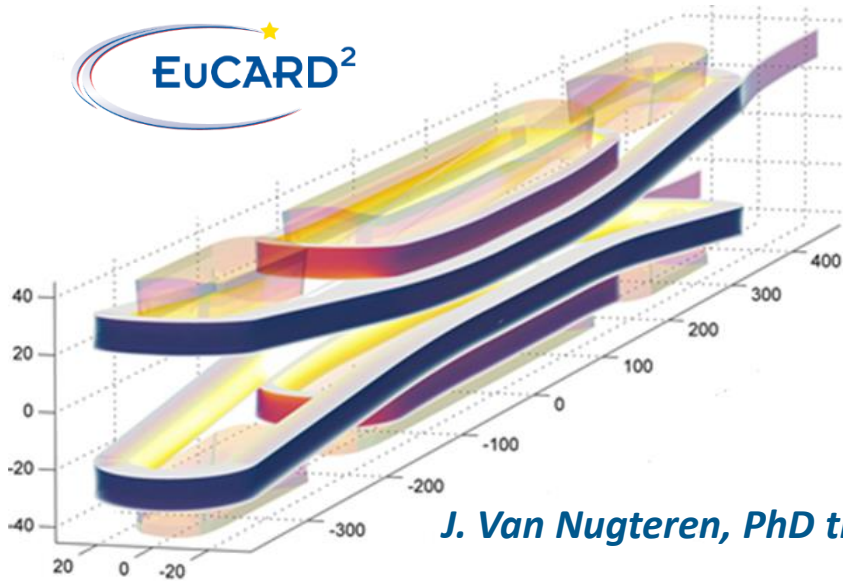


J. Van Nugteren, PhD thesis, 2016

