

Uncertainties in the measurement of the mechanical properties of Nb₃Sn coil constituents and cable stacks

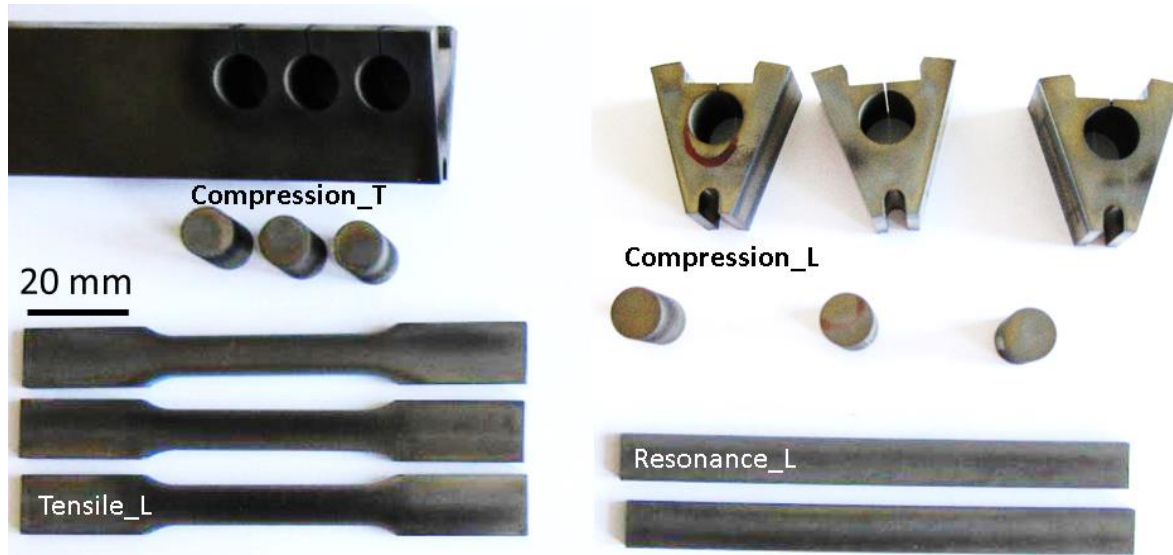
C. Scheuerlein, F. Lackner

*Nb₃Sn Rutherford cable characterization for accelerator magnets workshop,
17th November 2017, CIEMAT, Madrid, Spain*

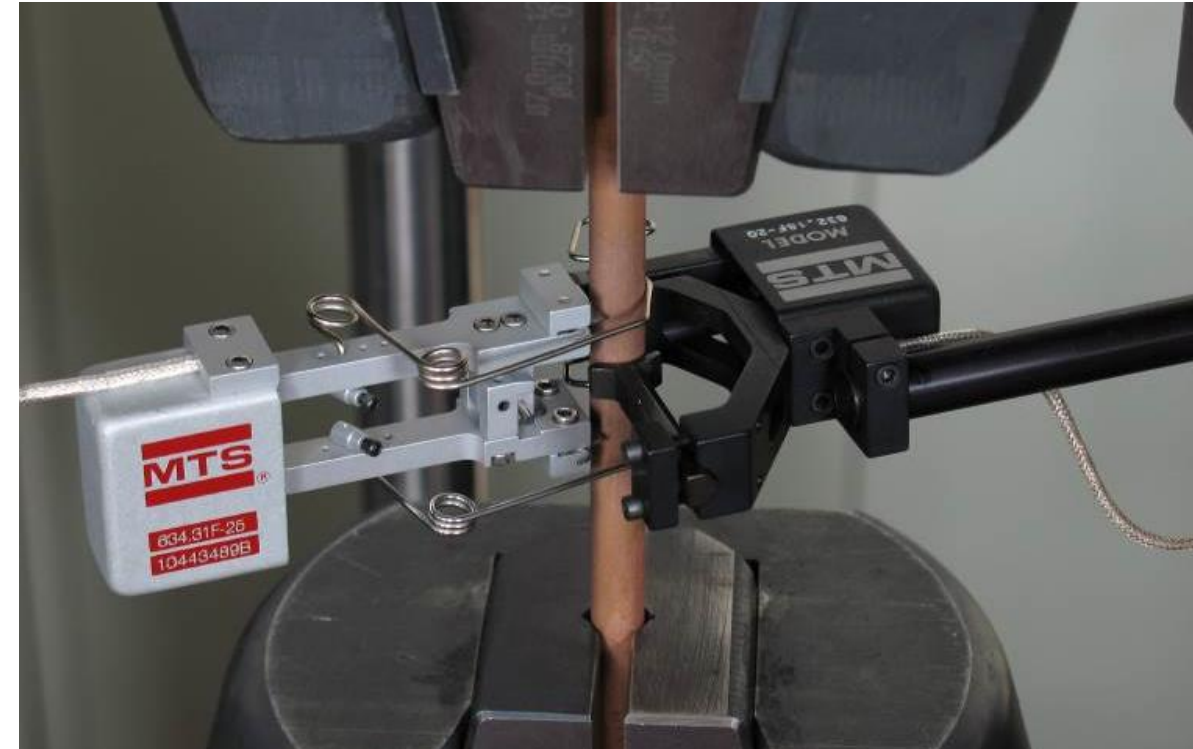
Outline

- Test sample geometries
- Uncertainties in dynamic Young's modulus measurements
- Uncertainties in stress-strain measurements using homogeneous samples with ideal testing geometry
 - Stress strain behaviour dependence on the load direction
 - Non linear stress-strain behaviour
 - Elastic anisotropy of the coil constituent materials
- Uncertainties in Nb₃Sn composite wire stress-strain measurements
- Uncertainties in the determination of the Nb₃Sn mechanical properties
- Uncertainties in stress-strain measurements of reacted and impregnated Nb₃Sn Rutherford cable stacks
- Conclusion

Sample geometries for standardised mechanical testing



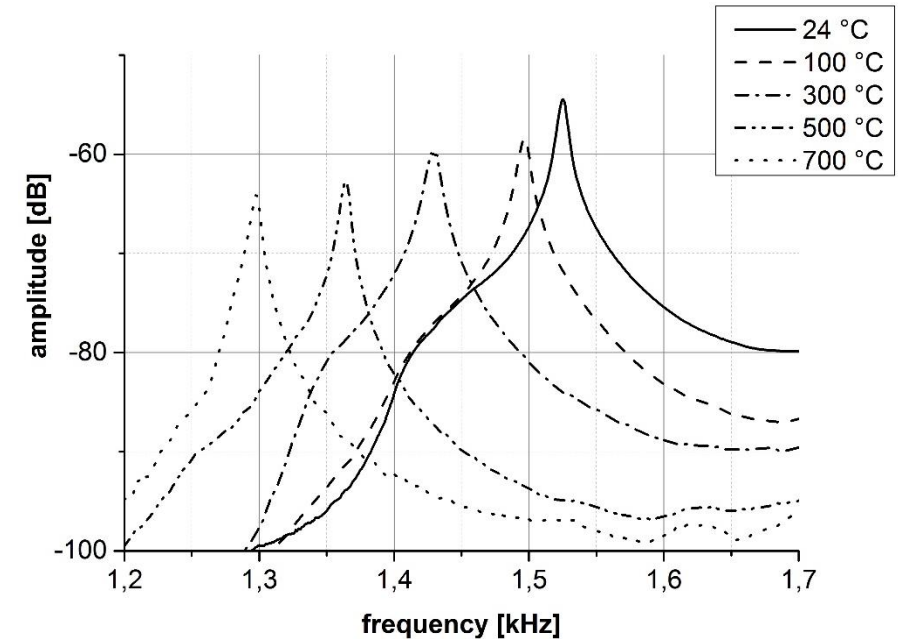
Standardized test samples for dynamic and static tensile and compression tests. All samples have been extracted from a Ti6Al4V pole wedge.



