



Nb₃Sn and Cu loading stress evolution in Rutherford cable stacks as a function of externally applied pressure

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Second workshop “Nb₃Sn Rutherford cable characterization for accelerator magnets”
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Outline

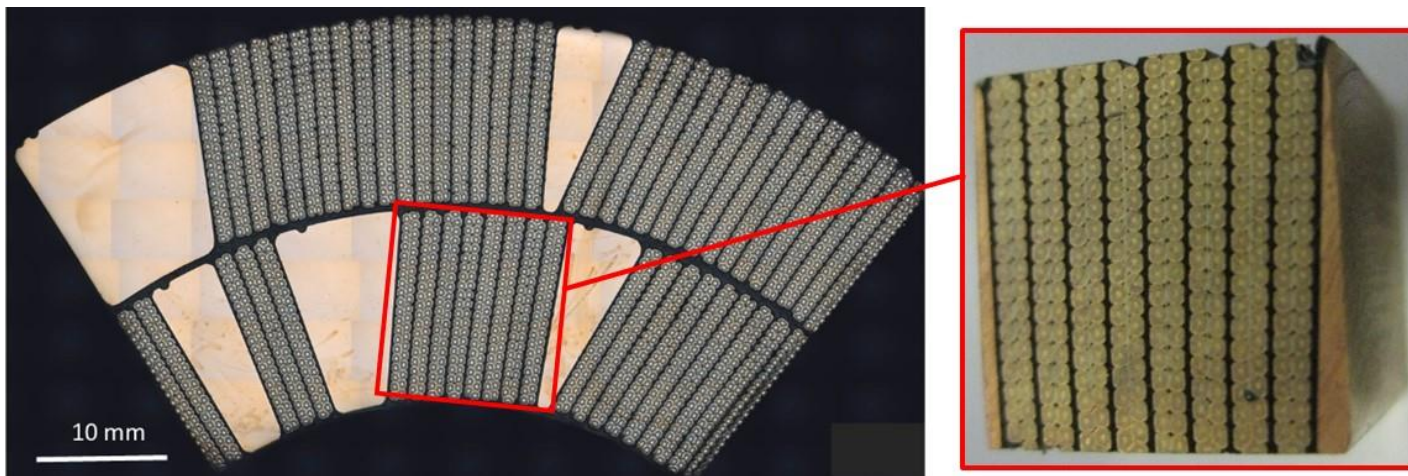
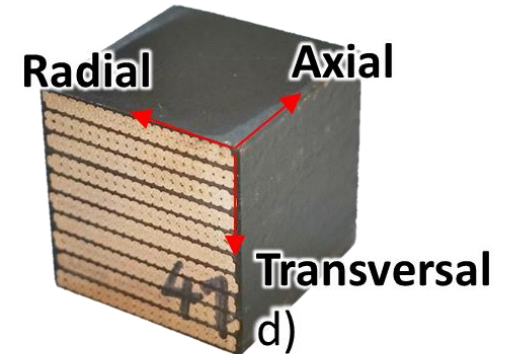
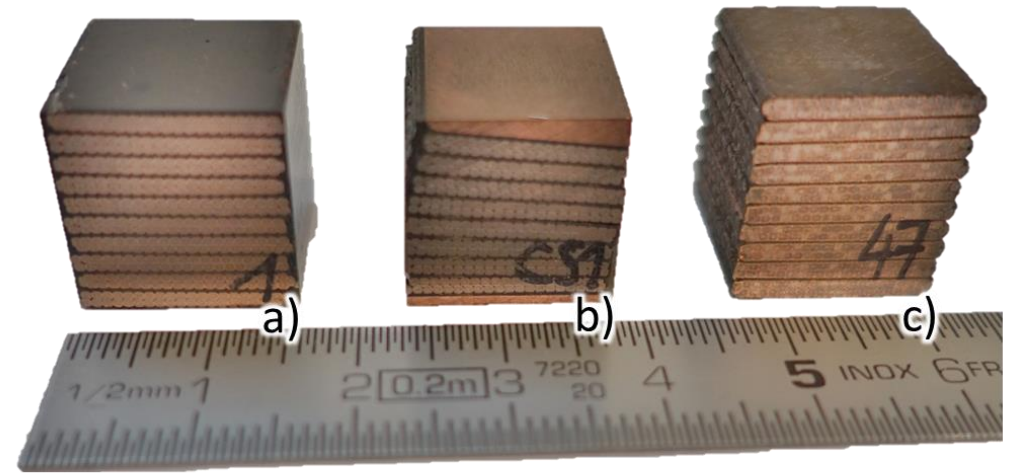
- Driving questions
- Loading strain and stress measurements in Nb₃Sn Rutherford cable stacks
 - Samples
 - Combined stress-strain and neutron diffraction measurements
- Nb₃Sn lattice strain and stress in 11 T dipole cable stacks
 - Transverse compressive loading
 - Loading in axial direction
- Discussion and conclusion

Driving questions

- What is the stress-strain behaviour of Nb₃Sn coils in accelerator magnets?
- What are the Nb₃Sn and Cu loading stresses at 150 MPa macroscopic transverse pressure (the assumed Nb₃Sn cable irreversible stress limit)?
- What would be the ideal mechanical properties of the impregnation material?

Samples for applied strain and stress measurements

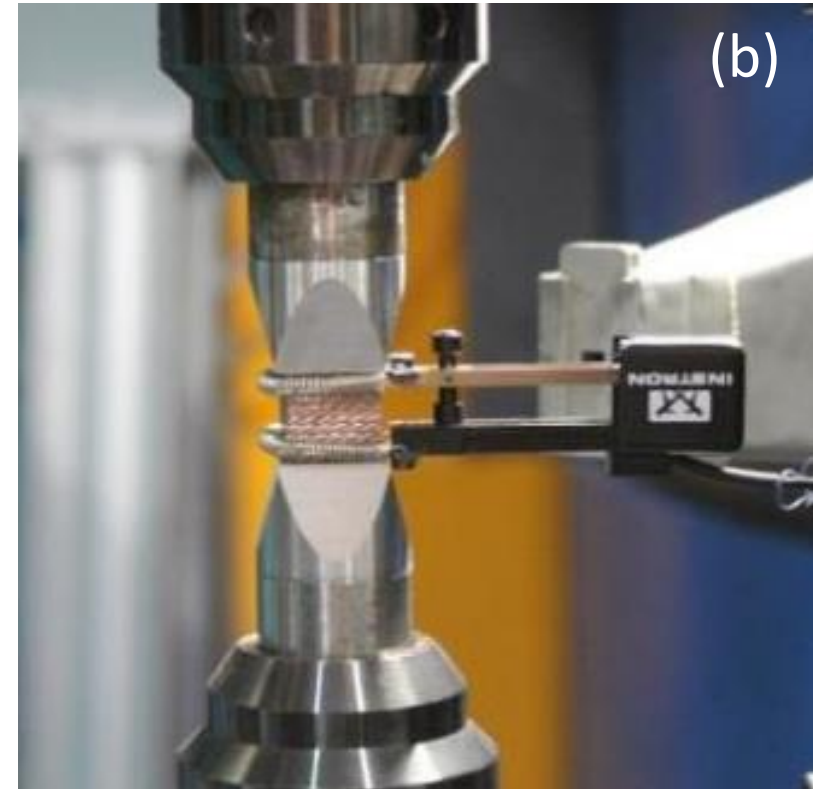
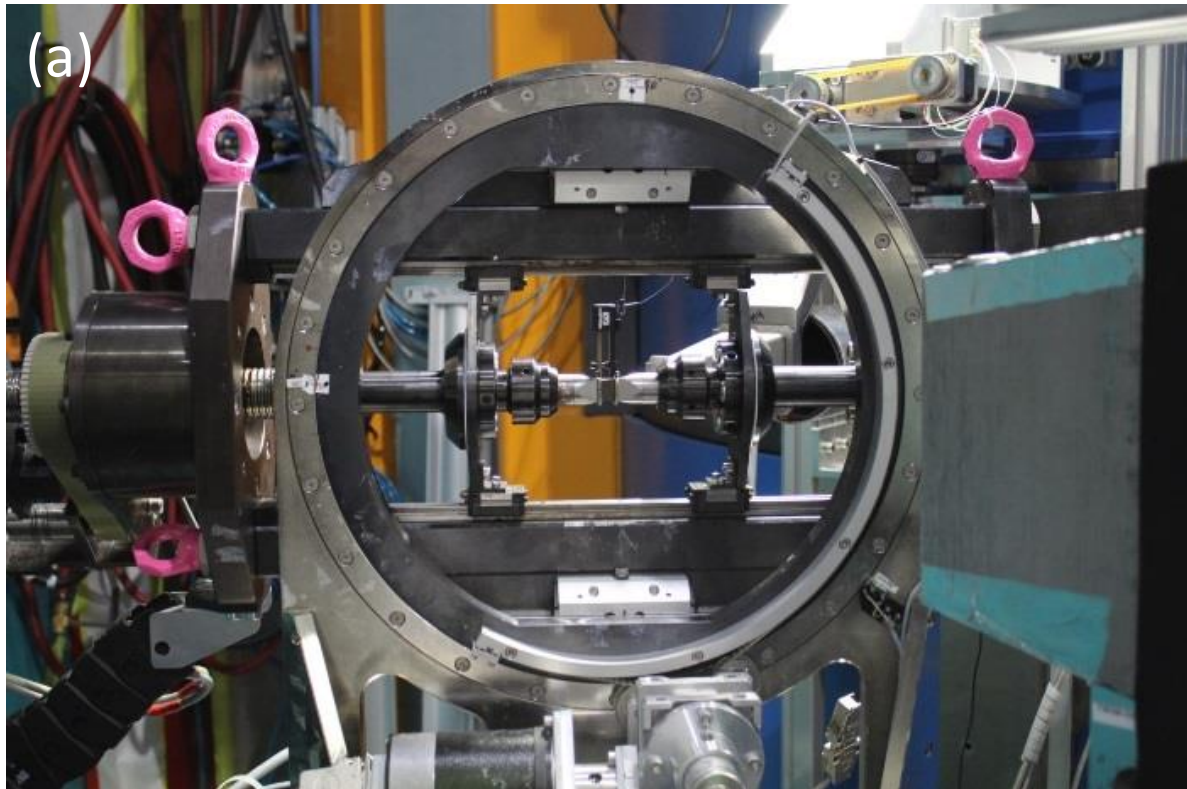
- All samples are made of Nb₃Sn 11 T dipole Rutherford cables with mica and S2 glass insulation.
- (a) Ten-stack samples impregnated with epoxy resin CTD-101K with different epoxy volume fractions.
- (b) Cubes machined out of 11 T dipole conductor block, containing 8 Rutherford cables and part of the adjacent coil wedges to compensate for the cable keystone angles.
- (c) Non impregnated ten-stack samples.



