



In-situ EXAFS investigation of N_2 - treatment of Nb

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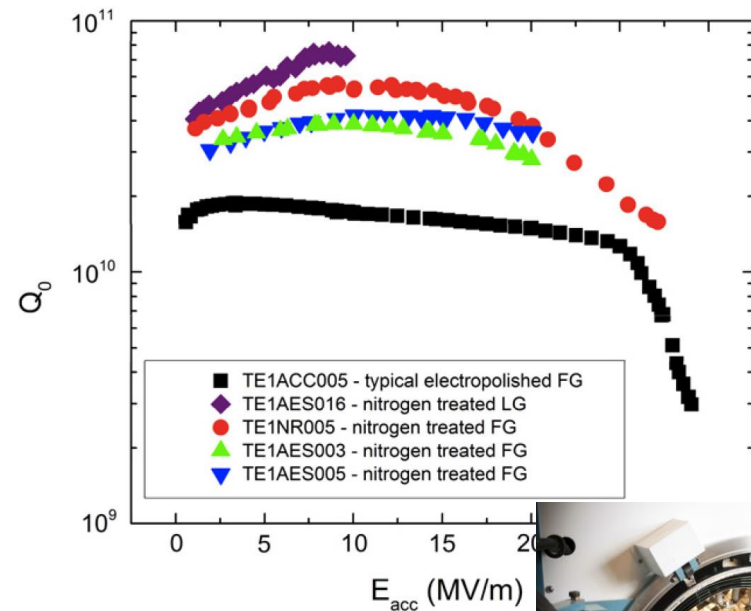


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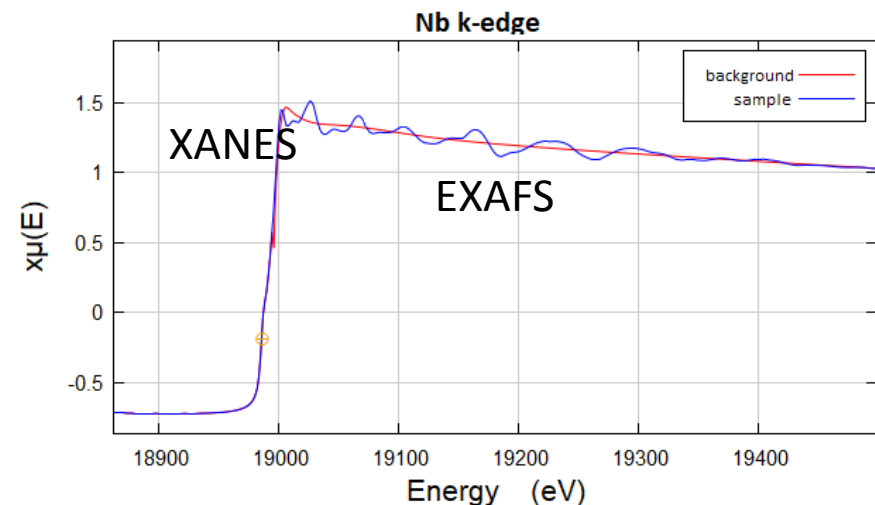


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- 1) Motivation
- 2) Methodology
 - EXAFS spectroscopy
 - Experimental details
- 3) Results and discussion
- 4) Conclusions and outlook



- Improvements of Nb-cavity quality factors by N_2 -treatments at elevated temperatures
⇒ “nitrogen-doping”
- Structural details of N_2 -doping still not understood in detail:
 - Position of nitrogen in the lattice?
 - Distortions of the lattice?
 - ...
- Environment (N_2 - atmosphere, high temperatures, ...)
⇒ difficult to investigate with standard techniques (XRD, XPS, SEM/TEM, ...)
⇒ alternative techniques!
- **Here:** in - situ investigations with EXAFS



XANES = X-ray Absorption Near-Edge Spectroscopy

EXAFS = Extended X-ray Absorption Fine-Structure

Element Specific: Absorption edges are element-specific

Valence Probe: XANES gives chemical state and formal valence of selected element.

Local Structure Probe: EXAFS gives atomic species, distance, and number of near-neighbor atoms around a selected element..