Sample for Poisson's ratio measurement and extensometers for axial and transverse strain measurements. The sample has been extracted from a DISCUP coil wedge.

Dynamic Young's modulus measurements

- For dynamic test methods (resonance and impulse excitation) the key elements defining the measurement uncertainty are the mass and physical size of the specimen [i].
- For both test methods a measurement uncertainty $< 1\%$ is realistic [ii,iii].
- Typical differences between dynamic and static Young's moduli (adiabatic vs isothermal modulus) are $< 1\%$ [ii].
- Since dynamic Young's modulus measurements are non destructive temperature dependent measurements are possible [iv].



Resonance peak of the first bending oscillation of rectangular 316LN specimen at different temperatures [iv].

[i] J.D. Lord, R. Morrell, NPL Good Practice Guide No. 98 "Elastic Modulus Measurement", NPL London 2006

[ii] F.H. Newmann, V.H.L. Searle, The general properties of matter. Issue 5, Edward Arnold, 1962

[iii] ASTM E 1875 "Standard Test Method for Dynamic Young's Modulus, Shear Modulus, and Poisson's Ratio by Sonic Resonance". 2013, ASTM International.

[iv] C. Scheuerlein, F. Lackner, F. Savary, B. Rehmer, M. Finn, C. Meyer, "Thermomechanical behavior of the HL-LHC 11 Tesla Nb₃Sn magnet coil constituents during reaction heat treatment", IEEE Trans. Appl. Supercond., submitted

Stress-strain behaviour dependence on the load direction at the example of stainless steel

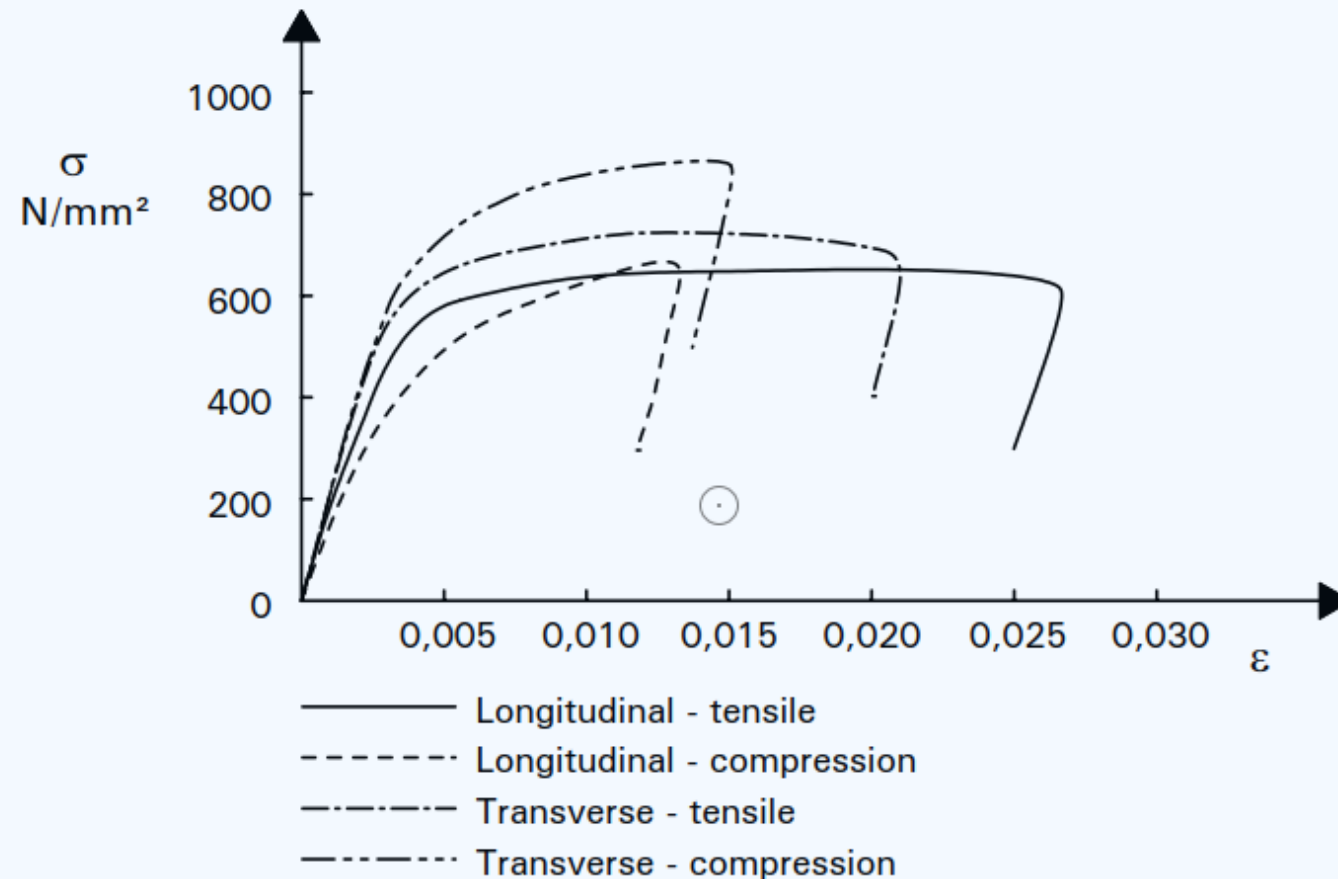
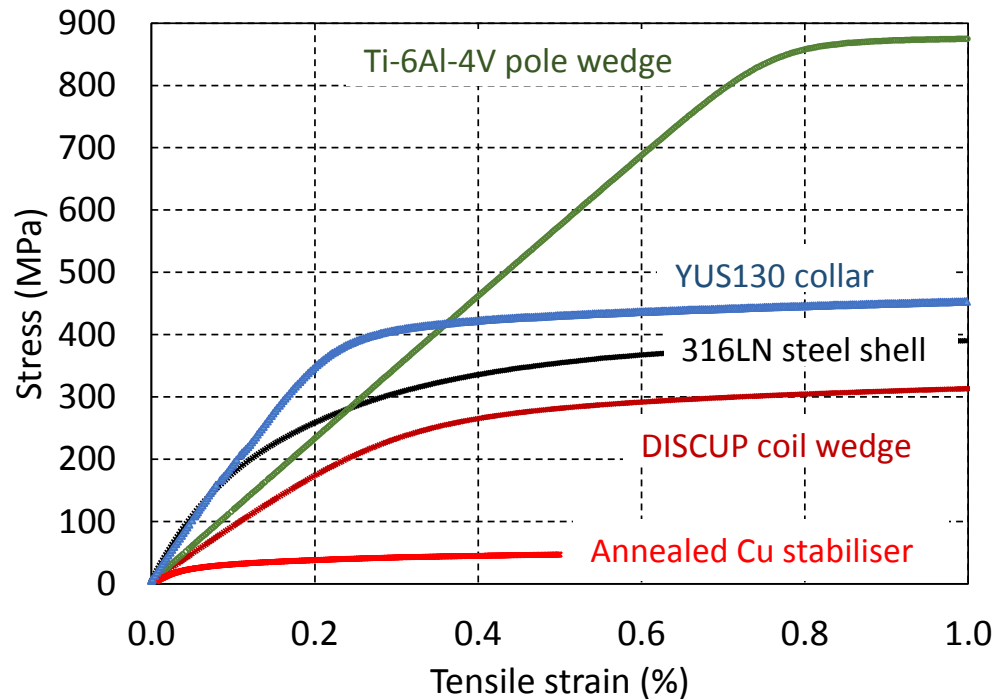


