

# ***Defect investigations of neutron irradiated high resistivity PiN and LGAD diodes***

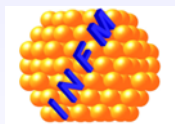
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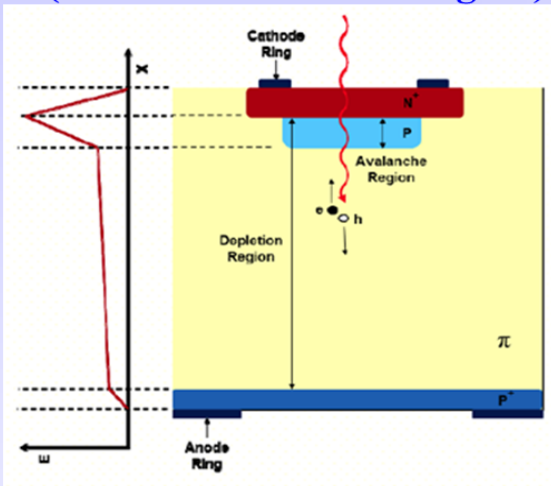
CERN

**Leonid Makarenko**, Belarus University



# Motivation: new structures based on p-type silicon

## LGAD, APD (Sensors with intrinsic gain)



LGADS - doping levels of  $10^{16} - 10^{17} \text{ cm}^{-3}$  in the gain layer and of  $10^{12} - 10^{15} \text{ cm}^{-3}$  in the rest of the structure

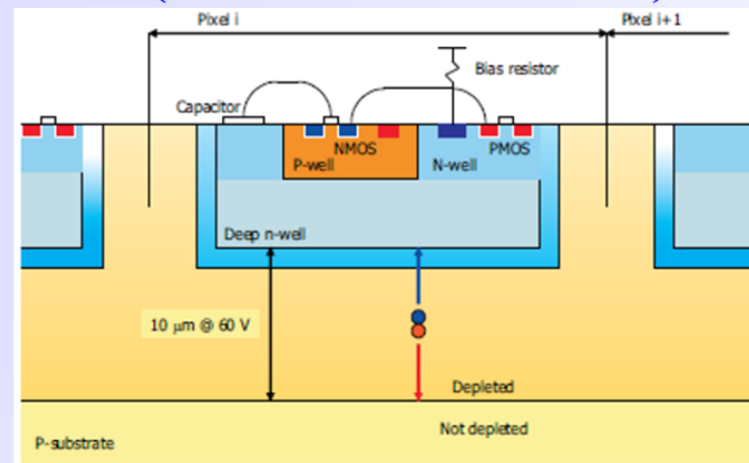
### Key properties

- Gain very sensitive to p+ layer doping and process parameters
- Gains of up to 100 achieved giving excellent timing resolution of 26 ps for thin LGADs
- Currently the best technology for achieving excellent timing measurement for mip – will be employed at ATLAS and CMS experiments after the upgrade

### Limitations:

- Radiation hardness – problem of acceptor removal which decreases the gain with fluence (intensive search for solution: Ga, C and understanding removal mechanism)
- Regions without the gain around the electrodes do not have gain – fill factor improvement

## HVCMOS (towards monolithic sensors)



HVCMOS - doping levels of  $10^{12} - 10^{15} \text{ cm}^{-3}$

### Key properties

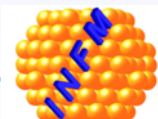
- Different substrates often limited by vendor – up to full depletion of 300 mm
- Excellent position resolution

### Limitations:

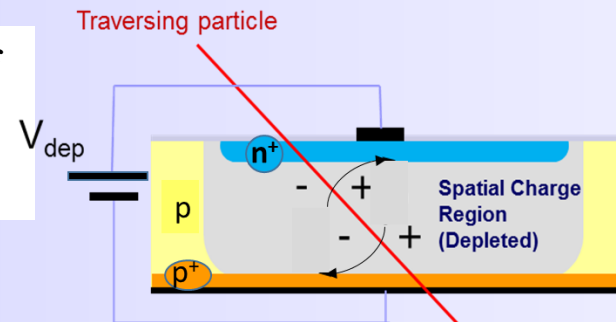
- Radiation hardness – problem of acceptor removal which changes detector performance
- Speed – for timing applications is not yet optimal
- SOI substrates or different other designs/processes including “Shallow Trench Isolation” affect charge collection

..optimizing for

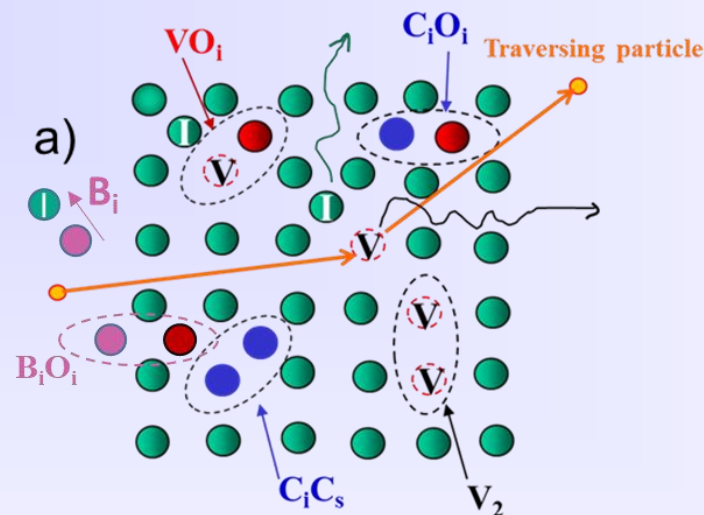
- Radiation hardness
- Time resolution
- Cost effectiveness



**Aim: to improve the radiation hardness** of different types of silicon sensors (single pads, pixel and strips, LGAD and HVCMOS) built on **p-type silicon**



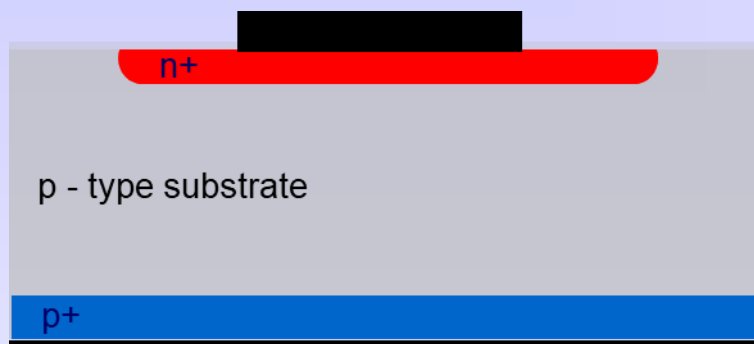
**How ? : By understanding the competing defect generation involving Boron, Carbon, Si<sub>i</sub> and O<sub>i</sub> and find ways for balancing them**



- - Silicon atom in the crystal pattern
- ⊥ - Interstitial Silicon atom
- ⊖ - Silicon vacancy
- - Oxygen impurity atom
- - Carbon impurity atom
- - Boron – dopant atom

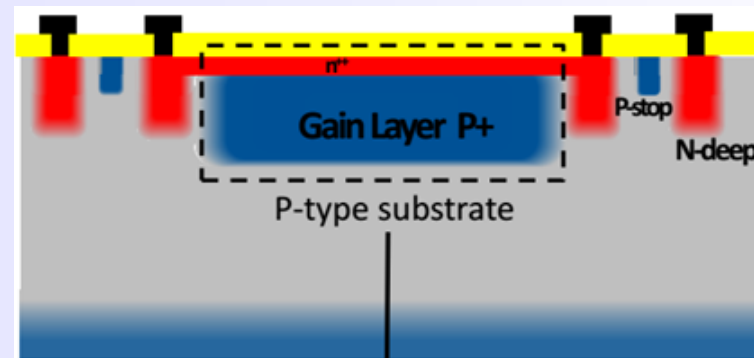
- point defects
- 1)  $V+O \rightarrow VO$
  - 2)  $V_2$
  - 3)  $I+C_s \rightarrow C_i$
  - 4)  $C_i+C_s \rightarrow C_iC_s$
  - 5)  $C_i+O_i \rightarrow C_iO_i$
  - 6)  $I+B_s \rightarrow B_i$
  - 7)  $B_i+O_i \rightarrow B_iO_i$
  - 8) .....?.....