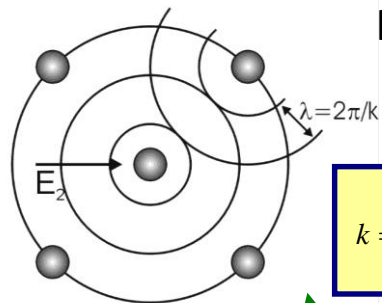
Low Concentration: concentrations down to 10 ppm for XANES, 100 ppm for EXAFS detectable!

Variability: samples can be solids, solutions, amorphous solids, soils, surfaces, etc.

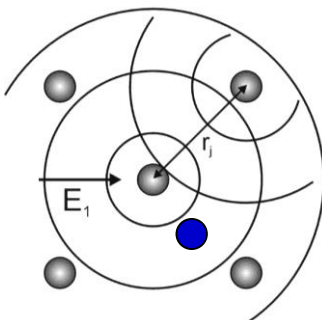
In-situ measurements: Suited for *real in-situ studies* with *time resolution!*

Continuous X-ray source: Synchrotron needed! Here: PETRA III / DESY, SLS / PSI (Ch), DELTA (Dortmund)

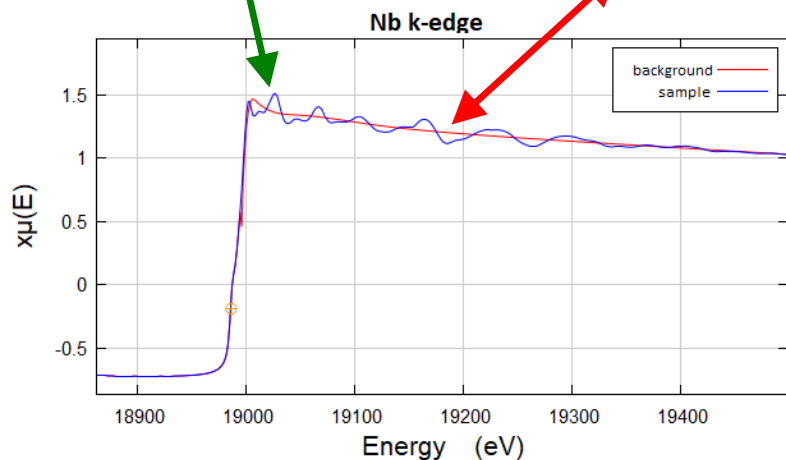
Constructive



Destructive
Interference

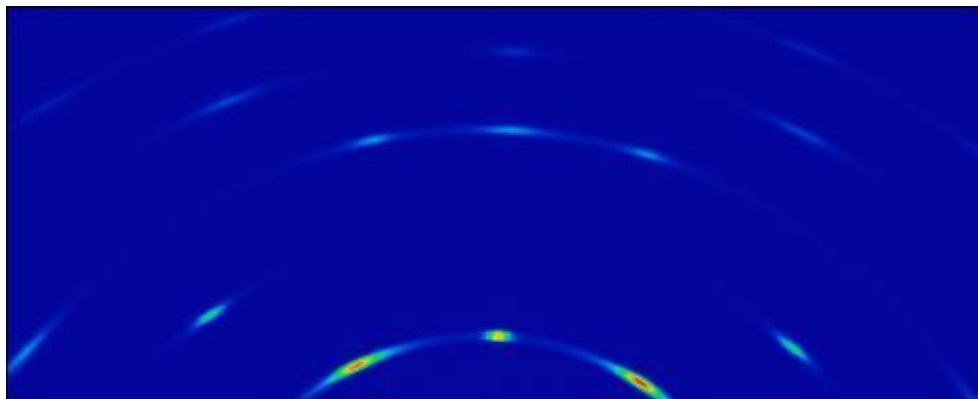


$$k = \sqrt{\frac{2m_e(E - E_0)}{\hbar^2}}$$

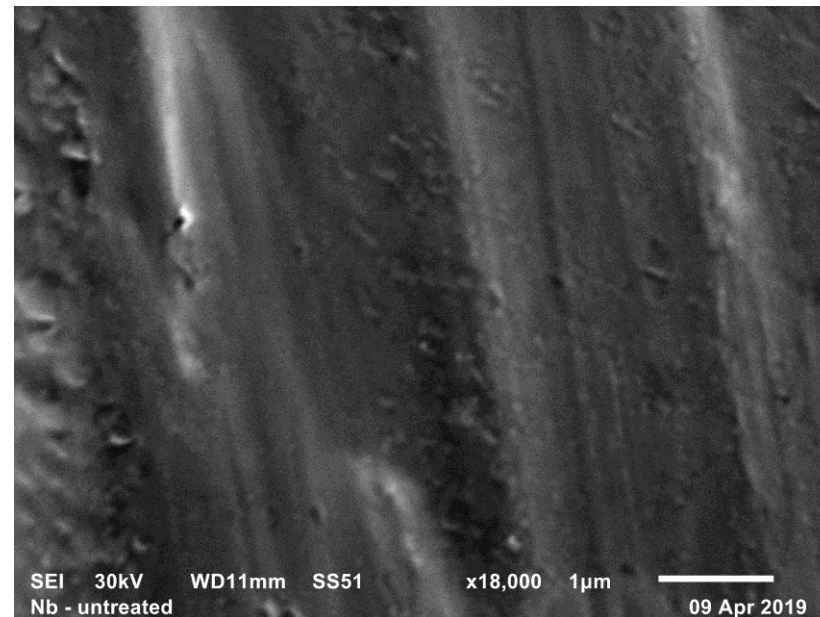


