

## The boron-oxygen ( $B_iO_i$ ) defect complex induced by irradiation with 23 GeV protons in p-type epitaxial silicon diodes

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- I. Motivation
- II. Experimental details
- III. Example of TSC spectra on  $B_iO_i$  and analysis method
- IV. Annealing properties of  $B_iO_i$
- V. Indication of the new defect  $\rightarrow$  X-defect
- VI. Analysis of the X-defect
- VII. Introduction rate of  $B_iO_i$
- VIII. Summary and outlook

Radiation damages of LGADs  
(Low Gain Avalanche Detectors) [1]:

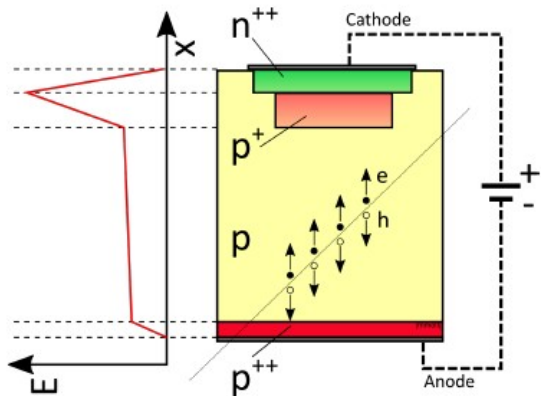


Fig 1. Schematic of LGAD

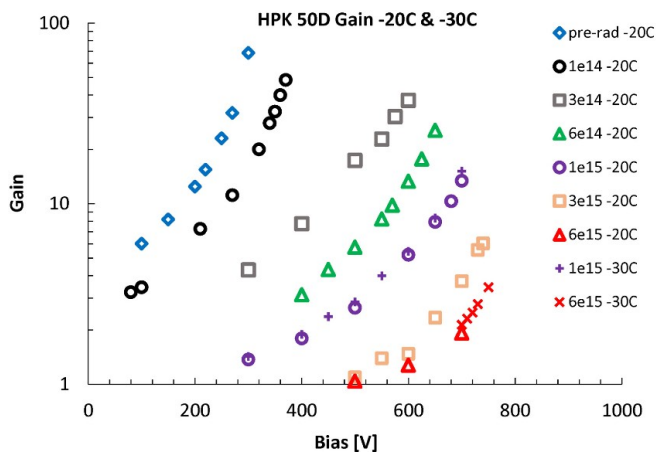


Fig 2. Gain value vs. bias for different fluence value

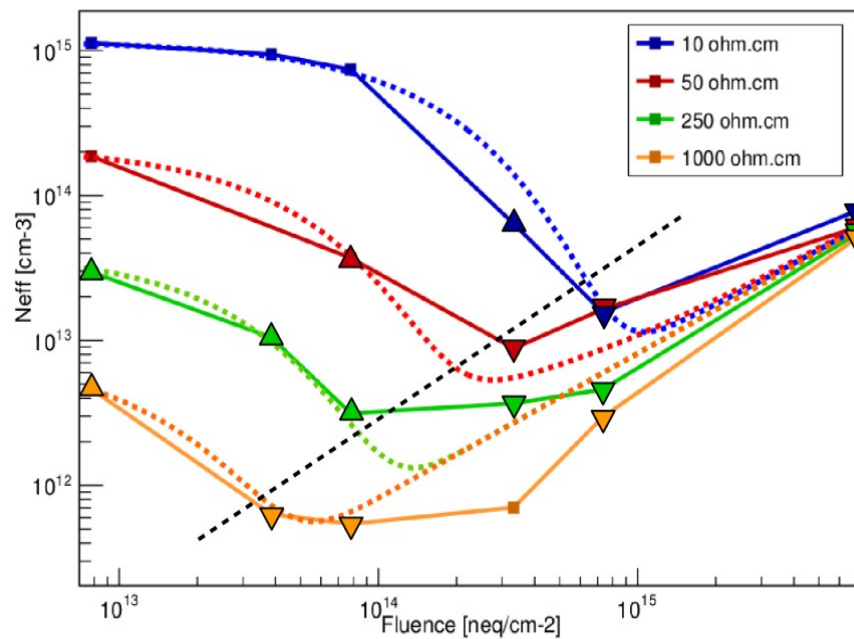


Fig 3.  $N_{eff}$  vs. fluence for different initial doping concentration

Radiation damage of p-type diodes is dominated by acceptor removal in the beginning and afterwards by acceptor generation.[2]

**$B^-$  turn to  $B_iO_i^+$**

Change in  $N_{eff}$  is a factor of 2

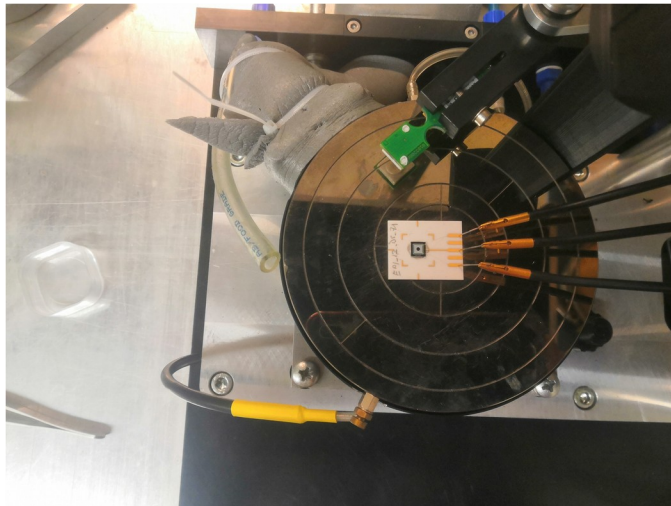
[1] Kramerberger, G., et al. "Radiation effects in Low Gain Avalanche Detectors after hadron irradiations." Journal of Instrumentation 10.07 (2015): P07006.

[2] Y. Gurimskaya, 31st RD50 Workshop, 20-22 of November, 2017, CERN, Geneva, Switzerland.

## Information of measured epitaxial silicon diodes (PIN)

Label	EPI50P_01_DS_73	EPI50P_06_DS_71	EPI50P_09_DS_73	EPI50P_12_DS_74
Resistivity	10 Ωcm	50 Ωcm	250 Ωcm	2000 Ωcm
Irradiation	23 GeV proton, $\Phi_{eq} = 4.28E13 \text{ cm}^{-2}$			
Area	6.927E-2 cm <sup>2</sup>			
Thickness	50 μm			

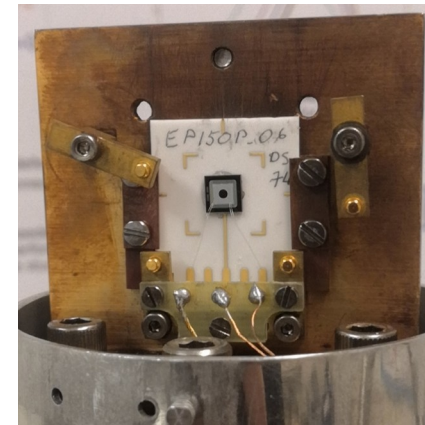
C-V, I-V:



