

Unambiguous identification of the split-vacancy configuration of the Sn V^- defect in diamond

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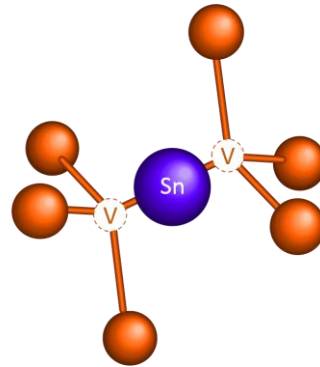
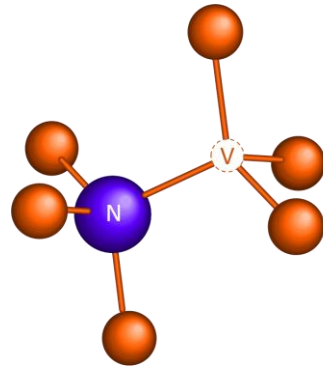
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Motivation: Sn V in diamond as single photon center

- NV, Si V, Ge V, Sn V [1,2] and Pb V centers in diamond are intensively investigated for their applications in processing and communication of quantum information and metrology.
- Two possible configurations for impurity-vacancy centers:

“full-vacancy”,
assumed for NV

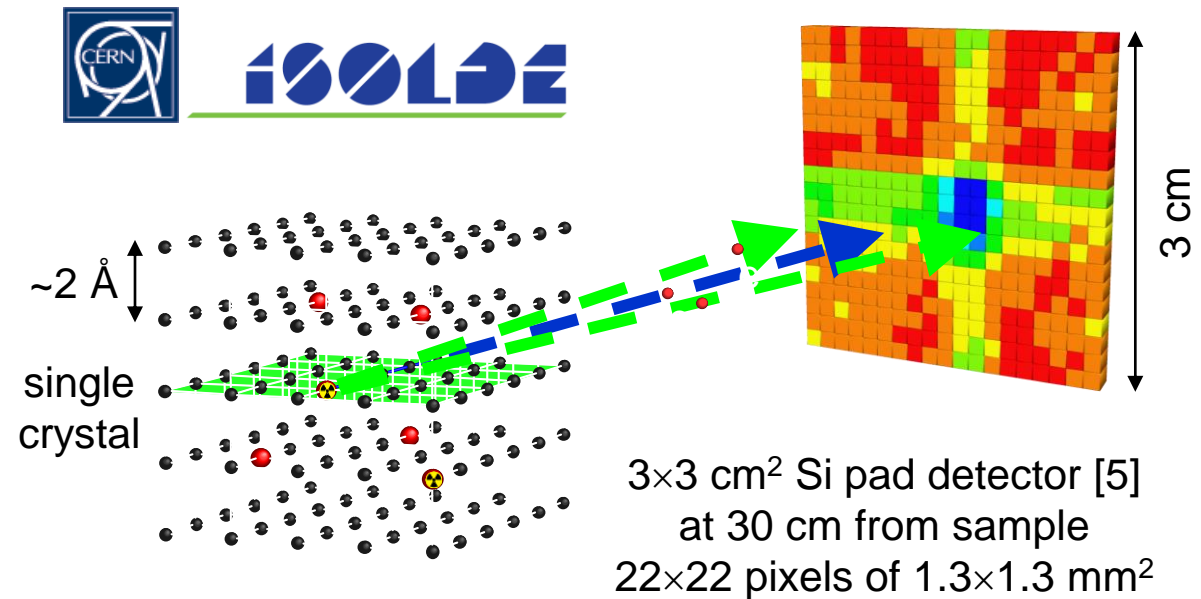


“split-vacancy” [3,4],
assumed for Si V,
Ge V, Sn V, Pb V

- Superior optical properties of the group-IV-vacancy centers are to a large extent a consequence of the D_{3d} inversion (mirror) symmetry.
- So far, no direct **structural** evidence available that these configurations are actually formed.
- Emission channeling lattice location experiments are uniquely suited to study this problem.