- Measurement of the energy-dependent X-ray absorption coefficient $\mu(E)$ of a core-level of a selected element (here: **Nb K-edge**)
- Good example of wave-particle dualism: Photoelectron acts like a wave \Rightarrow energy-dependent interference of outgoing and scattered waves \Rightarrow modulation of $\mu(E)$
- Fourier-Transform of the oscillations gives bond length in R-space
- **Main Idea:** Inserted N-atoms will slightly modify interference pattern of Nb

High purity Nb-materials:
99.99% Nb foils of 6 – 25 μm thickness

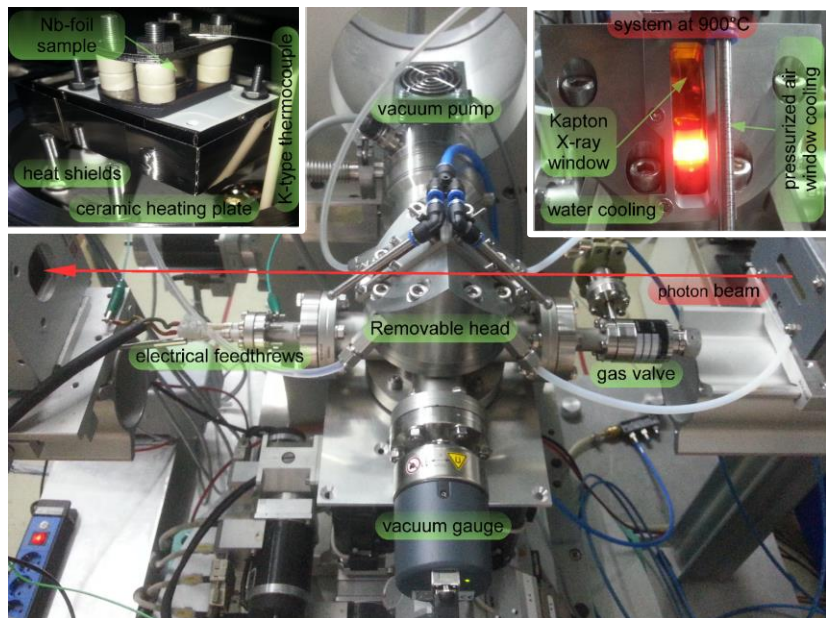


X-ray diffraction, $E = 16 \text{ keV}$:
 \Rightarrow Polycrystalline, textured foils