From Felix Wolf et al, *“Effect of epoxy volume fraction on the stiffness of Nb₃Sn Rutherford cable stacks”*

Set-up for combined stress-strain and diffraction measurements

- Lattice strain in Nb₃Sn and Cu is measured by neutron diffraction in the three principal stress directions upon compressive loading.
- Gauge volume 5×5×5 mm³ in the center of the samples.
- Load frame combined with an Eulerian cradle enables rotation of the sample load axis with respect to the scattering geometry is integrated in the Stress-Spec neutron diffraction beamline of MLZ.
- Macroscopic sample strain is measured simultaneously with an extensometer (Instron 2620-602 with a gauge length of 12 mm).



(a) Load frame mounted in Eulerian cradle in the Stress-Spec beamline. (b) Ten-stack sample and extensometer mounted for combined stress-strain and diffraction measurements.

Calculation of loading strain and stress

- The Nb₃Sn and Cu loading strains caused by the externally applied stress are determined from the Nb₃Sn (321) and Cu (220) scattering angles.
- Loading strains in transverse, axial, and radial directions are calculated with respect to the assumed stress free scattering angle.

$$\varepsilon_{hkl} = \frac{d_{hkl} - d_{0,hkl}}{d_{0,hkl}} = \frac{\sin(\theta_{0,hkl})}{\sin(\theta_{hkl})} - 1$$

- Loading stresses are calculated assuming that the transverse, axial and radial directions are the principal stress directions in the sample, and that there are no shear stresses in the sample.
- Elastic constants and Poisson ratios have been calculated from single crystal elastic constants ($E_{Nb3Sn(321)}=131$ GPa, $\nu_{Nb3Sn(321)}=0.363$, $E_{Cu(220)}=138.9$ GPa, $\nu_{Cu(220)}=0.333$) [1,2,3].
- Nb loading stresses could not be calculated. Because of the strong Nb texture there are no peaks that could be detected in all directions.

$$\sigma_{ii} = \frac{E_{hkl}}{1 + \nu_{hkl}} \left(\varepsilon_{ii} + \frac{\nu_{hkl}}{1 - 2\nu_{hkl}} (\varepsilon_{11} + \varepsilon_{22} + \varepsilon_{33}) \right)$$

[1] K.R. Keller, J.J. Hanak, "Lattice softening in single crystal Nb₃Sn", Phys. Lett., vol. 21(3), (1966), 263-264

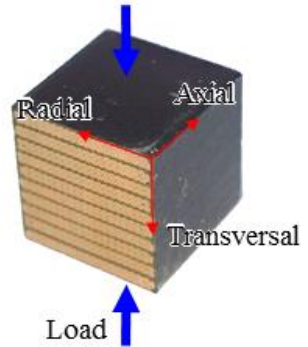
[2] H. Wern, R. Johannes, H. Walz, "Dependence of the X-Ray Elastic Constants on the Diffraction Plane", phys. stat. sol. (b) 206(2), (1998), 545-557

[3] E. Kröner, "Berechnung der elastischen Konstanten des Vielkristalls aus den Konstanten des Einkristalls", Zeitschrift für Physik, 151(4), (1958), 504–18

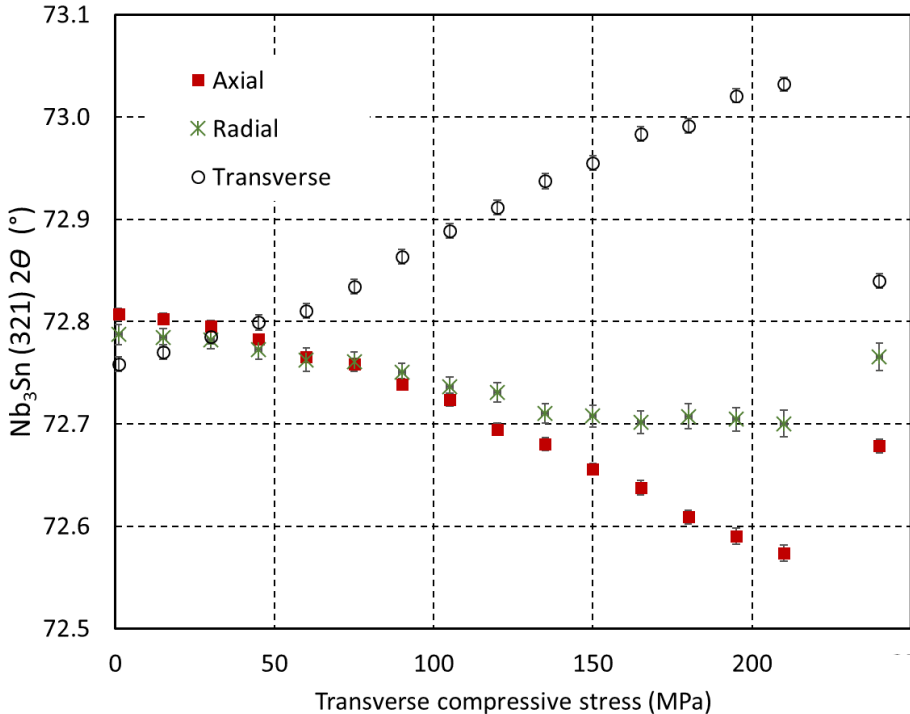
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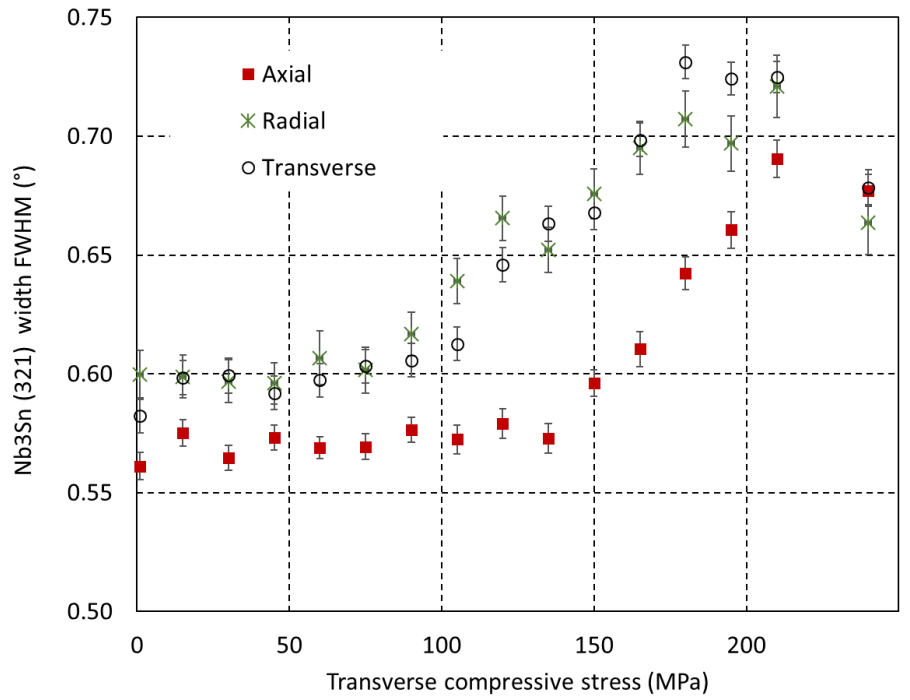
Transverse compressive loading of non-impregnated cable stack: Nb_3Sn diffraction angle and peak width changes