Experimental parameter (C-V, I-V):

Temperature: 20 °C  
 Humidity: below 10%  
 Frequency for C-V: 230 Hz, 455 Hz, 1k Hz, 10k Hz  
 AC voltage for C-V: 0.5 V

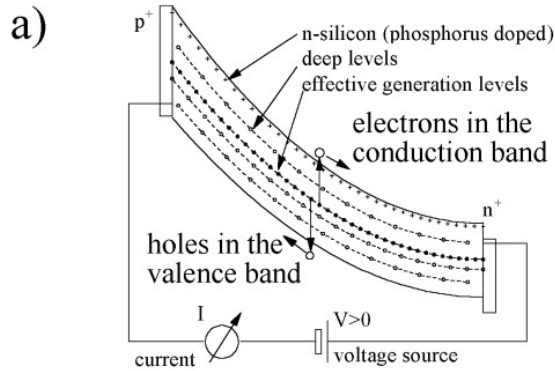
Thermally stimulated current-TSC:



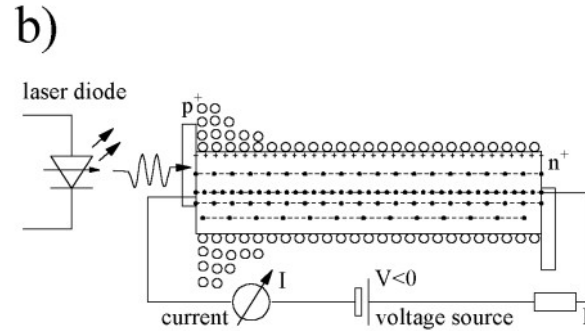
Experimental parameter (TSC):

Cooling down bias: 0 V  
 Filling temperature: typical 10 K  
 Filling: Forward bias filling, 0 V filling or light injection  
 Filling time: 30 s  
 Delay time: 30 s  
 Heating rate: 0.183 K/s  
 Heating up bias: depend on effective space charge concentration after irradiation

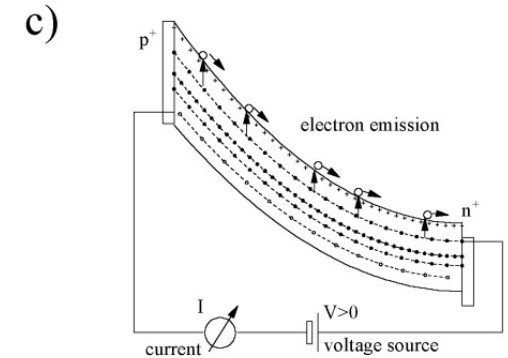
Basic Principle of Thermally Stimulated Current-(TSC) [2]:



**a) Cooling:**  
Room temperature → filling temperature



**b) Injection:**  
Forward, 0 V or light injection



**c) Heating up:**  
Reverse biased diodes, constant heating rate

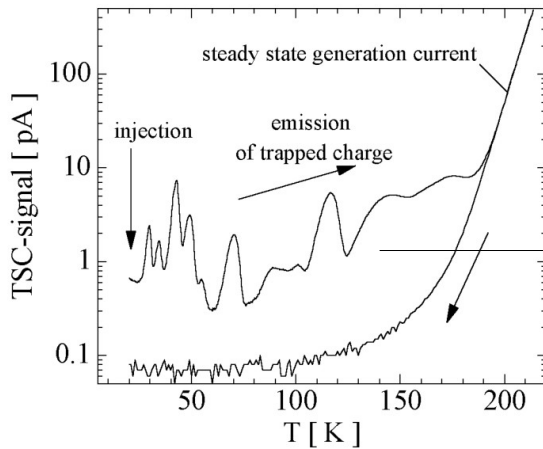


Fig 4. TSC spectrum [2]

$$I_{tsc} = \frac{1}{2} q_0 A d(T) N_t e_n \exp\left(-\frac{1}{\beta} \int e_n(T) dT\right)$$

$$e_n = \sigma_n v_{th,n} N_c \times \exp\left(\frac{-E_a}{K_b T}\right)$$

$$E_a = E_C - E_T \quad [1]$$

$N_t$  is defect concentration;  $\beta$  is heating rate;  $\sigma_n$  is capture cross section;  $E_a$  is activation energy;  $A$  is diodes area;  $d(T)$  depleted thickness;

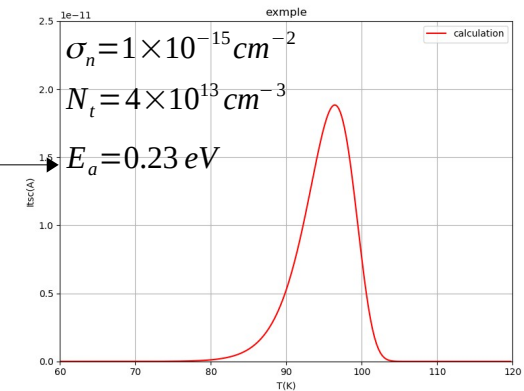


Fig 5. Example of calculated TSC peak

[1] Buehler, M. G. Solid-State Electronics 15.1 (1972): 69-79.

[2] Moll, Michael. No. DESY-THESIS-1999-040. DESY, 1999.