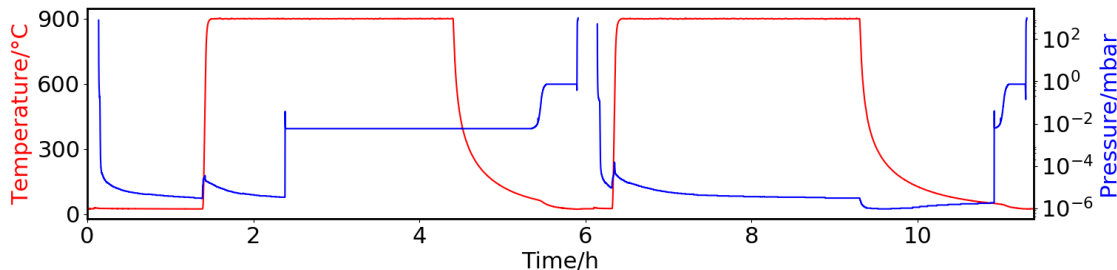


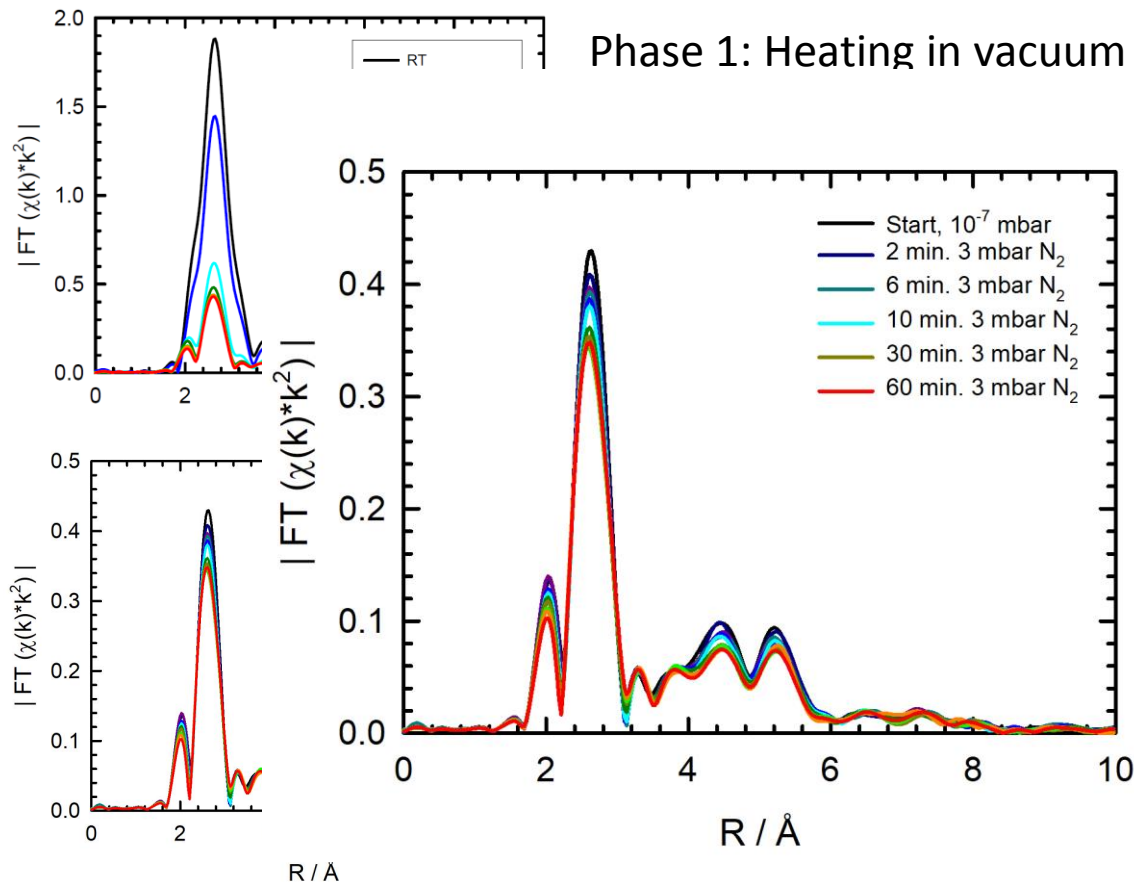
SEM:

\Rightarrow Smooth, wavy surfaces (rolling of the foils)