- Average diffraction angles within the $5 \times 5 \times 5 \text{ mm}^3$ gauge volume are recorded.
- Diffraction peak broadening indicates that the Nb_3Sn loading strain distribution becomes more inhomogeneous upon application of external stress.
- In axial direction the Nb_3Sn (321) peak is narrower than it is in radial and transverse directions, and the axial Nb_3Sn (321) peak width increase starts at higher stress.



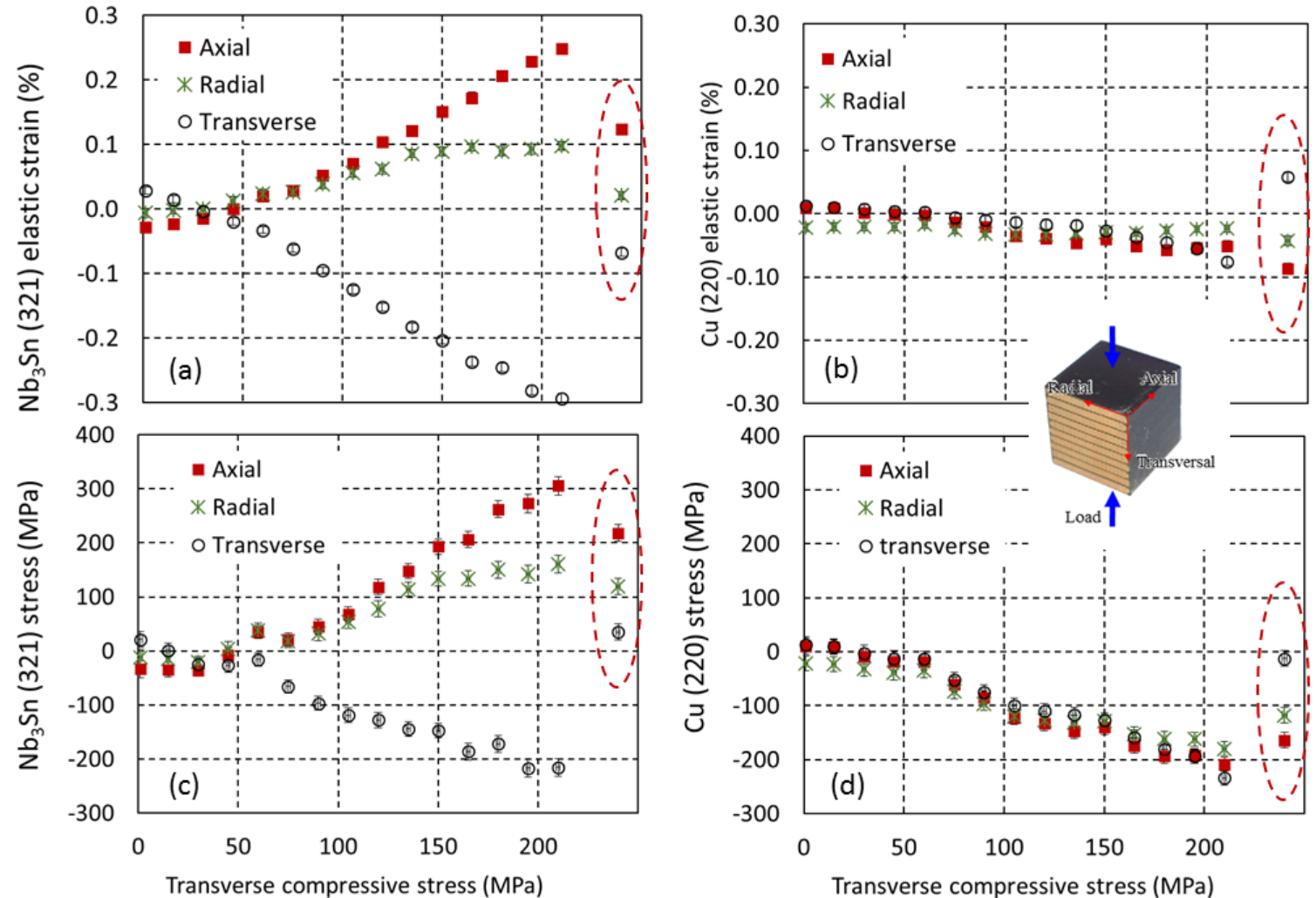
Nb₃Sn (321) diffraction angles as a function of externally applied transverse compressive stress.



Nb₃Sn (321) diffraction peak width evolution as a function of externally applied transverse compressive stress.

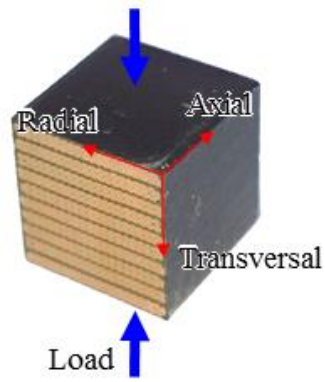
Nb₃Sn and Cu loading strains and stresses

- Loading strains and stresses are calculated from the relative changes of the diffraction angles with respect to the diffraction angle measured in the assumed stress free state.
- Cu is under similar compressive stress in all three directions.
- Transverse compression induces an important axial tensile stress in the Nb₃Sn.
- The transverse Nb₃Sn loading stress is an average value, the maximum Nb₃Sn stress is likely higher, as indicated by the Nb₃Sn diffraction peak broadening.

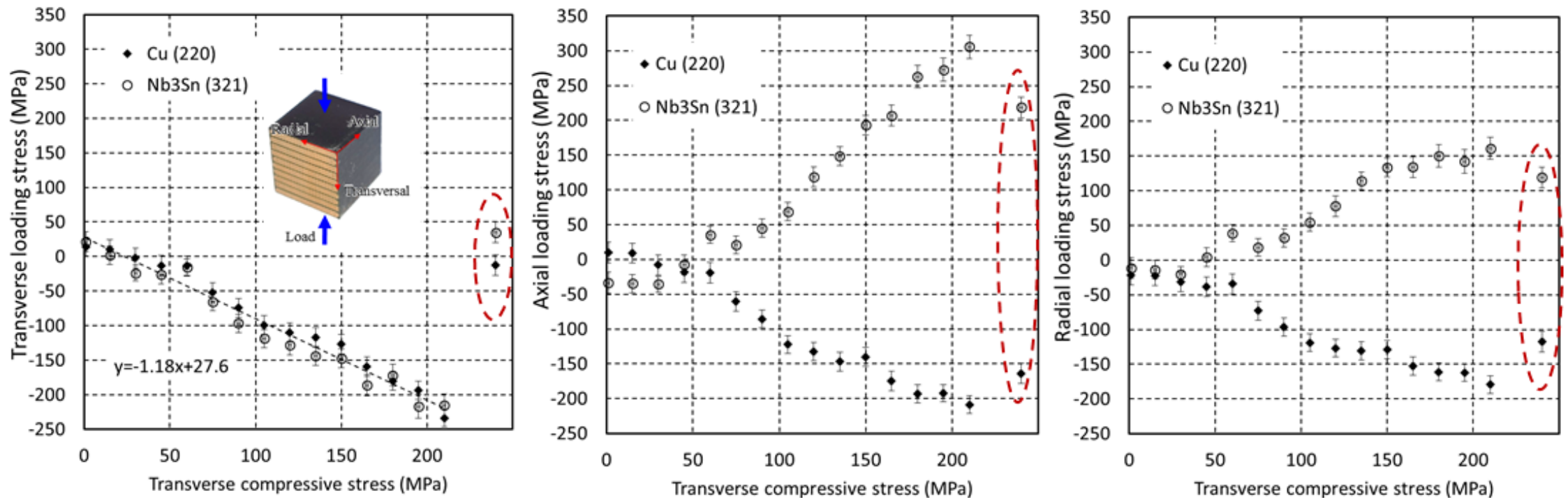


(a) Nb₃Sn (321) and (b) Cu (220) lattice strains and (c) Nb₃Sn (321) and (d) Cu (220) loading stresses as a function of externally applied transverse compressive stress.

