



Phase structure and superconducting properties of RHQT Nb₃Al wires fabricated by static and dynamic rapid heating

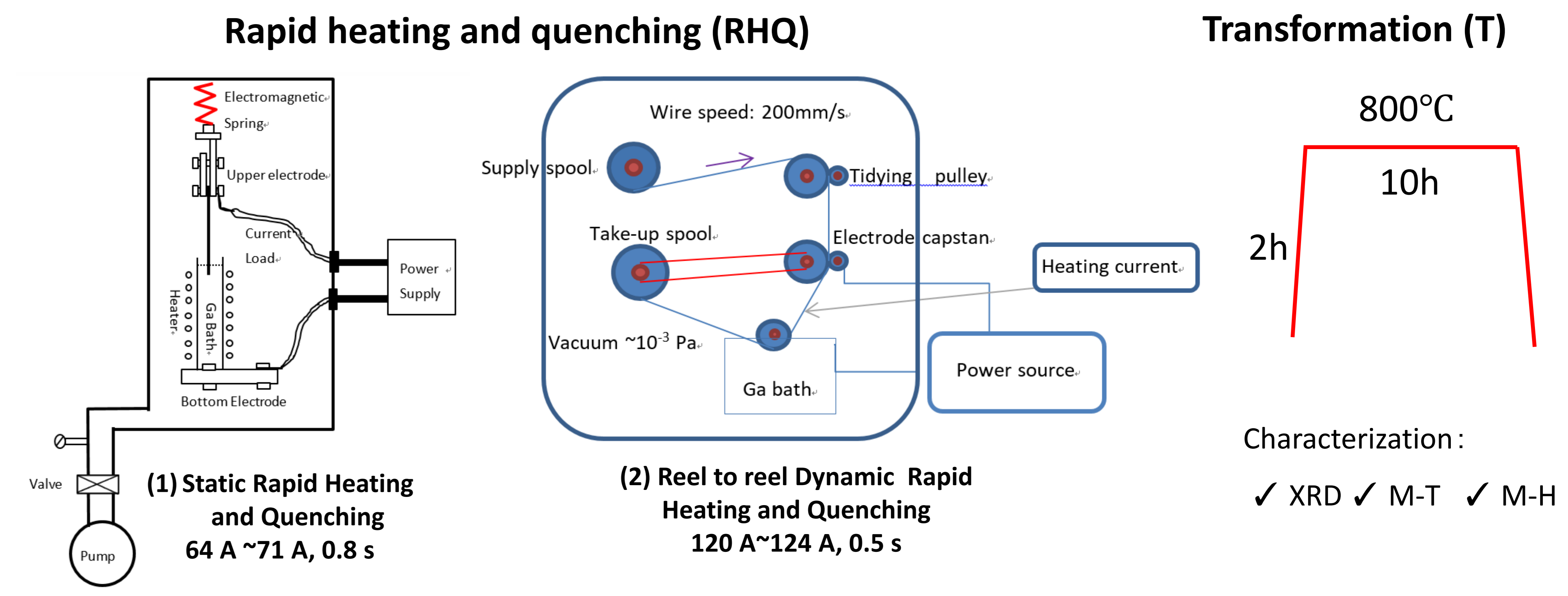
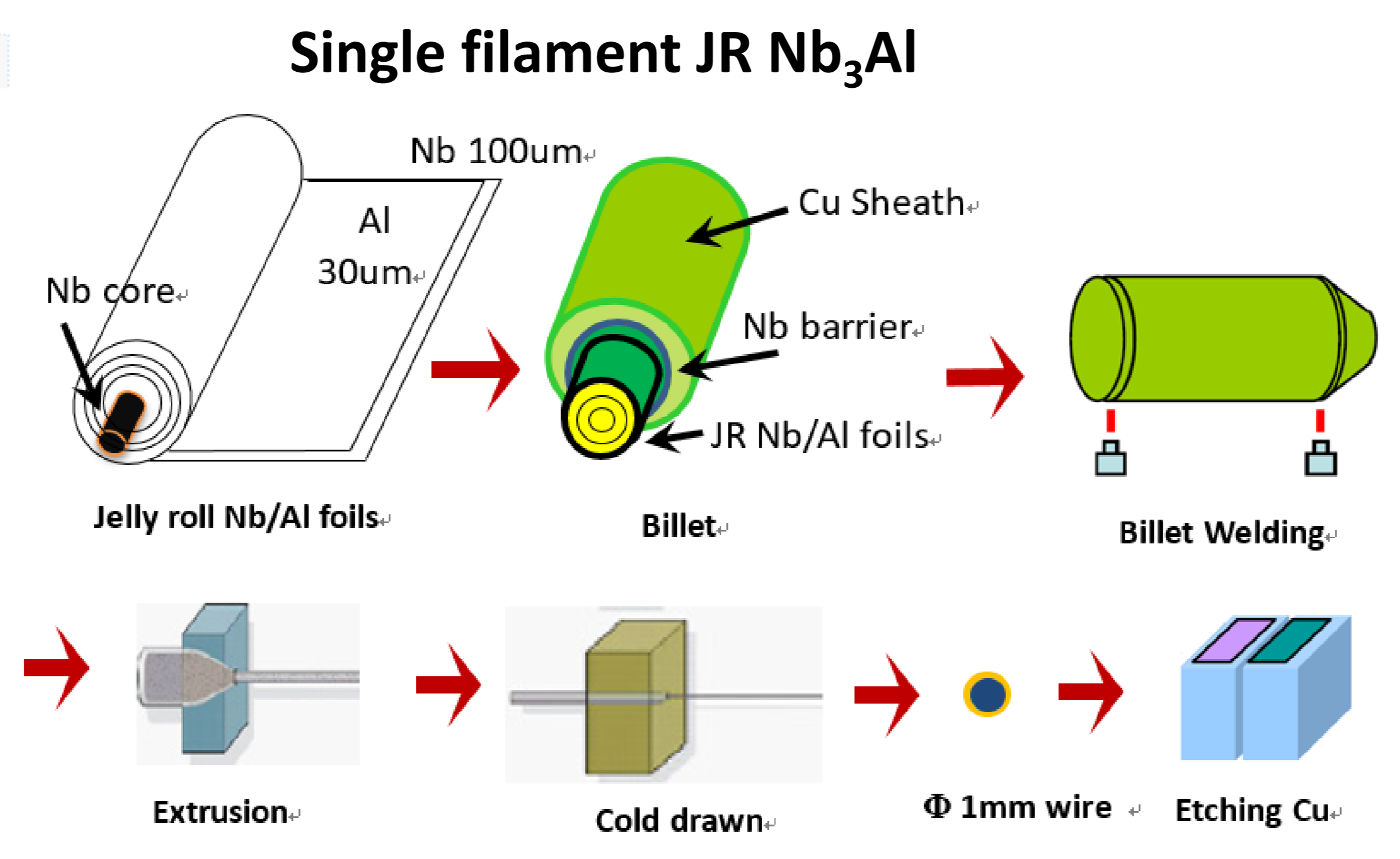
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