- In-situ heating chamber, fully remote controlled
- Kapton windows for X-ray measurements
- $RT < T < 1200^{\circ}\text{C}$
- $p < 10^{-6}$ mbar (at 900°C ca. 2×10^{-6} mbar)
- Transmission X-ray measurements
- Treatment with up to 30 mbar N_2 (Kr, Ar, ...) for minutes ... to ... several hours

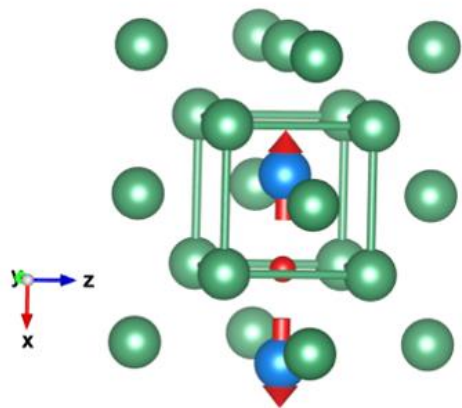




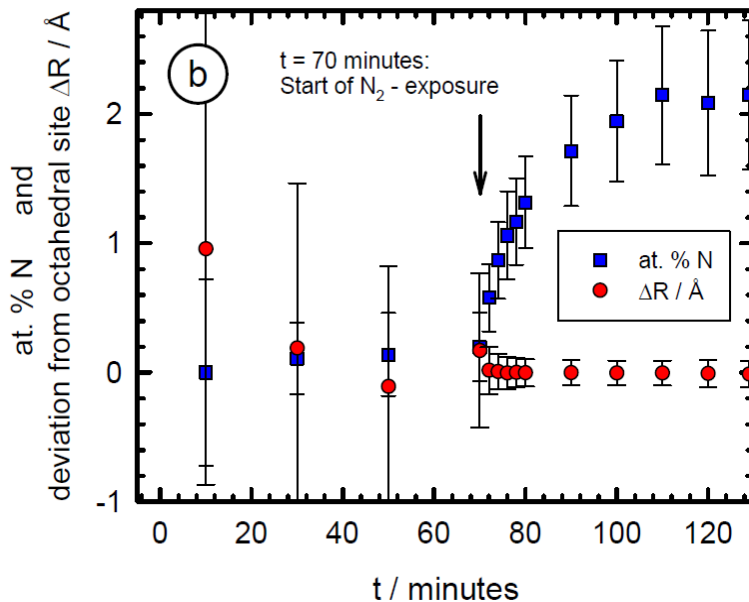
- Plot Fourier-Trafo in R-Space
- Reduction in NN-amplitude
⇒ caused by lattice vibrations
- Constant after first few minutes of baking
⇒ no effects of e.g. poor vacuum conditions etc.

Phase 2: N_2 - exposure

- Further reduction in NN-amplitude caused by N-gas exposure
⇒ Effect of N_2 visible
- Uptake causes blurring