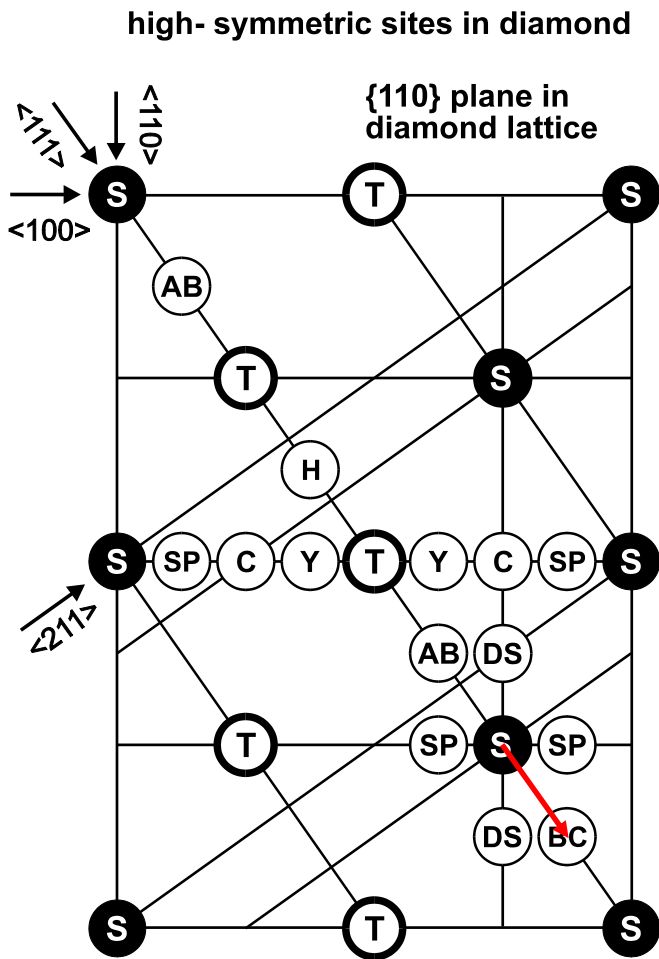
Emission Channeling with Short-Lived Isotopes (EC-SLI)

- Radioactive ^{121}Sn ($t_{1/2}=27$ h) probe atoms are produced at CERN's ISOLDE on-line isotope separator facility.
- 60 keV ion implanted ($2 \times 10^{12} \text{ cm}^{-2}$) into natural type IIa diamond, measured RT as-implanted and 920°C annealed
- Position- and energy sensitive detector [5] is used to detect emission channeling [6] effects of β^- decay particles from ^{121}Sn in the vicinity of major crystallographic directions.

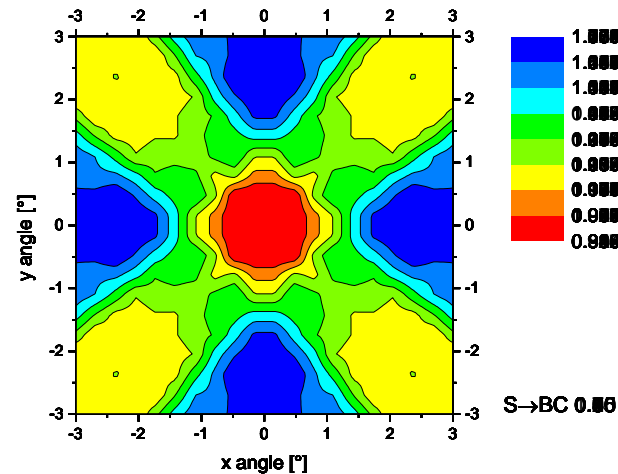


- Angular dependent β^- emission patterns characterize the lattice site distribution of the ^{121}Sn emitter atoms.