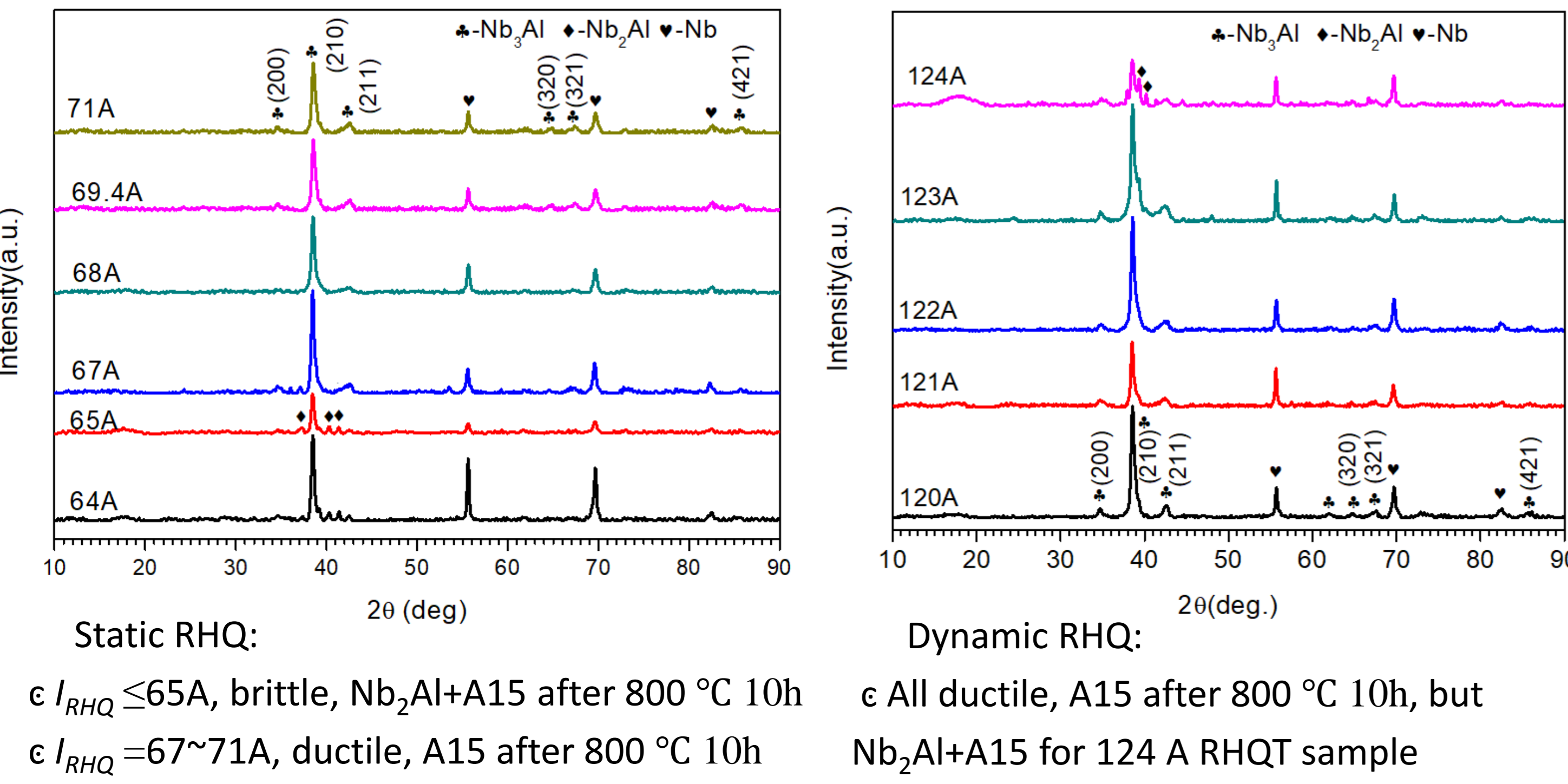
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Introduction

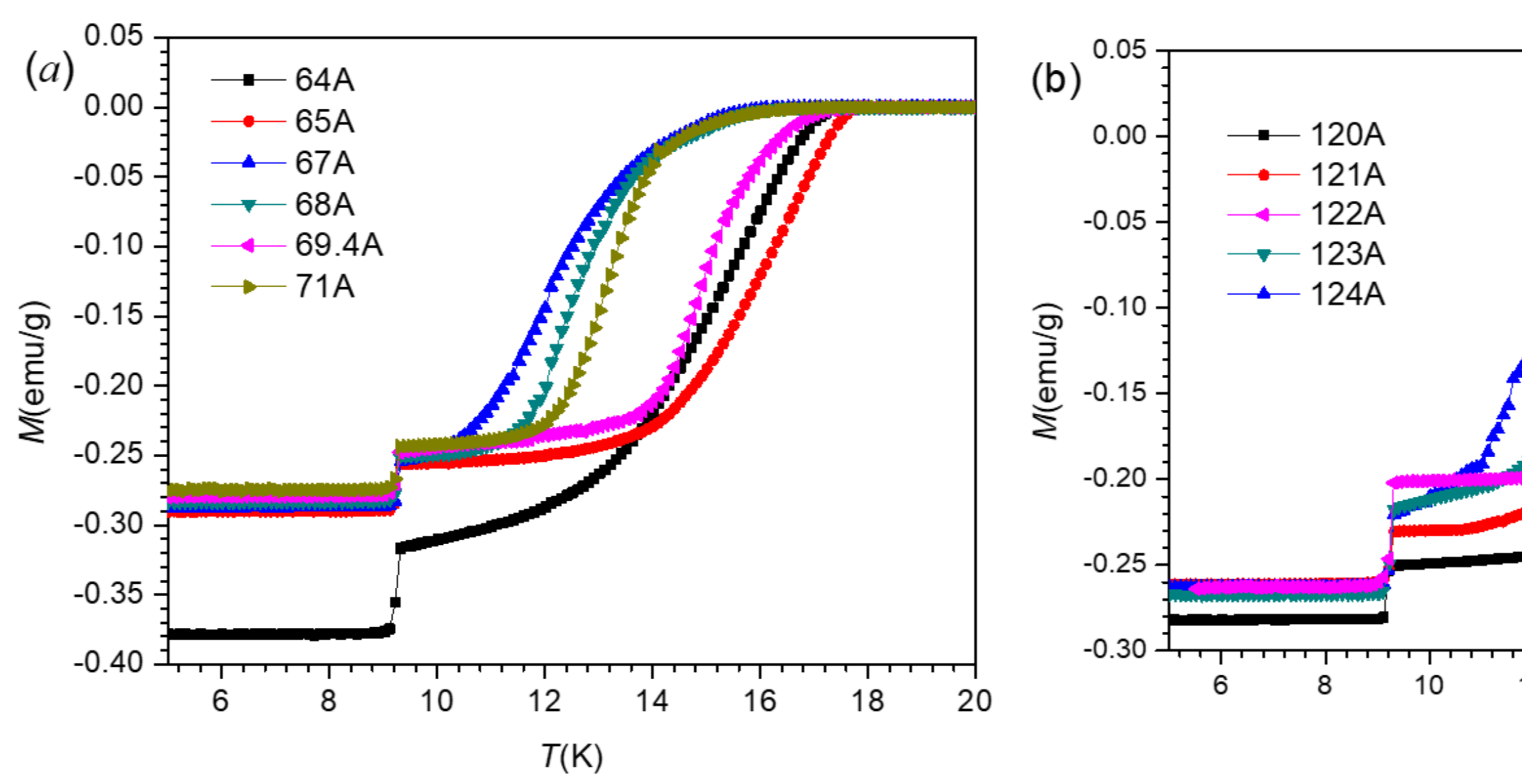
- Nb₃Al prohibits higher T_c (18.9 K), higher B_{c2} (29.5 T@4.2 K) and better stress-strain tolerance than Nb₃Sn.
- Bronze process cannot fabricate Nb₃Al phase for non-superconducting of Nb-Al-Cu compounds, stoichiometric Nb₃Al can only be formed beyond 1900 °C.
- Rapid heating, quenching and transformation (RHQT) process can fabricate stoichiometric Nb₃Al with fine grain size, exhibiting excellent J_c over whole magnetic fields.
- Right now, it is still lack of the comparison between phase structure and superconducting properties of RHQT Nb₃Al wires fabricated by static and dynamic rapid heating and quenching.