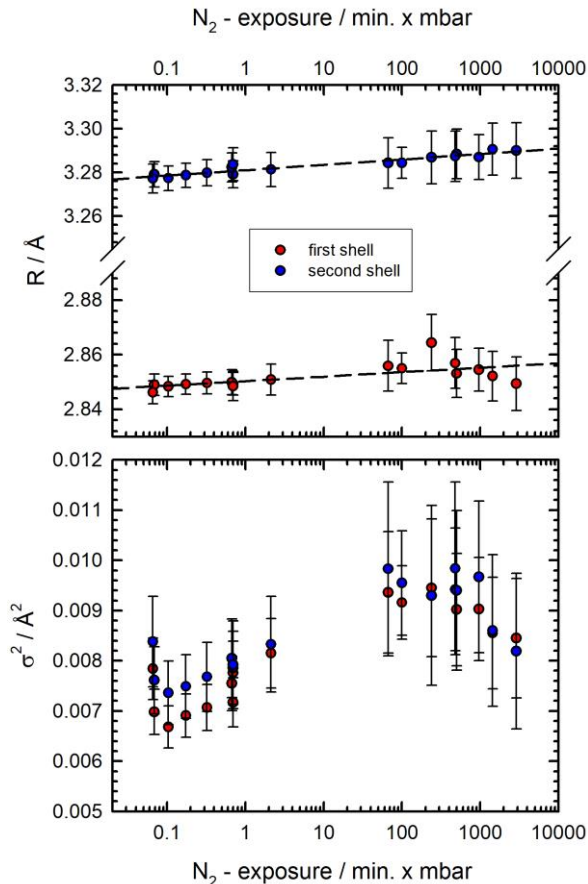
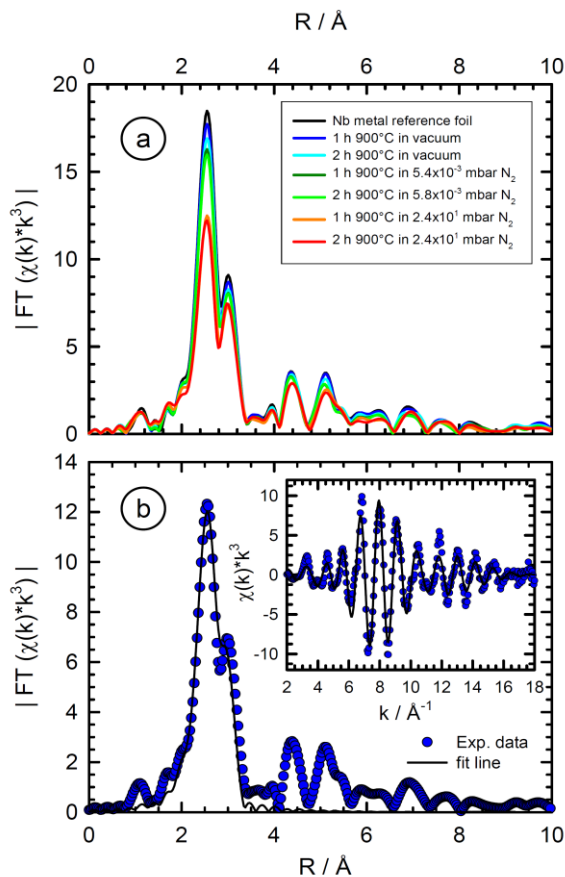


- Nb: bcc-crystal
- N₂-uptake (●) at octahedral interstitial sites
- Neighboring atoms (●) are (slightly) displaced

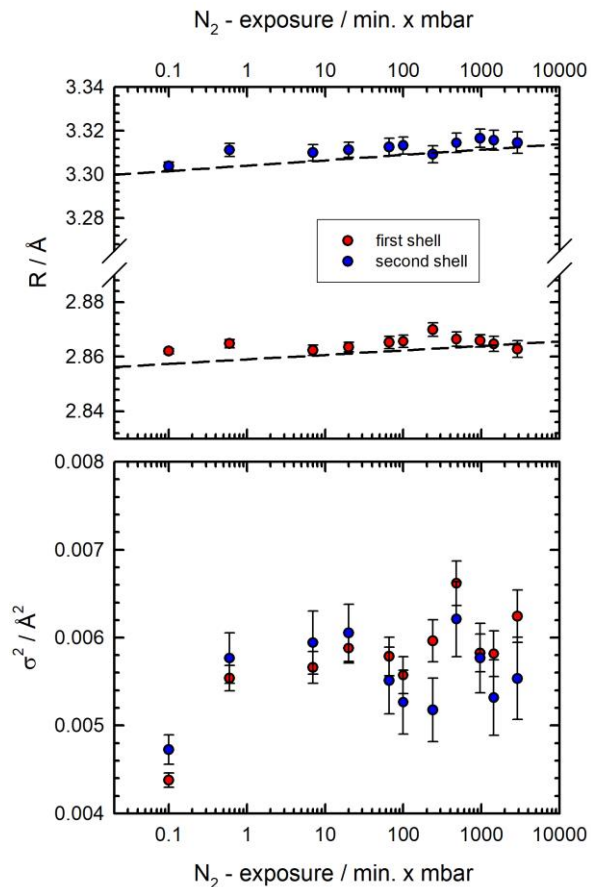


- Model of hard spheres
- Octahedral site competitive with ionic N-size of 70 pm
- Linear combination fits of unit cells with & without N-atoms
- bcc – lattice: 2 Nb atoms/unit- cell)
- up to ca. % of unit cells with N₂