## PiN pads

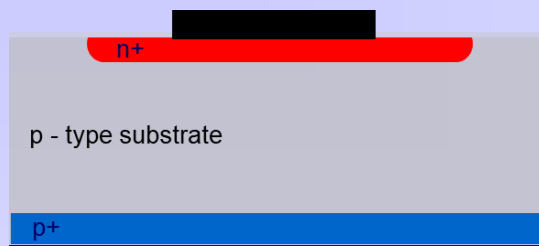


**2 p-type bulk Boron doped STFZ diodes** from *National Center for Micro-electronics (CNM)*—irradiated with 1MeV neutrons, fluences of  $1 \times 10^{14}$  and  $1 \times 10^{15} \text{ cm}^{-2}$

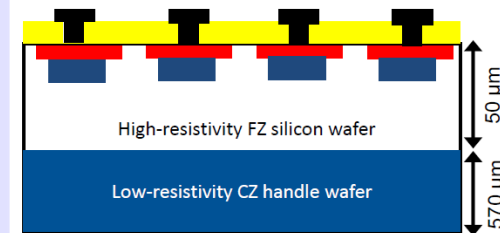
## LGADs



**4 LGAD structures** produced on STFZ silicon from *National Center for Micro-electronics (CNM)* with Boron implanted gain layer— irradiated with 1MeV neutrons, fluences of  $1 \times 10^{14}$  and  $1 \times 10^{15} \text{ cm}^{-2}$  - however, **only 2 could be measured**



$$[B]_{\text{bulk}} = 2-5 \times 10^{12} \text{ cm}^{-3}$$



$$[B]_{\text{gain layer}} \sim 4 \times 10^{16} \text{ cm}^{-3}$$

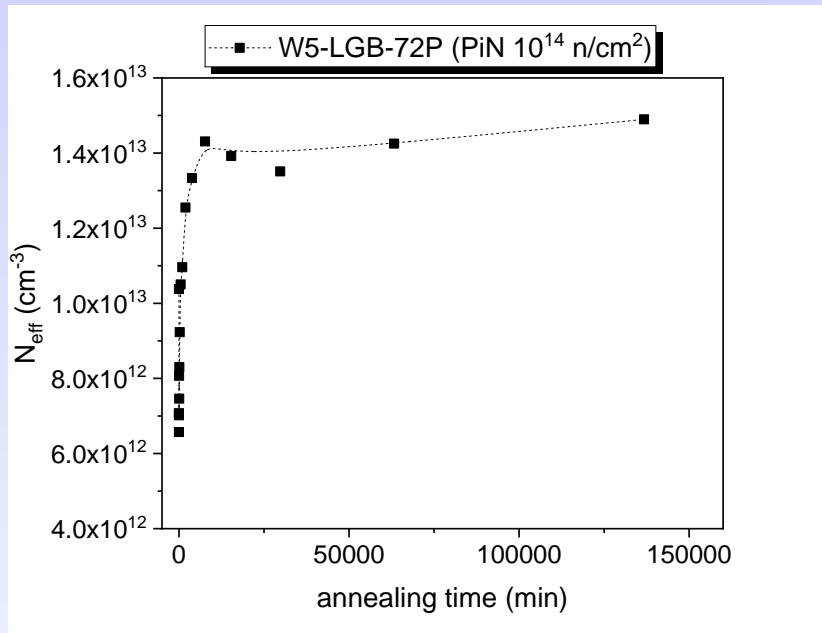
**RUN 9088 neutron irradiated-for TSC measurements**

NR.	Name	Fluence n/cm <sup>2</sup>	Type	Thickness	Full depletion voltage (V)	Depletion voltage of the gain layer (V)	Leakage current at V <sub>dep</sub> (20 C) (A)
1	W5-LGB-82P <i>RUN9088, H6TD, 10.06.2016</i>	1E15	PIN- no gain layer	50 μm	16.8		1.64x10 <sup>-5</sup>
2	W5-LGB-72P <i>RUN9088, H6TD, 10.06.2016</i>	1E14	PIN- no gain layer	50 μm	5.6		1.73x10 <sup>-6</sup>
3	W3-LGB-71, dose 1.8E13 <i>RUN9088, 29.06.2016</i>	1E14	LGAD	49 μm bulk, 1 μm B implanted layer	28.95	26.1	3.78x10 <sup>-6</sup>
4	W11-LGB-74, dose 2.0E13 <i>RUN9088, 30.06.2016</i>	1E15	LGAD	49 μm bulk, 1 μm B implanted layer	28.5	15.9	2x10 <sup>-5</sup>