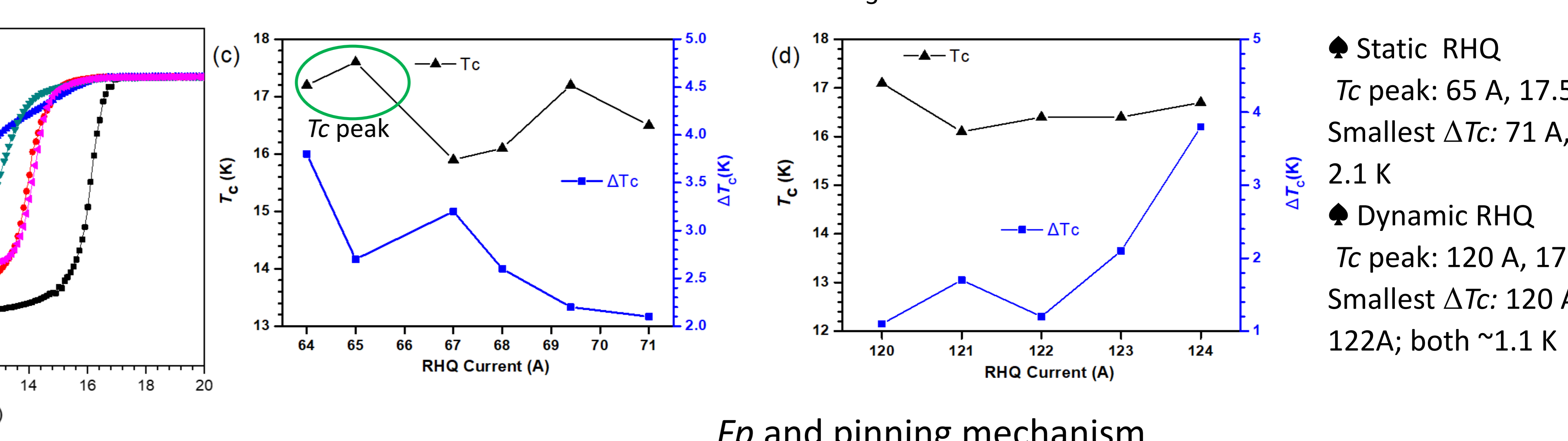
XRD of static and dynamic RHQT Nb₃Al wires



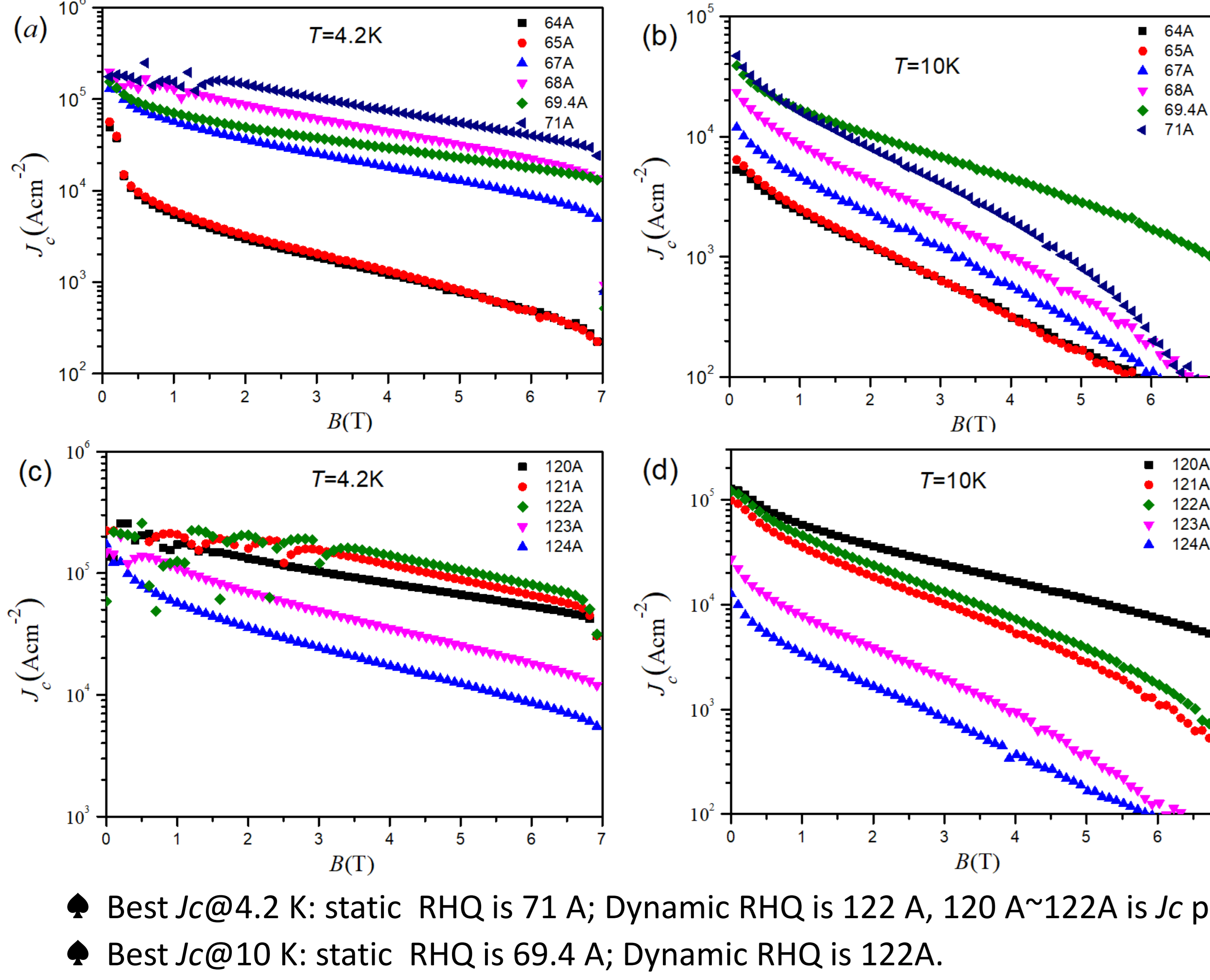
M-T curves of RHQT Nb₃Al wires



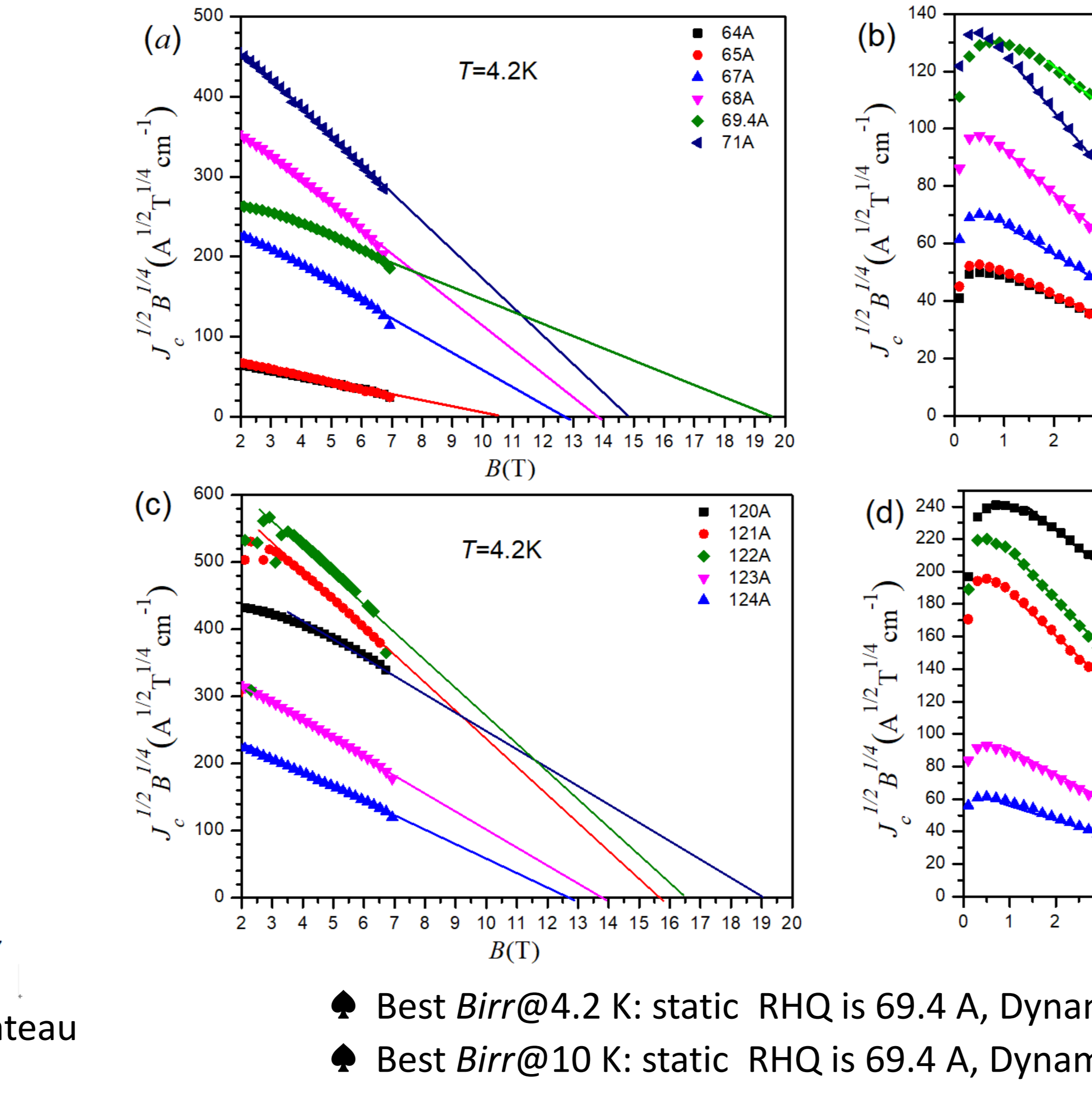
T_c -onset and ΔT_c of RHQT Nb₃Al wires



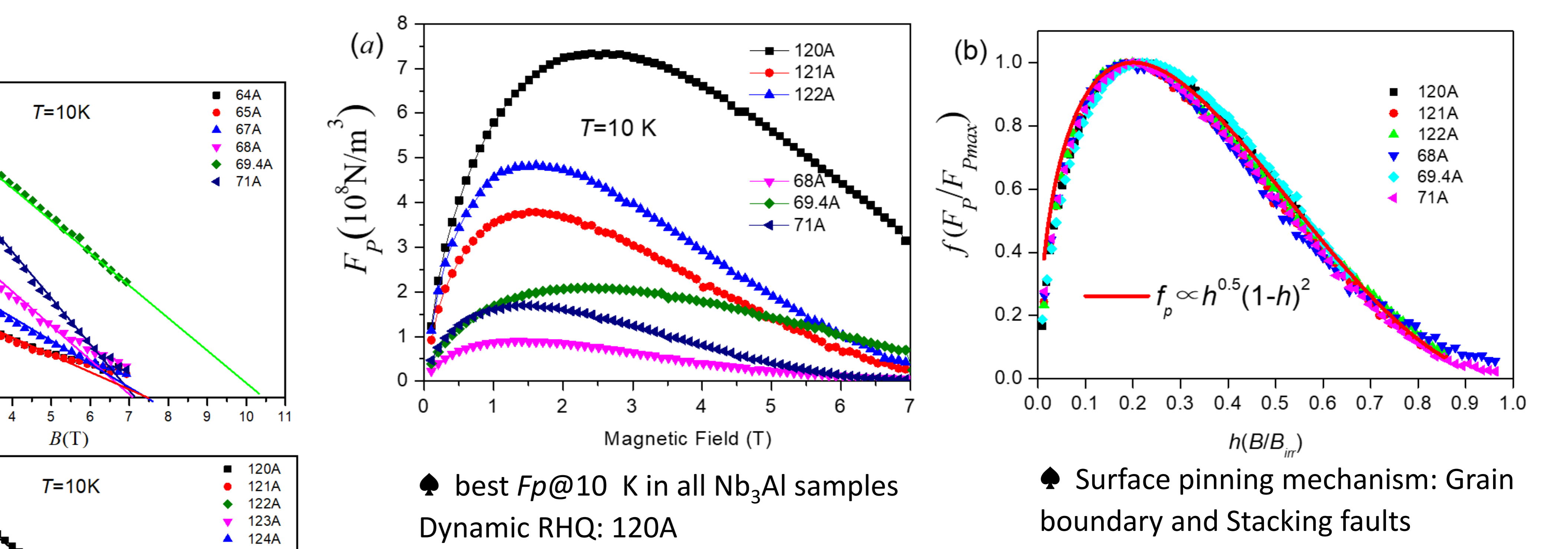
J_c calculated by Bean model from M-H



Kramer Plot at 4.2 K and 10 K



F_p and pinning mechanism



Conclusion

- Static RHQT Nb₃Al wires have wider ΔT_c (>2.1 K) than dynamic RHQT Nb₃Al (~ 1.1 K).
- At 4.2 K, 6.5 T, J_c range of dynamic RHQT Nb₃Al is $4.7 \sim 6.9 \times 10^4 A/cm^2$, much narrower than the static RHQT samples of $1.8 \sim 3.4 \times 10^4 A/cm^2$.
- Nb₂Al impurity phase formed when I_{RHQ} deviated from optimum condition: static RHQ of 64A \sim 65A (lower I_{RHQ}) and dynamic RHQ of 124A (higher I_{RHQ}).

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