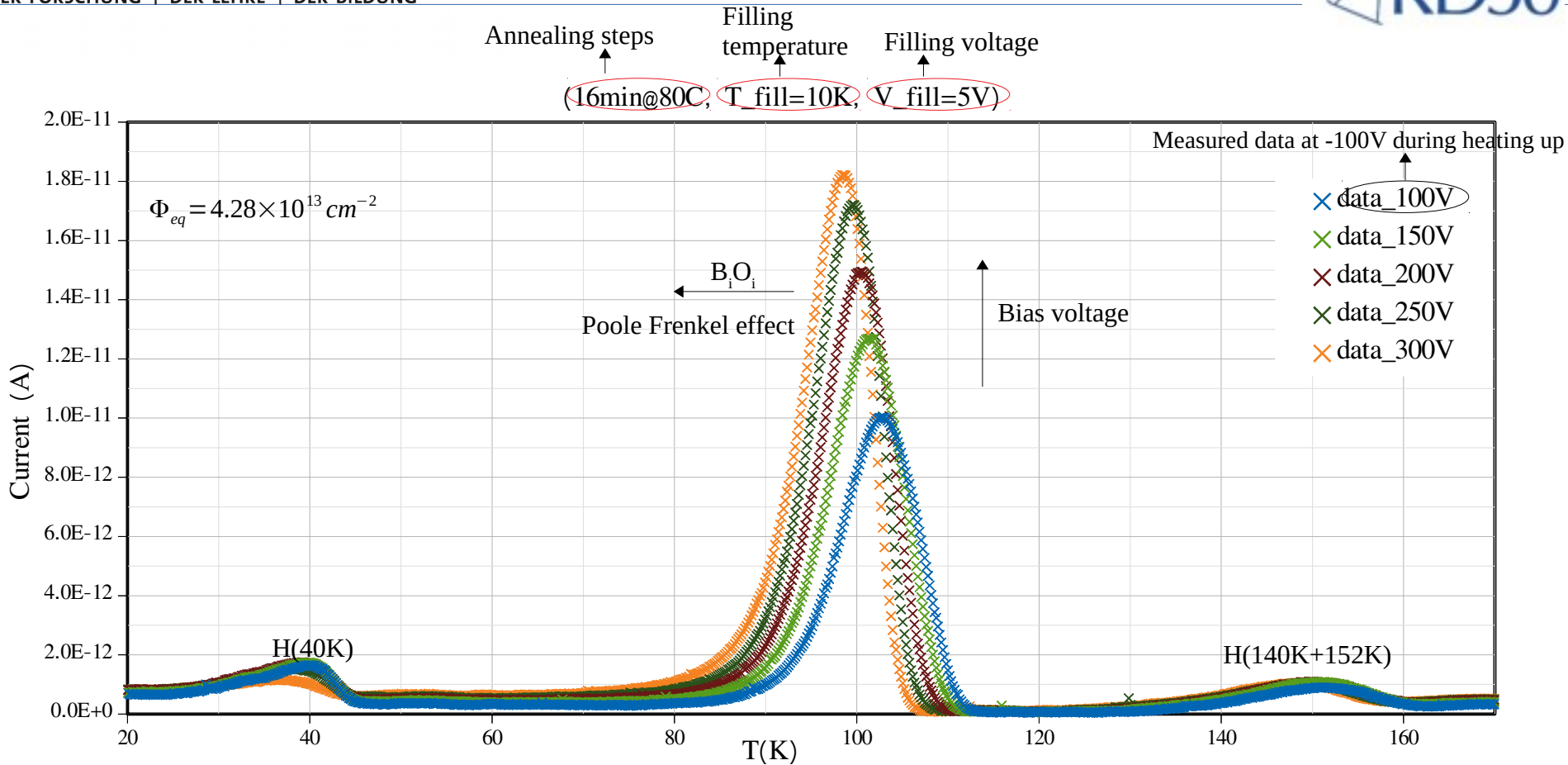


Fig 6. TSC spectra for different bias voltages of 50  $\Omega$ cm diode after 23 GeV proton irradiation

- Dominant  $B_iO_i$  signal
- Shift of peak maximum with  $V_{bias} \rightarrow$  Poole-Frenkel effect; electron trap  $B_iO_i$  (o/+) donor defect
- Peak amplitude increases with bias voltage due to increasing depletion depth and after full depletion extending into the p+ region

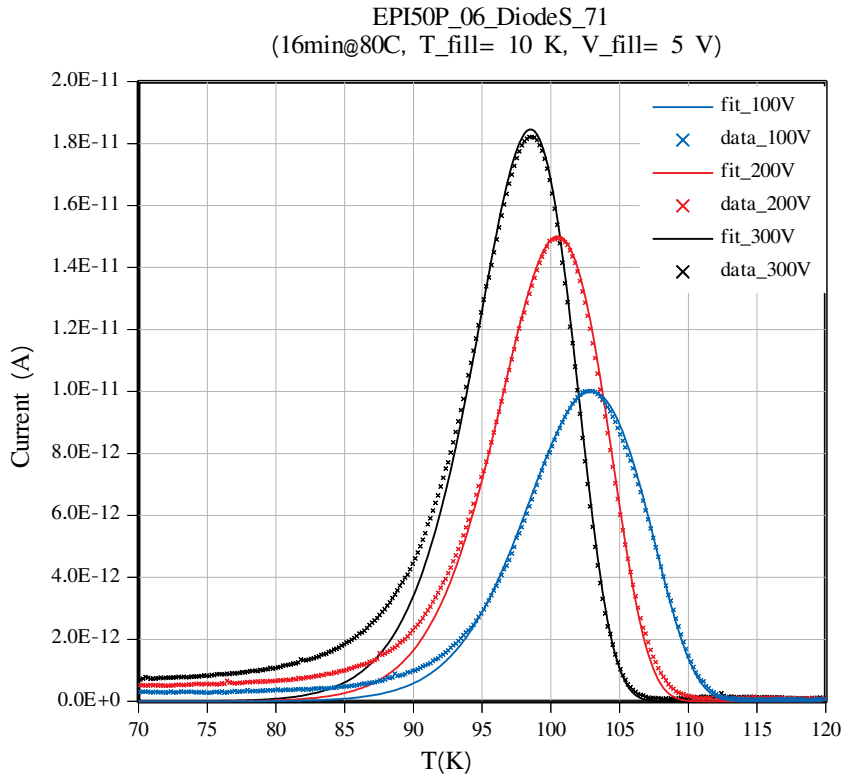


Fig 7. Results of fitting TSC Peak

Analysis method (3-D Poole Frankel effect) [1-3]:

$$I_{tsc} = \int_0^d \frac{1}{2} q_0 A x N_t e_n(T, x) \exp\left(-\frac{1}{\beta} \int e_n(T, x) dT\right) dx$$

$$e_n = \sigma_n v_{th,n} N_c \times \exp\left(\frac{-E_{a0}}{k_B T}\right) \left[\left(\frac{1}{\gamma^2}\right) (e^\gamma (\gamma - 1) + 1) + \frac{1}{2}\right]$$

$$\gamma = \left(\frac{q_0 E(x)}{\pi \epsilon_0 \epsilon_r}\right)^{\frac{1}{2}} \times \frac{q_0}{k_B T}$$

$q_0$ : elementary charge; A: area; d: depleted thickness;  $N_t$ : defect concentration;  $e_n$ : emission rate; T: temperature; x: position;  $\beta$ : heating rate;  $\sigma_n$ : electron captured cross section;  $v_{th,n}$ : thermal velocity of electron;  $N_c$ : density of state in the conduction band;  $E_{a0}$ : zero field activation energy;  $k_B$ : Boltzmann constant;  $E(x)$ : electric field.

[1] Buehler, M. G. Solid-State Electronics 15.1 (1972): 69-79.

[2] J. L. Hartke, J. Appl. Phys. 39, 4871 (1968).

[3] Pintilie, I., E. Fretwurst, and G. Lindström. Applied Physics Letters 92.2 (2008): 024101.

## Isothermal annealing

( $T_{\text{filling}}=10\text{ K}$ ,  $V_{\text{filling}}=5\text{ V}$ ,  $V_{\text{bias}}=-300\text{ V}$ )