“Many-beam” calculation of β^- emission yields

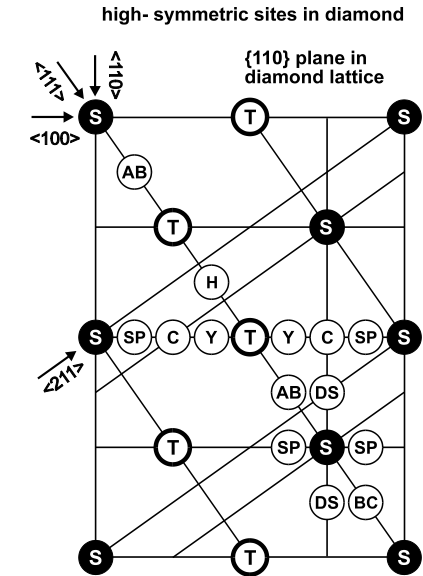
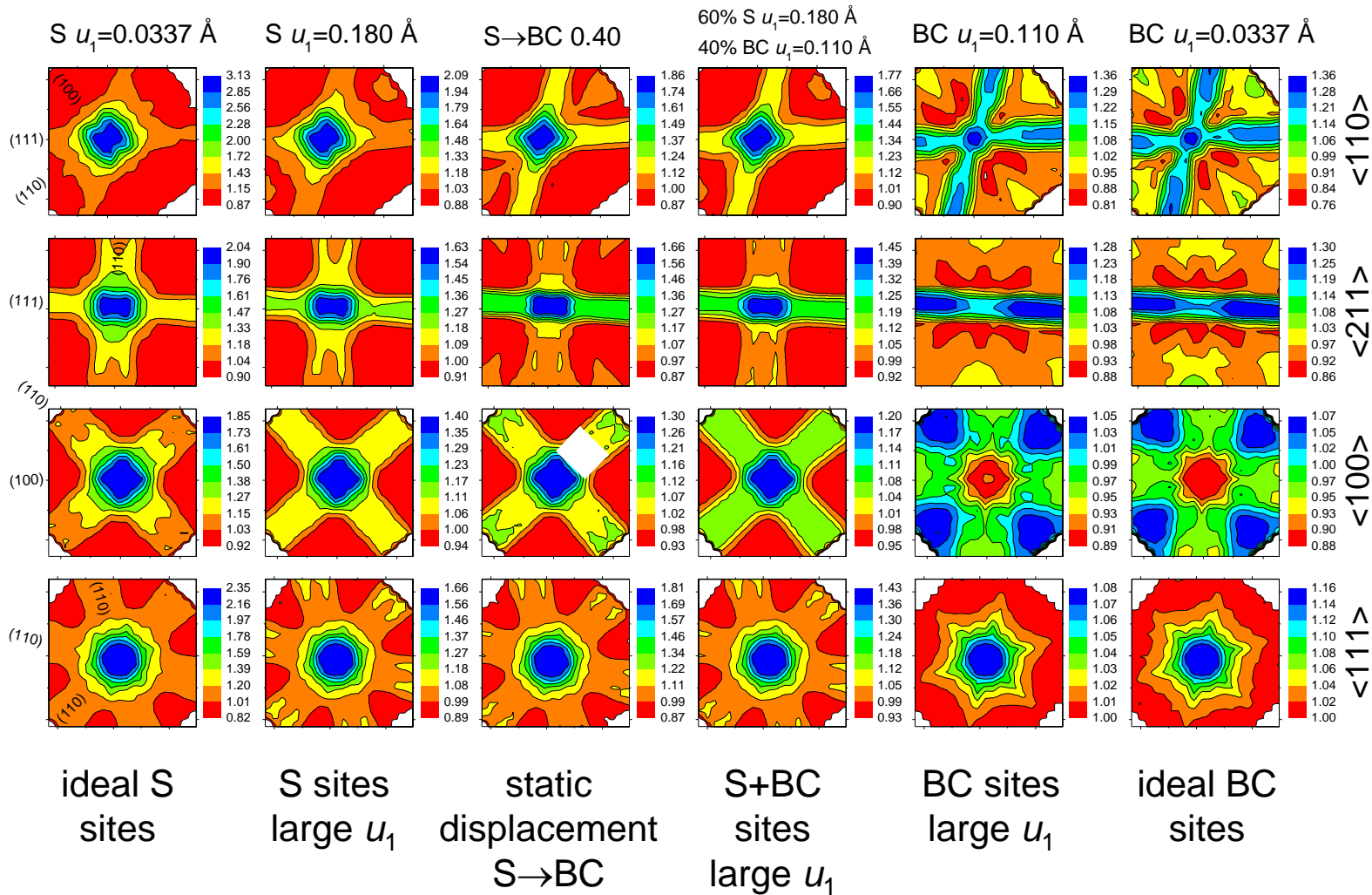