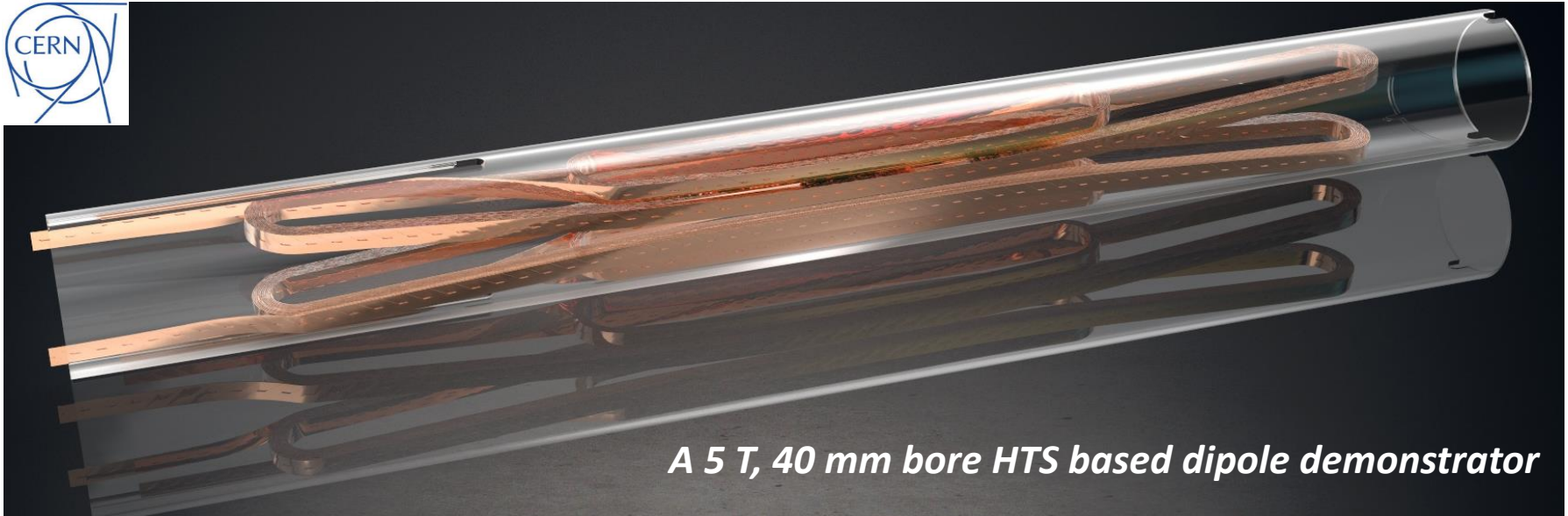
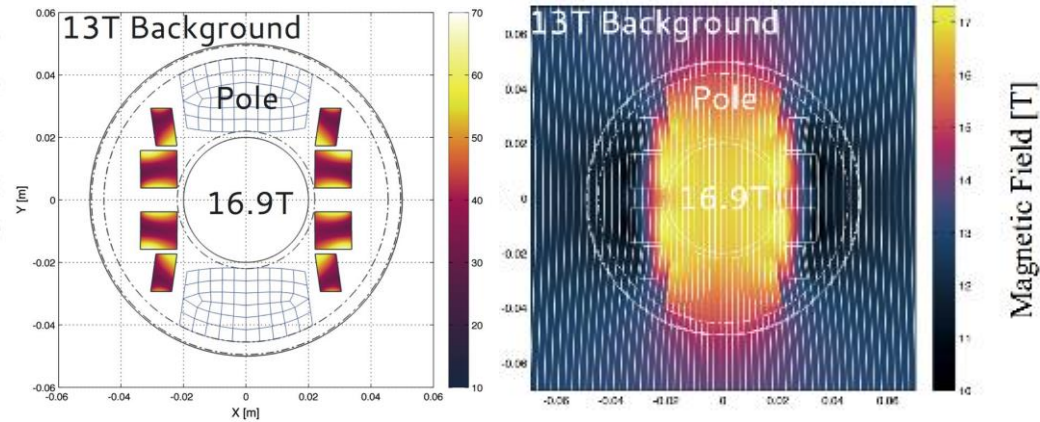