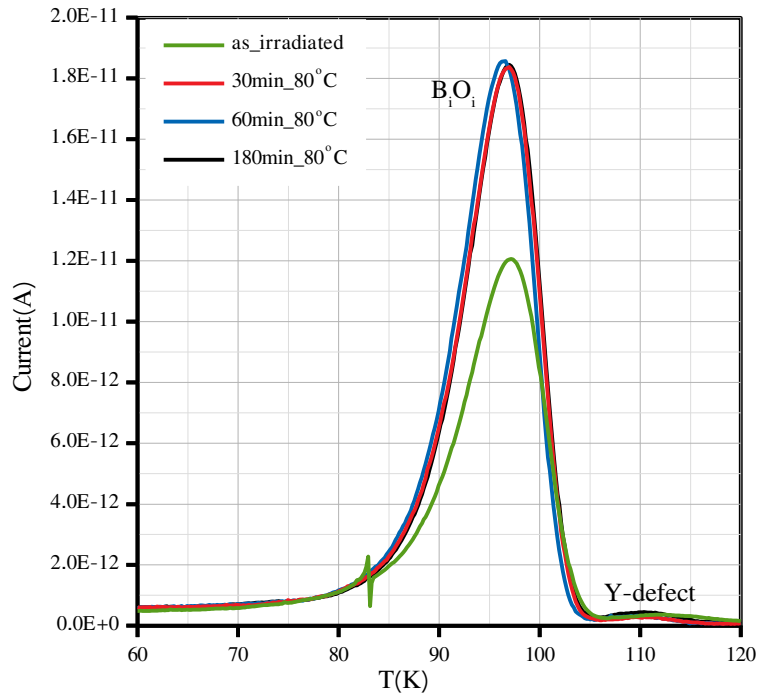


Fig 8. TSC for isothermal annealing

## Isochronal annealing

( $T_{\text{filling}}=10\text{ K}$ ,  $V_{\text{filling}}=5\text{ V}$ ,  $V_{\text{bias}}=-300\text{ V}$ )

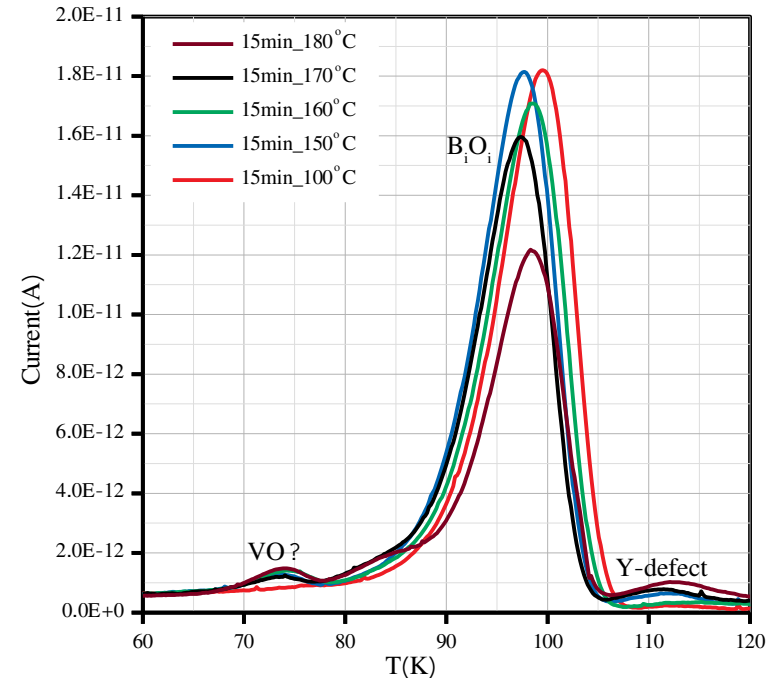


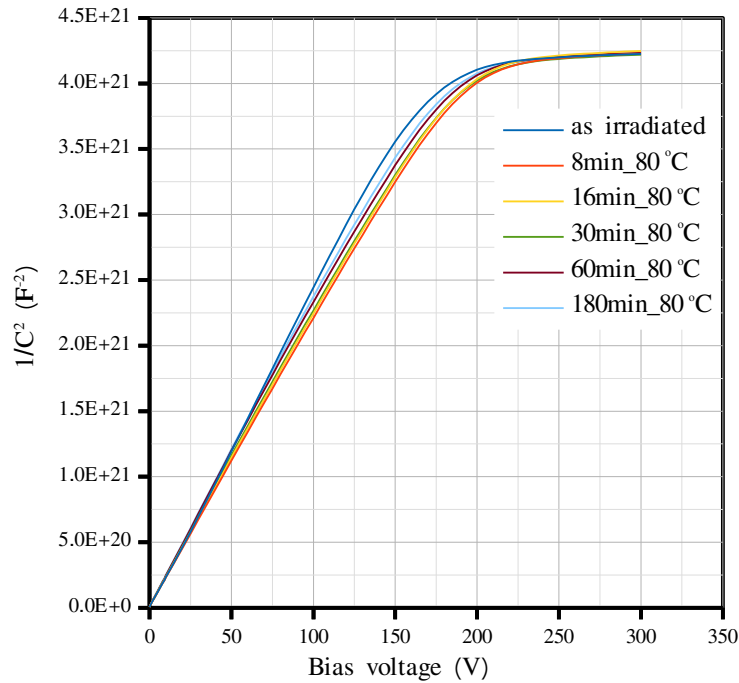
Fig 9. TSC for isochronal annealing

- $B_iO_i$  increase after first annealing step and turn to stable during all annealing steps.

- $B_iO_i$  is constant up to an annealing temperature of  $T_{\text{ann}} = 150\text{ }^\circ\text{C}$ ;
- For  $T_{\text{ann}} > 150\text{ }^\circ\text{C}$ , [ $B_iO_i$ ] decrease;
- Peak temperature slightly changes (unknown reason).
- Peak near  $B_iO_i$  at lower temperature range start to appear, possibly it is VO (didn't show Poole Frenkel effect).



Isothermal annealing



T=20 °C  
 Freq: 10 kHz  
 AC:0.5 V  
 $\Phi_{eq} = 4.28 \times 10^{13} \text{ cm}^{-2}$

Isochronal annealing

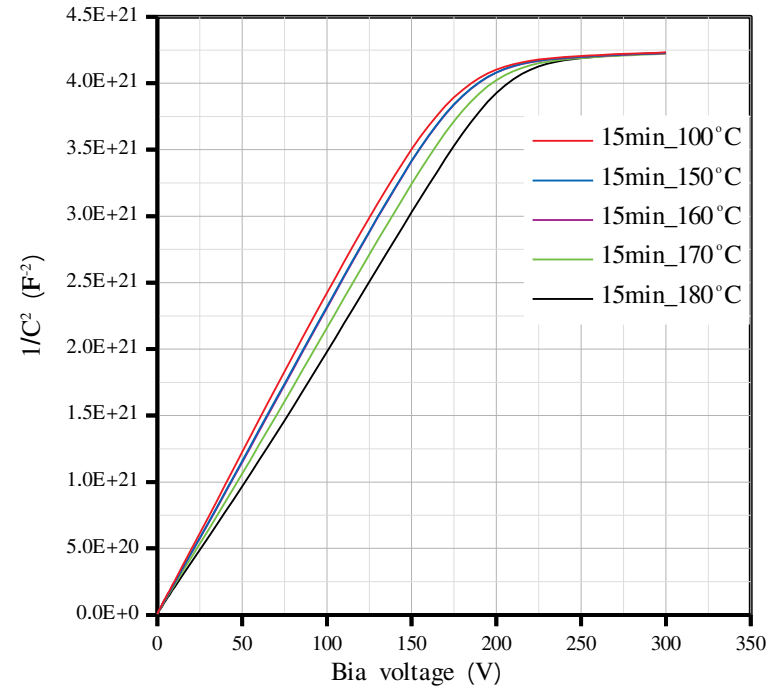


Fig 10. C-V for isothermal annealing at 80 °C up to 180 min

Fig 11. C-V curve for isochronal annealing steps (100 °C~ 180 °C)

Changes of capacitance are only small changes for different annealing steps due to most of shallow defects being nearly stable after annealing below 160 °C.