- β^- angular emission yield patterns from ^{121}Sn are calculated for ~ 250 lattice sites in the diamond unit cell using the “many-beam” [6,9] approach.
- Anisotropy and contours of patterns change with position of ^{121}Sn in lattice, e.g. the $\langle 100 \rangle$ pattern when moving from S to BC sites:



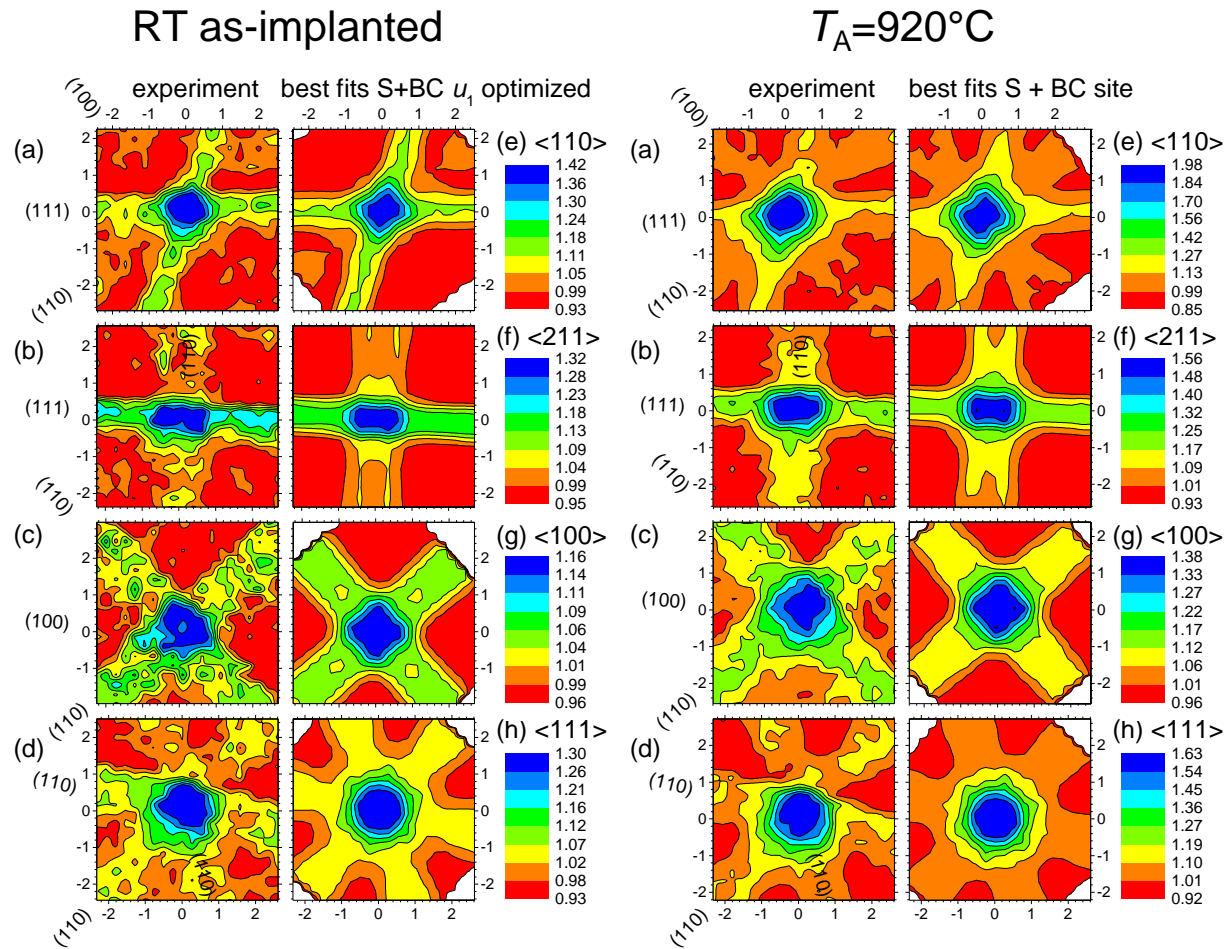
- For S and BC sites patterns also calculated for Gaussian distributions of rms $u_1(^{121}\text{Sn}) = 0.01 \dots 0.50 \text{ \AA}$