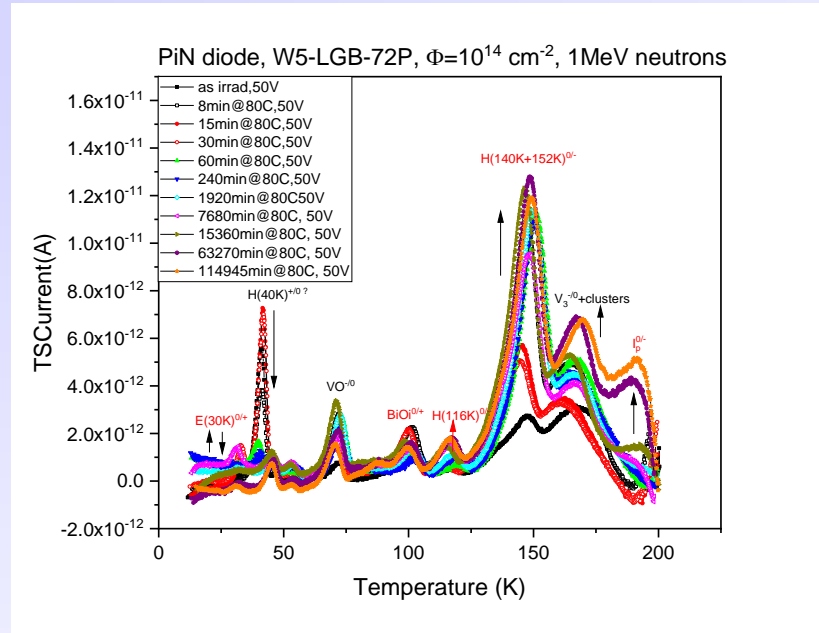


$$\phi = 10^{14} \text{ cm}^{-2}$$

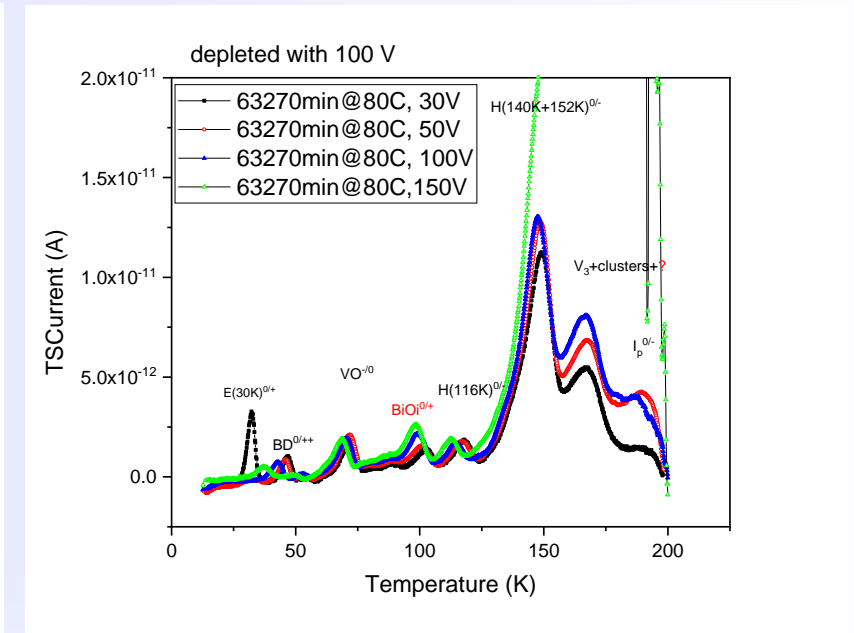
Diode depleted with 100V during TSC



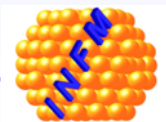
N<sub>eff</sub> from CV

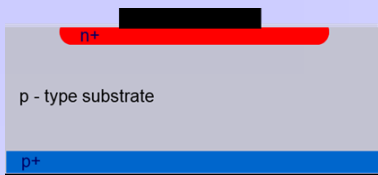


Some random variations in BiO<sub>i</sub> peak

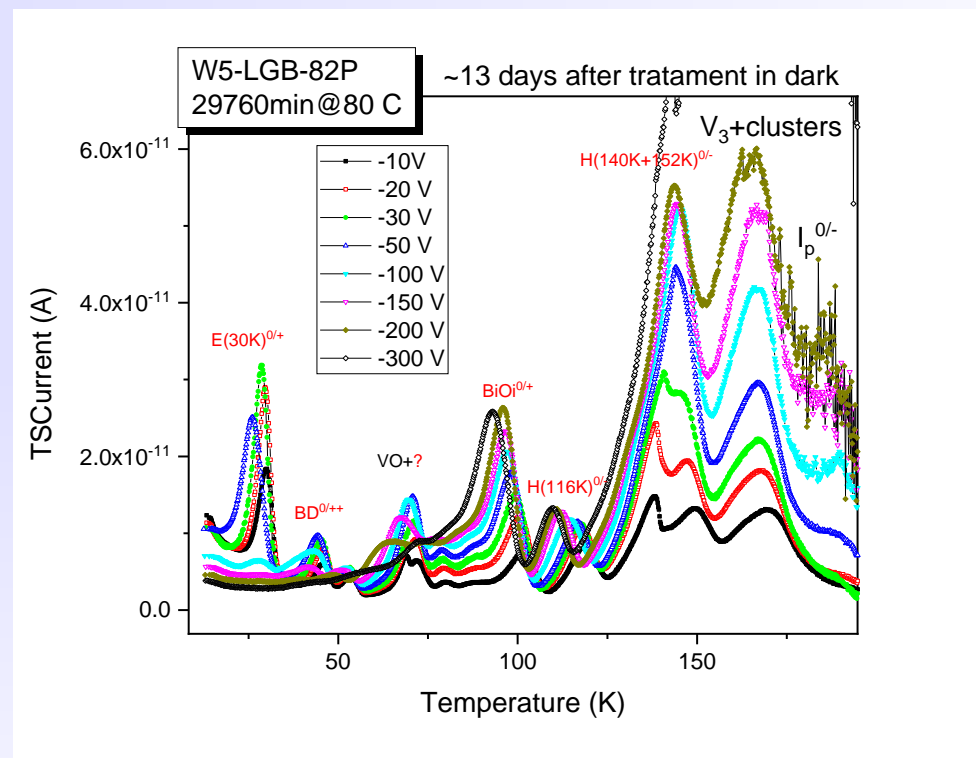
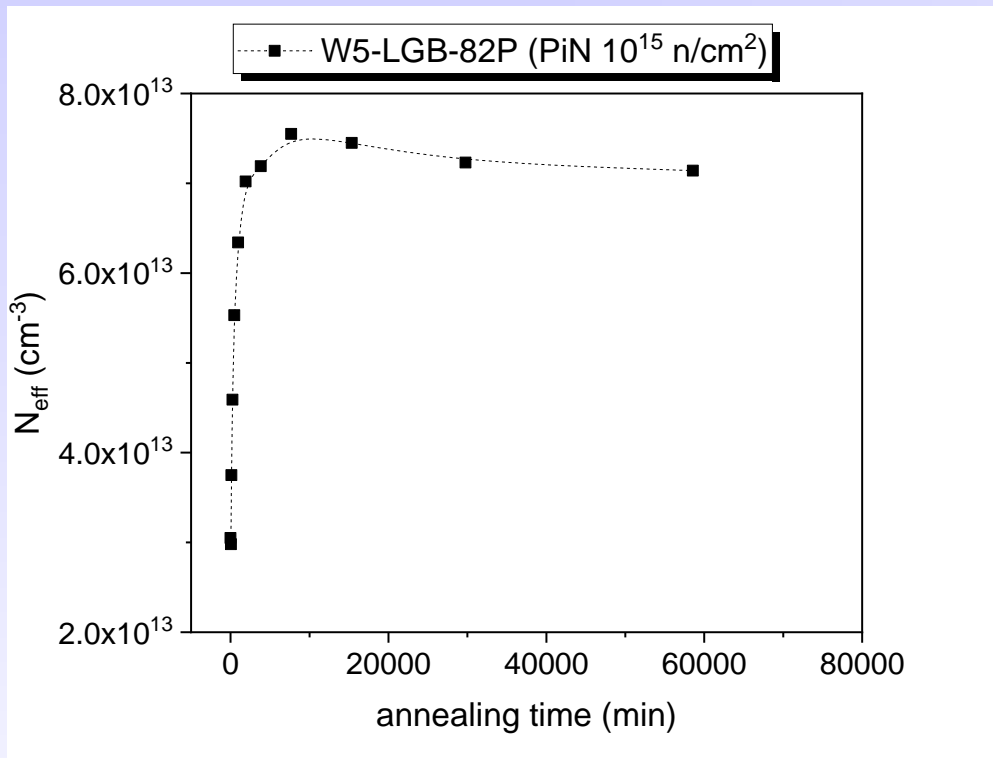


- changes between 100V-150 V due to the actual time passed after the annealing/light exposure

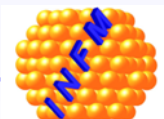


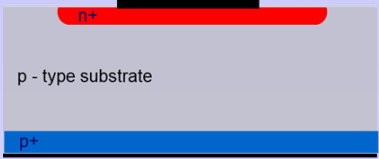


$$\phi = 10^{15} \text{ cm}^{-2}$$



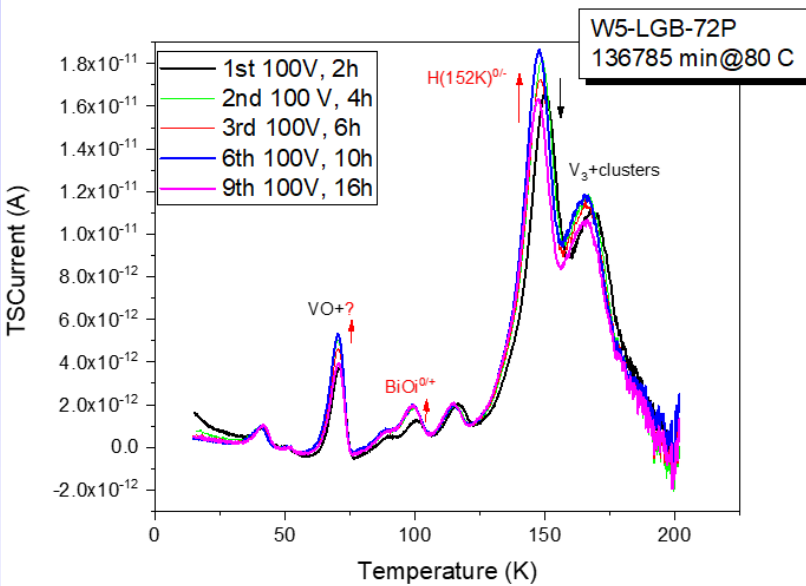
No clear indication for a type-inversion, **diode depleted with 300 V**



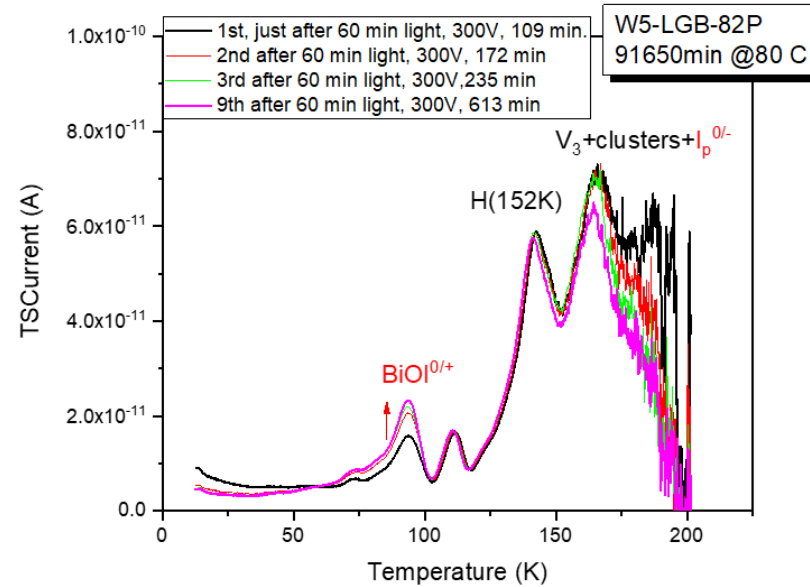


## Consecutive measurements after exposure to light

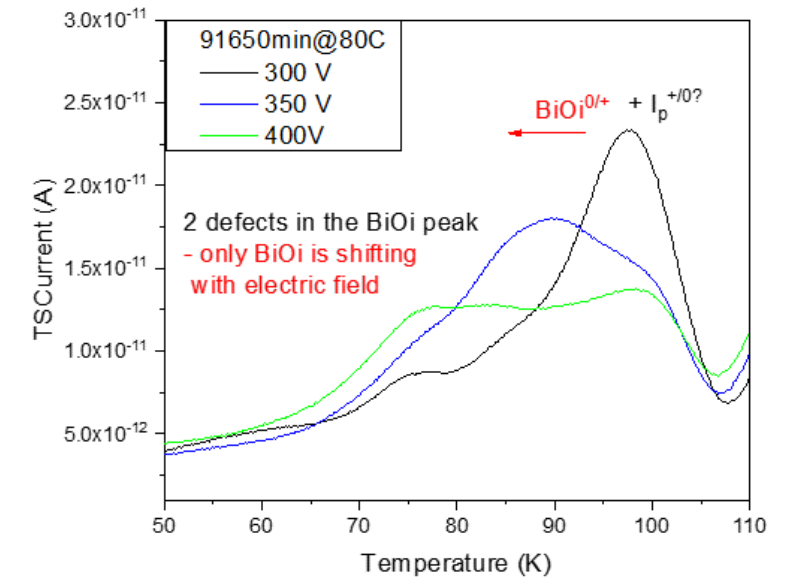
$\phi=10^{14} \text{ cm}^{-2}$



$\phi=10^{15} \text{ cm}^{-2}$



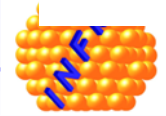
$\phi=10^{15} \text{ cm}^{-2}$



- Increase in the BiOi peak amplitude in time after heating or exposure to light

→ **Bistable electrical activity**

- Only half of the peak at 100 K, when measure with biases < 300 V, belongs to BiOi