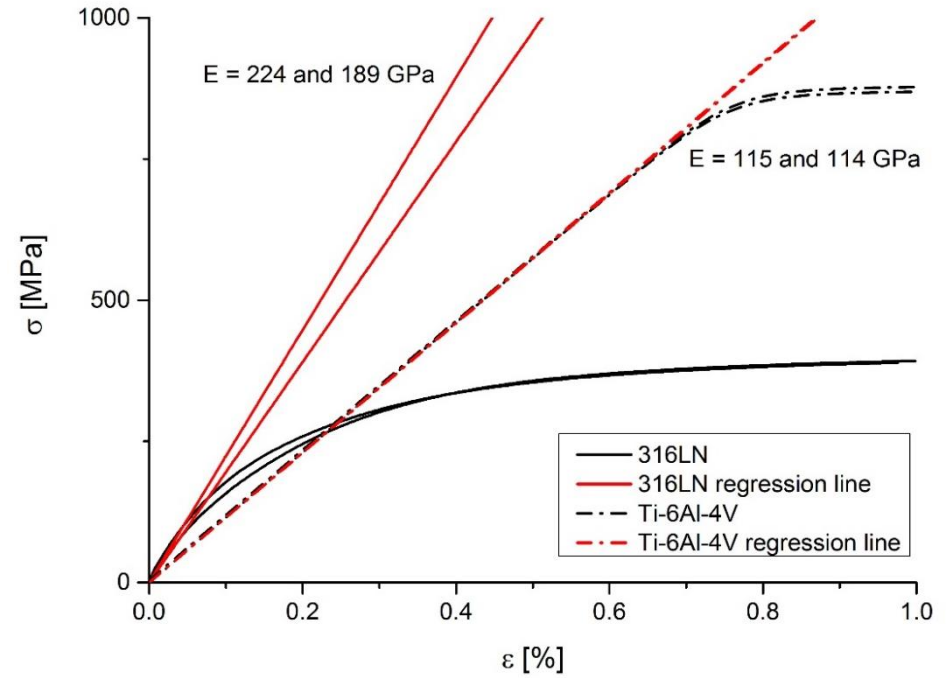
Figure 3.2 Typical stress-strain curves for grade 1.4318 cold worked to strength level C850

Uncertainties in static Young's modulus measurements

- Precise Young's moduli can be derived from stress-strain measurements of materials that exhibit pronounced linear elastic behaviour (e.g. Ti6Al4V).
- Comparatively large uncertainty in the Young's modulus results of materials that do not exhibit linear behaviour, like stainless steels.



Comparison of the RT engineering stress-strain curves of different coil and magnet materials [iv].



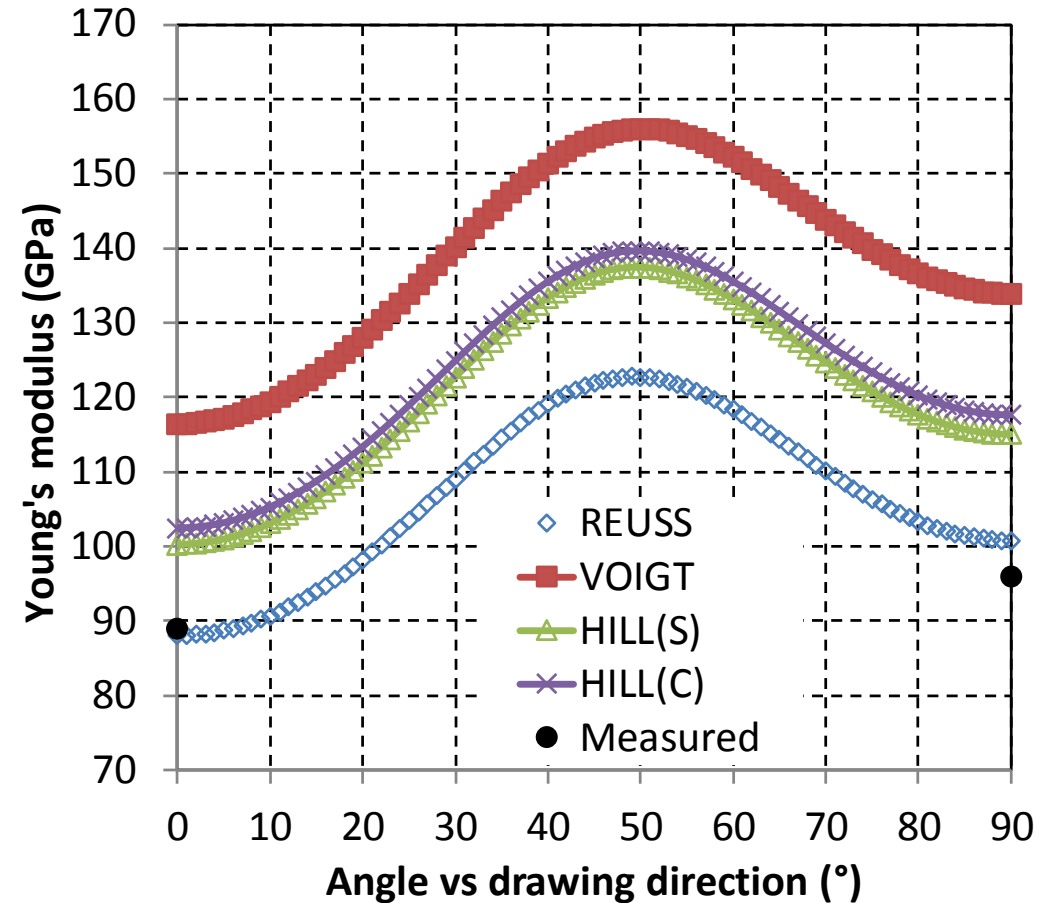
Comparison of 316LN and Ti6Al4V stress-strain curves and linear fits for the determination of Young's modulus [v].