Theoretical β^- emission yields from some relevant sites



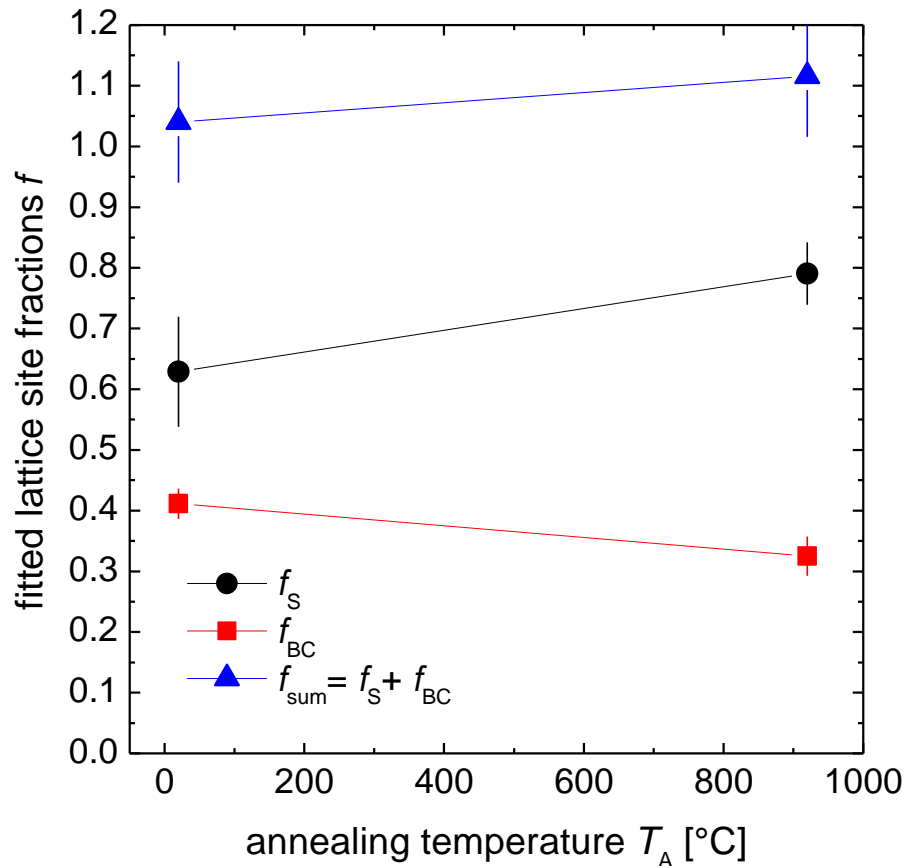
- Increasing the rms displacements $u_1(^{121}\text{Sn})$ from the value $u_1=0.0337 \text{ \AA}$ expected for ideal lattice sites or introducing static displacements reduces the anisotropy of patterns and changes their contours.
- Experimental patterns are analyzed by means of fitting linear superpositions [10] of up to 2 theoretical patterns.