- ~1000 m of REBCO tape*
- ~70 m of Roebel cable*
- 5 T in standalone configuration*
- 40 mm aperture*

A coil of Roebel cable: a Short Dipole Demonstrator

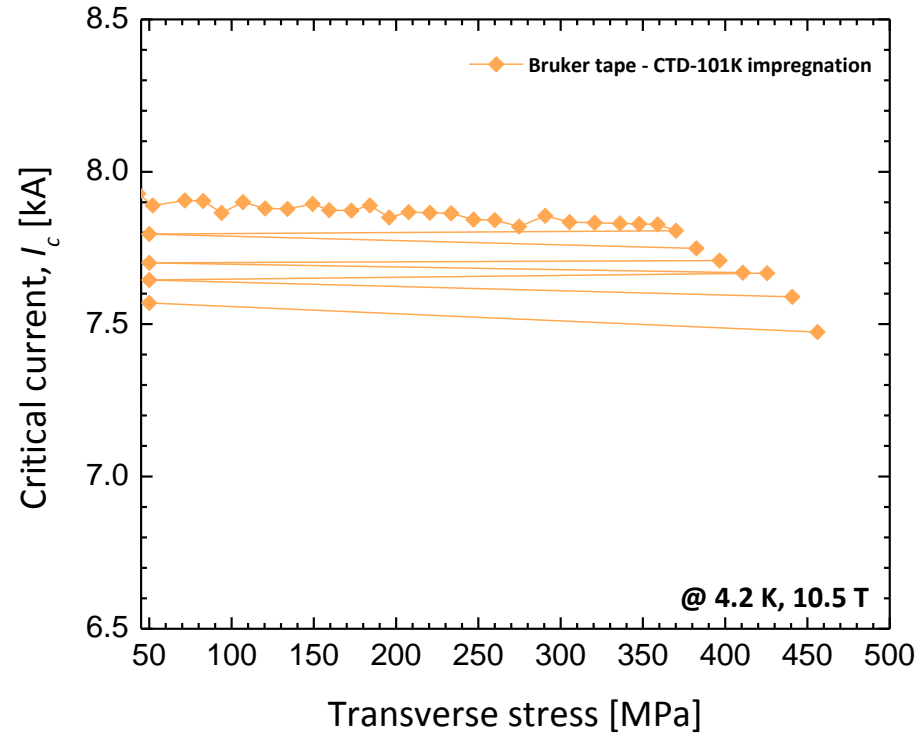
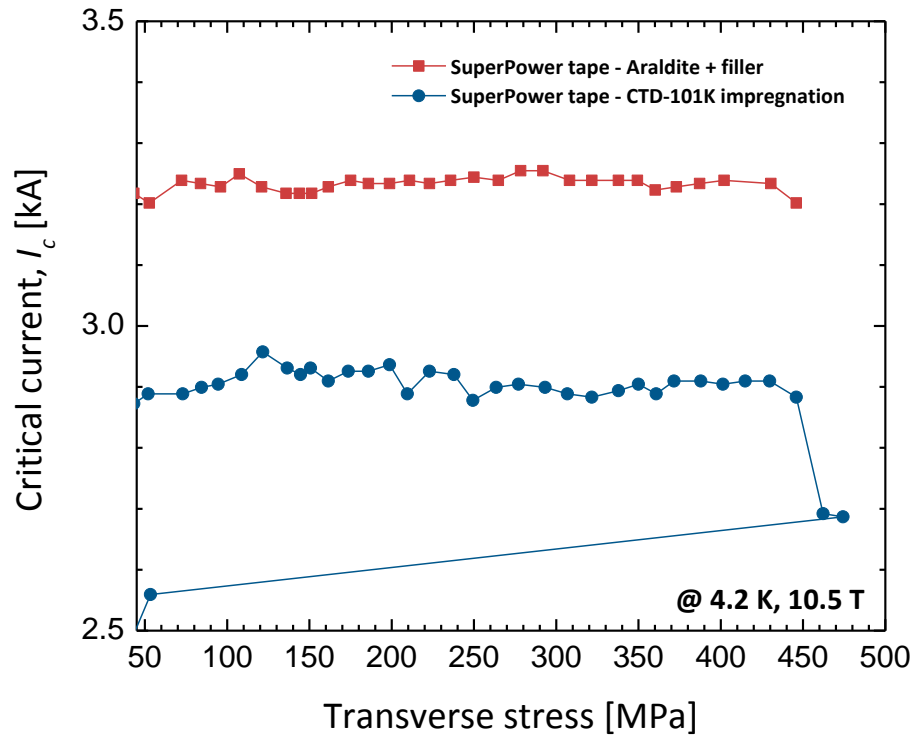