[iv] C. Scheuerlein, F. Lackner, F. Savary, B. Rehmer, M. Finn, C. Meyer, "Thermomechanical behavior of the HL-LHC 11 Tesla Nb₃Sn magnet coil constituents during reaction heat treatment", IEEE Trans. Appl. Supercond., submitted

[v] C. Scheuerlein, F. Lackner, F. Savary, B. Rehmer, M. Finn, P. Uhlemann, "Mechanical properties of the HL-LHC 11 Tesla Nb₃Sn magnet constituent materials", IEEE Trans. Appl. Supercond., 27(4), (2017), 4003007

Elastic anisotropy in the 11 T dipole DISCUP coil wedges

- The DISCUP Young's moduli derived from stress-strain compression tests are 89 GPa in the wedge extrusion direction and 96 GPa perpendicular to the extrusion direction [v].
- These values are substantially lower than the values between 115 to 130 GPa found for ODS Copper in literature.
- The angular dependence of Young's modulus has been calculated from texture data obtained by neutron diffraction and from Cu single crystal elastic constants.
- The DISCUP wedges are strongly textured (multiples of random orientation MRD=16), which causes a strong elastic anisotropy of about 30%.



Angular DISCUP Young's modulus dependence with respect to the wedge extrusion direction. Calculated assuming equal strains (Voigt) and equal stresses (Reuss) in all grains, respectively. Measurement results from [v] are shown for comparison. Courtesy of W. Gan, Helmholtz-Zentrum Geesthacht.

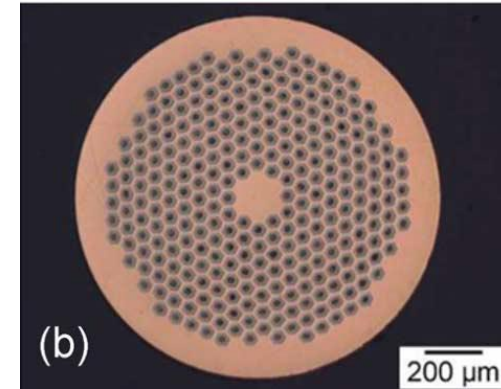
Summary: Uncertainties in Young's modulus measurement with ideal test samples, according to well established standards

- Dynamic Young's modulus measurement uncertainty <1 %.
- Static Young's modulus measurement by tensile tests uncertainty is typically 3% [ii], provided the material exhibits pronounced linear elastic behaviour.
- Static Young's modulus measurement by compression tests uncertainty is typically 5% [iii].
- For materials that do not exhibit pronounced linear stress strain behaviour (e.g. stainless steel and annealed copper), the uncertainty of the E-modulus derived from static tests can be >10% [v,vi].
- Important elastic anisotropy of textured coil materials (as an example DISCUP elastic anisotropy is >30%).

[vi] Ch. Weißmüller, H. Frenz, "Messunsicherheit bei der Ermittlung des E-Moduls im Zugversuch an Stahl", *Materials Testing*, vol. 55, no. 9, 2013, pp. 643-647

Uncertainties in stress-strain measurements of Nb₃Sn/Cu composite wires

- The most easy to study wire load case is uniaxial tensile loading in the wire drawing direction.
- Nb₃Sn/Cu composite wires are not suitable tensile test samples.
 - the wire constituents are not stress free
 - the filaments contain porosity
 - at zero axial load the wires are always bended
- No linear part in the loading stress-strain curve. Therefore, the wire stiffness is estimated from the linear slope of unloading curves.
- The Nb₃Sn/Cu wire stiffness is stress and strain dependent, even in the wire drawing direction.



Strain (%)	Stiffness (GPa)
0.02	139
0.04	120
0.06	117
0.08	118
0.10	118
0.12	116
0.14	117
0.16	117

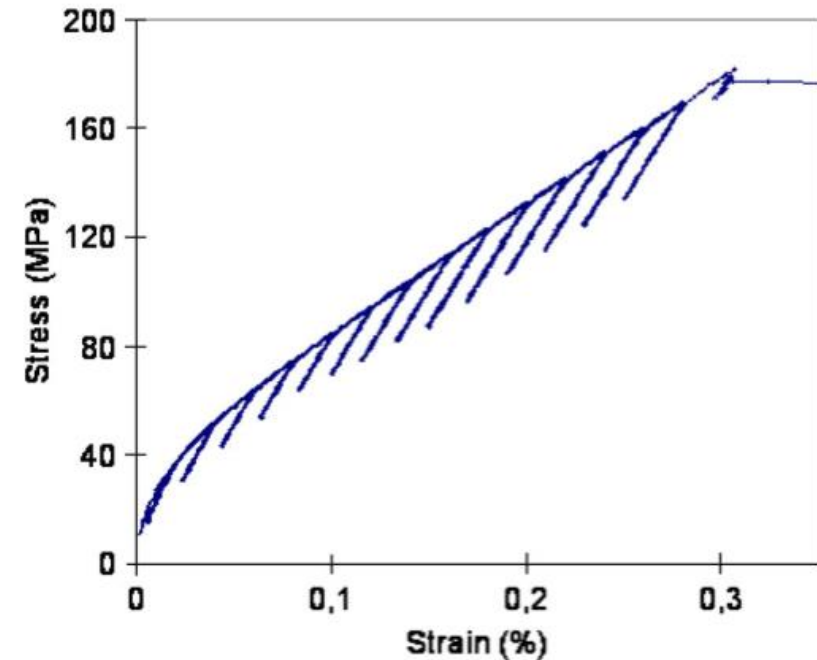
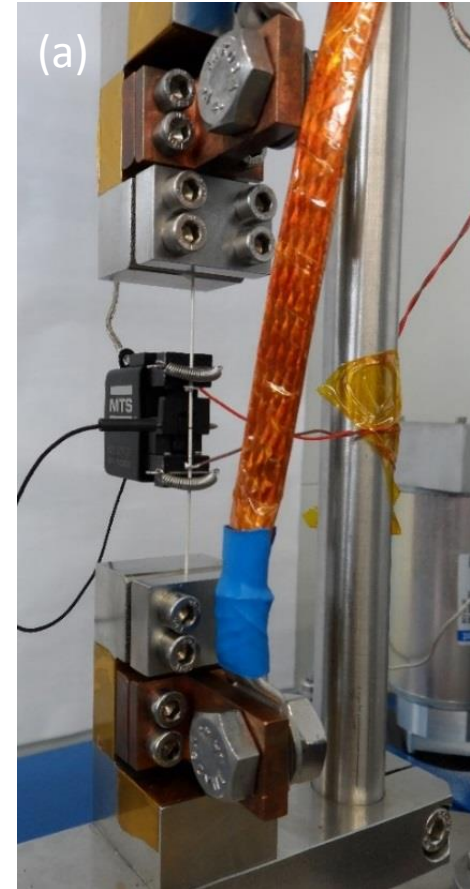
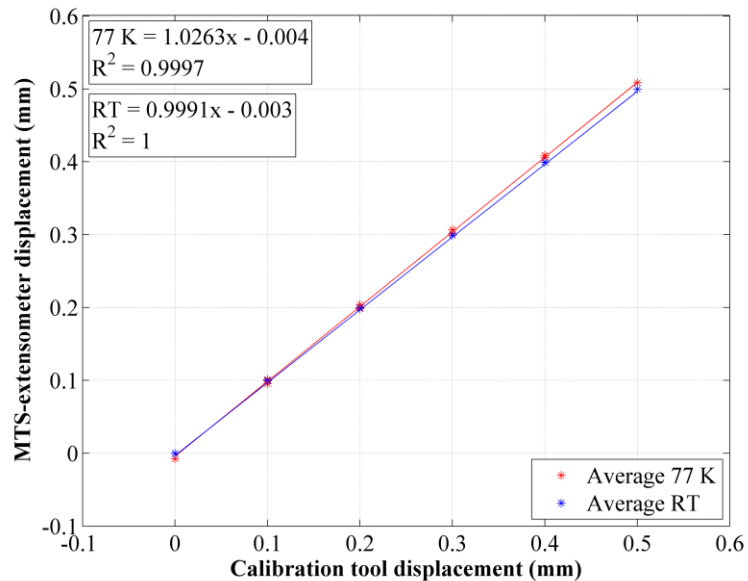


Fig. 2. Engineering stress-strain curve of the reacted PIT B215 composite strand measured at RT. The stress has been partly released every 0.02% in order to measure the apparent composite E-modulus as a function of axial composite strain. Courtesy B. Rehmer and M. Finn, Federal Laboratory for Materials Research, Berlin.