Emission channeling lattice location results



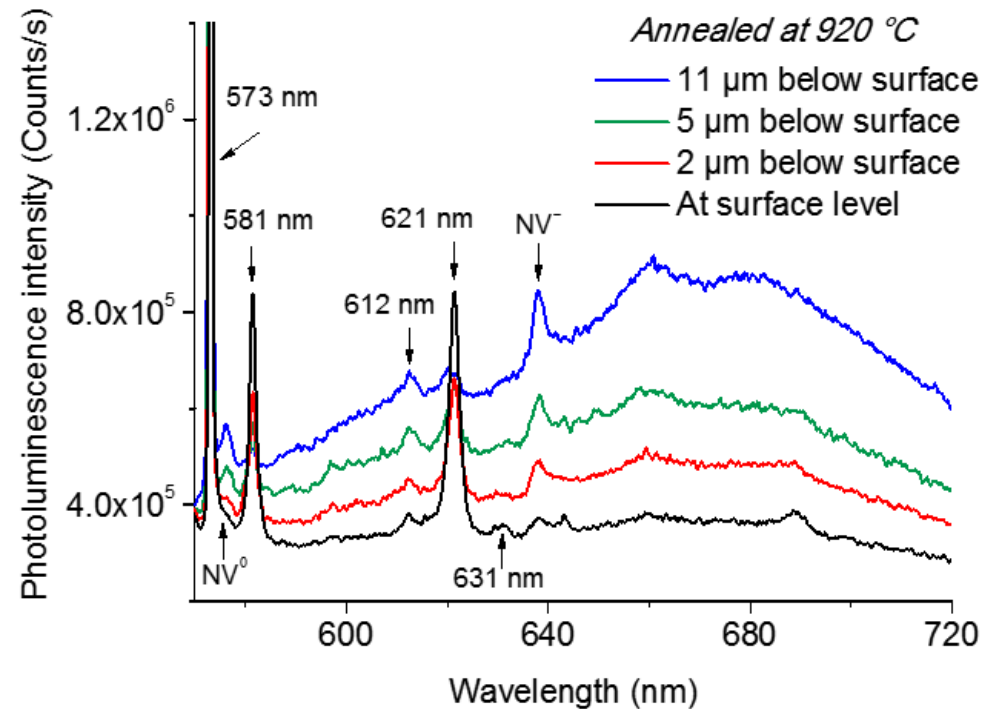
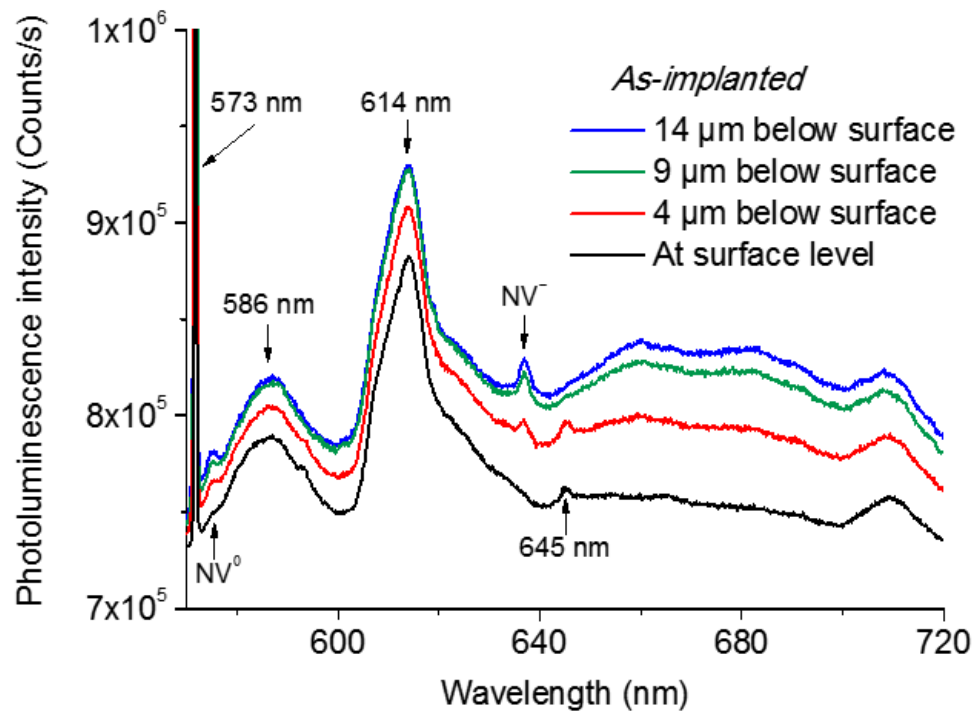
- Strong channeling effects along all axial and planar directions indicate that ^{121}Sn substitutional sites must be involved.
- 920°C annealing ~doubles the maximum yield (β^- anisotropy) of all patterns. A minority fraction is found on BC sites.
- RT as-implanted: best fits obtained for 63% S with $u_1=0.18 \text{ \AA}$
41% BC with $u_1=0.11 \text{ \AA}$
- $T_A=920^\circ\text{C}$: best fits obtained for 79% ideal S with $u_1=0.034 \text{ \AA}$
32% ideal BC with $u_1=0.034 \text{ \AA}$