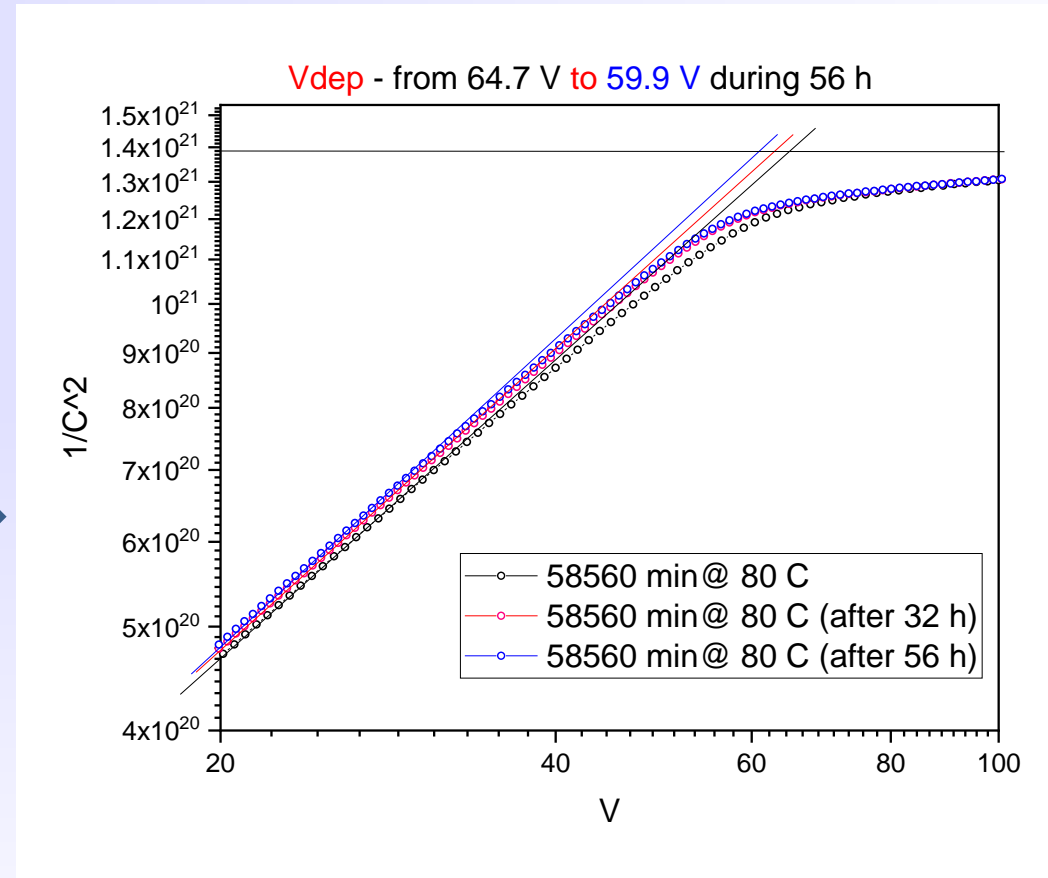
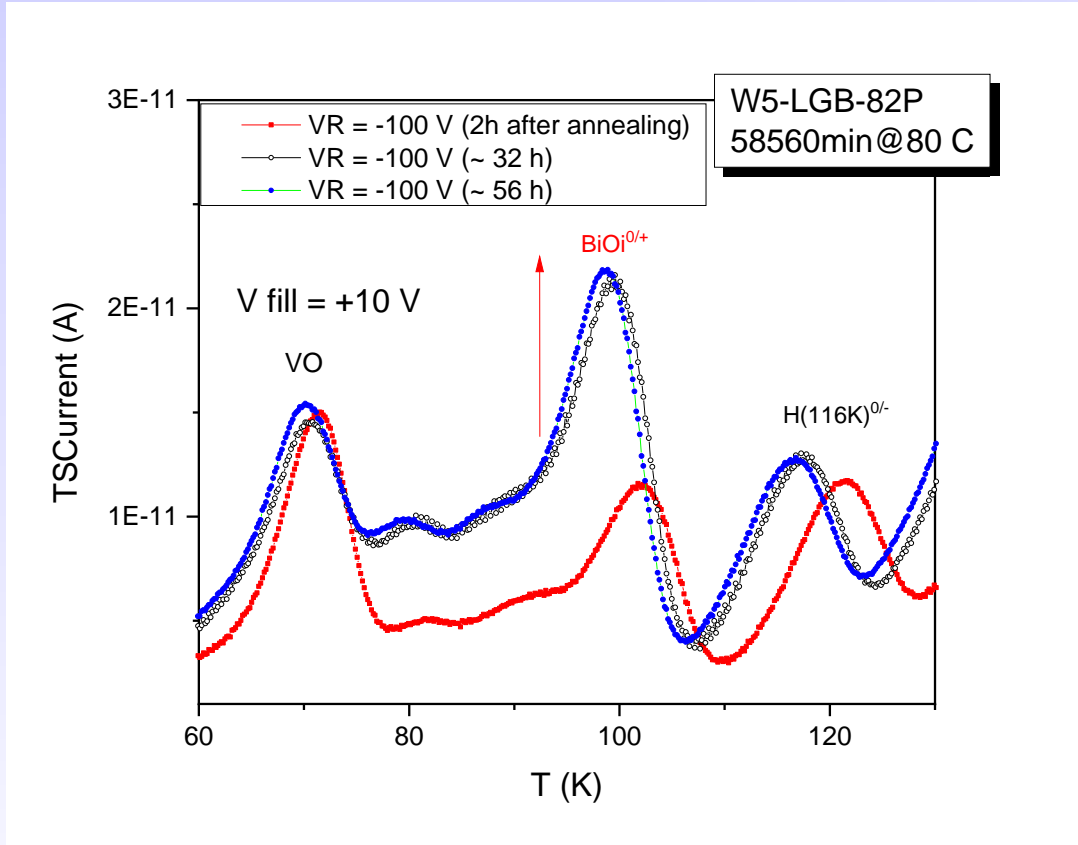




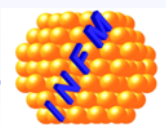
# Instability of BiOi electrical activity after annealing/exposure to light

$$\phi = 10^{15} \text{ cm}^{-2}$$

## Consecutive measurements after heat treatment

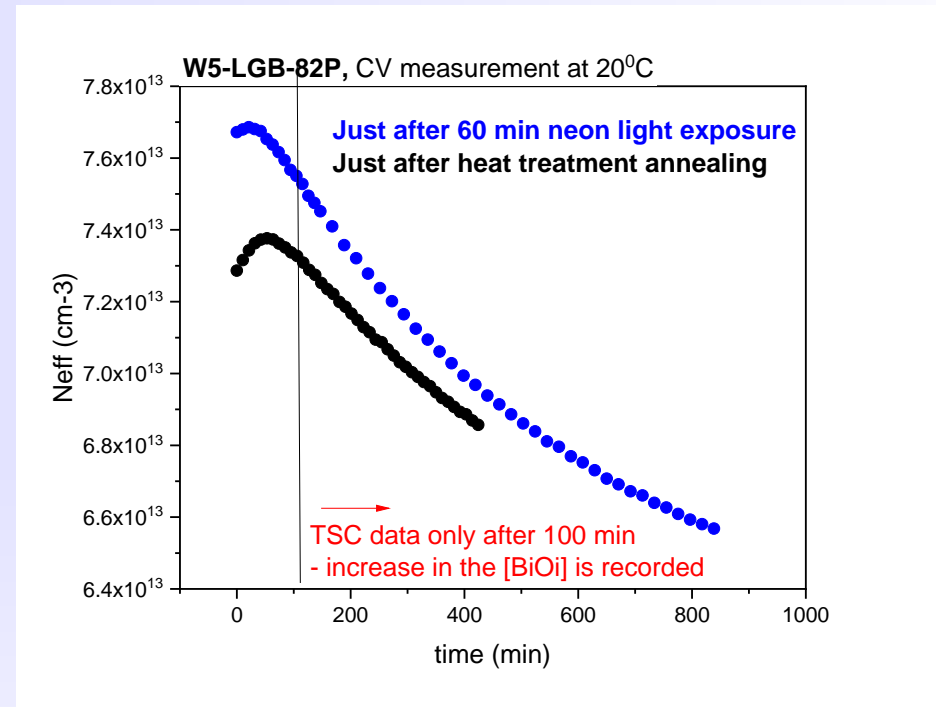
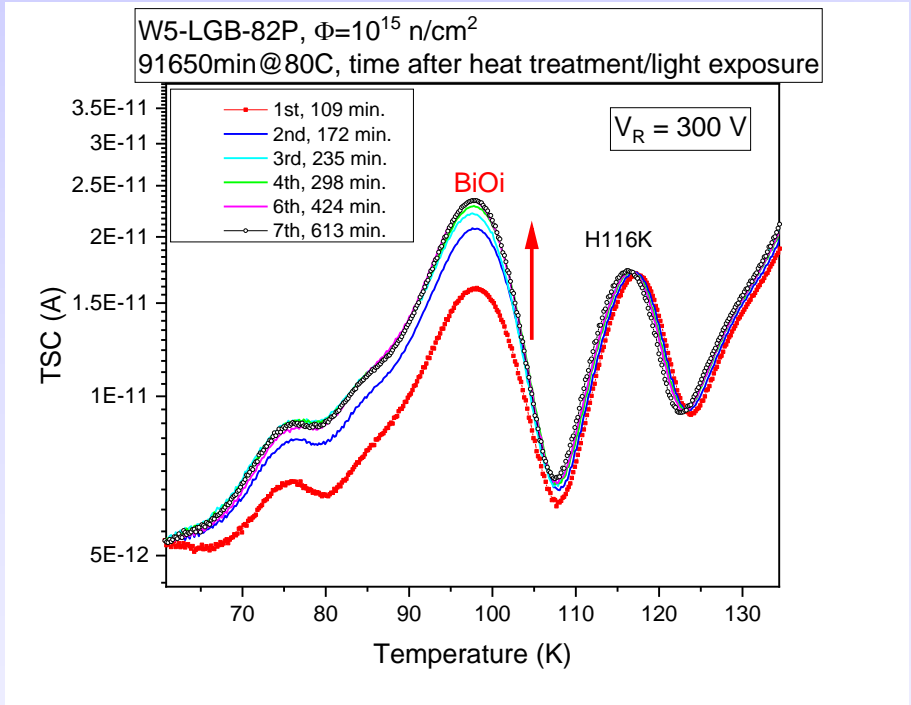