Nb₃Sn and Cu loading stresses comparison



- The transverse Nb₃Sn and Cu loading stress increases linearly with the externally applied transverse stress (iso-stress conditions in the composite in transverse load direction).
- The transverse loading stress increase is about 20% higher than the externally applied stress increase, which can at least partly be explained by the presence of porosity in the sample.
- In the sample that is not constrained in axial and radial directions, the isotropic Cu pressure induces an important axial and radial tensile stress in Nb₃Sn.
- In magnet coils that are constraint in all directions, the axial and radial Nb₃Sn stress may be much smaller.

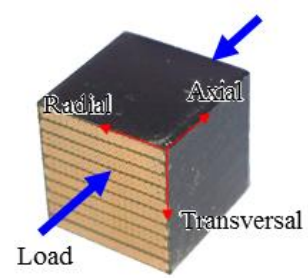


Comparison of Nb₃Sn (321) and Cu (220) loading stress evolution in (a) transverse, (b) axial and (c) radial direction.

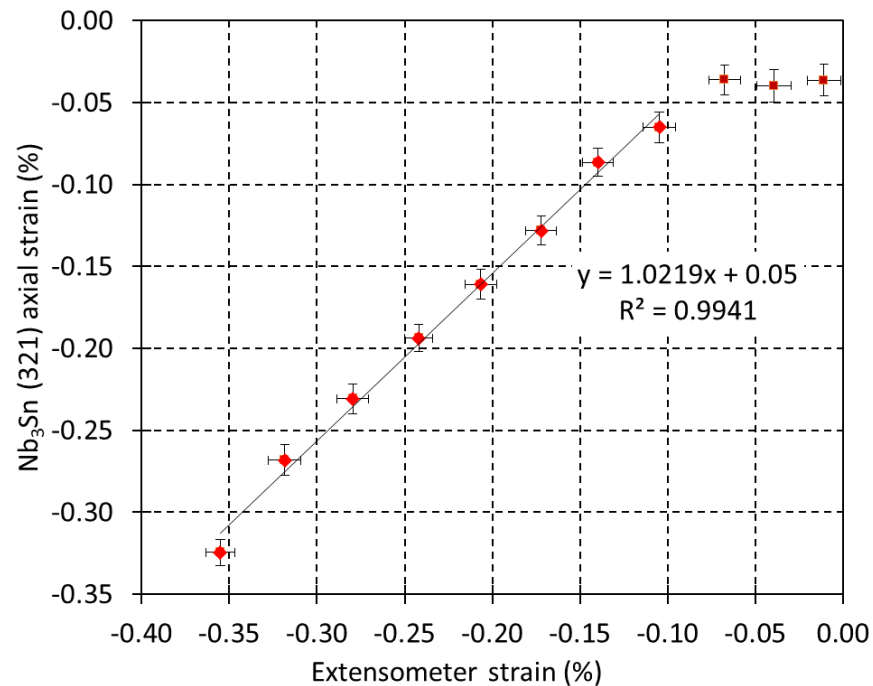
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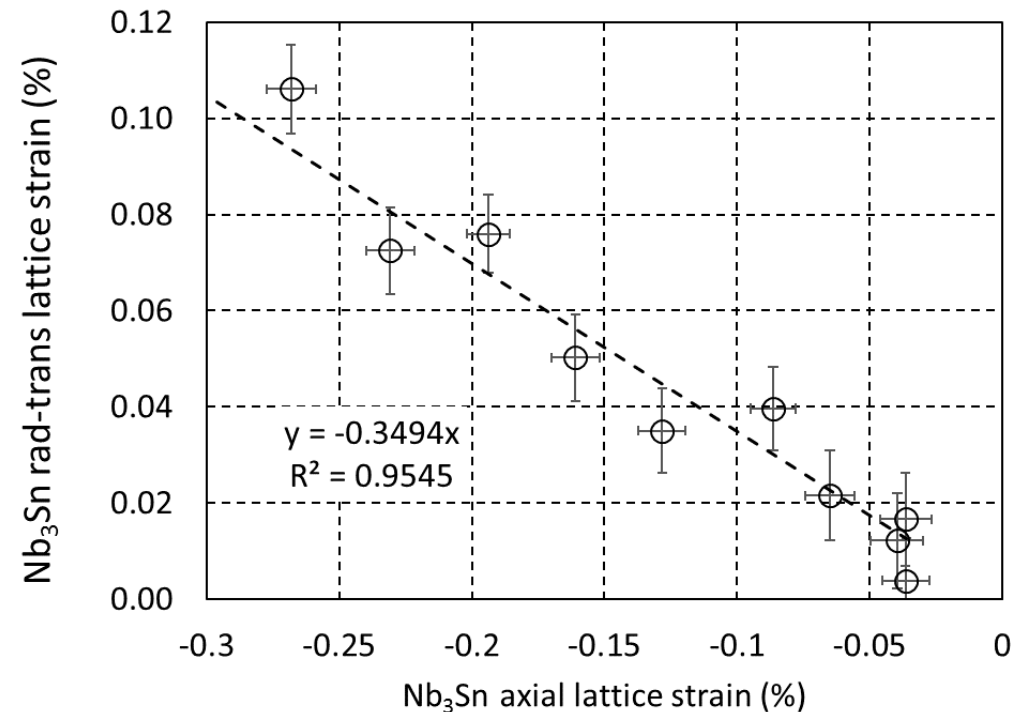
Monotonic axial loading: Nb₃Sn loading strain vs macroscopic sample strain



- When the macroscopic strain exceeds 0.1%, the elastic Nb₃Sn strain increases with a slope close to unity.
- Transverse and radial Nb₃Sn lattice spacing changes are mainly caused by the Poisson effect (assuming a Poisson's ratio of about 0.36).