^{121}Sn site fractions as function of annealing temperature



- Surprisingly high fraction of ^{121}Sn found on BC sites (= “split-vacancy” configuration) already in the as-implanted state.
- Besides reducing the rms displacements from S and BC sites, annealing at 920°C also converts some of the ^{121}Sn from BC sites to S sites.

Confocal depth-resolved photoluminescence



- Same sample was implanted with stable ^{120}Sn at KU Leuven (60 keV, $2 \times 10^{12} \text{ cm}^{-2}$).
- PL excitation by 532 nm laser 1 mW

- RT as-implanted state dominated by broad lines around 586 nm and 614 nm (damage-related)
- 920°C annealed: characteristic sharp PL line (FWHM 2.3 nm) from SnV^- at 621 nm [1,2] near the surface
- Sharp line at 581 nm seems to be the so-called L1 center [11] (of yet unidentified origin)

Isotopes for possible future EC-SLI experiments on colour centers in diamond

isotope	$t_{1/2}$	yield [ions/ μC]	target + ion source	remarks
³¹ Si	157 min	³¹ Al: 2.5×10^5	UC _x + Al RILIS	low yield, ³¹ Si recoil implant
⁷⁵ Ge	82.8 min	⁷⁵ Ga: 3×10^7	UC _x + Ga RILIS	⁷⁵ Ge recoil implanted
²⁰⁹ Pb	3.25 h	?	UC _x + Pb RILIS + LIST?	possible ²⁰⁹ Fr contamination
²⁷ Mg	9.46 min	1×10^7	Ti + Mg RILIS	
⁴⁵ Ca	164 d	⁴⁵ K: 1×10^7	UC _x + W	⁴⁵ Ca recoil implanted
⁶⁵ Ni	2.52 h	7×10^7	UC _x + Ni RILIS	
⁶ He	807 ms	5×10^7	UC _x or BeO	All noble gases
²³ Ne	37.2 s	1.6×10^6	UC _x	from plasma
⁴¹ Ar	109 min	3.2×10^7	TiO ₂ or UC _x	ion sources
⁸⁷ Kr	76.3 min	2×10^8 - 2×10^9	UC _x or PbBi	
¹³³ Xe	5.24 d	6×10^7	PbBi	
¹³⁵ Xe	9.14 h	1.5×10^8	ThC	

- Yields are mostly “book values” from new ISOLDE yield data base.
- In red: $t_{1/2}$ also suitable for radioactive PL;
- For He and Ne: no PL radiotracers;
- For Mg: ²⁸Mg (20.9 h) for PL.

Conclusions

- First direct structural evidence and quantification for implanted Sn in the Sn V “split-vacancy” configuration
- Clarification of the Sn V production via ion implantation:
- RT as-implanted state characterized by high fraction of Sn V (~40%) but in a distorted configuration;
- 920°C annealing results in reconfiguration of Sn in ideal S (~70%) and BC sites (Sn V, ~30%).
- PL signature of Sn V⁻ detected: sharp line at 621 nm.
- Radioactive isotopes for EC-SLI and radiotracer PL can be found for most “colour center elements”.
- Colour centers in diamond will be subject of a proposal to INTC in 2020.