The increase in the BiOi concentration may cause the decrease in  $V_{dep}$

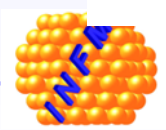


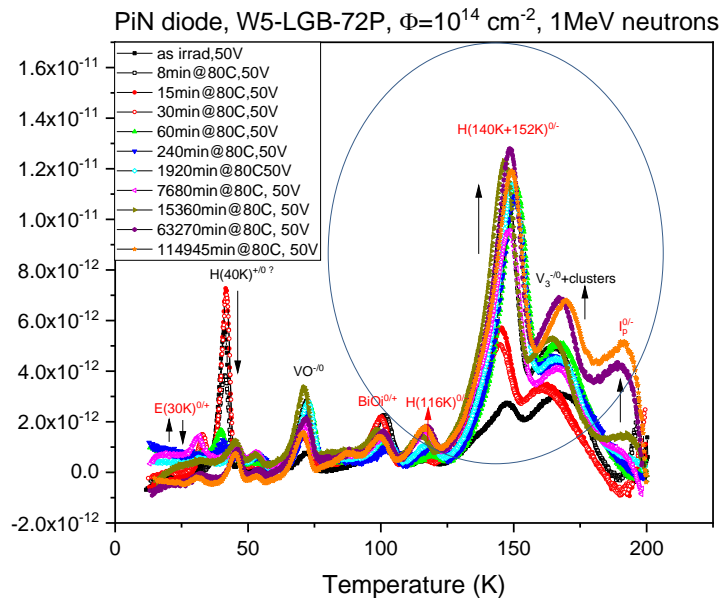
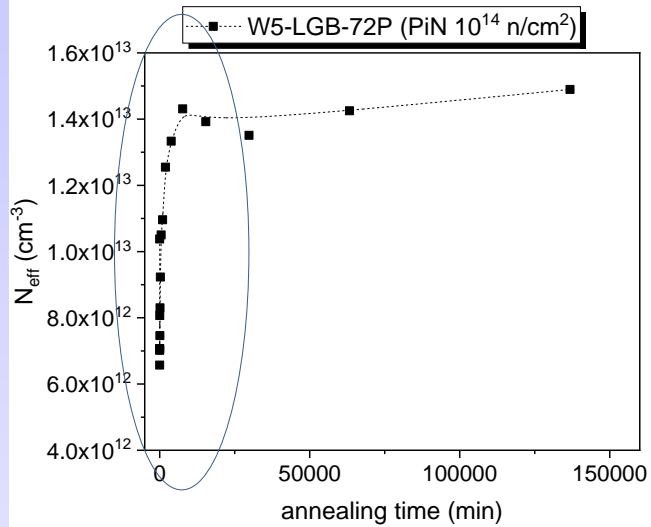
# Instability of the electrical activity of BiOi after annealing/ light exposure

n+  
p - type s  $\phi = 10^{15} \text{ cm}^{-2}$   
p+



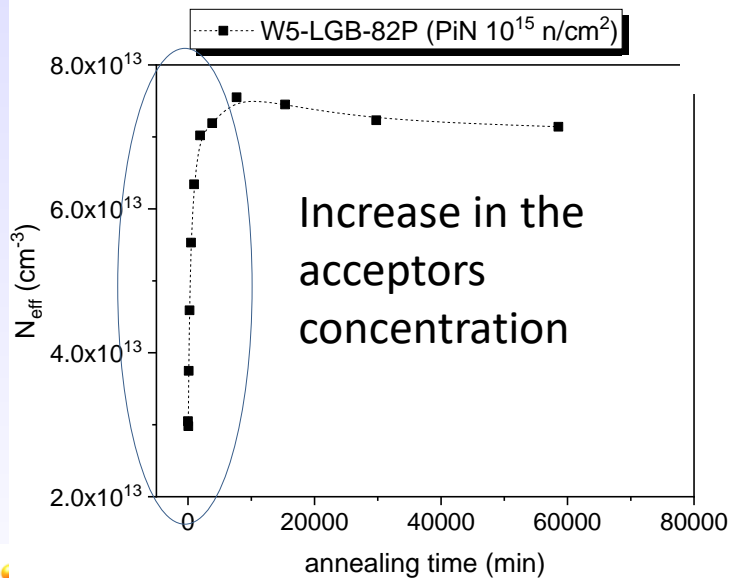
- The increase in BiOi concentration accounts for the variation in  $V_{dep}$
- the diode is not inverted ?





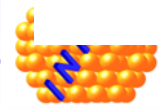
$\Phi = 10^{14} \text{ cm}^{-2}$  of high resistivity p-type STFZ Si

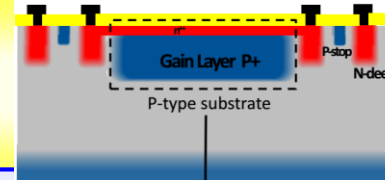
- **Generation of donors:** generation rate of BiO<sub>i</sub> is 0.03 cm<sup>-1</sup> (maximum) and of E(30K) donor is 0.017 cm<sup>-1</sup>. The amount of acceptor removal caused by BiO<sub>i</sub> is 6x10<sup>12</sup> cm<sup>-3</sup> (counted as twice the amount of BiO<sub>i</sub>) and of 1.7x10<sup>12</sup> cm<sup>-3</sup> by E(30K)
- **Generation of acceptors:** The amount of H(11K), H(140K+152K) acceptors created by irradiation ~1.3x10<sup>13</sup> cm<sup>-3</sup> (total generation rate of ~ 0.13 cm<sup>-1</sup>) / I<sub>p</sub> not counted



$\Phi = 10^{15} \text{ cm}^{-2}$  **Generation of donors:** generation rate of BiO<sub>i</sub> is 0.016 cm<sup>-1</sup> (maximum) and of E(30K) donor is 0.017 cm<sup>-1</sup>. The amount of acceptor removal caused by BiO<sub>i</sub> is 3.2x10<sup>13</sup> cm<sup>-3</sup> (counted as twice the amount of BiO<sub>i</sub>) and of 1.7x10<sup>13</sup> cm<sup>-3</sup> by E(30K)

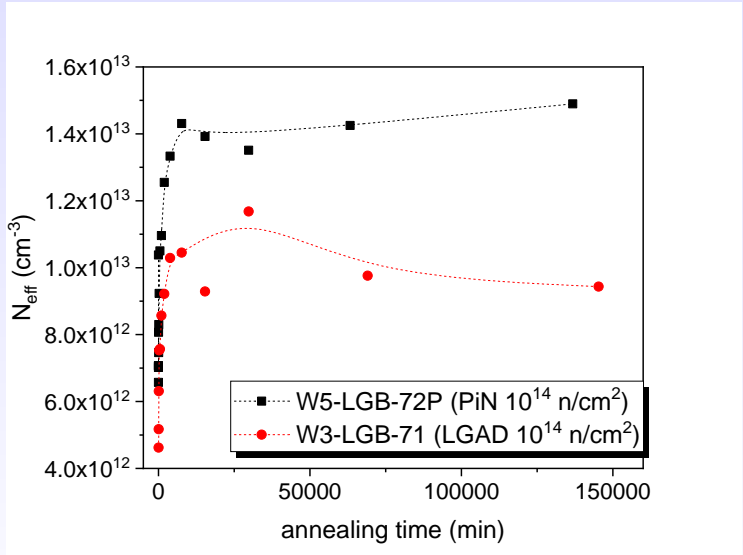
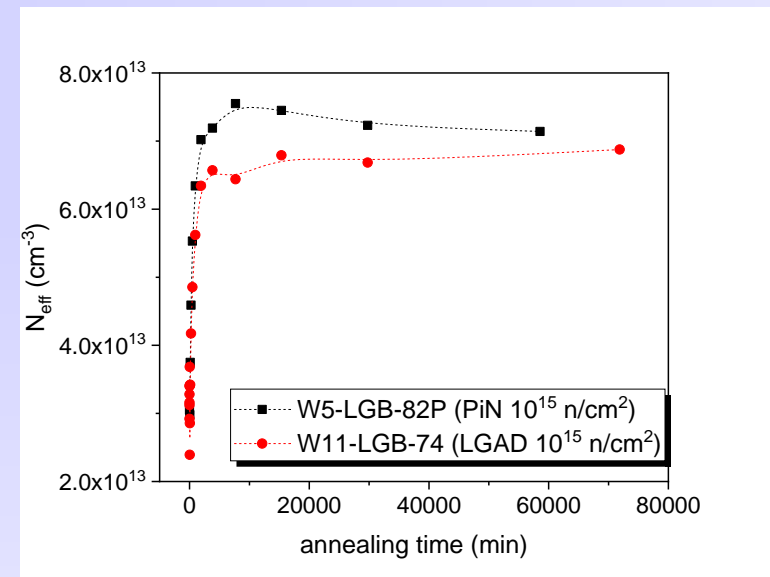
- **Generation of acceptors:** The amount of H(11K), H(140K+152K) acceptors created by irradiation 1.23x10<sup>14</sup> cm<sup>-3</sup> (total generation rate of ~ 0.123 cm<sup>-1</sup>); I<sub>p</sub> not counted