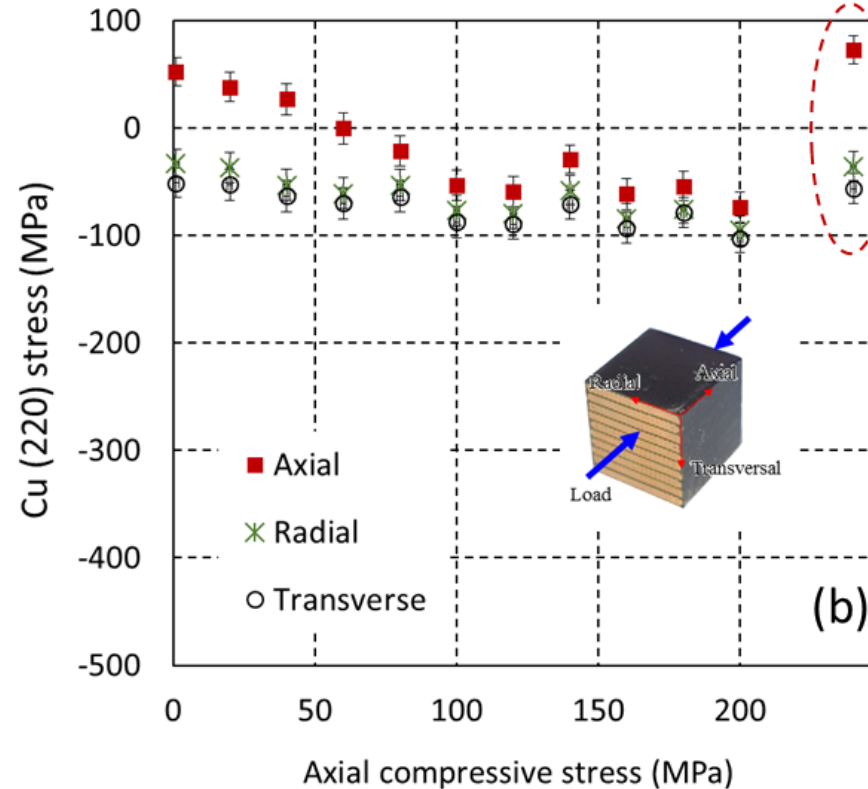
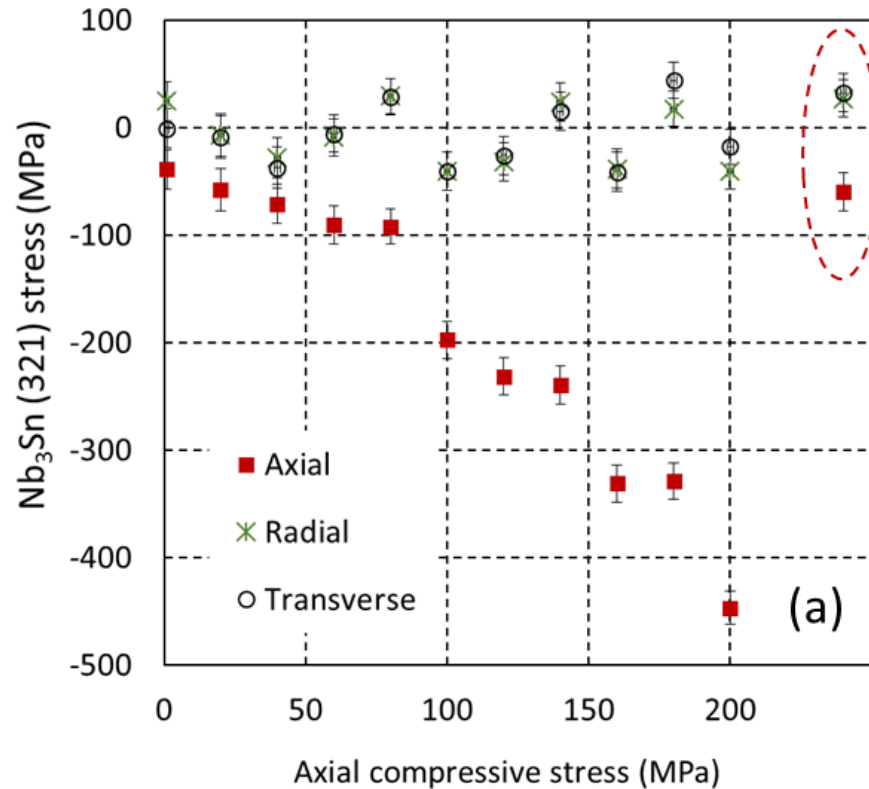
Elastic axial strain derived from neutron diffraction data as a function of the macroscopic sample strain measured with an extensometer.



Average transverse and radial Nb₃Sn (321) strain as a function of axial strain.

Monotonic axial loading: Nb₃Sn and Cu loading stresses

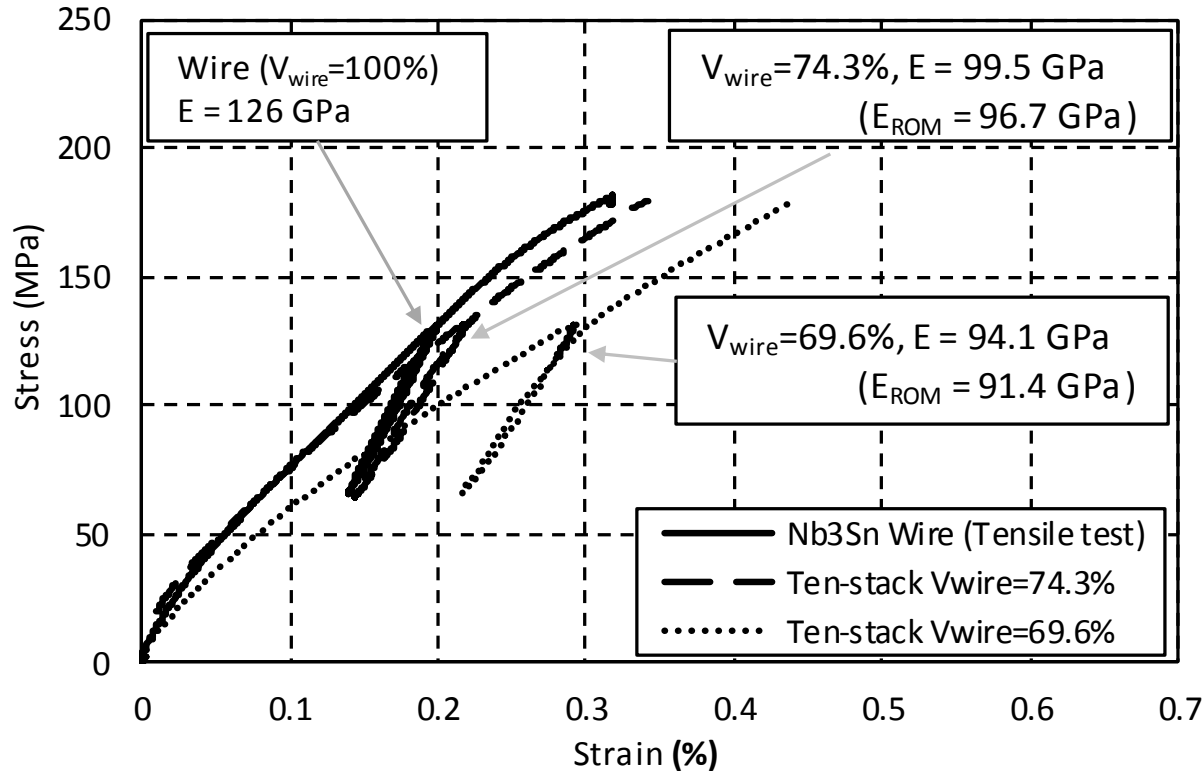
- When the macroscopic axial pressure on the cable stack exceeds about 100 MPa, the Cu elastic strain remains constant and the load is mainly carried by Nb₃Sn (and probably also by Nb).
- At 150 MPa externally applied pressure the axial Nb₃Sn (321) and Cu (220) loading stresses are about 300 MPa and 100 MPa, respectively.



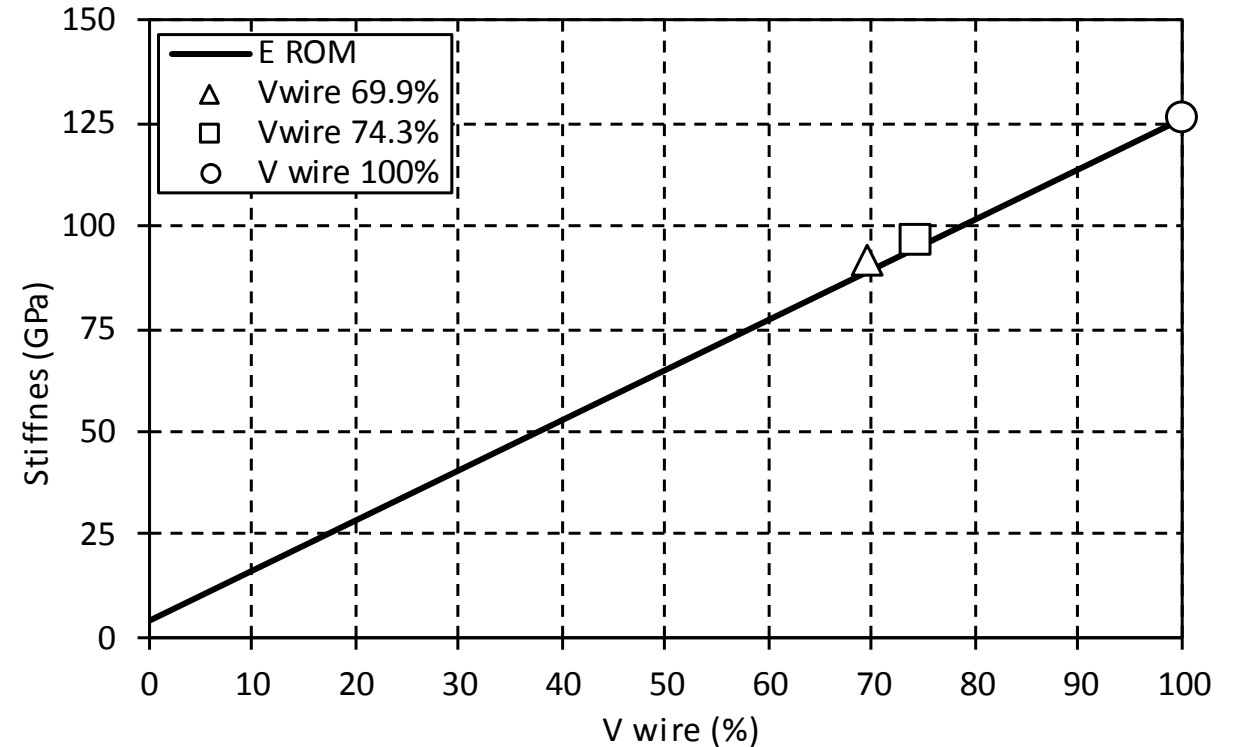
(a) Nb₃Sn and (b) Cu loading stresses in impregnated cable stack as a function of axial compressive stress. Encircled data points show the residual strain when the external stress is released.