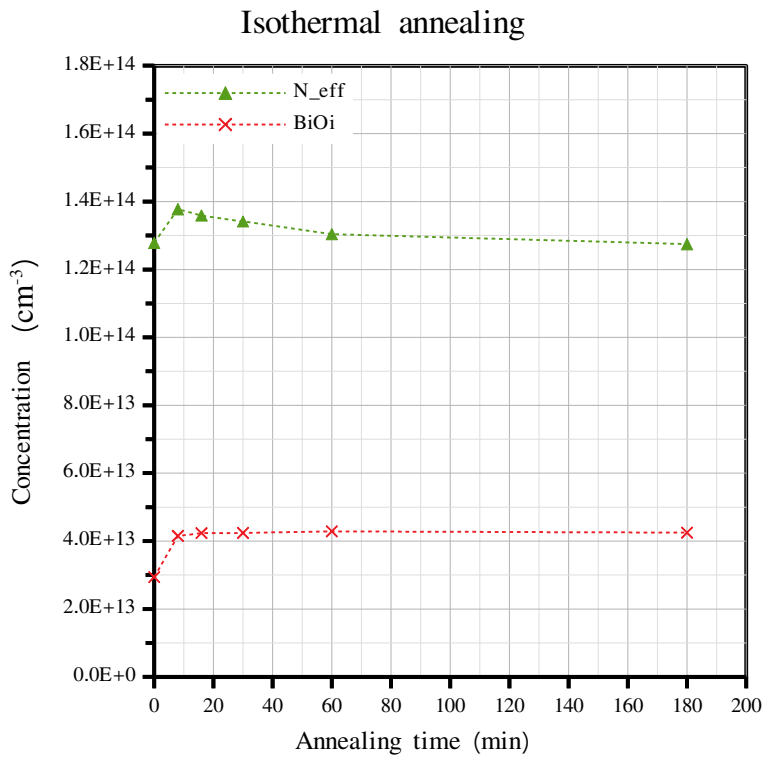


Fig 12. Concentration of  $B_iO_i$  and  $N_{eff}$  vs annealing time for isothermal annealing

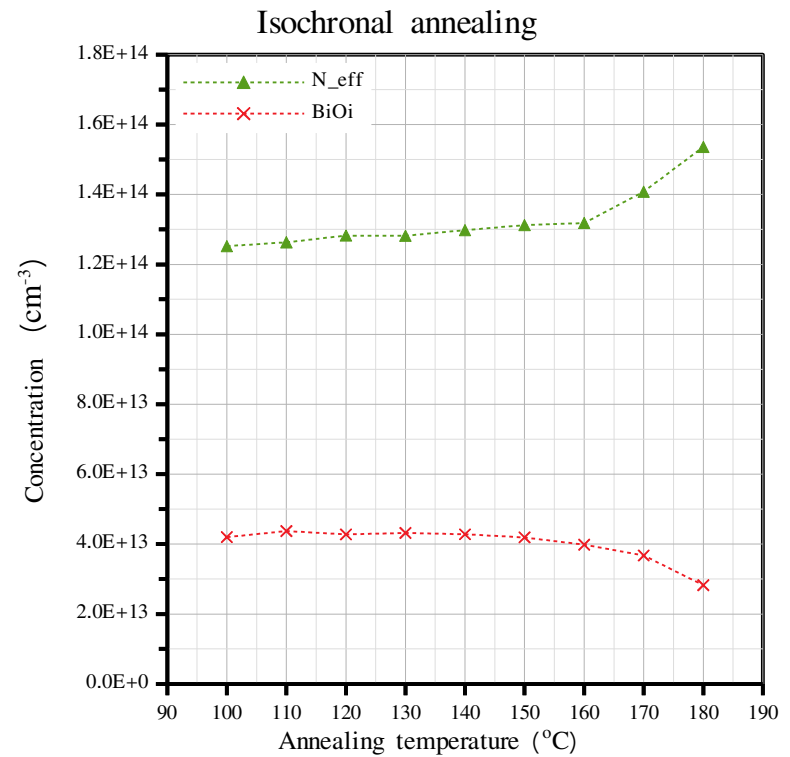


Fig 13. Concentration of  $B_iO_i$  and  $N_{eff}$  vs annealing temperature for isochronal annealing

$$N_{eff}(C-V) = N_s + 2 \times [B_iO_i] \pm others \quad N_s = -[B_s]$$

- $N_{eff}$  is given by C-V measurement;
- For isothermal annealing the change of  $N_{eff}$  is affected by defects ( $B_iO_i$  is constant);
- For isochronal annealing the change of  $N_{eff}$  is mainly affected by  $B_iO_i$ . And  $N_{eff, 180} - N_{eff, 170} = 2 \times ([B_iO_i]_{170} - [B_iO_i]_{180})$ .

16min@80C,  $V_{filling}=5$  V,  $T_{fill}=60$  K,  $V_{bias}=-250$  V

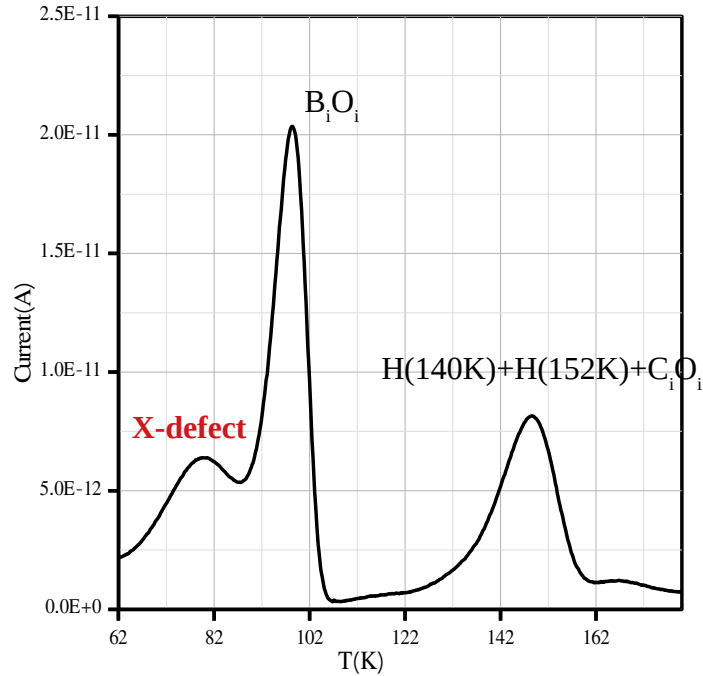


Fig 14. TSC spectrum of 50 Ωcm p-type diodes after filling at  $T_{fill} = 60$  K with forwards bias  $V_{fill} = 5$  V.