Extensometry for wire uniaxial stress-strain measurements

- The precision of stress-strain measurements relies mainly on careful extensometer strain measurements.
- Extensometers for strain measurements in wire drawing direction are commercially available.
- Extensometer calibration at RT can be performed by the supplier.
- Calibration at 77K and 4.2 K is not easily available and requires the setting up of dedicated calibration tools [vii].

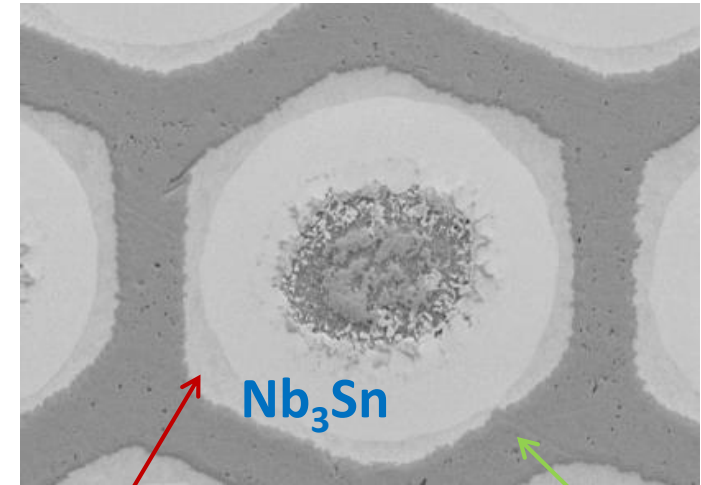
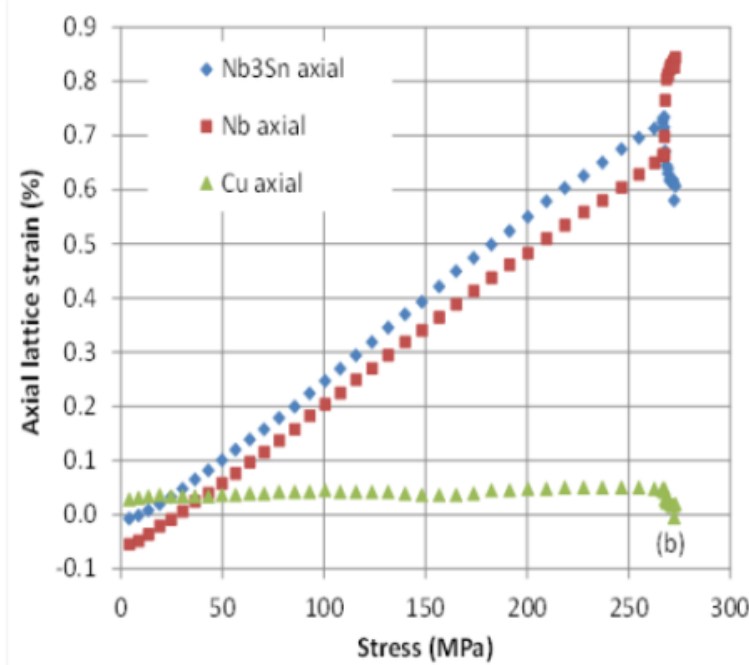
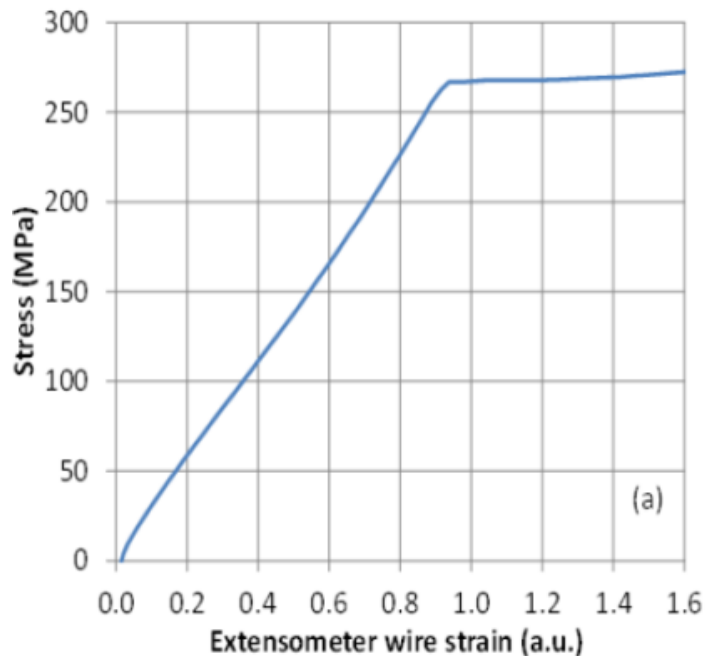


(a) MTS-clip on extensometer type 632.27F-21 with gauge length of 25 mm and (b) calibration tool.

Mechanical behaviour of Nb₃Sn/Cu PIT wire constituents studied by simultaneous stress-strain-XRD measurements at 4.2 K

The effect of different wire constituents on the wire mechanical behaviour can be understood from combined X-ray diffraction-stress-strain results :

- Almost no load is carried by the annealed Cu matrix
- Linear elastic behaviour of Nb and Nb₃Sn
- Nb and Nb₃Sn axial pre-compression in the unloaded wire
- Load transfer from Nb₃Sn to Nb at a wire stress of about 270 MPa and a strain of about 0.7%



Nb barrier

Cu matrix

[viii] C. Scheuerlein, M. Di Michiel, F. Buta, B. Seeber, C. Senatore, R. Flükiger, T. Siegrist, T. Besara, J. Kadar, B. Bordini, A. Ballarino, L. Bottura, "Stress distribution and lattice distortions in Nb₃Sn/Cu multifilament wires under uniaxial tensile loading at 4.2 K", Supercond. Sci. Technol. 27, (2014), 044021.

(a) 4.2 K stress-strain curve and (b) axial lattice strain vs uniaxial tensile stress [viii].

Summary: Sources of uncertainty in uniaxial tensile stress-strain measurements of Nb₃Sn/Cu composite wires

- Nb₃Sn/Cu composite wires are not suitable for precise tensile tests. Their constituents are not stress free, the filaments contain porosity and at zero axial load the wires are always bended.
- There is no linear part in the wire loading stress-strain curves.
- The wire stiffness is commonly estimated from the linear slope of unloading stress-strain curves. The slope of these curves is stress and strain dependent.
- Wire stiffness results depend strongly on test and data analysis procedures.
- Wire stress-strain results cannot be easily compared with results obtained with alternative test methods.
- Only few laboratories perform the challenging stress-strain measurements at 4.2 K.