Monotonic axial loading: Experiment vs rule of mixture (ROM) estimate

- Stress strain measurements in axial compression with ten stack samples with different epoxy volume fraction.
- Assuming $E_{\text{wire-axial}}=126$ GPa and $E_{\text{epoxy}}=4$ GPa.
- Stiffness can be predicted assuming iso-strain conditions. The load is essentially carried by the wire, and the stress in the epoxy remains small.



Comparison of stress-strain curves of Nb₃Sn wire (axial tension) and ten-stack samples wire different volume fraction (V_{wire}) (axial compression).



ROM estimation (upper bound) of the axial compression stiffness dependence on V_{wire} and comparison with experiment.

From Felix Wolf et al, "Effect of epoxy volume fraction on the stiffness of Nb₃Sn Rutherford cable stacks"

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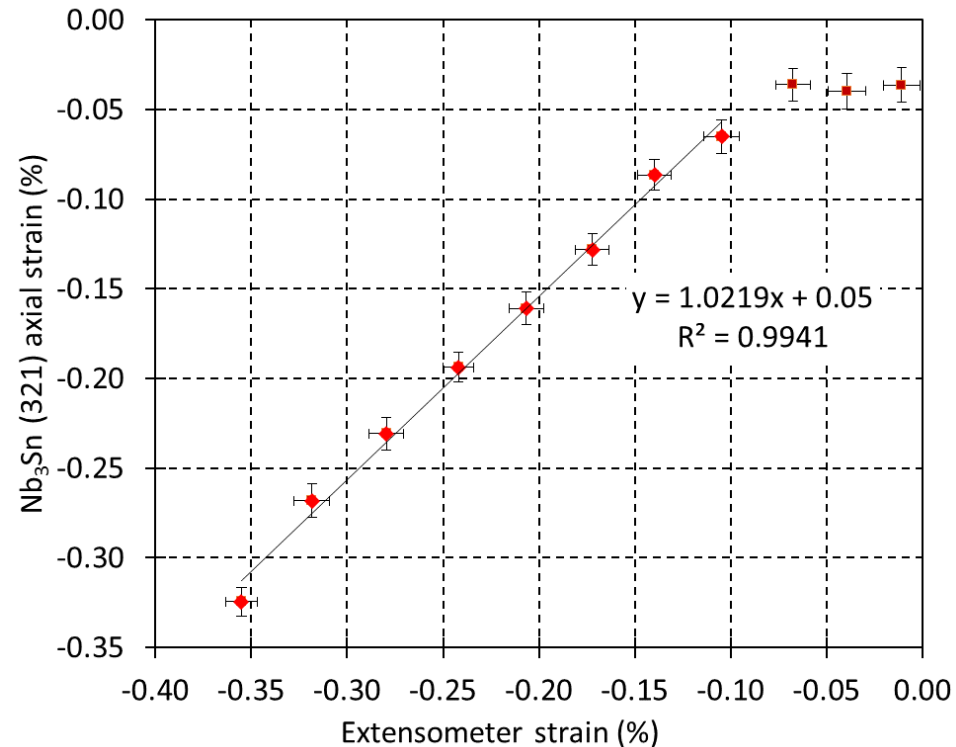
Stress-strain behaviour of Nb₃Sn coils in accelerator magnets

- For the first time it was possible to determine experimentally Nb₃Sn and Cu loading strains and stresses in Rutherford cable stacks.
- These results are a new opportunity to validate FE models.
- Ten-stack samples are representative for coils made of the same conductor with the same epoxy volume fractions (see presentation of F. Wolf).
- The load case of transverse compression of the unconstrained cable stacks is probably not representative for the mechanical loading in magnets where the coils are constrained in axial and radial directions.
- Nb₃Sn coils can be considered as a fiber reinforced composite. The three main results of this study are valid for magnets too:
 - Iso-strain conditions in axial load direction
 - Iso-stress conditions in transverse compression
 - Isotropic compressive stress in the annealed Cu stabilizer

Stress-strain behaviour of Nb₃Sn coils in axial and transverse load direction

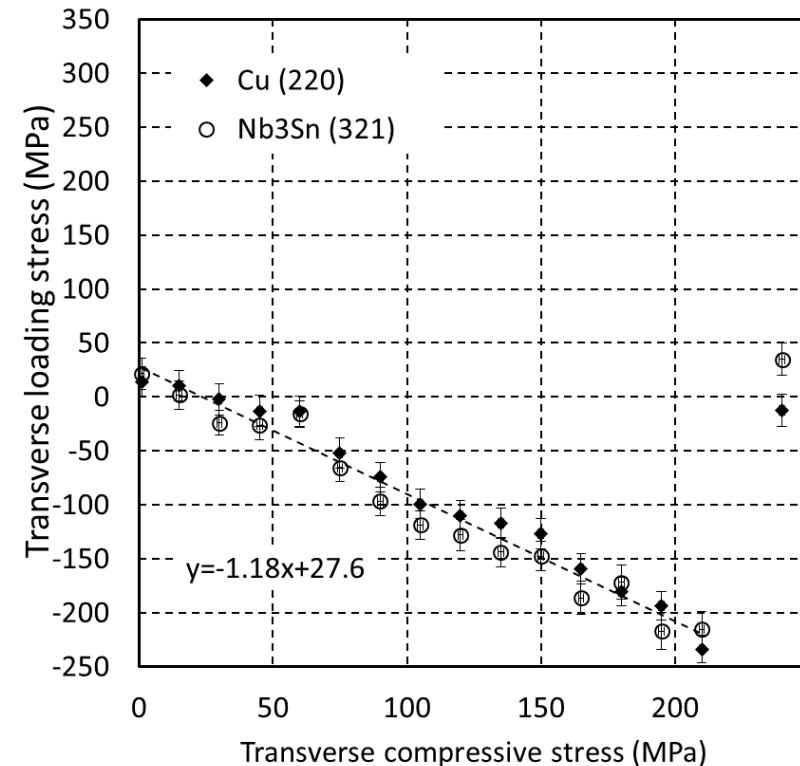
Axial loading

- Iso-strain conditions
- Transverse and radial Nb₃Sn lattice spacing changes are mainly caused by the Poisson effect.
- Axial coil stiffness can be predicted from the epoxy volume fraction and the wire stiffness (at RT the 11 T dipole coil axial stiffness is about 95 GPa).



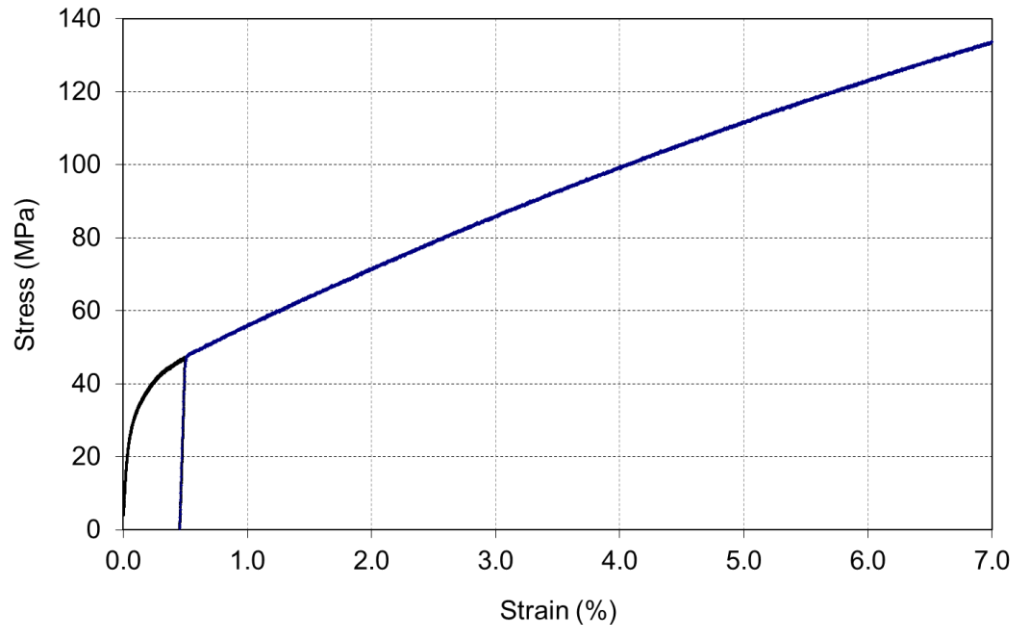
Transverse loading

- Iso-stress conditions
- Cu is under similar compressive stress in all three directions.
- Transverse compression induces an important axial tensile stress in the Nb₃Sn.