J. Van Nugteren, PhD thesis, 2016



A 5 T, 40 mm bore HTS based dipole demonstrator

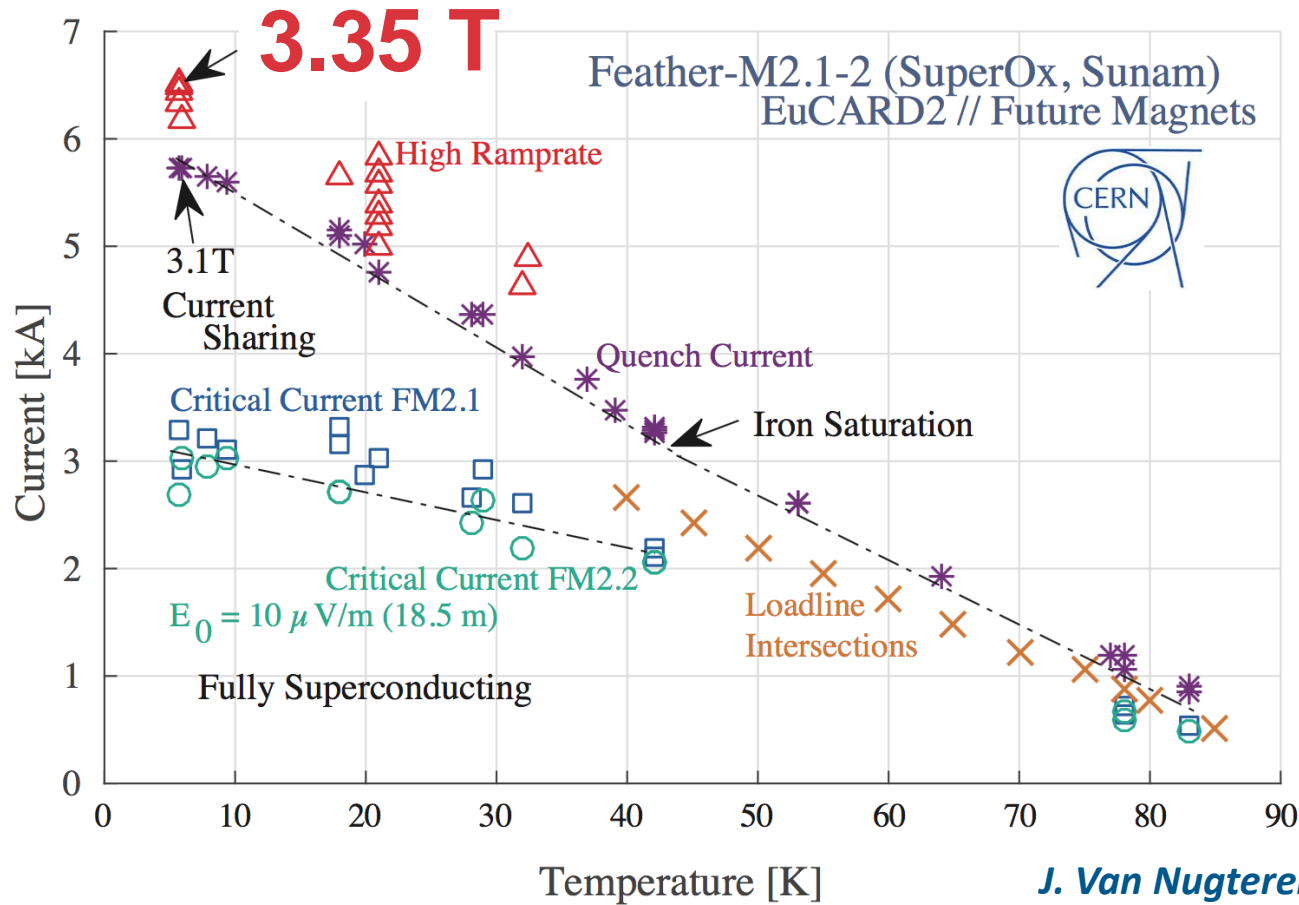
Are Roebel cables able to withstand large transverse loads?



The answer is YES: Irreversible stress limit ≥ 400 MPa

Short Dipole Demonstrator results

1st coil wound with moderate I_c SuNAM cable (~40% of the Bruker cable)



J. Van Nugteren @ EUCAS2017

New coil with ultra-high I_c cable from Bruker already wound
Expected field in stand-alone test ~7T

What we have learned...

- *Why we care about mechanical loads*
Electromagnetic forces in solenoids and accelerator magnets
- *What hides behind the reversible degradation of I_c*
Deviatoric strain, intrinsic anisotropy,...
- *Which are the intrinsic mechanisms that lead to the irreversible degradation of I_c*
Detailed study of the stress concentration at the voids in Nb_3Sn
- *How the stress tolerance depends also on factors extrinsic to the wire*
Load geometry, impregnation...
- *An overview of the electromechanical properties of Nb_3Sn and YBCO*



Thank you for the attention !