Nb₃Sn mechanical properties at RT

- Published RT Nb₃Sn Young's modulus values vary between 32 GPa to 165 GPa.
- Measuring the mechanical properties of Nb₃Sn is very tricky because:
 - The intermetallic phase Nb₃Sn is stable with Sn contents ranging from about 18-25 at.%.
 - Preparation of Nb₃Sn bulk samples which properties are representative for the Nb₃Sn filaments in multifilament wires is extremely difficult (e.g. because of different Sn gradients, alloying elements, grain size and orientation distribution, and presence of contaminating phases).
 - Determining Nb₃Sn/Nb sub-element Young's moduli from tensile tests, for instance using extracted (brittle) Nb₃Sn filaments, is very delicate.
 - The filaments are a composite, containing Nb₃Sn, unreacted Nb diffusion barriers, porosity and other material. For such materials rule-of-mixture estimates may not be appropriate.

The unusual temperature dependence of the Nb₃Sn mechanical properties and calculation of Nb₃Sn elastic anisotropy in PIT and RRP type wires

- During cooldown Nb₃Sn undergoes a phase transformation at roughly 40 K, which causes a lattice softening.
- This is for instance revealed by the slope of the stress-strain curves acquired at RT and at 4.2 K [viii].
- The Nb₃Sn elastic properties in PIT and RRP wires can be calculated using the Nb₃Sn grain orientation distribution and published Nb₃Sn single crystal elastic constants.
- The results illustrate the different elastic anisotropy in RRP and PIT type wires.
- How reliable are the Nb₃Sn single crystal elastic constants?

TABLE II
Nb₃Sn ELASTIC MODULI IN AXIAL AND TRANSVERSE DIRECTIONS
CALCULATED FOR THE RRP AND PIT WIRES AT RT AND AT 4.2 K

		PIT B215	RRP #7419
RT	E _{axial}	130	140
	E _{trans}	135	129
4.2 K	E _{axial}	106	127
	E _{trans}	116	104

From: "Elastic Anisotropy in Multifilament Nb₃Sn Superconducting Wires", IEEE Trans Appl. Supercond., 25(3), (2015), 8400605

Summary: Uncertainties of Nb₃Sn mechanical properties at RT and 4.2 K

- Preparation of Nb₃Sn bulk samples which are representative for the Nb₃Sn in wires for mechanical testing is nearly impossible.
- The mechanical Nb₃Sn properties in the filaments of IT, PIT and Bronze route wires are different.
- Nb₃Sn Young's moduli calculations rely on the availability of accurate single crystal elastic constants.
- During cooling to below 77 K a Nb₃Sn lattice softening occurs. 77 K mechanical test results do not allow to predict 4.2 K Nb₃Sn mechanical properties.

Transverse compression test of reacted and impregnated Nb₃Sn Rutherford cable stacks

- Transverse compressive loading of a composite is a highly complex load case.
- Transverse compression tests of reacted and impregnated Nb₃Sn Rutherford cable stacks can provide qualitative estimates of the stress-strain behaviour of the Nb₃Sn conductor in magnet coils under monotonic and cyclic azimuthal loading.
- Reacted and impregnated Nb₃Sn Rutherford cable stacks are sometimes also used to determine a so-called “Elastic modulus” of the Nb₃Sn conductor in magnet coils.
- Since Rutherford cable stacks do not exhibit linear elastic behaviour at any stress level, the use of the term “Elastic modulus” in the context of stress-strain measurements of Nb₃Sn cable stacks is misleading.

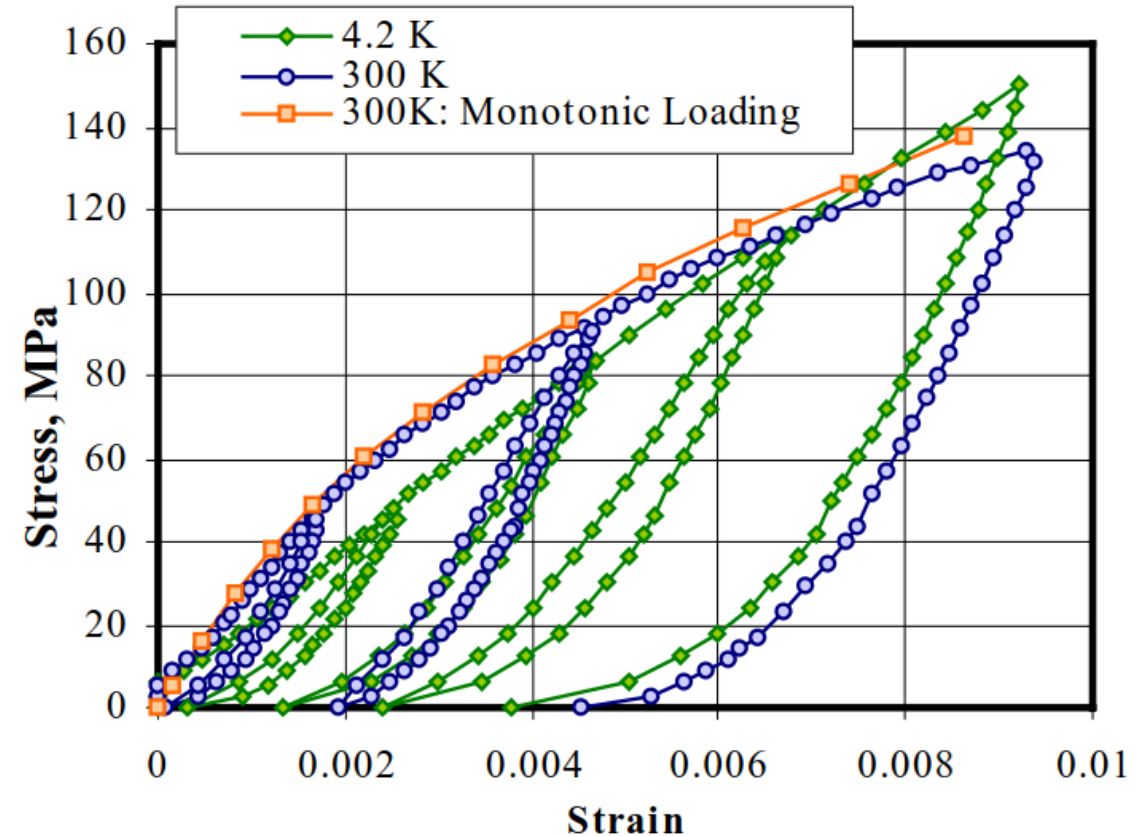


Figure 6: Load-Unload-Reload Tests.

From: D.R. Chichili, T.T. Arkan, I. Terechkine, J.A. Rice, “Niobium-Tin Magnet Technology Development at Fermilab”, Proceedings of the 1999 Particle Accelerator Conference, New York, 1999

Sources of uncertainties in transverse compression experiments using reacted and impregnated Nb₃Sn Rutherford cable stacks

- Geometry of Rutherford cable stacks is not suited for precise mechanical testing.
- Stiffness of Rutherford cable stacks is not uniform across the loaded area.
- Geometry of Rutherford cable stacks is only partly representative for the conductor in magnet coils.
- Strong influence of load rate and mechanical cycling on stress-strain behaviour.
- No standard procedures available. Very strong influence of data analysis procedures on stiffness results.
- Stiffness depends on the applied strain and stress.
- Very strong influence of the load direction on stress-strain behaviour.
- Integration of an extensometer in the test set-up is in many cases not possible.
- No easy cross check with other mechanical test methods is possible.