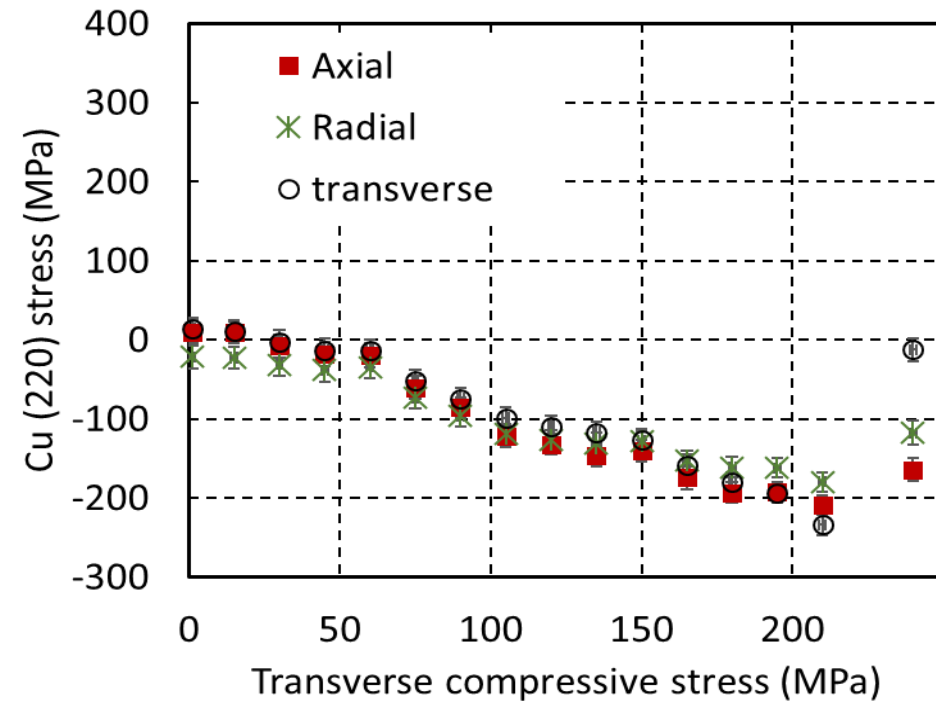


Stress-strain behaviour of the fully annealed Cu stabiliser

- The unconstrained annealed Cu stabiliser yields at about 20 MPa ($R_{p0.2}$ is 40 MPa).
- Therefore, under tension annealed Cu cannot carry high loads.
- Under compression in hydrostatic conditions the annealed Cu provides an isotropic pressure around the Nb_3Sn filaments.
- The magnitude of the Cu pressure corresponds approximately with the externally applied transverse compressive stress.
- In unconstrained cable stacks Cu creeps at high stresses.



Tensile stress-strain curve of fully annealed Cu wire.



Effect of the impregnation material mechanical properties

- Axial loading:

- At 150 MPa axial pressure the Nb₃Sn and Cu axial loading stresses are about 300 MPa and 100 MPa, respectively.
- Under axial loading the effect of the epoxy ($E_{epoxy}=4$ GPa) on coil stiffness and Nb₃Sn loading stress is negligible.
- A reinforcement in axial load direction should have highest possible E-modulus.

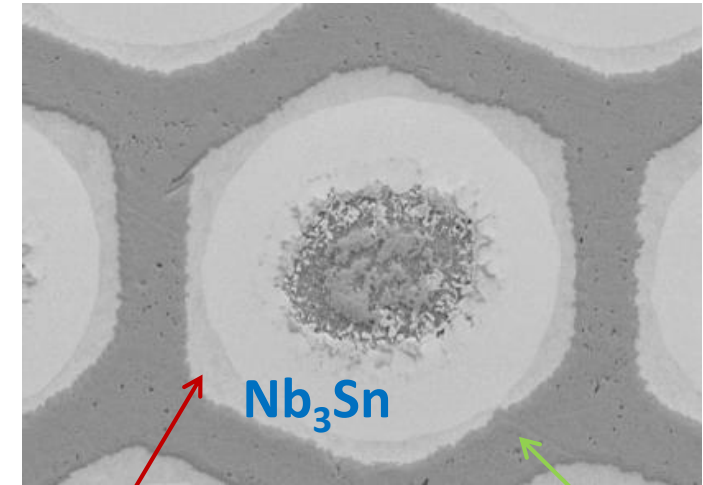
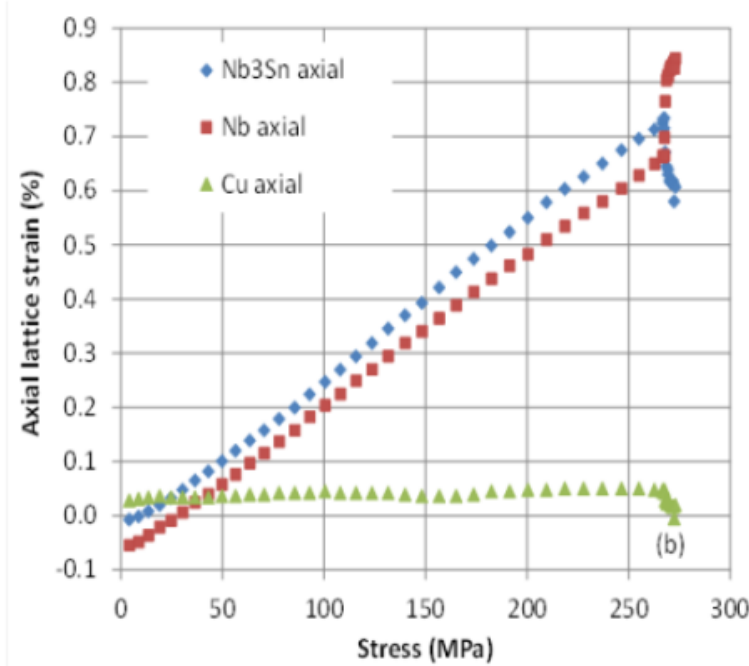
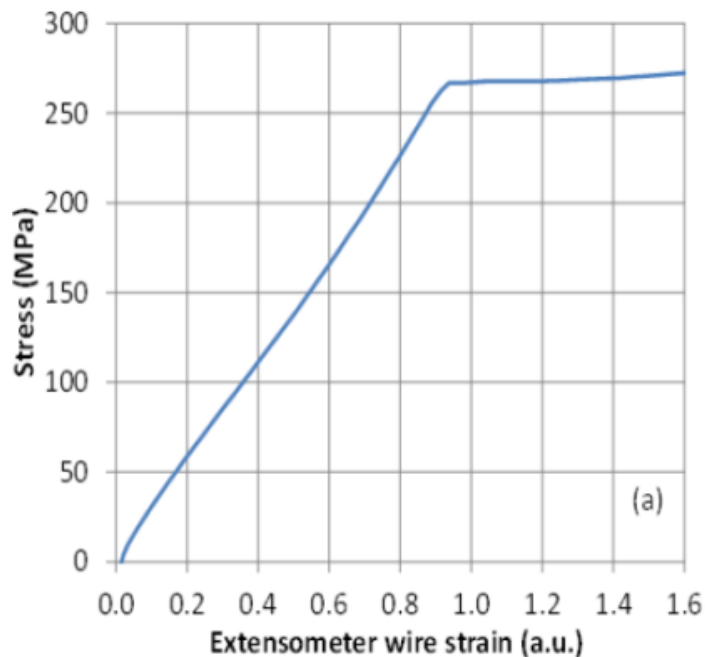
- Transverse compression:

- At 150 MPa transverse pressure the average transverse Nb₃Sn and Cu loading stresses are about 150 MPa.
- Under transverse compression the epoxy impregnation can reduce the Nb₃Sn loading stresses.
- Assuming iso-stress behavior, the elastic properties of the impregnation material should not strongly influence the Nb₃Sn coil stiffness.