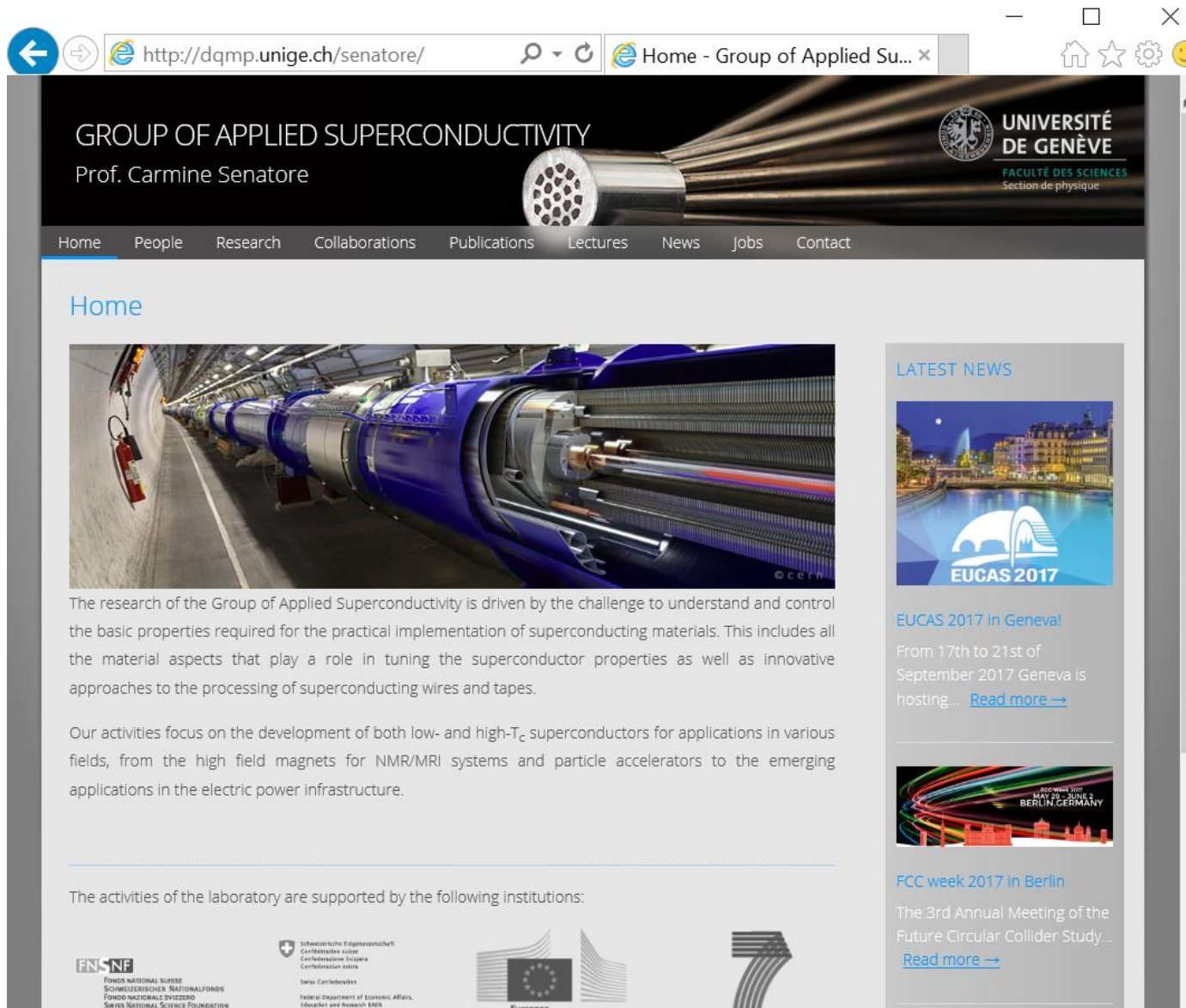
...time for questions...

Carmine SENATORE

carmine.senatore@unige.ch


<http://supra.unige.ch>

If you want to know more about applied superconductivity in Geneva, visit <http://supra.unige.ch>



The screenshot shows a web browser window with the address bar displaying <http://dqmp.unige.ch/senatore/>. The page header features the text "GROUP OF APPLIED SUPERCONDUCTIVITY" and "Prof. Carmine Senatore" on the left, and the "UNIVERSITÉ DE GENÈVE" logo and "FACULTÉ DES SCIENCES" on the right. A navigation menu includes "Home", "People", "Research", "Collaborations", "Publications", "Lectures", "News", "Jobs", and "Contact".


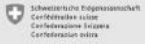


Home




The research of the Group of Applied Superconductivity is driven by the challenge to understand and control the basic properties required for the practical implementation of superconducting materials. This includes all the material aspects that play a role in tuning the superconductor properties as well as innovative approaches to the processing of superconducting wires and tapes.

Our activities focus on the development of both low- and high- T_c superconductors for applications in various fields, from the high field magnets for NMR/MRI systems and particle accelerators to the emerging applications in the electric power infrastructure.

The activities of the laboratory are supported by the following institutions:

-  FONDS NATIONAL SUISSE / SCHWEIZERISCHER NATIONALFONDS / FONDO NAZIONALE SVIZZERO / SWISS NATIONAL SCIENCE FOUNDATION
-  Schweizerische Eidgenossenschaft / Confœderaziun svizra / Confœderaziun Svizzera / Confederaziun elvira / Swiss Confederation
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- 


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