Conclusion

- Nb_3Sn coil and magnet materials properties are anisotropic and load direction dependent.
- Highly accurate Young's moduli can be obtained from dynamic measurements.
- In favourable cases (e.g. Ti6Al4V) Young's moduli can be derived from static stress-strain measurements, using standardised test sample geometries and test procedures.
- Transverse compression tests of reacted and impregnated Nb_3Sn Rutherford cable stacks can provide qualitative estimates that may be representative for the stress-strain behaviour of the Nb_3Sn conductor in magnet coils under monotonic and cyclic loading.
- Results of transverse compression tests of reacted and impregnated Nb_3Sn Rutherford cable stacks cannot be used to precisely predict stress levels in magnets.
- The use of the term Young's modulus or elastic modulus in the context of stress-strain measurements of Rutherford cable stacks is misleading.
- Stress-strain tests at 77 K do not allow to predict Nb_3Sn behaviour at 4.2 K.

Back-up slides

Calculation of Young's modulus from resonance frequencies

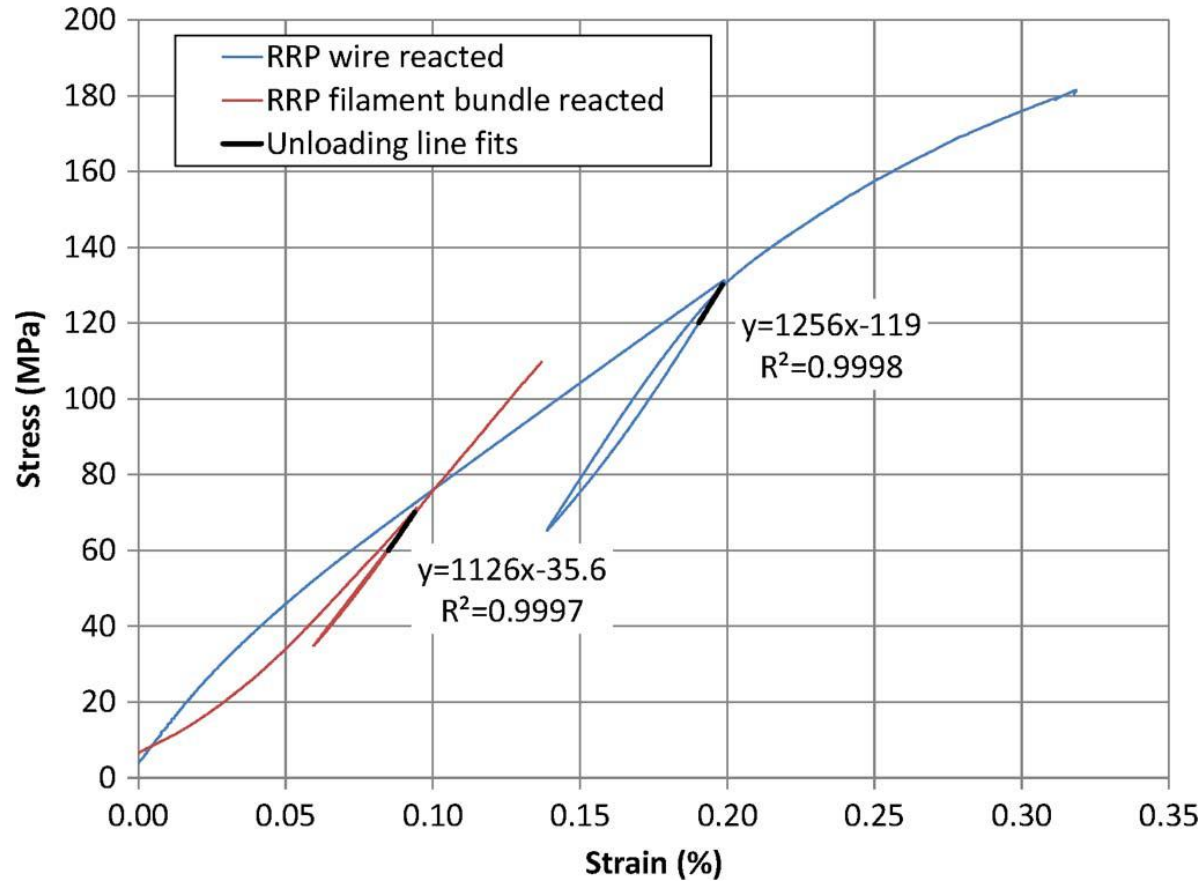
For long thin beams with rectangular cross section the Young's modulus (E) is calculated from the mass (m), length (L), width (b) and thickness (t) of the beam, and its fundamental frequency in bending (f_f).

$$E = 0.9465 \left(\frac{mf_f^2}{b} \right) \left(\frac{L^3}{t^3} \right) T$$

where

$$T = 1 + 6.858 \left(\frac{t}{L} \right)^2$$

Uniaxial tensile testing of extracted Nb₃Sn filaments



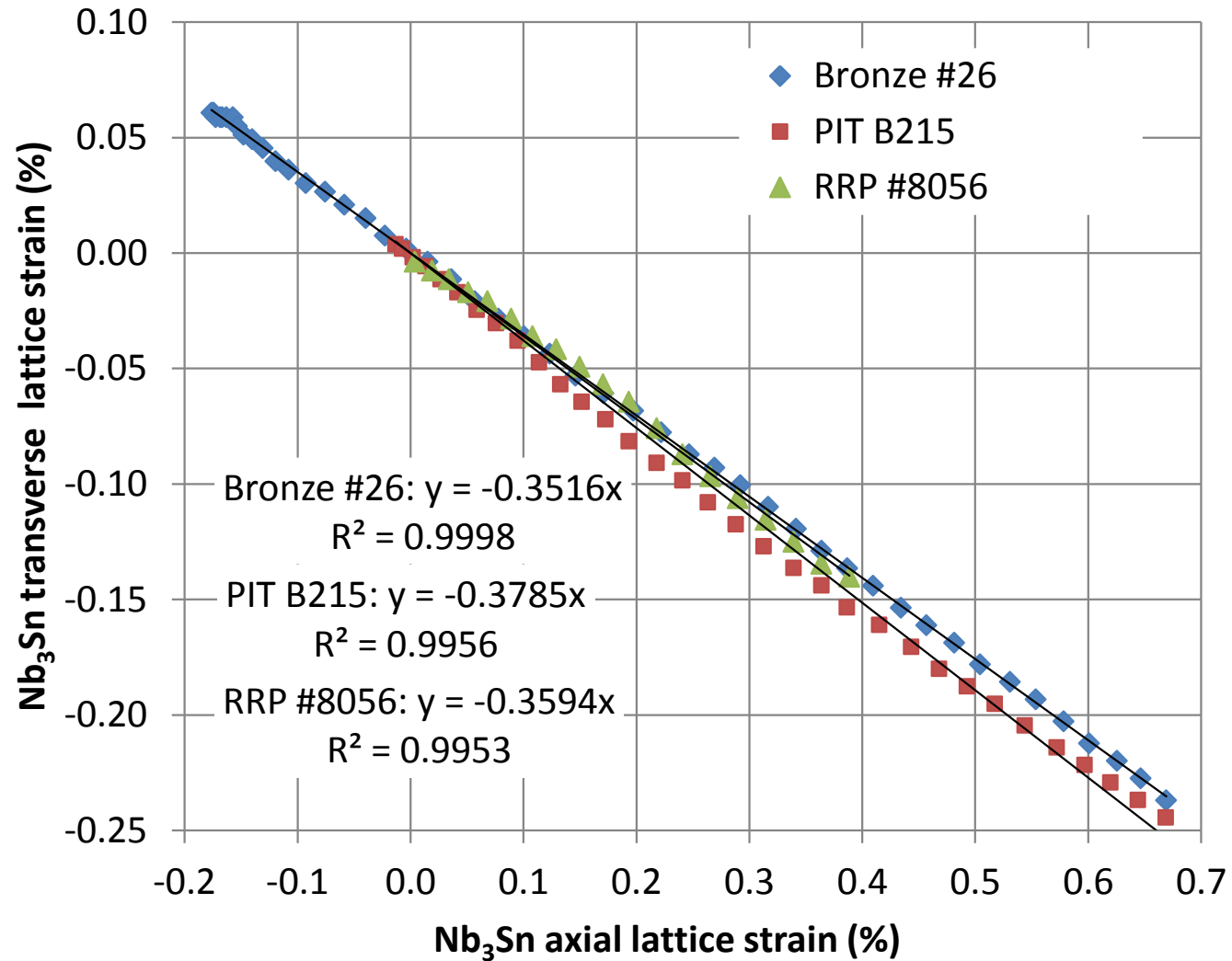
Stress-strain curves of a reacted RRP wire and its extracted filaments measured at room temperature. From : IEEE TAS, 25(3), (2015), 8400605