3. Ex-situ measurements at 300 K: Exposure dependence



- Verification of in-situ measurements
- Fit of the first two coordination shells
- Investigation on effect of different exposure intensities on Nb-Nb bond distances (R_1 , R_2) and mean squared displacement (σ_1^2 , σ_2^2)
- Observed increase of R_1 , R_2 , σ_1 , σ_2 agree qualitatively with model
- **But:** large uncertainties

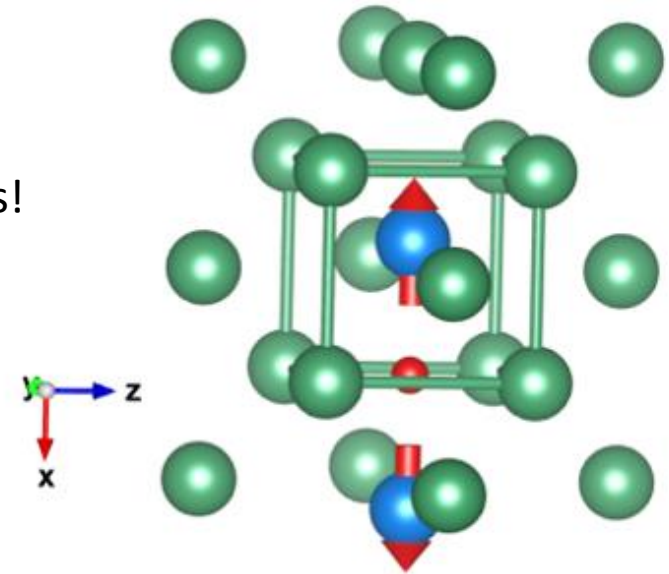


- **Measurements at LN2-temperature:**
 - ⇒ Reduction of thermal lattice vibrations
 - ⇒ larger EXAFS amplitudes, increased accuracy
 - Smaller uncertainties in R
 - Smaller σ^2
- **But:** static disorder increases with N₂- exposure!
& is larger than bulk Nb reference
- **Main results:**
Trends from RT measurements are fostered

- N_2 -uptake in Nb on interstitial octahedral sites, increasing with time and pressure
- In- and ex-situ measurements agree with each other
- precision measurements foster trends (slightly increased average Nb-Nb-distance & disorder)
- Kr- and Ar-uptake: do not show any detectable effects!

Next steps:

- Improvement of cell: T-measurement, MS, ...
- Measurements at 4 K – even more precise ...
- Improvement of the fit model:
 - More shells / paths
- Measurements with samples from real cavities using surface sensitive EXAFS
⇒ **samples from the community are welcome!**





- Department in Wuppertal: R. Frahm, B. Bornmann, P. Pagel, R. Wabnitz, ...



- DELTA (TU Dortmund, Germany)



- PSI – Villingen, Swiss Light Source



- DESY photon science Hamburg



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- MWF NRW: financial support

... and you for you attention!!!



EXAFS - Function $\chi(k)$:

K-edges, Gaussian pair-distribution function between neighboring atoms:

$$\chi(k) = \sum_i N_i |f_i(\pi, k)| \frac{1}{k r_i^2} e^{-2r_i/\lambda} e^{-2\sigma_i^2 k^2} \cdot \sin[2k r_i + \phi_i(k)]$$

Scattering power
damping
therm. + stat. disorder
oscillations

k	wave-number of the photo-electron:	$k = \sqrt{\frac{2m_e(E - E_0)}{\hbar^2}}$
r	distance to neighboring atom	
$\phi(k)$	phase-shift: absorber $\phi_A(k)$, scattering atom $\phi_B(k)$ (sensitive for the type of neighboring atom)	
N	coordination number of neighboring atom	
$ f(\pi, k) $	scattering amplitude of neighboring atom (element - sensitive)	
λ	mean free path of the photo-electron	
σ^2	mean squared displacement (stat. & therm.!) of the neighbor	