$$p_t = \frac{N_t}{\left(1 + \frac{c_n}{c_p}\right)}$$

60min@80C,  $V_{fill}=0$  V,  $T_{fill}=60$  K,  $V_{bias}=-250$  V

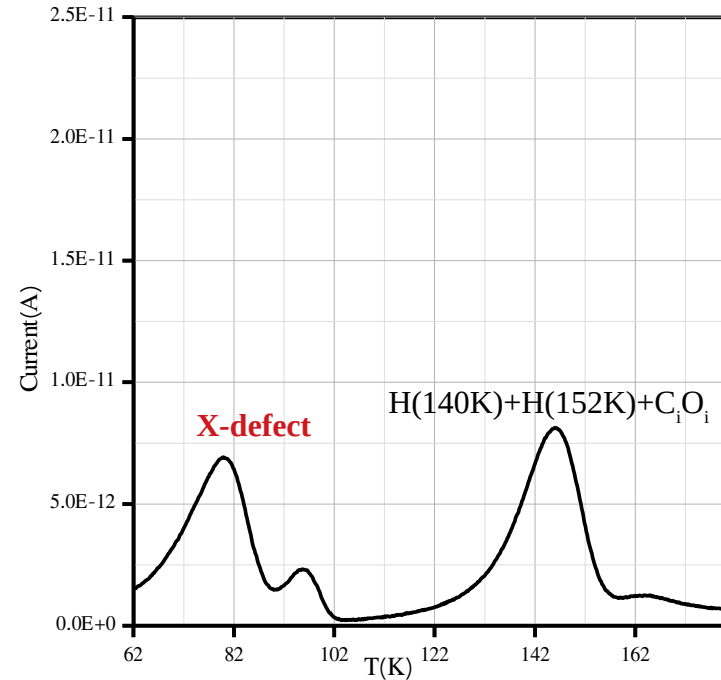


Fig 15. TSC spectrum for 0 V filling (majority carriers filling) and  $T_{fill} = 60$  K

$$p_t = N_t (1 - \exp(-c_p \cdot p \cdot t_{filling}))$$

Compare two procedure:

- The capture cross section of X-defect is strongly temperature dependent
- X-defect is hole trap.

60min@80C, V\_fill=0V, T\_fill=10K

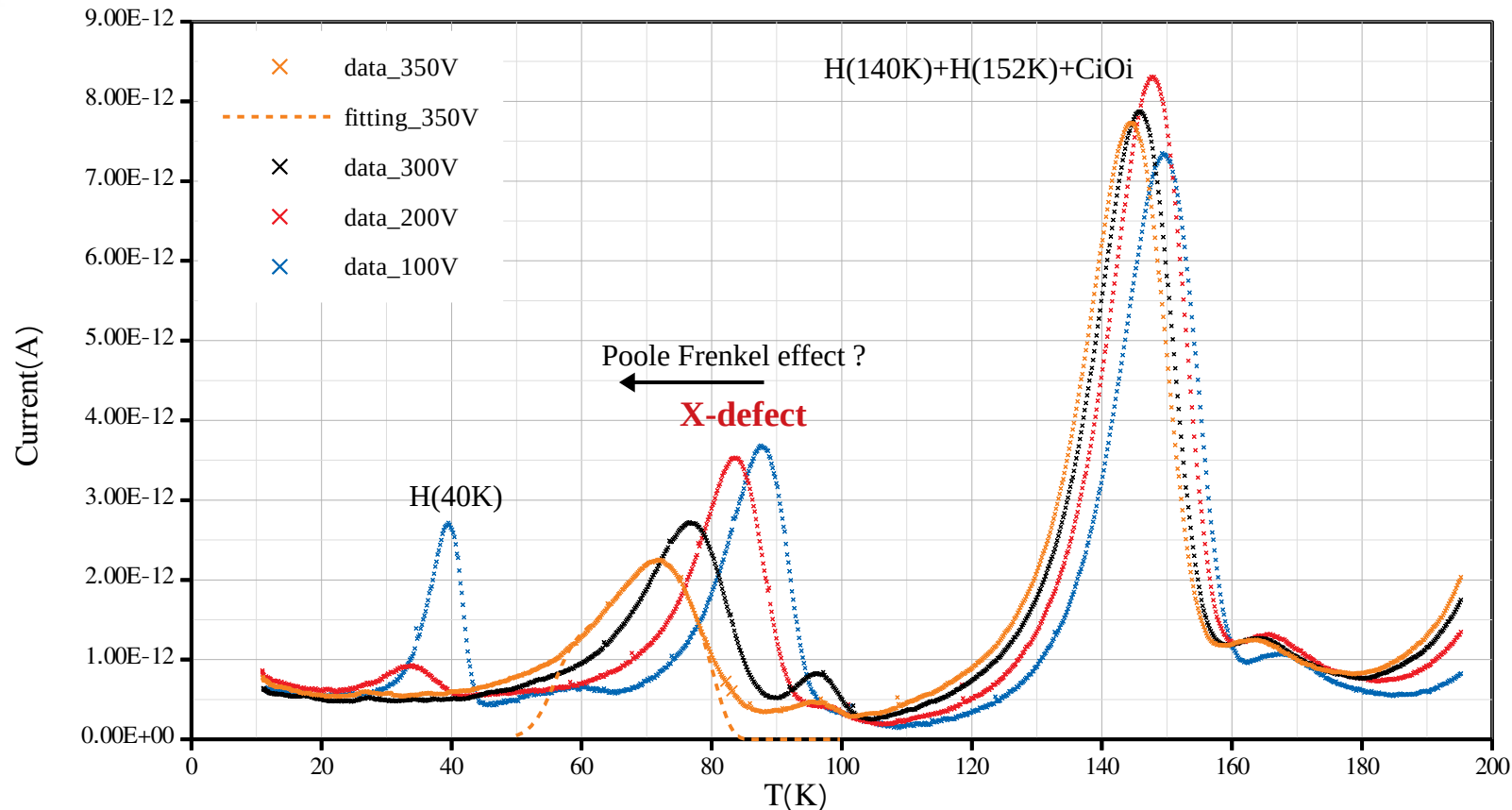


Fig 16. TSC spectra for different bias

- Special procedure: zero volt filling, only holes can be captured
- X-defect: hole trap, PF-effect, shallow acceptor → X(o/-), i.e. neutral if filled; T-shift very large ~ 15K for  $V_{bias} = 100\sim 350$  V. Long tail at higher bias voltage.
- H(140K)+H(152K)+C<sub>i</sub>O<sub>i</sub>:  $E_t \approx E_v + 0.42\text{eV}$