Back-up slides

Mechanical behaviour of Nb₃Sn/Cu PIT wire constituents studied by simultaneous stress-strain-XRD measurements at 4.2 K

- Loading strain is measured in two directions (axial and transverse/radial)
- Linear elastic behaviour of Nb and Nb₃Sn
- Nb and Nb₃Sn axial pre-compression in the unloaded wire
- Load transfer from Nb₃Sn to Nb at a wire stress of about 270 MPa and a strain of about 0.7%.



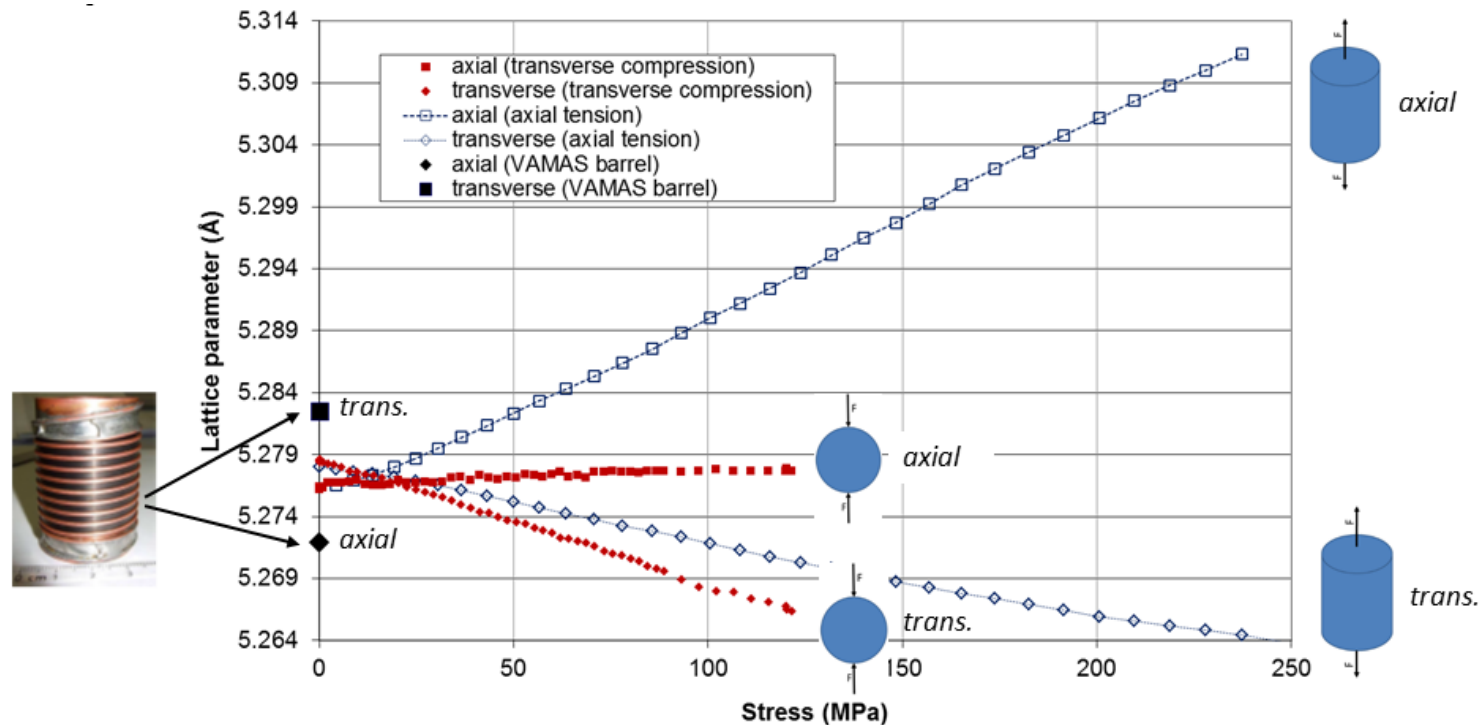
Nb barrier

Cu matrix

Supercond. Sci. Technol. 27, (2014), 044021.

Axial and transverse Nb₃Sn lattice parameter changes in a PIT wire as a function of externally applied axial tensile and transverse compressive stress

- Measured by high energy synchrotron X-ray diffraction
- Loading strain was measured in two principal directions (axial and transverse), but not in radial direction.
→ Loading stresses cannot be calculated.
- Nearly linear lattice spacing vs stress dependence.
- Slight d-spacing increase in axial direction under transverse compression.



Supercond. Sci. Technol. 27, (2014), 044021.

IEEE Trans. Appl. Supercond. 19(3), (2009), 2645-2648

Nb₃Sn lattice parameter in axial and transverse direction as a function of axial tensile or transverse compressive stress. The lattice parameters for the same PIT wire on a VAMAS barrel are shown for comparison.

Elastic modulus of RRP type Nb₃Sn wire

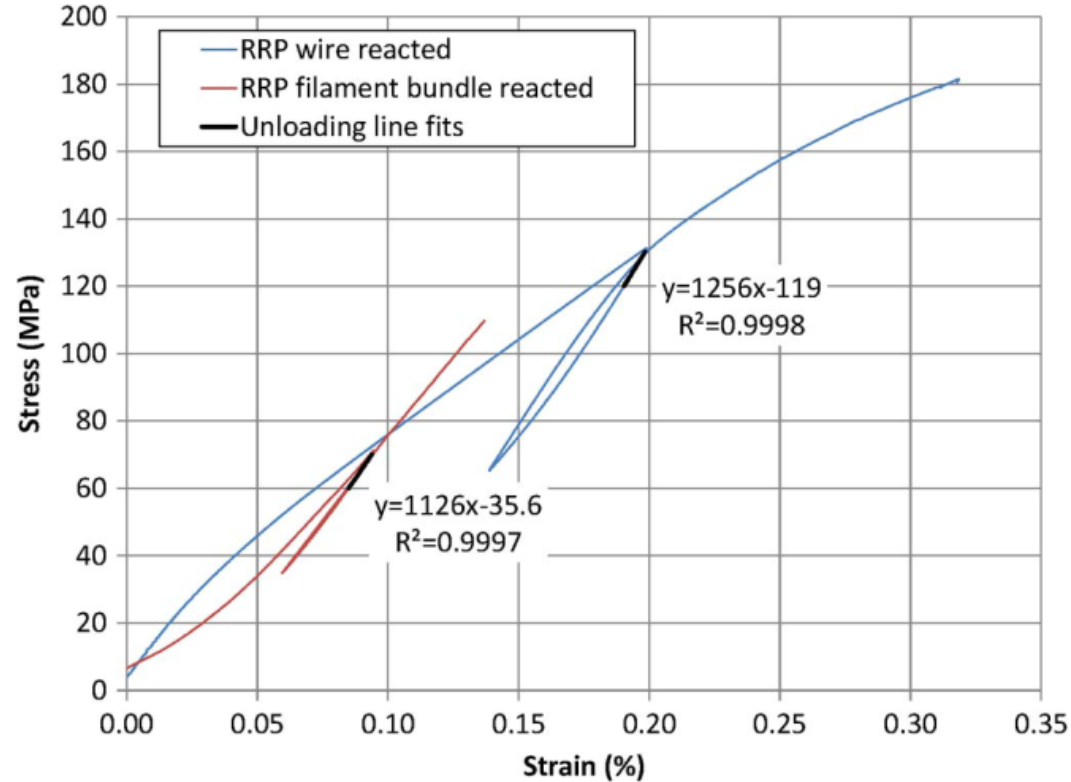


Fig. 3. Stress-strain curves measured at room temperature on a reacted RRP wire and its extracted filaments.

- E is defined as the initial linear slope of the unloading curve.
- Determined elastic modulus of the reacted RRP wire: **126 GPa**

TABLE II
Nb₃Sn ELASTIC MODULI IN AXIAL AND TRANSVERSE DIRECTIONS
CALCULATED FOR THE RRP AND PIT WIRES AT RT AND AT 4.2 K

		PIT B215	RRP #7419
RT	E _{axial}	130	140
	E _{trans}	135	129
4.2 K	E _{axial}	106	127
	E _{trans}	116	104