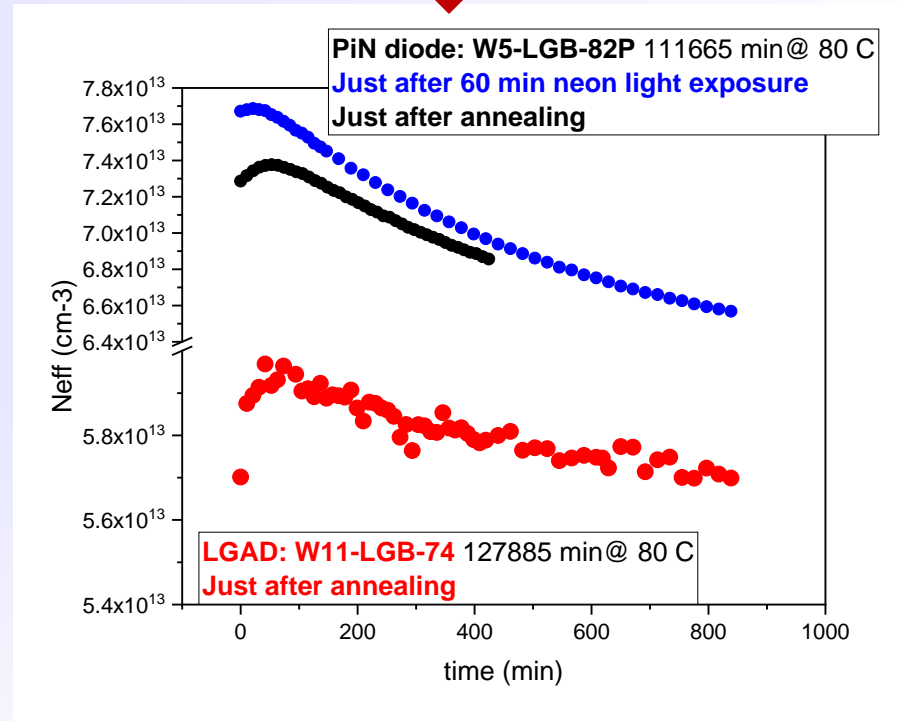


In the bulk of the LGADs

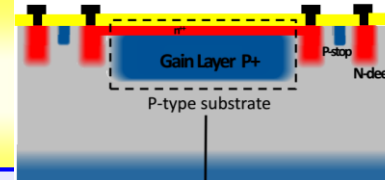
Dependence on the annealing time at 80 C



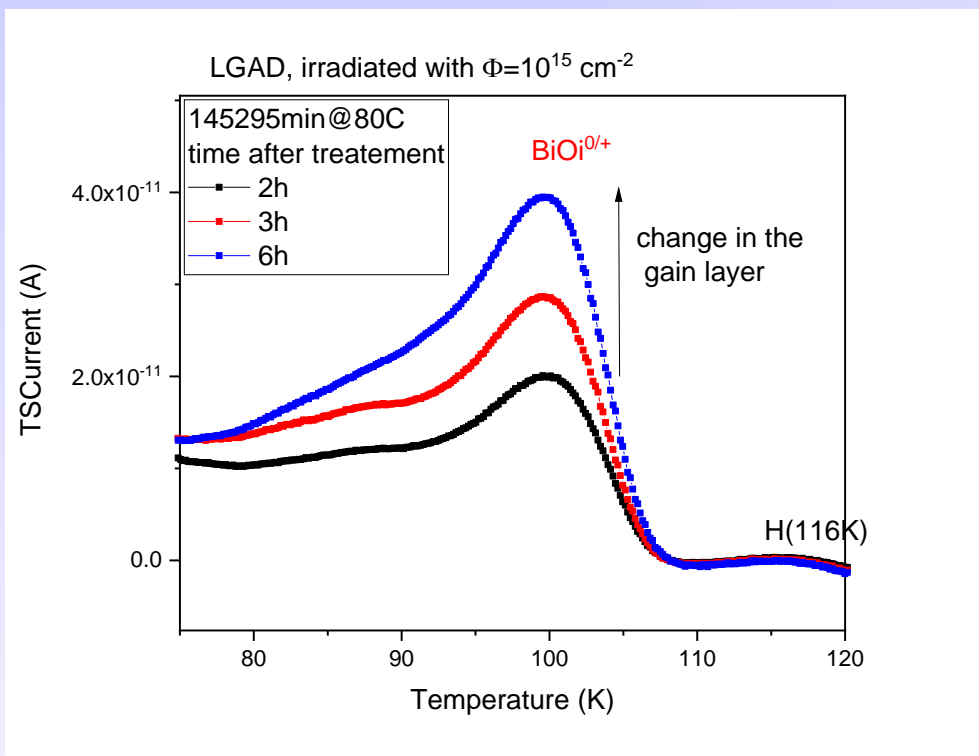
Dependence on the time after annealing at 80 C or exposure to light



Instability of BiOi in the bulk – reflected in  $N_{eff}$  variations (from CV)



In the gain layer – differs only the generation of BiOi



### Generation of BiOi donors:

#### $\Phi=10^{14} \text{ cm}^{-2}$ LGAD

- generation rate of BiOi is  $8.6 \text{ cm}^{-1}$ . The amount of acceptor removal caused by BiOi in the gain layer is  $1.72 \times 10^{15} \text{ cm}^{-3}$  (counted as twice the amount of BiOi)

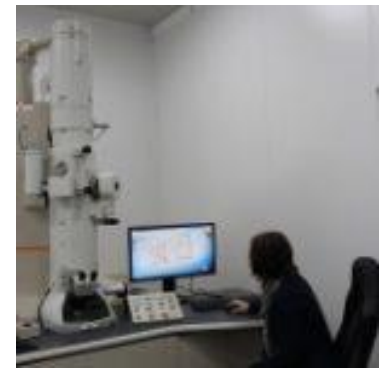
#### $\Phi=10^{15} \text{ cm}^{-2}$ LGAD

- generation rate of BiOi is  $\sim 8 \text{ cm}^{-1}$ . The amount of acceptor removal caused by BiOi in the gain layer is  $1.6 \times 10^{16} \text{ cm}^{-3}$  (counted as twice the amount of BiOi)

Similar instability of BiOi

# TEM on irradiated Silicon : current issues and future challenges

$$\phi=10^{15} \text{ cm}^{-2}$$



TEM perspective on the sample (after cross-sectional preparation):

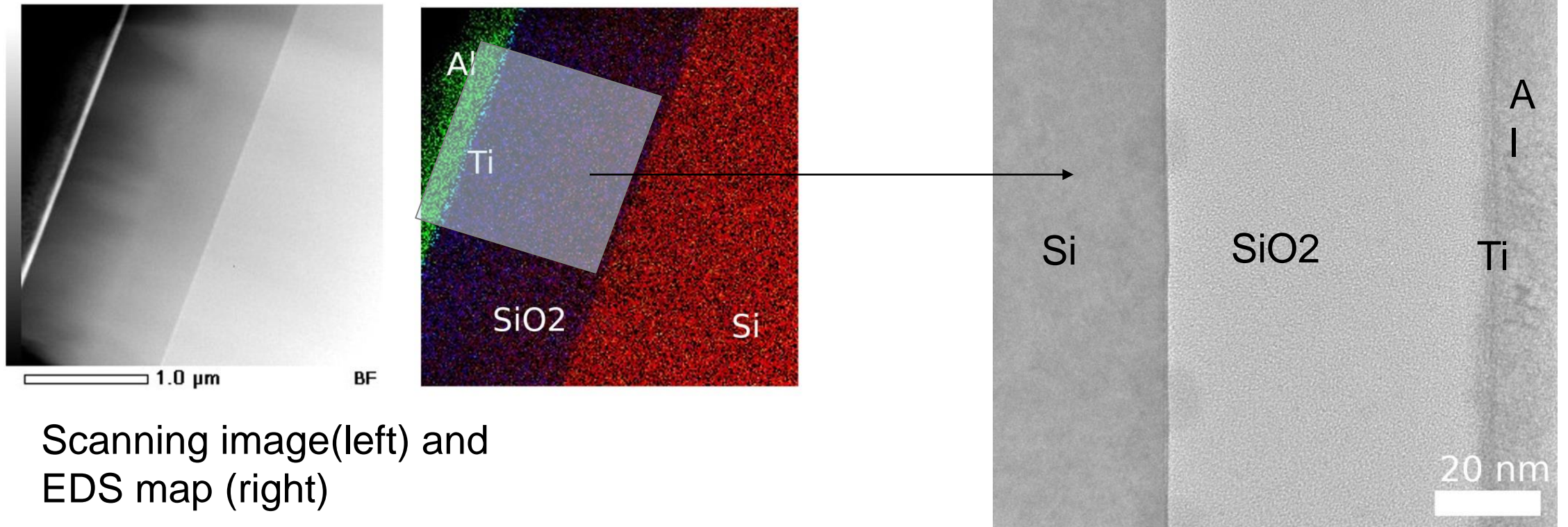
Al (~nm)
Ti (~nm)
SiO <sub>2</sub> (~nm)
Si (~nm)