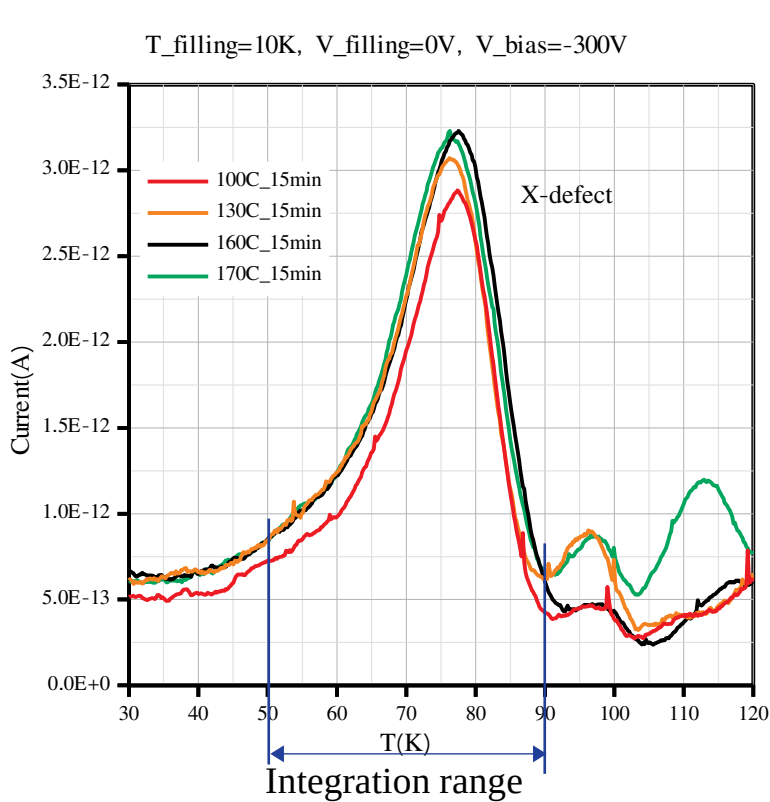


Fig 17. TSC spectra for 0 V filling and for different isochronal annealing steps

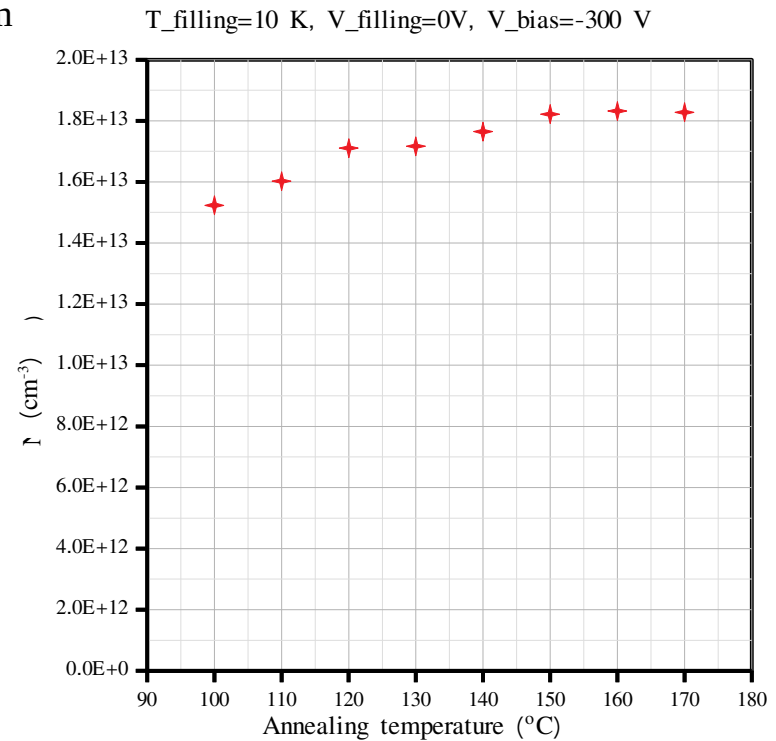


Fig 18. Defect concentration extracted from Fig 17 vs annealing temperature

$$N_t = \frac{2 \int I dt}{q_0 A d(C)}$$

Concentration of X-defect slightly increases with annealing temperature

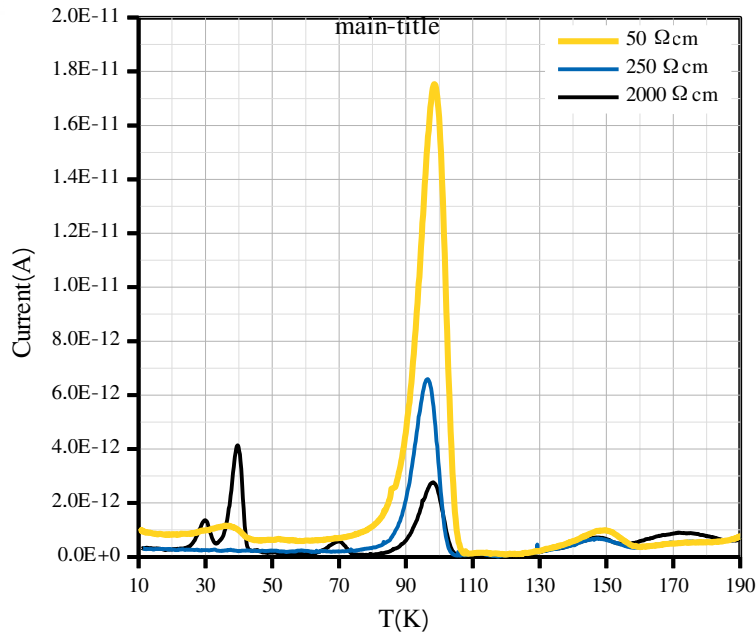


Fig 19. TSC spectra for diodes with different resistivity 50  $\Omega$ cm, 250  $\Omega$ cm and 2000  $\Omega$ cm