Summary of 11 T dipole materials elastic and plastic properties

Material	R _{p0.2} (MPa)	R _m (MPa)	A (%)	Z (%)	Young's modulus dynamic (GPa)		Young's modulus static (GPa)	
					Resonance	Impulse	Tensile	Compr.
316LN_L	351±12	674±2	54±5	63±2	191±0.3	191±0.3	183±7	n.m.
316LN_T	324±4	658±1	53±5	63±1	n.m.	n.m.	203±15	n.m.
Magnetil_L	117±3	241±1	n.m.	73±3	196±0.8	196±0.3	208±19	n.m.
Magnetil_T	124±2	267±1	47±2	71±4	219±0.1	218±0.3	213±3	n.m.
YUS-130_L	401	793	71	66	196±0.9	196±0.3	196	n.m.
YUS-130_T	415	749	51	66	193±1.1	192±0.3	189	n.m.
Ti-6Al-4V_L	868±5	930±12	17±1	28±2	115	114	115±1	120±2
Ti-6Al-4V_T	n.m.	n.m.	n.m.	n.m.	n.m.	n.m.	n.m.	116±2
DISCUP no-HT_L	332±2	387±1	22±2	43±4	92.9±0.2	93.8±0.3	87±1	89±1
DISCUP no-HT_T	n.m.	n.m.	n.m.	n.m.	n.m.	n.m.	n.m.	96±2
DISCUP HT_L	284±1	376±1	26±1	48±1	96.7	96.3	91±1	87±2

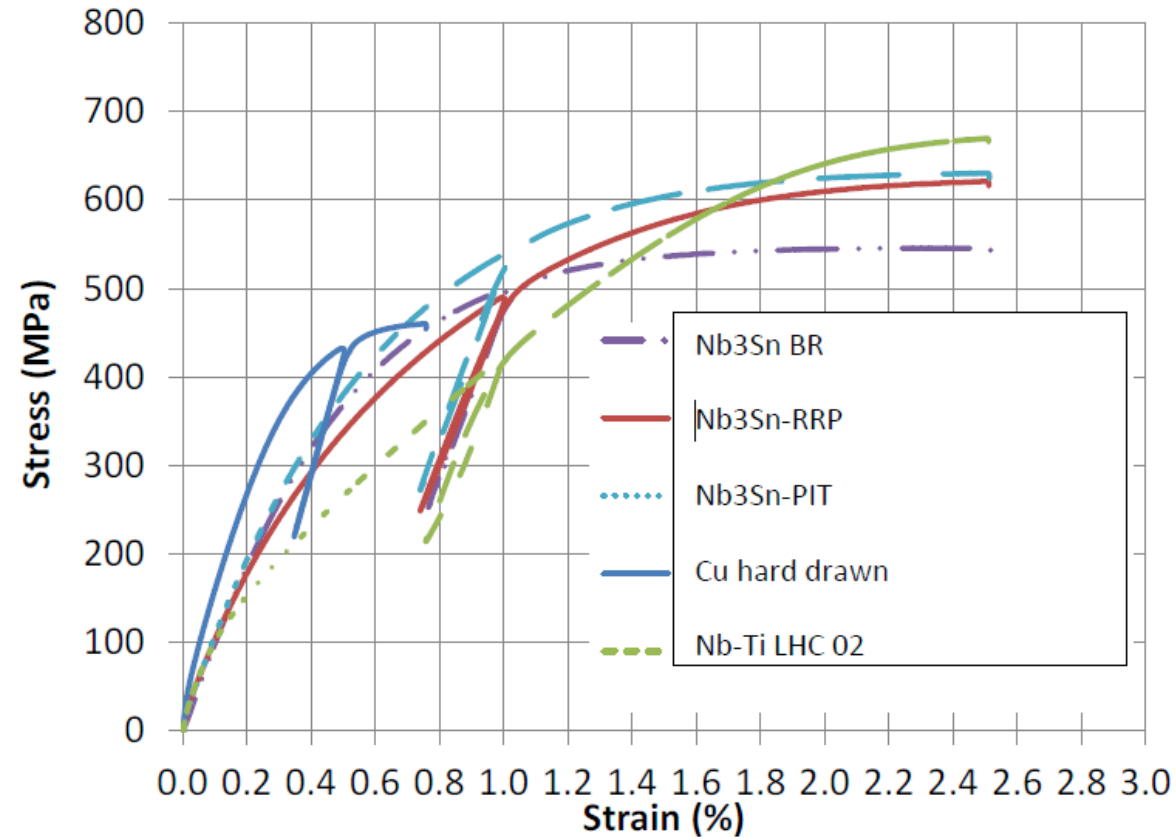
[v] C. Scheuerlein, F. Lackner, F. Savary, B. Rehmer, M. Finn, P. Uhlemann, "Mechanical properties of the HL-LHC 11 Tesla Nb₃Sn magnet constituent materials", IEEE Trans. Appl. Supercond., 27(4), (2017), 4003007

Nb₃Sn Poisson ratio at 4.2 K



Comparison of the 4.2 K Nb₃Sn Poisson ratio in the BR, PIT and RRP wire.
From: Supercond. Sci. Technol. **27** (2014) 044021 (7pp)

Mechanical properties of unreacted Nb₃Sn wires



Comparison of the stress-strain curves of unreacted Nb₃Sn (PIT, RRP and BR), hard drawn Cu and LHC Nb-Ti wire.