Important: Prior to observations, sample has been treated for 80 min @ 160 C

## Transmission Electron Microscope JEOL2100

- Scanning Transmission Electron Microscopy (STEM) unit with Annular Dark Field and Bright Field detectors;
- Energy dispersive X-ray spectrometer (EDS),
- Movie-recording CCD camera
- Analytical holder with two tilting axis (double tilt);
- Holder with two tilting axis (double tilt) for in-situ heating of the sample;
- Tomography kit for TEM/STEM including electron tomography holder
- ASTAR crystallographic mapping unit
- Accelerating voltage range: 80kV-200kV
- LaB6 filament

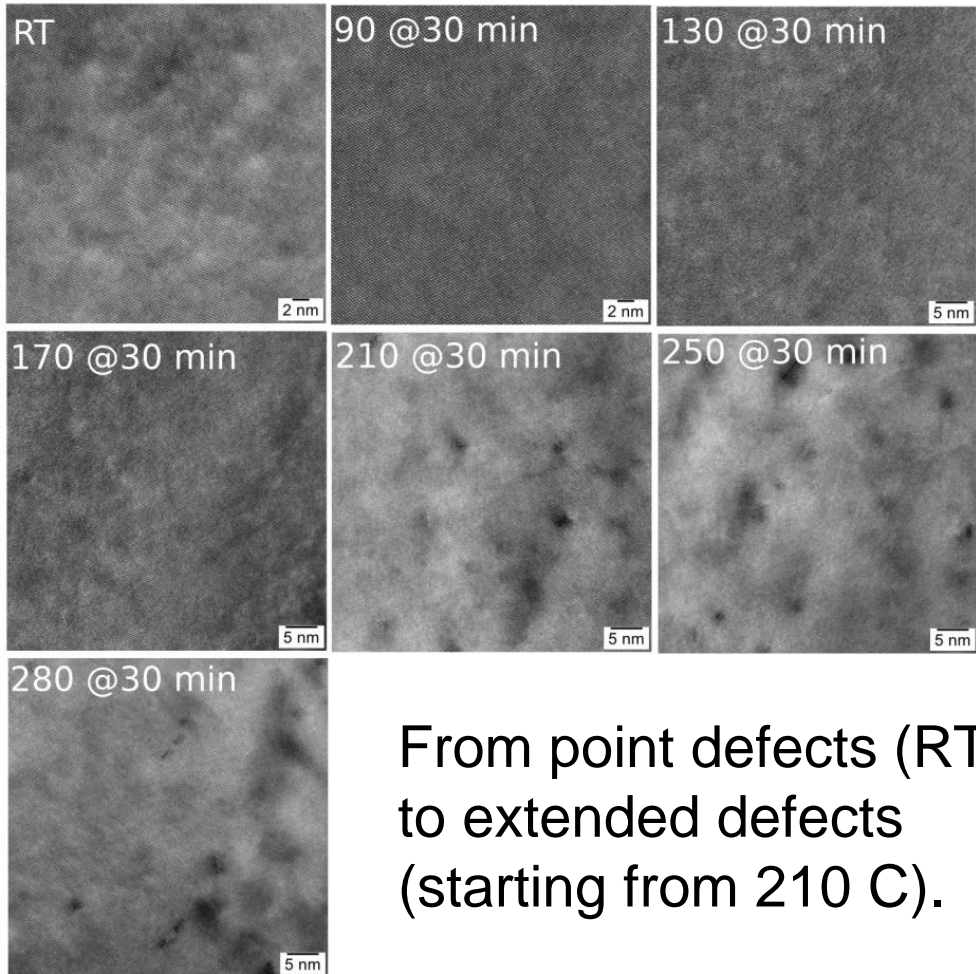
# Choosing area @200kV



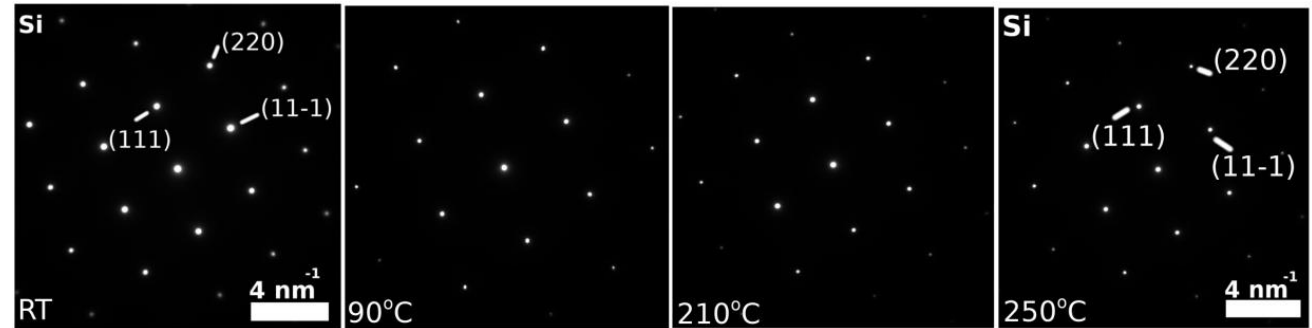
Scanning image(left) and  
EDS map (right)

# In situ evolution with temperature.

## Q&A: Reality or beam damage?



From point defects (RT) to extended defects (starting from 210 C).



No structural changes observable by Selected area electron diffraction – Si is still crystalline

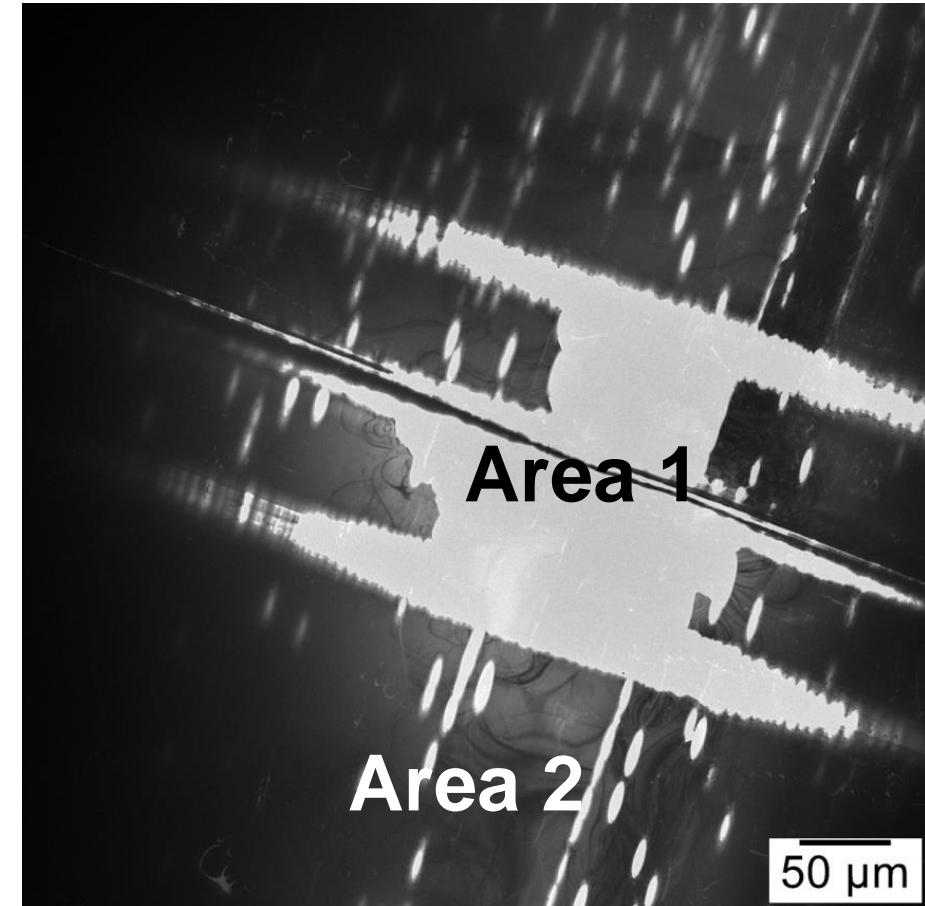
**On a fresh sample, electron beam has been focused on a clean silicon area. Beam damage has been observed within seconds (defect accumulation). Thus, the structural evolution @200kV is not conclusive !!!**