$\phi_{eq} = 4.28 \times 10^{13} \text{ cm}^{-2}$   
8min@80°C

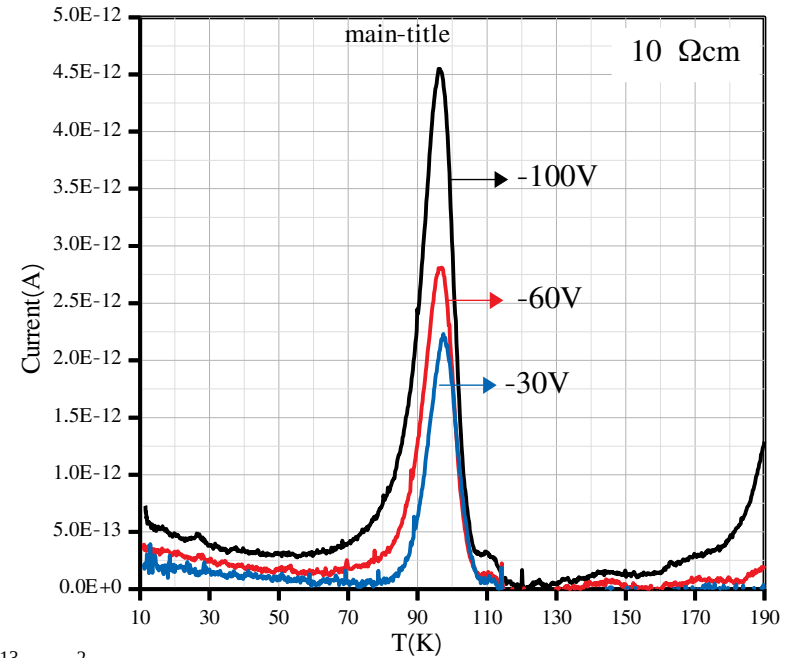
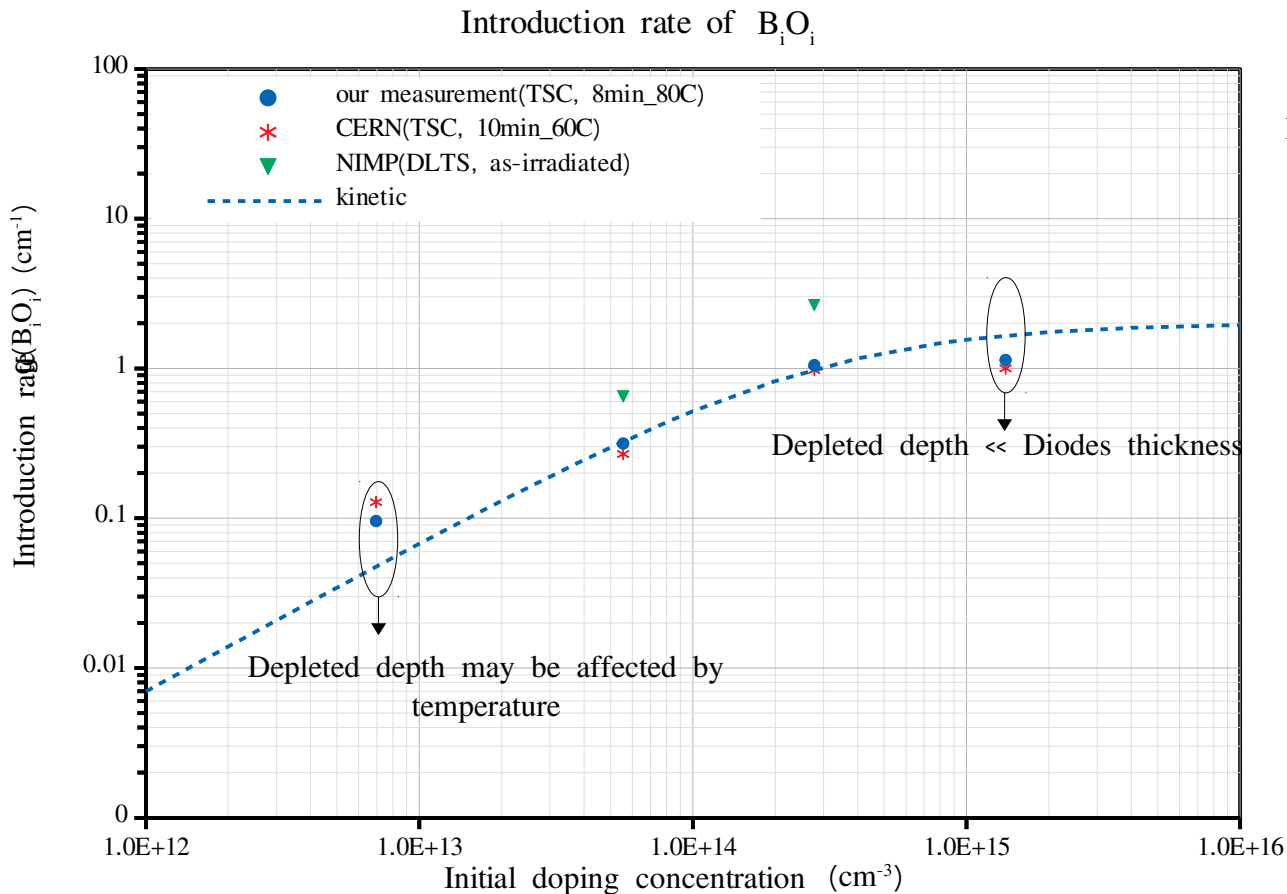


Fig 20. TSC spectra for 10  $\Omega$ cm diodes

Resistivity of measured diodes	$V_{fill}$	$V_{bias}$	$T_{fill}$	$V_{fd}$ non-irradiation	$V_{fd}$ (C-V)	Depleted depth( $\mu$ m)
50 $\Omega$ cm	5V	-300V	10K	-372.0V	-180V	42.8
250 $\Omega$ cm	10V	-300V	10K	-74.4V	-20V	42.2
2000 $\Omega$ cm	10V	-100V	10K	-9.3V	-4V	41.8

Resistivity of measured diodes	$V_{fill}$	$V_{bias}$	$T_{fill}$	$V_{fd}$ non-irradiation	$V_{fd}$ (C-V)	Depleted depth( $\mu$ m)
10 $\Omega$ cm	2V	-30V	10K	-1860.2V	-1674.2V	5.74
	2V	-60V	10K			8.03
	2V	-100V	10K			12.62



Definition of  $g(B_iO_i)$ :

$$g(B_iO_i) = \frac{N_{t,B_iO_i}}{\phi_{eq}}$$

Defect Kinetics Model [1-2]:

$$g(B_iO_i) \approx g_I \times \left(1 + \frac{k_{IC}[C_s]}{k_{IB}[B_s]}\right)^{-1}$$

$$g_I \approx 2 \text{ cm}^{-1}$$

$$\frac{k_{IB}}{k_{IC}} \approx 7$$

$$[C_s] \approx 2 \times 10^{15} \text{ cm}^{-3}$$

Fig 21. Introduction rate vs initial doping concentration

For 250 and 2000  $\Omega\text{cm}$ , lowest defect concentration was used for calculated introduction rate (depleted depth range into the p+ region), defect concentration of 10  $\Omega\text{cm}$  given by average value of -30 V, -60 V and -100 V TSC measurements.

[1] Makarenko, Leonid F., et al. physica status solidi (a) 216.17 (2019): 1900354.

[2] Moll, Michael. "Acceptor removal-Displacement damage effects involving the shallow acceptor doping of p-type silicon devices." (2019).

I.  $B_iO_i$  defect for 50  $\Omega\text{cm}$  diodes irradiated with 23 GeV protons

Describe  $B_iO_i$  peak on TSC spectra:

3-D Poole Frenkel effect (shift with bias voltage)

Annealing behaviors:

Isothermal annealing 80 °C:

- $B_iO_i$  is stable

Isochronal annealing:

- If  $T_{\text{ann}} > 150$  °C,  $[B_iO_i]$  decrease
- $T_{\text{ann}} = 170$  °C and  $T_{\text{ann}} = 180$  °C shows that the change  $\Delta N_{\text{eff}} \approx 2 \times \Delta N_t(B_iO_i)$  as expected from  $B_s(-) \rightarrow B_iO_i(+)$

II. X-defect:

- Hole trap, shift with bias voltage, a long tail at T below  $T_{\text{max}}$
- After isochronal annealing, X-defect slightly increases
- Temperature dependent capture cross section of holes

III. Measurements for diodes with different resistivity :

- $V_{\text{fd}}$  after irradiation to  $\Phi_{\text{eq}} = 4.28 \times 10^{13} \text{ cm}^{-2}$  significantly decrease for high resistivity material (50  $\Omega\text{cm}$ , 250  $\Omega\text{cm}$ , 2k  $\Omega\text{cm}$ ).
- Higher initial doping concentration leads to higher  $B_iO_i$  introduction rate after the same fluence value. But the increase is not linear .

Further plan : the degradation of the gain in LGAD device seems to be dominated by the boron removal in the gain layer. This has to be validated microscopic studied on irradiated LGADs.