**Answer: Beam damage is significant @ 200kV.**

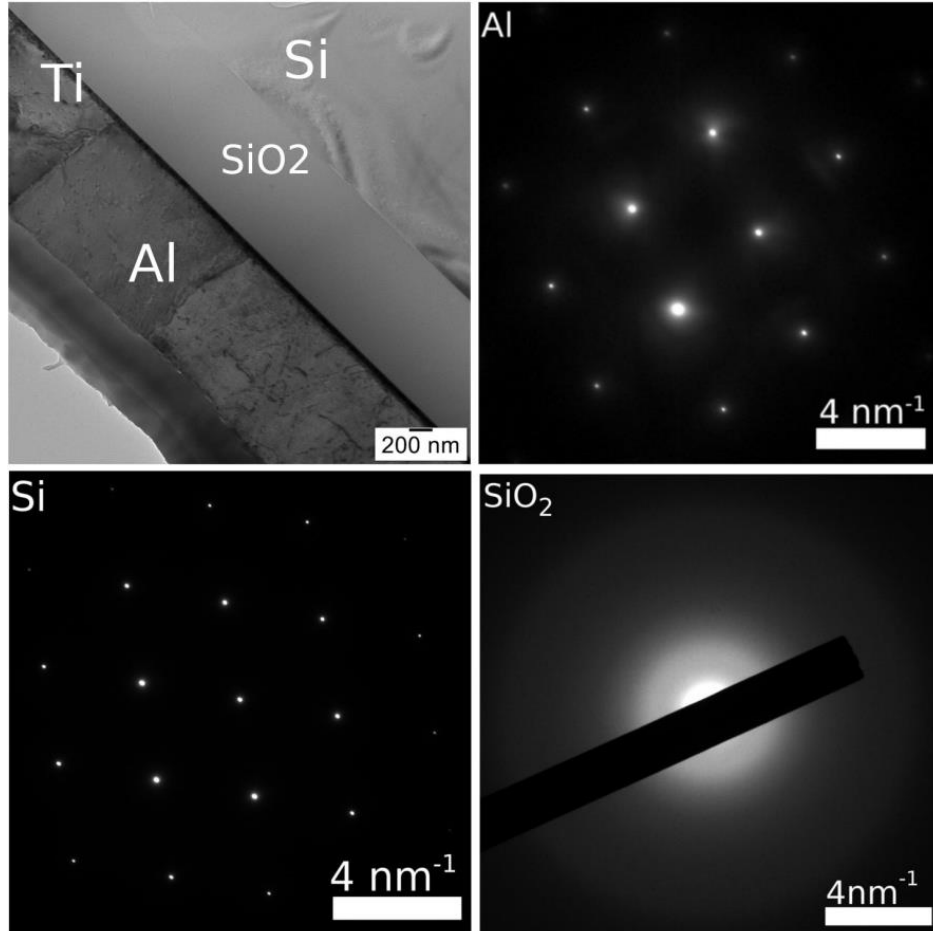


# Switching to 80kV accelerating voltage

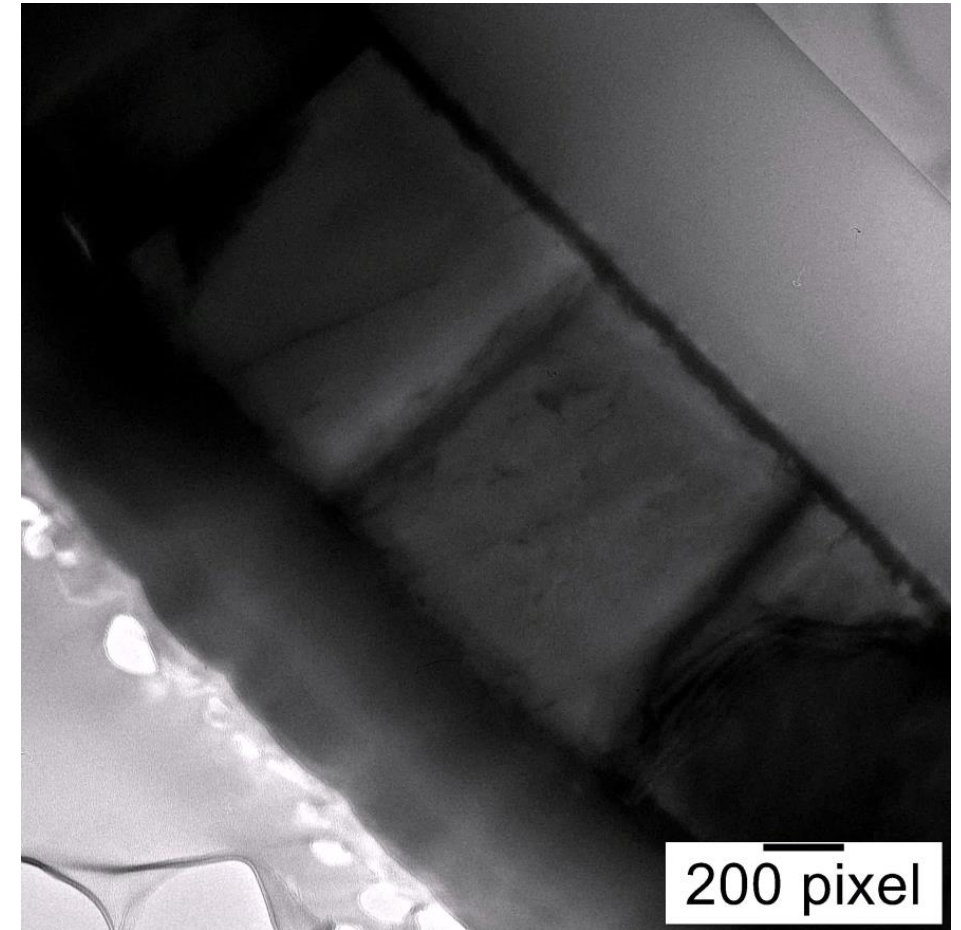
- No beam damage has been observed at 80kV.
- In-situ heating has been observed both at the Al-Ti-SiO<sub>2</sub> interface (area1) and deep inside the Si substrate (area 2, ~100  $\mu\text{m}$  depth)
- Sample has been heated for 30 min @ 150 C, then the temperature has been increased @250 and kept constant for another 30 min.



# At the surface (area1)



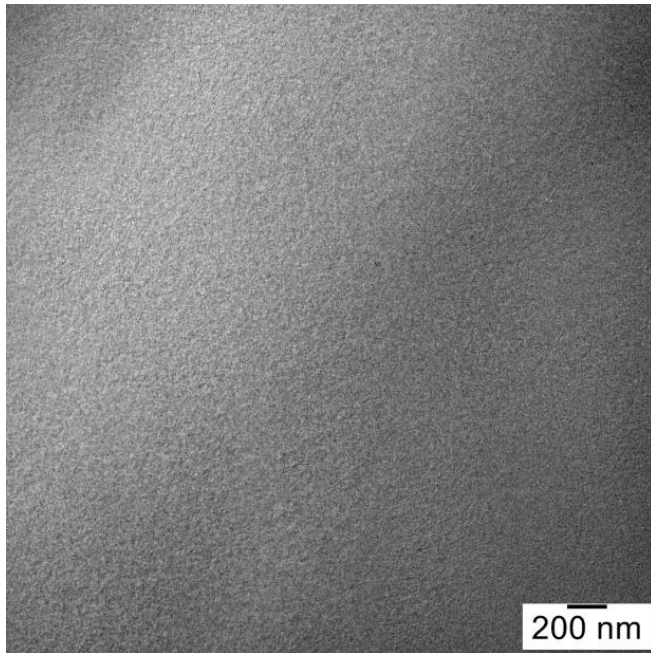
150C-250C  
transition



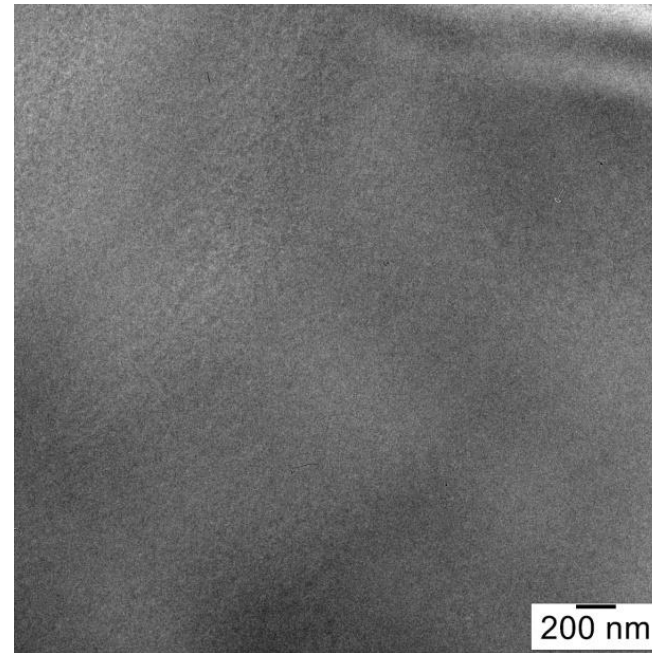
Extended defects (dislocations) dynamics in the Al coating as observed during 150-250 C heating.

# Deep (area2)

After 30 min @ 150



After 30 min @ 250



**Apart from lattice bending (sample is extremely thin, electron transparent), no defects have been observed.**

## **Future perspectives and challenges:**

1. Observe the cross-sectional sample at higher temperatures, for longer times
2. In-situ observations of the Silicon film at atomic resolution

- We have performed a long –term annealing at 80 C
- The investigated high resistivity PiN diodes are not inverted after a fluence of  $10^{15} \text{ cm}^{-2}$
- The variation of  $N_{\text{eff}}$  in these samples are due to the increase of acceptor concentration during the annealing rather than to a generation of donors.
- Due to the instability of the BiOi defect, it is difficult to precisely follow the annealing behavior, the CV measurements are usually performed just after heat treatment while the earliest TSC can be obtained after 2 h.
- TEM investigations could not reveal changes due to irradiation (too low the fluence)

## Further work:

- look for the reasons of BiOi instability and find a way to correlate with the CV results
- similar studies on different resistivity materials (higher fluences for TEM) and LGADs

Thank you for your attention !